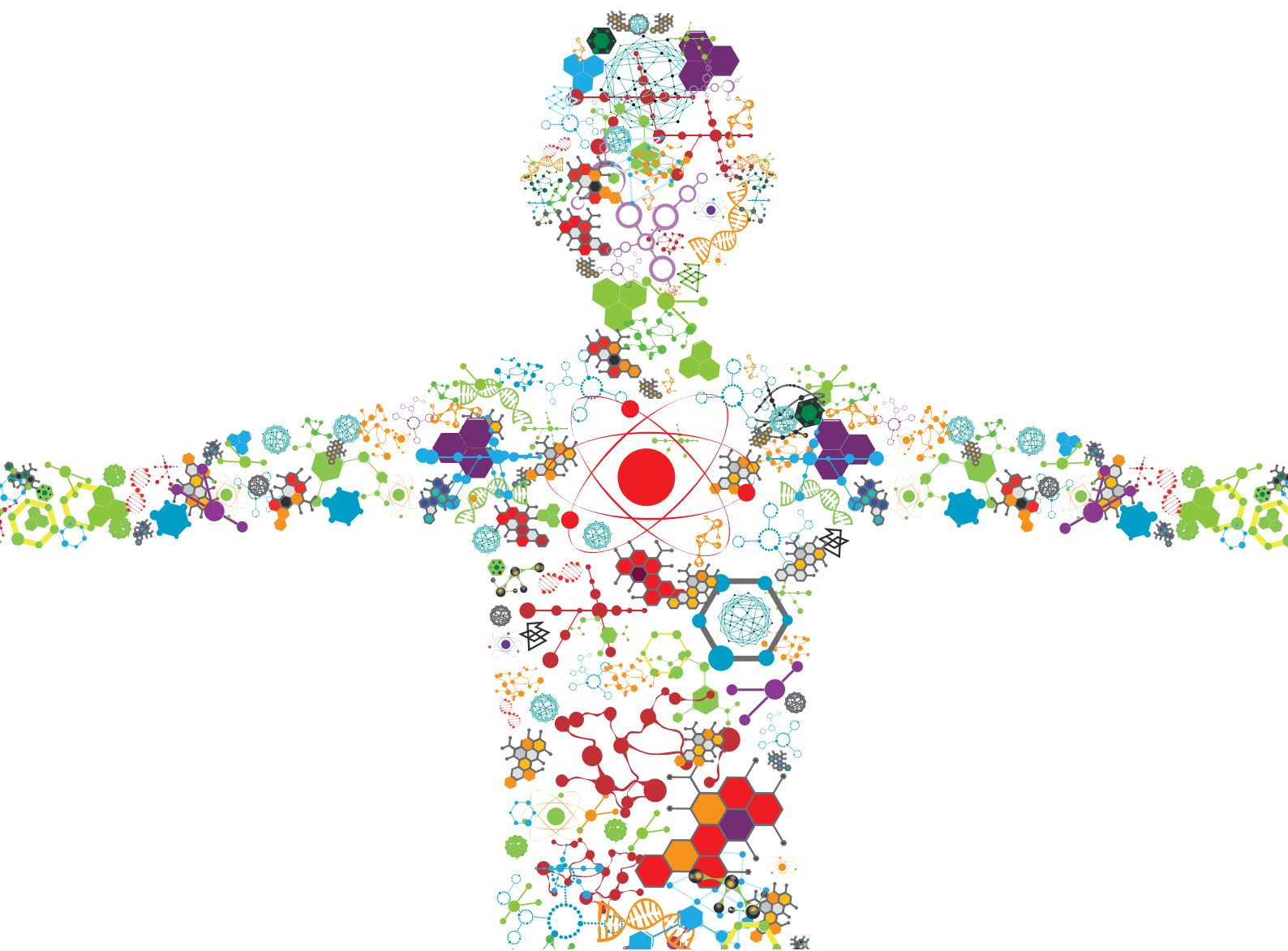


# BIOMATERIALS FOR ENGINEERING CELLULAR ENVIRONMENTS IN TISSUE ENGINEERING

EDITED BY: Hon Fai Chan, Hae-Won Kim and Xin Zhao

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# BIOMATERIALS FOR ENGINEERING CELLULAR ENVIRONMENTS IN TISSUE ENGINEERING

Topic Editors:

**Hon Fai Chan**, The Chinese University of Hong Kong, China

**Hae-Won Kim**, Institute of Tissue Regeneration Engineering (ITREN), South Korea

**Xin Zhao**, Hong Kong Polytechnic University, Hong Kong, SAR China

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# A Polydopamine-Functionalized Carbon Microfibrous Scaffold Accelerates the Development of Neural Stem Cells

Yanru Yang<sup>1</sup>, Yuhua Zhang<sup>2</sup>, Renjie Chai<sup>2,3,4,5\*</sup> and Zhongze Gu<sup>1\*</sup>

<sup>1</sup> State Key Laboratory of Bioelectronics, Southeast University, Nanjing, China, <sup>2</sup> Key Laboratory for Developmental Genes and Human Disease, Ministry of Education, Institute of Life Sciences, Southeast University, Nanjing, China, <sup>3</sup> Co-innovation Center of Neuroregeneration, Nantong University, Nantong, China, <sup>4</sup> Institute for Stem Cell and Regeneration, Chinese Academy of Science, Beijing, China, <sup>5</sup> Jiangsu Province High-Tech Key Laboratory for Bio-Medical Research, Southeast University, Nanjing, China

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### Edited by:

Xin Zhao,  
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Baolin Guo,  
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Yun Qian,  
Shanghai Jiao Tong University, China

### \*Correspondence:

Renjie Chai  
renjiechai@seu.edu.cn  
Zhongze Gu  
gu@seu.edu.cn

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Neuroregenerative medicine has witnessed impressive technological breakthroughs in recent years, but the currently available scaffold materials still have limitations regarding the development of effective treatment strategies for neurological diseases. Electrically conductive micropatterned materials have gained popularity in recent years due to their significant effects on neural stem cell fate. Polydopamine (PDA)—modified materials can also enhance the differentiation of neurons. In this work, we show that PDA-modified carbon microfiber skeleton composites have the appropriate conductivity, three-dimensional structure, and microenvironment regulation that are crucial for the growth of neural stem cells. The design we present is low-cost and easy to make and shows great promise for studying the growth and development of mouse neural stem cells. Our results show that the PDA-mediated formation of electrically conductive and viscous nanofiber webs promoted the adhesion, organization, and intercellular coupling of neural stem cells relative to the control group. PDA induced massive proliferation of neural stem cells and promoted the expression of Ki-67. Together, our results suggest that the composite material can be used as a multifunctional neural scaffold for clinical treatment and *in vitro* research by improving the structure, conductivity, and mechanical integrity of the regenerated tissues.

**Keywords:** three-dimensional scaffolds, biomaterials, polydopamine, carbon microfiber, neural stem cell

## INTRODUCTION

Neurodegenerative diseases lead to severe sensory and motor damage in the human body and are among the leading causes of social disability and death globally, and the lack of transplantable nerve tissue is the prime hindrance to clinical nerve repair. Due to their multipotency and self-renewal, neural stem cell (NSC) transplantation therapy is one of the most promising strategies for treating traumatic injuries and neurodegenerative disorders. *In vitro* NSC expansion (to increase NSC resources) through nerve tissue engineering (NTE) methods might provide the most effective way to solve this clinical shortage. The ideal tissue engineered nerves are composed of scaffold materials, a microenvironment that induces or promotes cell growth, and NSCs. Biomaterials and

their biocompatible modifications, for example, oriented substrates like electrospun microfibers, conductive materials such as graphene, and ordered materials like photonic crystals are gaining widespread use in the field of NTE applications because the scaffolds have shown the unique characteristics of topological cues and bioelectric properties for mimicking the extracellular matrix (ECM) or for performing other functions (Martino and Pluchino, 2006; He et al., 2012; Limongi et al., 2018).

In recent years, electrospinning technology has generated considerable interest in fabricating tissue engineering biomaterials, mainly due to the morphological and structural properties of such materials, which almost perfectly mimic the fibrillary 3D architecture of the natural ECM (Carlberg et al., 2009; Zhu et al., 2018). Furthermore, an increasing body of evidence suggests that NSC fate—including cell adhesion, differentiation, migration, and proliferation—is greatly affected by the pattern of biochemical, physical, and topographical properties that constitute the essential elements of their niches *in vivo* (Kam et al., 2008; Heo et al., 2011; Yang et al., 2012; Ma et al., 2016). The electrospun scaffolds with aligned fibers display the versatility of ECM mimicry, which required in NTE, and several studies have been performed using this feature of electrospun fibers to induce the differentiation of NSCs. Gregory et al. fabricated electrospun polyethersulfone fiber mesh to study the influence of different fiber diameters of electrospun substrates on rat hippocampus-derived adult NSC differentiation and proliferation (Christopherson et al., 2009). Qi et al. produced a versatile substrate for insulin-like growth factor 1 delivery in a graphene oxide-incorporated poly (lactic-co-glycolic acid) electrospun nanofibrous mat and found improved NSC survival, diffusion, and differentiation (Qi et al., 2019). The electrospun micro- or nanofibers that the cells are interfacing with influence NSC growth, and the physical properties of the spun materials (such as electrical conductivity, mechanical strength, etc.) guide the development of NSCs. Neurons are electroactive cells, and cell dysfunction is also closely related to electrical activity. Electrical stimulation can induce depolarization of excitatory cell membranes and induce functional neuronal responses, and thus conductive materials could be a promising option for applications in NTE. To date, conductive biomaterials have been widely used for tissue engineering, such as they can apply in skin wound healing (He et al., 2000), 3D myoblast guidance (Wang et al., 2015), peripheral nerve tissue engineering (Wu et al., 2016) and engineered 3D cardiac anisotropy (Wu et al., 2017) and so on. Neuronal cells are quickly excited and transmit electrical signals in one direction, and the use of conductive electrospun nerve scaffolds to promote autologous nerve repair has become a trend in recent years. The electrical stimulation on these scaffolds might also be an important factor for facilitating neurite extension and differentiation (Koppes et al., 2016; Granato et al., 2018; Uz and Mallapragada, 2019). Cellulose has excellent conductivity after carbonization, so it is a great original material for conductive electrospinning technology.

Polydopamine (PDA) is a highly viscous and highly adsorbent polymer membrane alkaline formed by the self-polymerization of the neurotransmitter dopamine. PDA has the characteristics

of a polyelectrolyte in a primary environment due to the deprotonation of the catechol group. Thus, in the presence of alkaline pH, an electrostatic interaction is generated between the PDA and cationic polyelectrolytes. Alternating deposition on the surface of the material allows PDA to self-assemble the biomaterials layer by layer (Lee et al., 2007; Wei et al., 2010; Faure et al., 2013). PDA can form a tightly adhering composite layer on the surface of almost any solid material, and this one-step modification is efficient, versatile, cost-effective, and environmentally friendly. Thus, PDA-mediated surface coating methods can be used to modify a wide variety of biomaterials for a diverse array of functional applications in stem cell-based tissue engineering. As an example, Wu et al. demonstrated that a PDA-based coiled polycaprolactone scaffold could promote mesenchymal stem cell differentiation (Shin et al., 2011). Another group studied the combined effects of PDA functionalization and collagen on micro-grated polydimethylsiloxane and showed that the material improved the culture of human mesenchymal stem cells (Sharma et al., 2019). These studies have shown that PDA-modified biomaterials can stimulate neurons and guide axonal extension even in the absence of some functions such as conductivity, topography, and microenvironment control. However, as Yang et al. (2018) proposed, the existing studies have not considered adding such factors together to regulate NSC growth behavior and thus identify more efficient and economical methods for use in NTE. Therefore, the combination of a PDA functionalized scaffold and carbonized electrospun fiber with unique electrochemical properties might be a new strategy for the design of biomaterials and microenvironments for NTE.

The goal of this study was to develop an electrically conductive and adherent 3D scaffold and to examine the effect on NSC development by combining the effects of PDA functionalization and electrical stimulation with 3D structures. Our results indicate that compared to the untreated carbon fiber (CF) group, the PDA-mediated collection of electrically conductive and viscous microfiber webs improved the adhesion, organization, and intercellular coupling of NSCs. The presence of PDA resulted in a significantly increased proliferation of NSCs and increased expression of Ki-67. Representative 3D reconstructions of confocal microscope images demonstrated that NSCs quickly grew into the interior of the PDA-CF but only accumulated across the surface of the CF scaffold. The expression of vinculin by NSCs was also significantly enhanced in NSCs grown on PDA-CF compared to those grown on CF. These results indicate that PDA promotes the adherence and proliferation of NSCs on the conductive 3D CF skeleton and thus that PDA-CF might be used to prepare multifunctional neuron scaffolds for both treatments and for *in vitro* research.

## MATERIALS AND METHODS

### Fibrous Scaffold Fabrication

To prepare microfibers, surgical cotton batting (DP of 1140) was ultrasonically washed with absolute ethanol for at least 2 h and dried under vacuum at 50°C and then ground into a powder. The electrospinning solution (1.1 wt%) was prepared

by dissolving the fine cellulose particles in lithium chloride/N, N-dimethylacetamide (Kim et al., 2005) for 12 h at room temperature. The cellulose solution was loaded into a 30 ml syringe and pushed through a 0.86 mm inner diameter stainless-steel needle with a set flow rate of 0.2 ml/min and a 20-kV high-voltage source. The distance between the needle and the collector was set at 20 cm. The electrospun fibrous scaffolds were finally peeled off from the screen (the aluminum foil), run through a solid-phase extraction column (Dong qi Bio-technology Co., China), and dehydrated in a CO<sub>2</sub> critical point dryer (HCP-2, Hitachi Ltd) before subsequent carbonization.

## Characterization of the CF Scaffold

The physicochemical properties of the surfaces of the tested materials were determined by Raman spectrometry (RAMAN, inVia, Renishaw Inc.) with a 532 nm laser. To conduct the wettability test, the samples were placed on a microscope slide and pressed slightly to smooth the surface and the contact angle was measured (JC2000D, Powereach, China). The morphological features of the materials were determined by sputter coating with a gold layer using a Hitachi E-1010 ion sputter (Hitachi, Ltd., Japan) and then imaging by scanning electron microscopy (SEM, Zeiss Ultra Plus, Germany). The elemental composition of the material was determined in real-time using an X-Max Energy Dispersive Spectrometer (EDS). The particular PDA-CF and CF groups of the sample were identified by Vertex 70-type Fourier-transform infrared spectroscopy (FT-IR). A 5–10 mg sample was mixed with potassium bromide and pressed into thin slices, and the test was performed in transmissive mode with 32 scans.

## Isolation and Culture of NSCs

The NSCs were isolated from the hippocampus zones of embryonic day 18.5 wild-type mouse brains. The hippocampus was removed and collected in a 1.5 ml centrifuge tube and then dissociated by Accutase (Gibco, USA) for 25 min in a 37°C incubator. The tissue was triturated mechanically by using pipet tips. NSCs were suspended in DMEM-F12 medium (Gibco, USA) containing 2% B-27, 20 ng/ml EGF, and 20 ng/ml FGF (R&D Systems, USA) and cultured in a humidified atmosphere with 5% CO<sub>2</sub> at 37°C. The passage of NSCs was performed every 7 days during culturing. Before seeding, the CF and PDA-CF scaffolds were coated with laminin (10 µg/ml, 37°C overnight). After purifying three generations, the NSCs for the proliferation studies were seeded at a concentration of  $5 \times 10^5$  cells/ml in proliferation medium consisting of DMEM-F12 medium with 2% B-27 supplemented with 20 ng/ml EGF and 20 ng/ml FGF. For the differentiation studies, NSCs were seeded at a similar concentration in DMEM-F12 medium containing 2% B-27 supplemented with fetal bovine serum (GIBCO, USA) and 1 µM retinoic acid (Sigma). The care and use of animals in these experiments followed the guidelines and protocols approved by the Care and Use of Animals Committee of Southeast University. All efforts were made to minimize the number of animals used and their suffering.

## SEM Observation of NSCs Cultured on PDA-CF

After culturing for 5 days, NSC cultures were washed with PBS two times and then fixed with 2.5% glutaraldehyde overnight at 4°C. The samples were dehydrated by an ethanol gradient series (30, 50, 70, 80, and 90%), lyophilized, sputter-coated with gold for 2 min, and then analyzed by SEM (Zeiss Ultra Plus, Germany).

## Cell Viability/Proliferation Assay

Cell viability and proliferation were assessed by CCK-8 assay. After culturing in proliferation media for 3 days, NSCs were incubated with CCK-8 reagent (diluted 20-fold in culture medium) for 2 h before the media were collected. The tetrazolium salt in CCK-8 is reduced by dehydrogenase in the cytoplasm to produce a yellow color. The intensity of the yellow color—which corresponds to the number of viable cells—was analyzed by a plate reader at 450 nm.

At the same time, after 5 days of culture cell viability was tested using the LIVE/DEAD viability/cytotoxicity kit for mammalian cells (Invitrogen, USA) according to the manufacturer's instructions. The percentage of living cells was calculated by determining the percentage of calcein-AM-positive cells out of the total cell number.

## Immunofluorescence Staining of Cultured NSCs

NSCs were washed with PBS and fixed in 4% paraformaldehyde (PFA) for 30 min at room temperature, followed by blocking with PBS (pH 7.4) containing 5% donkey serum, 0.1% Triton X-100, 0.02% sodium azide (NaN<sub>3</sub>), and 1% bovine serum albumin for 1 h at room temperature. This was followed by overnight incubation with primary antibodies at 4°C. Immunostaining was resumed by washing with PBS and incubating with secondary antibody (Invitrogen) for 1 h at room temperature followed by washing and then mounting in antifade fluorescence mounting medium (DAKO). The antibodies used in this study were anti-Ki-67, anti-TuJ1, and anti-nestin (Abcam, USA). Cell proliferation was measured with the Click-it EdU imaging kit (Invitrogen). Three independent experiments were repeated for each assay.

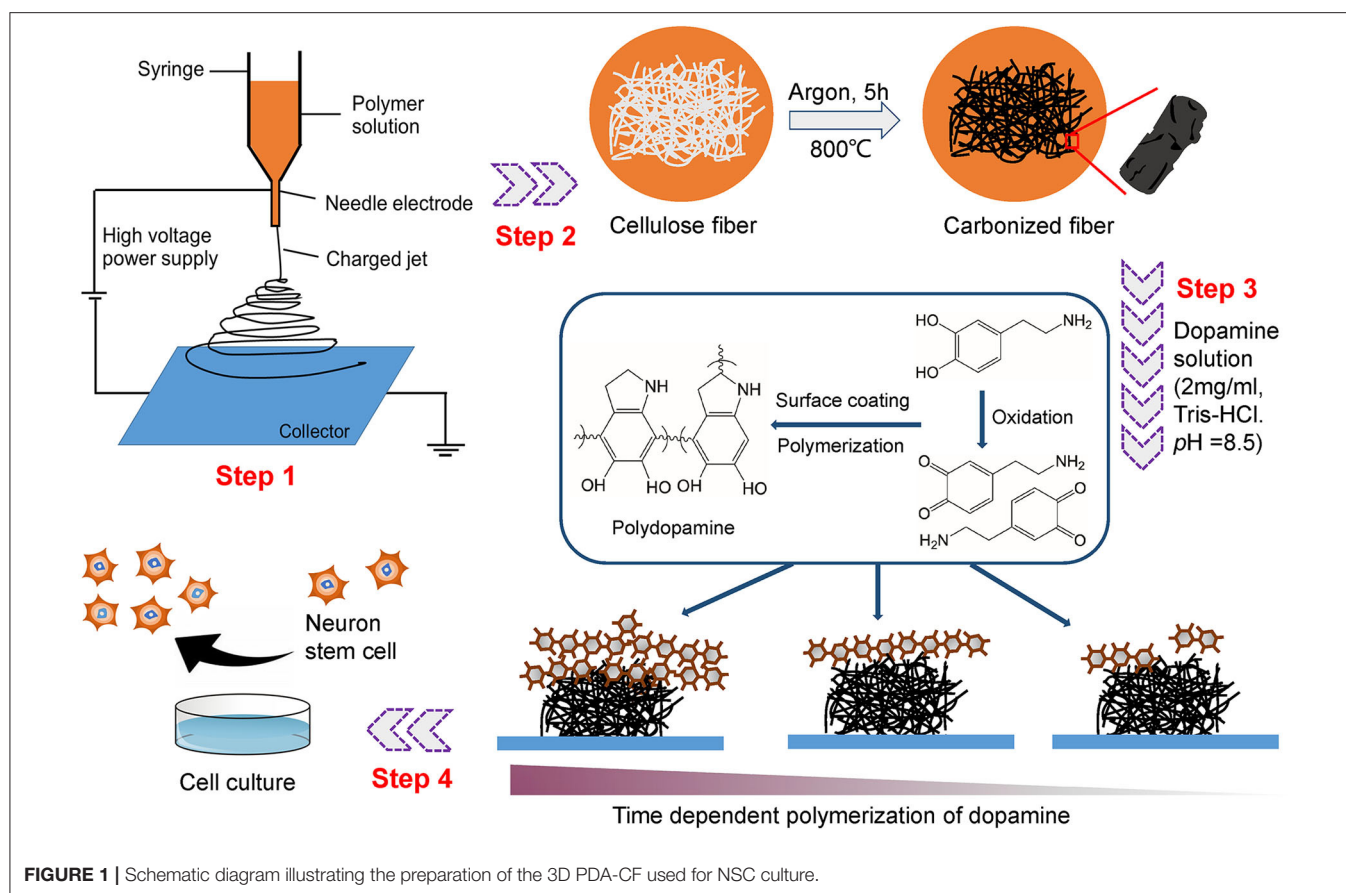
## RNA Extraction and RT-PCR

The RNA isolation, reverse transcription, and PCR analysis were performed following standard protocols. Briefly, total cellular RNA was isolated using TRIzol reagent (Life Technologies), 1 µg of RNA was reverse transcribed using TaqMan Reverse Transcription Reagents (Applied Biosystems), and the mRNA levels of the indicated genes were measured in triplicate using the SYBR Green master mixture (Applied Biosystems) and a Chromo-4 real-time RT-PCR instrument (MJ Research). The mRNA levels were normalized to GAPDH (internal control), and gene expression was presented as fold changes (Ct method).

## Statistical Analysis

All experiments were repeated three times in triplicate for each assessment, and all data are expressed as the mean ± standard deviation. Data were analyzed by one-way by Tukey's tests, and a *p* of <0.05 was considered statistically significant.





**FIGURE 1** | Schematic diagram illustrating the preparation of the 3D PDA-CF used for NSC culture.

## RESULTS AND DISCUSSION

### Fabrication and Characterization of the PDA-CF Scaffold

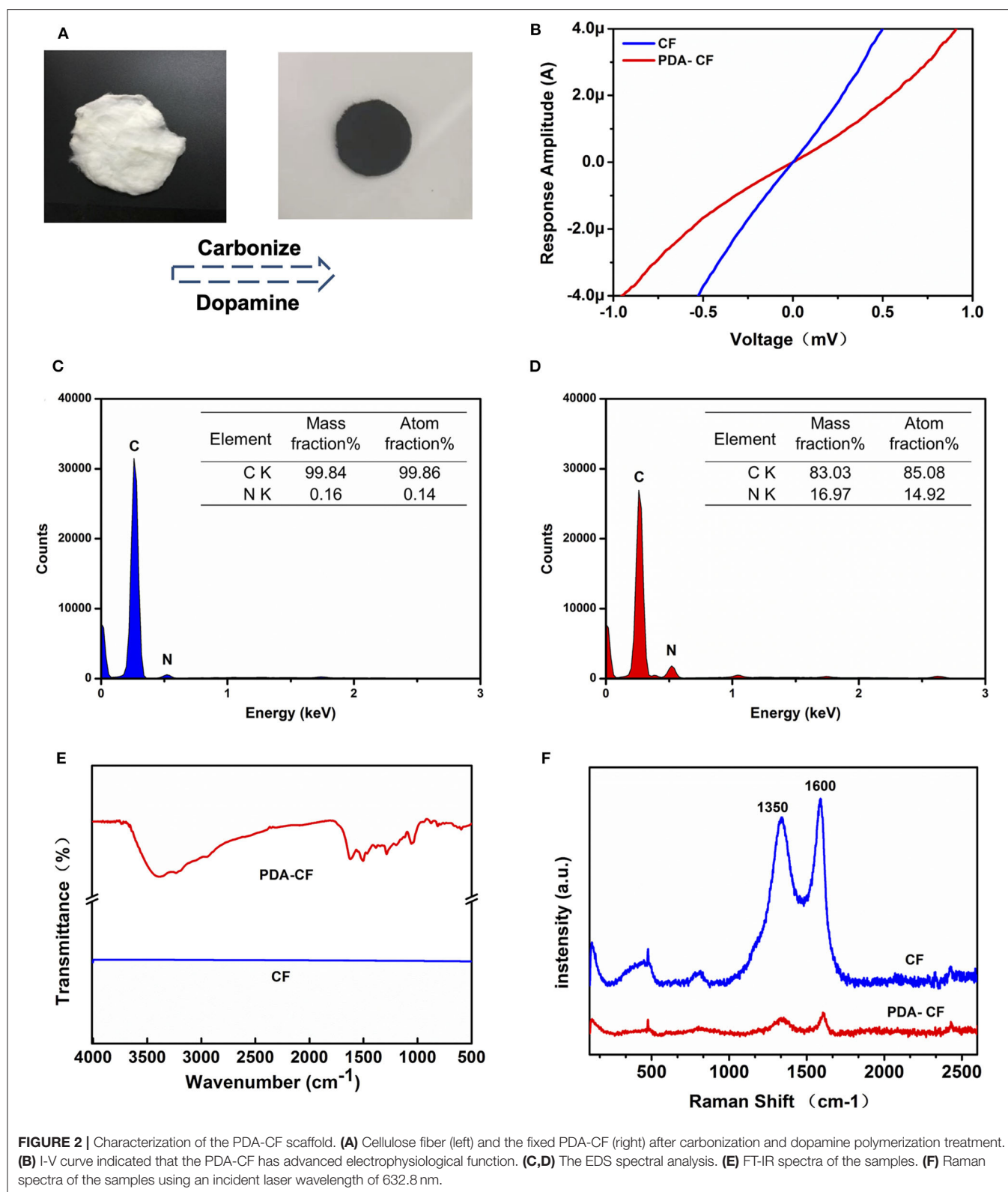
**Figure 1** shows the overall procedure for the fabrication of the PDA-CF scaffold and its application in NSC culture. First, the cellulose microfibers were fabricated from natural cellulose by electrospinning, and the cellulose fibers formed carbon aerogels after high-temperature carbonization (heat treatment in the tube furnace at 800°C for 5 h). The synthesis of PDA is depicted in step 3. Dopamine (2.0 mg/ml) was used as the precursor, and CFs doped with PDA self-assembled under alkaline conditions. The carbonized fibers show excellent flexibility which can be bent into different angles and folded as Li et al. (2016) presented in their study, the material was finally fixed on a circular cover glass with a diameter of 1 cm (**Figure 2A**).

The morphological features of the tested materials before and after dopamine treatment were characterized by SEM. The CFs (**Figure 3A**) retained almost all the topological cues from electrospinning before thermal annealing. In the image of PDA-CF (**Figure 3B**), it can be seen that the CF surface was very rough and that the fibers were covered with clusters of particles that we speculated were the PDA layer. Further evidence for this came from the EDS analysis and images (**Figures 2C,D**). The carbon content of the untreated CF was 99.84%, and the nitrogen

content was only 0.16%. After dopamine treatment, the nitrogen content increased to 16.97%, and because the additional nitrogen mainly derived from the PDA layer, this indicated that PDA had effectively been deposited onto the CF.

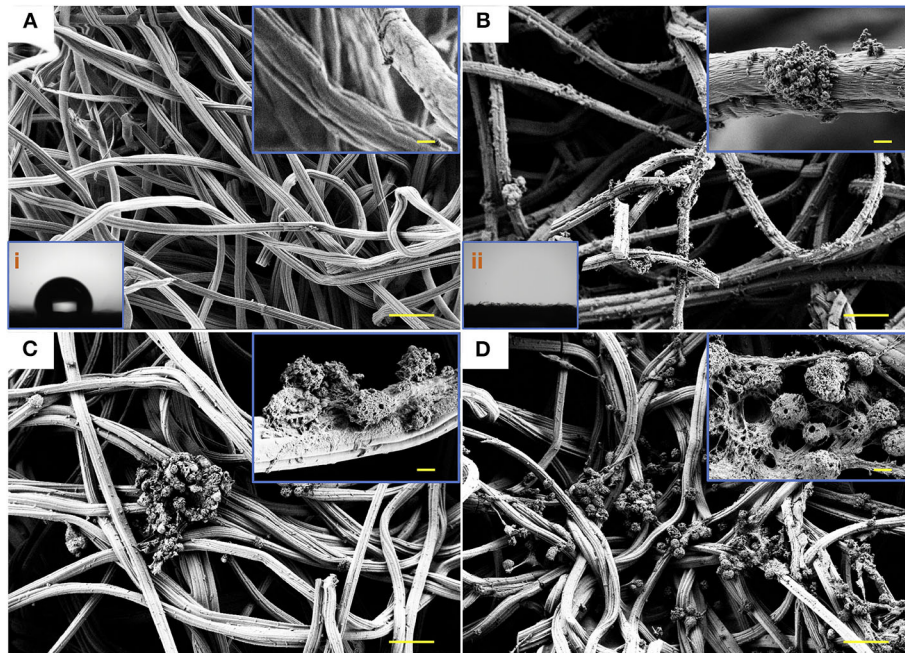
FT-IR (**Figure 2E**) also illustrated this point. The analysis of the material by infrared spectroscopy showed a straight line for CF with no absorption vibration peaks. This result is consistent with the characteristics of the material itself because the cotton cellulose material is a pure carbon sample after high-temperature carbonization. The infrared absorption of PDA-CF varied between 1200 and 1600  $\text{cm}^{-1}$ . According to the literature, 1512  $\text{cm}^{-1}$  is the absorption peak of N-H, and 1284  $\text{cm}^{-1}$  is the tensile vibration peak of a typical polymer C=O, both of which are characteristic absorption peaks in the PDA structure (Kim et al., 2005; Wang et al., 2008; Tang et al., 2015; Li et al., 2016). Taken together the results of SEM, EDS, and FI-IR analysis confirmed that PDA successfully coated the surface of the CFs.

The Raman analysis (**Figure 2F**) showed two peaks for graphite, one at 1,596  $\text{cm}^{-1}$  corresponding to the G band and one at 1,350  $\text{cm}^{-1}$  corresponding to the D band (Li et al., 2016). The conductivity test (**Figure 2B**) showed that the PDA-CF was electrophysiologically functional. Furthermore, we calculated the resistance from the I-V curve of the CF and PDA-CF scaffolds and found that the PDA-CF had a greater resistance (around 1,200  $\Omega$  in an area of 500 mm  $\times$  500 mm) compared to the



CF (540  $\Omega$ ). The main reason for this difference might be that the adhesion of the PDA reduces the conductivity of the material. As carbonized cellulose microfibers are in highly dense

and hydrophobic before PDA modification, the solution is not entirely covering the surface, leading to the PDA layer coat on CF as particle clusters, which is the best method to protect the



**FIGURE 3 |** SEM micrographs and water contact angle. SEM images illustrating the networked surfaces of CF (A) and PDA-CF (B) and the water contact angle (i and ii) of the two materials. The growth of NSCs cultured on CF (C) and PDA-CF (D). Scale bars in the high-magnification images are 20  $\mu\text{m}$ , and in the low-magnification images are 0.5  $\mu\text{m}$ .

conductive properties of the materials. We further analyzed the water contact angle [Figures 3A(i),B(ii)] on the CF and PDA-CF substrates. Due to the hydrophobic nature of the CF material itself, the water does not wet it and instead forms beads on the surface of the sample. In contrast, water droplets on the PDA-CF surface quickly penetrate the sample layer, and no beads of water are visible on the surface of the sample. Thus, the PDA modification transforms the relatively hydrophobic CF material into a highly hydrophilic material, which is beneficial for the adhesion and growth of cells.

## The Biocompatibility of PDA-CF

We isolated and amplified NSCs from the hippocampus of fetal mice and seeded them on the PDA-CF and CF scaffolds. In most of the SEM observations, as illustrated in Figures 3C,D, the PDA-CF scaffold seemed to enhance NSC proliferation. The calcein acetoxymethyl ester (calcein-AM) and ethidium homodimer-1 (EthD-1) staining assays were used to evaluate the cytotoxicity of PDA-CF. As shown in Supplementary Figure 1A, almost 83% of the cells cultured on PDA-CF for 5 days were alive, and there were significant differences between PDA-CF and CF in terms of cell viability (Supplementary Figures 1B,C), thus suggesting that PDA-CF has exceptional biocompatibility.

The cells were stained with DAPI after culturing on the scaffolds for 1 day, 5 days, and 10 days, and representative 3D reconstructions were made of the confocal microscope images with different colors showing cells at different depths. From the z-axis, we found that NSCs grew quickly into the interior of the PDA-CF after 5 days of cell culturing (Figure 4A), while NSCs

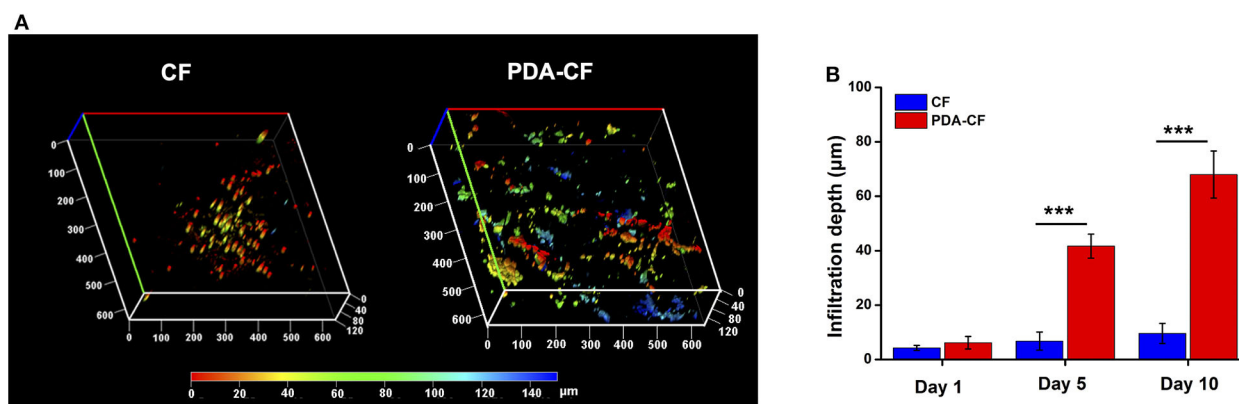
primarily expanded along the surface of the CF scaffold. We also measured the cell infiltration depth on the two materials (Figure 4B), and the results were consistent with what we observed directly from the images. The greater infiltration of NSCs grown on PDA-CF suggests that PDA-CF might promote cell migration and cell viability.

## NSC Proliferation on PDA-CF

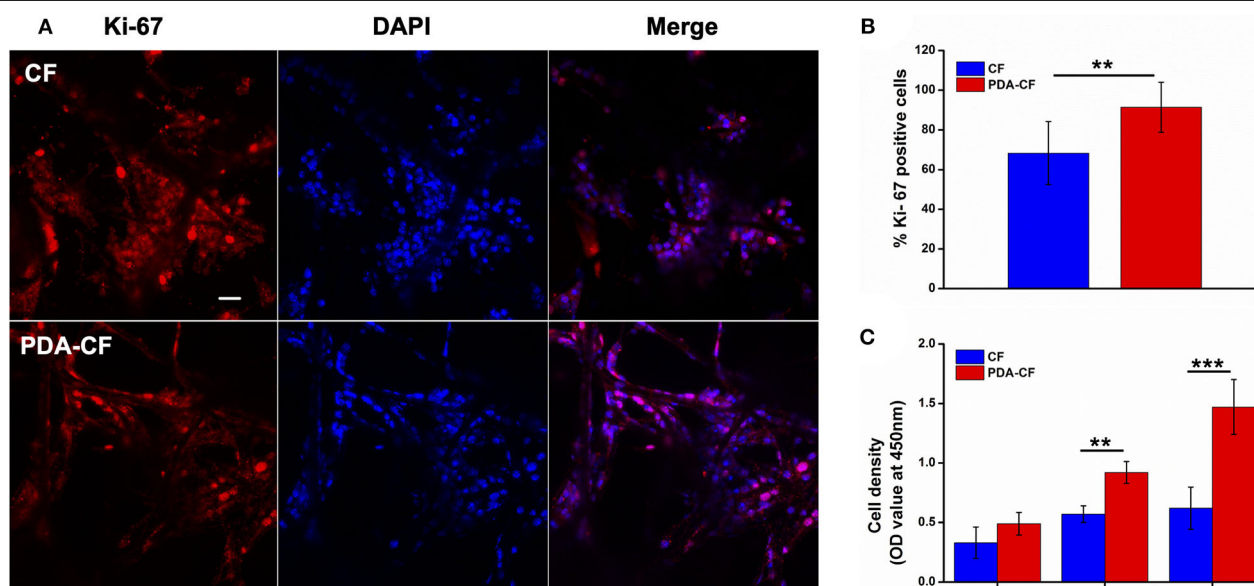
The effect of PDA-CF on NSC proliferation was assessed by Ki-67, nestin, and 5-ethynyl-2'-deoxyuridine (EdU) immunofluorescence staining. The Ki-67 protein, which is required for cell division, is an active mitotic marker of proliferating cells during neurogenesis. Nestin is the intermediate filament protein for labeling and identifying NSCs. EdU is a thymidine analog that is incorporated into replicating chromosomal DNA during the S phase of the cell cycle and is widely employed to measure cell proliferation in the nervous system (Scholzen and Gerdes, 2000; Chehrehasa et al., 2009; Park et al., 2010). The expression of Ki-67 protein after 5 days of culture is presented in Figure 5A, and quantification (Figure 5B) showed that NSCs cultured on PDA-CF had a greater number of Ki-67-positive cells. The CCK-8 results (Figure 5C) suggested that PDA-CF could promote cell proliferation better than CF.

The percentage of EdU-positive cells (Figure 6A) also showed that NSCs grew better on the PDA-functionalized scaffold. After the NSCs were cultured for 1 day, the NSCs were treated with 10  $\mu\text{M}$  EdU to label the mitotic cells. To quantify the proliferating NSCs, we stained the NSCs for the stem cell marker nestin.





**FIGURE 4 |** Representative 3D reconstructions of confocal laser scanning microscopy mapping (A) after 5 days of culture. The depth of cell growth on the materials can be seen from the z-axis. (B) Mean infiltration depths on the CF and PDA-CF scaffolds (\* $p < 0.05$ ; \*\* $p < 0.005$ ; \*\*\* $p < 0.0005$ ).



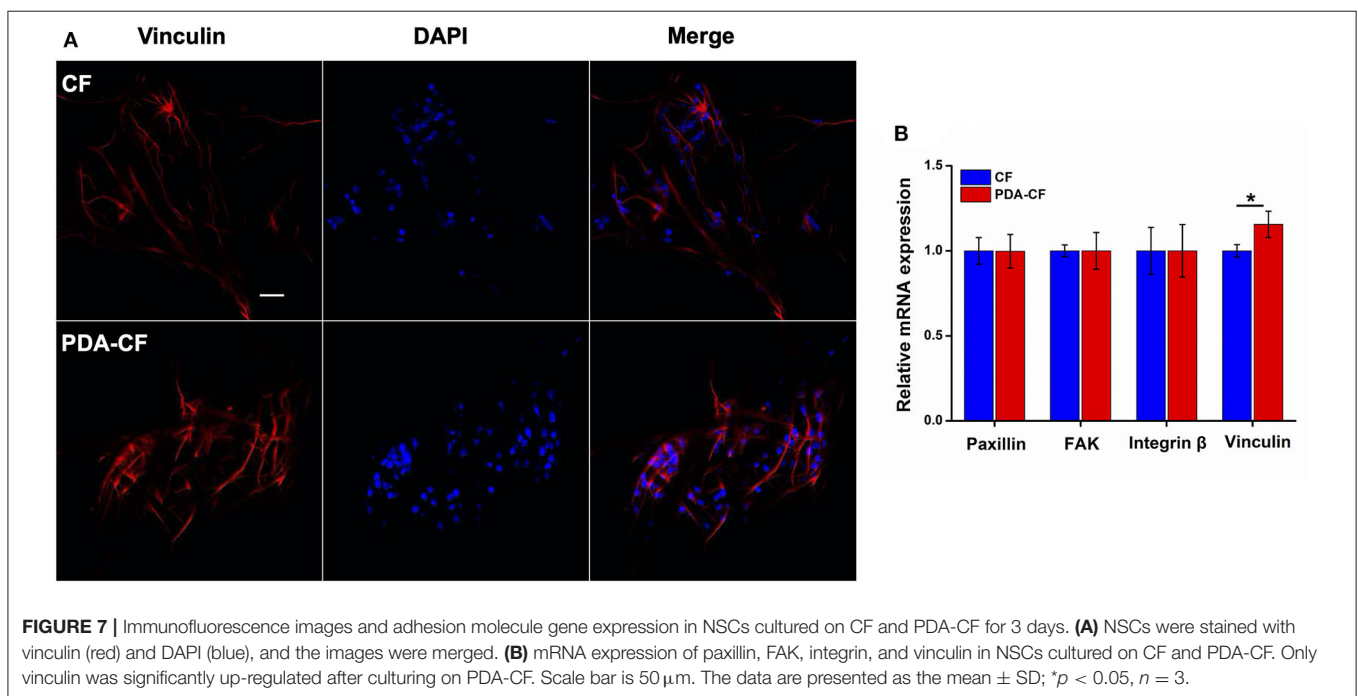
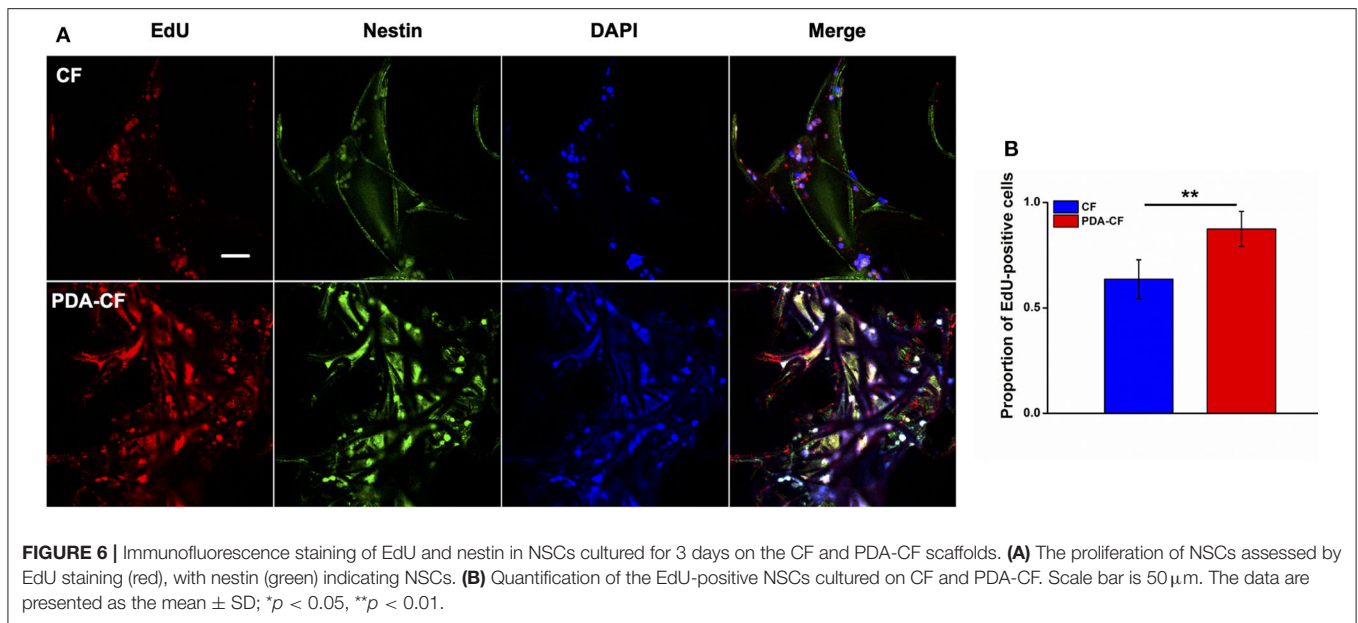
**FIGURE 5 |** PDA-CF increased the proliferation of NSCs compared to the CF control. (A) Laser scanning confocal microscopy in NSCs showing immunostaining of Ki-67 and DAPI staining. (B) The proportions of Ki-67-positive cells after 5 days of culture. (C) CCK-8 analysis of relative cell viability. Scale bar is 50 μm (\*\* $p < 0.005$ ; \*\*\* $p < 0.0005$ ).

After 3 days, the number of EdU-positive NSCs on the PDA-CF scaffold was significantly greater than on the CF scaffold. Quantification of the EdU-positive cells (Figure 6B) showed that the mean proportion of EdU-positive cells on the PDA-CF scaffold was 87.43%, but was only 68.9% on the CF scaffold. These results suggested that PDA-CF could substantially promote the proliferation of NSCs.

### Change in Expression of Adhesion Factors

Cell adhesion plays an essential role in regulating cell proliferation, differentiation, migration, and apoptosis. The SEM micrographs in Figure 3D show that NSCs adhered well to the scaffold surface. The association between NSCs proliferation

processes and adhesion factor expression needs to be verified. To assess NSC adhesion, the gene expression of paxillin, focal adhesion kinase (FAK), integrin  $\beta$ , and vinculin was measured by quantitative real-time PCR using GAPDH as the control. Vinculin is a cytoskeletal protein associated with focal adhesion that regulates cell proliferation (Bays and DeMali, 2000). For NSCs cultured on PDA-CF for 3 days, immunohistochemistry showed increased expression of vinculin protein (Figure 7A), and quantification of vinculin (Figure 7B) showed increased expression compared to the CF group, which suggests that PDA might promote NSC proliferation by regulating the expression of vinculin. The interaction between integrins and the ECM induces the expression of paxillin, which is a central component



of focal adhesion complexes and plays a vital role in cell attachment, spreading, and migration through the transduction of extracellular signals into intracellular responses (Lo'pez-Colome et al., 2017; Ma and Hammes, 2018). Cells attach to the ECM through trans-membrane integrin proteins and use these interactions to sense their environment (Kechagia et al., 2019). We measured the expression of integrin β by quantitative real-time PCR, and we found no significant changes between the two groups, which means that the proliferation of NSC on PDA-CF has no relationship with this biomolecule synthesis. In addition,

we also assayed the expression of FAK, which is a cytoplasmic non-receptor tyrosine kinase that enables activation by growth factor receptors or integrins in various types of human cancers. It is a critical regulator in promoting cancer proliferation (Bullard Dunn et al., 2010; Tai et al., 2015). While from the results, the characteristics of PAD-CF scaffold seems to have no effect on the expression of FAK. In conclusion, PDA-CF might influence vinculin protein and thus regulate NSC proliferation.

In addition, we explored the effect of the PDA-CF scaffold material on the differentiation of NSCs using βIII-tubulin (TuJ1),

the expression of which has been widely used as a marker of early neuron development (Ambasudhan et al., 2011). We measured the immunofluorescence staining of TuJ1 at 1 day, 5 days, and 10 days after NSC seeding, and during the period of differentiation detection TuJ1 was expressed weakly in both the PDA-CF and CF groups and there was no significant difference between the two groups (**Supplementary Figure 2A**). Meanwhile, quantitative analysis of the NSC differentiation-related genes *Gap*, *Map2*, and *TuJ1* indicated that there was no significant difference in the expression levels between the various genes in cells grown on the two scaffolds (**Supplementary Figure 2B**). As a result, NSCs did not differentiate into neural cells on the PDA-CF scaffold any more so than on the CF control scaffold.

## CONCLUSION

The difficulty in achieving functional recovery after nerve injury results in overwhelming medical costs and psychological burden, and thus there is a great need for methods to prevent neuronal cell loss and to promote nerve repair. The replacement of autologous nerves by nerve cells generated through NTE is a current research focus, and the development of biomimetic ECM is also of great importance. In addition, electrical stimulation of neurons in the absence of topographical features has been shown to guide axonal extension. In this paper, we have developed a low-cost and straightforward method for modifying 3D carbon microfibrous scaffolds with PDA to create a microenvironment that is more conducive to cell growth.

The first aim of this research was to fabricate and characterize a PDA-functionalized carbon microfibrous electrospun scaffold. The cellulose was made into a 3D network structure by electrospinning, and it maintained its electrical conductivity and mechanical flexibility and biocompatibility after heat treatment. The second part of this research was to evaluate the material's effects on the proliferation and differentiation of NSCs from the hippocampus of fetal mice. Our results suggest that PDA-CF acts as a neural scaffold that promotes the maturation of NSCs by accelerating cell development and adhesion as indicated by increased Ki-67 expression. Representative 3D reconstruction of confocal microscope images showed that NSCs grew quickly into the interior of the PDA-CF, while only expanding on the surface of the CF scaffold. In addition, the expression of vinculin was significantly enhanced on PDA-CF compared with the CF control group. This is very interesting, in some way, PDA seems to accelerating the development of NSCs by promoting the expression of adhesion protein vinculin.

These results indicated that PDA plays an essential role in promoting the spread of NSCs on the electrically conductive 3D CF skeleton and that the viscous microfibrous networks formed by PDA-CF are key for improving NSC attachment and development. However, the molecular mechanisms behind these

effects need to be studied further. NSCs did not differentiate into neural cells on PDA-CF any more so than they did on the CF control material, and we suspect that the reason for this is that NSCs maintained a state of self-renewal in the short time for which they were cultured on PDA-CF. Future work should extend the incubation time of NSCs to 3 weeks in order to further explore the differentiation of cells cultured on PDA-CF scaffolds. This method can be used to prepare multifunctional neuron scaffolds for nerve transplantation treatment and *in vitro* research, and these hybrid materials are also likely to find use with myocardial and other muscle cells to create tissue constructs with improved organization, electroactivity, and mechanical integrity.

## DATA AVAILABILITY STATEMENT

The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation, to any qualified researcher.

## ETHICS STATEMENT

The studies involving animal experiments were reviewed and approved by Care and Use of Animals Committee of Southeast University. The study does not involve any human sample or human data.

## AUTHOR CONTRIBUTIONS

RC, ZG, and YY designed experiments. YY and YZ carried out experiments and analyzed experimental results. YY wrote the manuscript. All authors contributed to the article and approved the submitted version.

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## SUPPLEMENTARY MATERIAL

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Building Osteogenic Microenvironments With Strontium-Substituted Calcium Phosphate Ceramics

Ben Wan, Renxian Wang\*, Yuyang Sun, Jingjing Cao, Honggang Wang, Jianxun Guo and Dafu Chen\*

Laboratory of Bone Tissue Engineering, Beijing Laboratory of Biomedical Materials, Beijing Research Institute of Traumatology and Orthopaedics, Beijing Jishuitan Hospital, Beijing, China

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### \*Correspondence:

Dafu Chen  
chendafujst@126.com  
Renxian Wang  
wrxpumc@126.com

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Bioceramics have experienced great development over the past 50 years. Modern bioceramics are designed to integrate bioactive ions within ceramic granules to trigger living tissue regeneration. Preclinical and clinical studies have shown that strontium is a safe and effective divalent metal ion for preventing osteoporosis, which has led to its incorporation in calcium phosphate-based ceramics. The local release of strontium ions during degradation results in moderate concentrations that trigger osteogenesis with few systemic side effects. Moreover, strontium has been proven to generate a favorable immune environment and promote early angiogenesis at the implantation site. Herein, the important aspects of strontium-enriched calcium phosphate bioceramics (Sr-CaPs), and how Sr-CaPs affect the osteogenic microenvironment, are described.

**Keywords:** calcium phosphate ceramics, strontium substitution, microenvironment, bone regeneration, biomaterials

## INTRODUCTION

Bone is a metabolically active specialized connective tissue with the capacity for continuous resorption and reformation, but the reconstruction process requires extra support in large bone defect cases, for example, injury, tumor excision, and age-related diseases (Loi et al., 2016; Ginebra et al., 2018). Clinically, autografts still remain the most common strategy to support and stimulate bone growth. While biologically desirable, the supply of autografts from a patient's own body cannot meet the amount required for large-sized bone defects (Wang and Yeung, 2017). Grafts from bone banks or other animal species can provide a vast range of grafting forms by virtue of their relatively easy availability (Wang and Yeung, 2017). Nevertheless, they are still subject to attenuated osteointegration and disease transmission.

Synthetic bioactive ceramics can overcome these limitations. They are obtained by chemical reagent deposition through a controlled process that allows their properties to be adapted to the specific requirements of different clinical situations. Calcium phosphate ceramics (CaPs) are basic bone repair materials with excellent osteoinductive and osteoconductive features; examples include hydroxyapatite (HAP),  $\beta$ -tricalcium phosphate ( $\beta$ -TCP), calcium polyphosphate (CPP), and biphasic calcium phosphate (BCP). They have been widely used in clinical applications in the form of implant surface coatings, the composition of cements, and scaffolds (Jeong et al., 2019). Moreover, CaPs encapsulated with pharmacologics and biologics in scaffolds have shown

multiple abilities, such as enhanced osteointegration, angiogenesis, and antibacterial properties (Bose and Tarafder, 2012; Marques et al., 2016). However, these biologically active molecules are only surface-loaded and show explosive release effects with little long-term benefit (Wu et al., 2014). This raises some concerns about their safety. For example, recombinant human bone morphogenetic protein-2 (rhBMP-2) is the first commercial synthetic osteoinductive molecule, and the combined use of rhBMP-2 and calcium phosphate ceramics has achieved many clinical successes. But it is worth noting that serious side effects related to ectopic or unwanted bone formation in certain situations have made the FDA increasingly resistant to approving the use of such materials (Vavken et al., 2016).

An alternative long-lasting and potentially safer strategy is the incorporation of trace metallic elements into CaPs. The mineral composition of our bone itself is not a homogenous and bioinert calcium phosphate-based material (Von Euw et al., 2019). Bone incorporates and releases various trace elements (such as  $Mg^{2+}$ ,  $Zn^{2+}$ , and  $Sr^{2+}$ ) into the microenvironment during bone metabolism (Bose et al., 2013). Previous studies have confirmed that CaPs modified with these bivalent trace metallic ions can lead to controlled degradation, increase the mechanical strength of the materials, and enhance bioactive properties. But among these bioactive ions, strontium (Sr) has gained great attention since strontium ranelate (SrRan) has been approved as an anti-osteoporotic drug for post-menopausal osteoporosis since it increases bone strength (Neves et al., 2017; Marx et al., 2020). Recent findings suggest that oral administration of SrRan may have no anabolic action on bone formation in humans, and even inhibits osteogenic differentiation under the *in vitro* experiment conditions (Wornham et al., 2014; Marx et al., 2020). However, the comprehensively described SrCaPs do exert positively influence on new bone formation and accelerate the healing process (Neves et al., 2017). These discrepancies may be partially due to the implant microenvironments where biomaterials interact with various cells. The local delivery of Sr ions from SrCaPs implanted in bone changes local microenvironment, which is involved in several biological processes such as osteogenesis, angiogenesis, osteoimmunomodulation.

Consequently, in this review, we present recent developments in strontium-substituted calcium phosphate ceramics (SrCaPs) and offer insights into how Sr ions released from implants influence the immune response, angiogenesis, and osteoblastic differentiation of bone marrow stromal cells (bMSCs).

## Sr-CaPs: CURRENT STATE OF THE ART

In the 1960s, surgeons valued the inertness of surgical materials in the human tissue environment, and first-generation bioceramics (alumina, zirconia, diverse forms of carbon) were developed as bone substitutes, mainly for femoral head fabrication (Smith, 1963). However, they elicited a foreign body reaction, forming fibrous capsules that isolated them from the body. In the 1980s, Larry Hench invented a bioactive glass with the ability to bond to living tissue, and Heughebaert reported that CaPs could induce bone formation (Hench and Wilson, 1984; Heughebaert

et al., 1988). Since then, osteoinductive and tissue-inducing materials have become the mainstream of biomaterial science and engineering (Hench and Polak, 2002). CaPs are the most studied and implanted bioactive ceramics in the clinic because of their cost-effective preparation from chemical reagents or natural resources. Various calcium phosphate compounds are in different crystalline phases and often classified on the basis of the Ca/P ratio. Of great interest is the subgroup of apatite compounds, which exhibit similarities to vertebrate hard tissues. Indeed, both bone and teeth are well-crystallized apatites (hydroxyapatite), and this structure can easily accommodate many cations and anions for substitution, thereby achieving adaptive biophysical functions (Drouet et al., 2017).

Strontium (Sr) is naturally deposited in the mineral phase of bones via consuming a normal diet, and it replaces approximately 0.035% of Ca content (International Programme on Chemical and Inter-Organization Programme for the Sound Management, 2010). The majority of *in vitro* experiments support a dual-acting mechanism in which Sr stimulates bone formation and hinders bone resorption (Saidak and Marie, 2012). The exact mechanism of the role of Sr in bone remains unclear, but it has been proposed that Sr acts on similar cellular targets as  $Ca^{2+}$  by activating the calcium sensing receptor (CaSR), thus interacting with Ca-driven signaling pathways related to bone metabolism regulation (Saidak and Marie, 2012). Furthermore, small animal trials and clinical trials have produced overwhelming evidence that Sr benefits bone remodeling and increases bone-mineral density, especially in the treatment of osteoporosis (Meunier et al., 2004; Reginster et al., 2012; Marx et al., 2020). Taking these beneficial effects into consideration, it is likely that local Sr ion delivery will enhance osteoinduction and osseointegration at the bone-implant interface, contributing to a faster healing microenvironment.

Recently, Sr has been widely incorporated into calcium phosphate-based materials for biomedical applications. Krishnan reported using Sr doped hydroxyapatite (SrHAP) particles to repair dental enamel with the potential to treat white spot lesions and incipient carious lesions (Krishnan et al., 2016). It is also an effective additive for toothpaste in the prevention of cariogenesis (Surdacka et al., 2007). These particles cannot only be used as a raw material, but also associated with other biomaterial compounds to facilitate new functions. Sr-CaPs are fabricated as coatings on the surfaces of metallic implants in order to accelerate bone healing at early implantation times (Li et al., 2010; Arcos and Vallet-Regí, 2020). Titanium and titanium-based alloys are commonly used for load-bearing applications (such as total joint replacement and fracture fixation elements), but carry a risk of loosening, especially when implanted in osteoporotic bones. Tao et al. (2016a) conducted an *in vivo* study on rats with ovariectomy-induced osteoporosis and reported that a strontium-containing HAP coating on shaped titanium implants was better than an Sr-free HAP coating in terms of new bone formation and push-out force. Furthermore, they conducted a comparative study that found that under the same the molar ratio [ $Zn$ ,  $Mg$ ,  $Sr/(Zn, Mg, Sr+Ca) = 10\%$ ], the SrHAP coating exhibited better osseointegration than zinc- and magnesium-substituted HAP coatings (Tao et al., 2016b).

SrCaP cements (SrCPC) consist of a combination of a precursor powder and liquid phase, allowing it to fill and heal bone defects during minimally invasive procedures. Kuang et al. (2015) prepared pre-set SrCPC for a metaphyseal defect in Sprague-Dawley rats and found early endochondral ossification at 2 weeks post-operation. Thormann et al. (2013) further demonstrated that Sr was deposited at newly formed tissues around the implanted SrCPC as visualized by Time-of-flight secondary ion mass spectrometry (ToF-SIMS) imaging and found a significant increase of mineralized extracellular matrix in the SrCPC group compared to the Sr-free calcium phosphate cement (CPC) and empty defect groups. Sr-CaP scaffolds can be further modified with drugs, especially antibiotics and growth factors. The chemical composition of SrHAP provides strong surface adsorption capacity and increases surface area, allowing drug release for a longer period of time (Xu et al., 2018). Lin et al. (2013) reported that SrHAP microspheres with 3D hierarchically mesoporous structures not only promoted an osteogenic response, but exhibited sustained vancomycin release. Yan et al. (2018) developed an Sr-nHAP/SF-Hep-BMP-2 scaffold system with controlled BMP-2 release that supported critical size calvarial defect healing in Sprague-Dawley rats. Tao et al. (2018) revealed synergistic bioactive properties of BMP-2 and Sr released from CPC composites, which achieved rapid bone healing in osseous defects in an ovariectomized rat model. These new applications for strontium-containing calcium phosphate extends their biomedical potential.

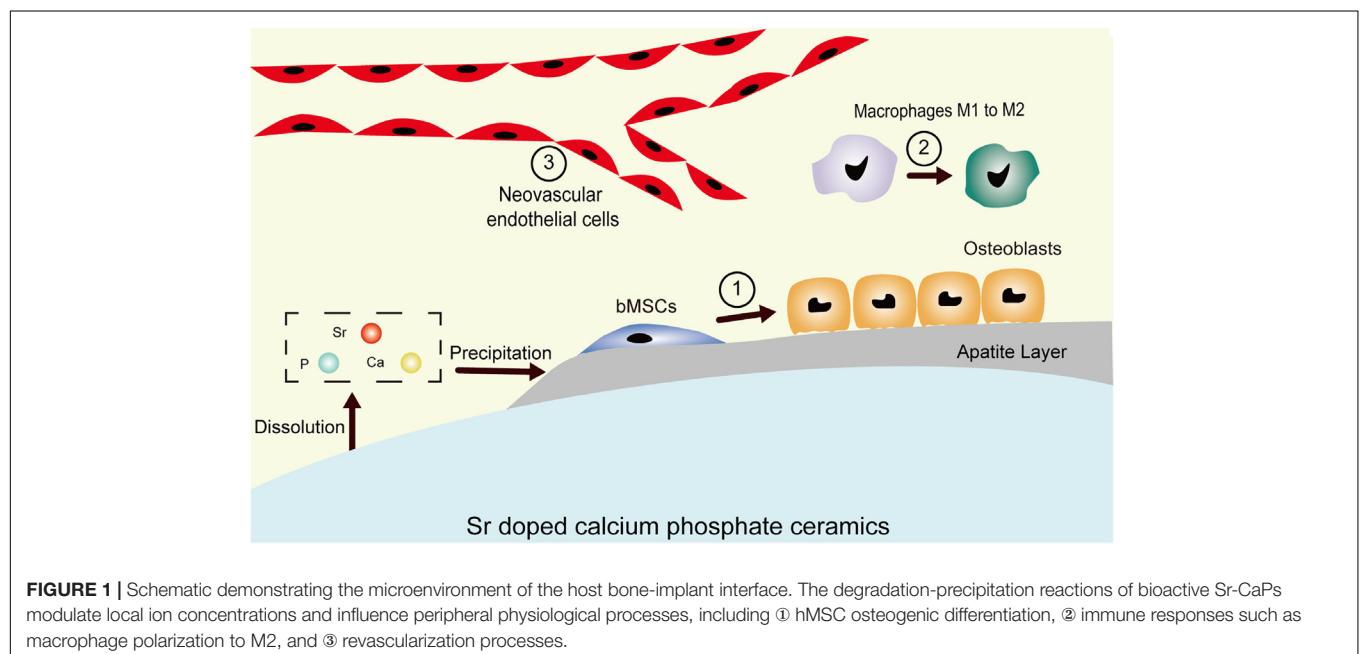
## FAVORABLE POST-IMPLANTATION ENVIRONMENTS FOR Sr-CaPs

Sr is chemically and physically close to calcium but has a larger ionic radius (112 vs. 99 pm). The partial substitution

of Ca by Sr results in higher solubility when compared with Sr-free CaPs owing to the enlargement of the unit cell (Zhu et al., 2018). The surfaces of SrCaPs are reactive and biodegradable, and bone-like apatite forms on them through continuous dissolution and precipitation. The newly formed apatite creates a strong bond between the living tissues and implants. However, it degrades over time when in contact with bodily fluids and it elevates the local concentrations of  $\text{Sr}^{2+}$ ,  $\text{Ca}^{2+}$ , and  $\text{PO}_4^{3-}$  ions. These released ions affect important components of the bone microenvironment, including the mineral deposited layer phase, the cellular phase (osteoblasts, macrophages, and vascular endothelial cells), and the soluble factor phase (growth factors and/or cytokines) and provide an extracellular osteogenic signaling network. The incorporation of an appropriate proportion of Sr into CaPs allows them to build an osteogenic microenvironment via modulating the inflammatory response, stimulating osteogenic differentiation of bone marrow mesenchymal stem cells, and promoting early angiogenesis (Figure 1).

## EFFECTS ON OSTEOGENESIS

SrCaPs is a promising biomaterial to help bone reconstruction. Mounting evidence shows that Sr replacement of Ca in CaPs leads to increased solubility due to the expansion of the crystal structure, and the high local  $\text{Sr}^{2+}$ ,  $\text{Ca}^{2+}$ , and  $\text{PO}_4^{3-}$  ionic concentrations are believed to be of great importance for osteoinduction and osteoconduction (Bose et al., 2013; Neves et al., 2017). Current research shows that both Sr and Ca mediate key cellular functions in osteogenesis-related cells via Ca-sensing receptor (CaSR)-dependent mechanisms (Marx et al., 2020). In the extracellular microenvironment, a high Ca concentration is





a potent chemical signal for bMSCs osteogenic differentiation (Dvorak and Riccardi, 2004; Barradas et al., 2012). A lack of calcium significantly retards cell growth and differentiation (Dvorak et al., 2004; Barradas et al., 2012). Interestingly, CaPs with Sr modification tend to have decreased Ca concentrations around the material due to induced apatite precipitation, though lower concentrations of Ca do not impair the proliferation or osteogenic differentiation of bMSCs (Schumacher et al., 2013; Kruppke et al., 2019b). In an *in vitro* study, Kruppke et al. (2019a) confirmed that Sr may compensate for the adverse effects of lowered Ca ion concentrations to a certain degree and potentially stimulate bone regeneration. They investigated the effect of varied Ca and Sr ion concentrations, respectively, on proliferation and differentiation of bMSCs and concluded that under low calcium concentrations culture conditions (0.3–0.4 mM due to the addition of fetal calf serum), Sr at 0.4–0.9 mM promoted differentiation of bMSCs while 0.9–1.8 mM stimulated the highest proliferation. And in a second study exploring the relationship between Sr and Ca on bone formation, they conducted *in vitro* and *in vivo* experiments to evaluate the osteogenesis effects of Sr together with two concentration gradients of Ca (normal calcium of 1.8 mM or high calcium of 9 mM) and concluded that Sr (1 mM) enhanced mineralization and ALP expression of MC3T3-E1 pre-osteoblast cells under a high dose of calcium (9 mM) (Xie et al., 2018). This may explain the results by Wornham et al. (2014) in which concentrations of Sr ranging from 10  $\mu$ M to 1 mM inhibit the mineralization ability of primary rat osteoblasts under 1.8 mM Ca present in the culture system. These experiments illustrate the importance of the implant microenvironment because bone has higher Ca levels than plasma. It also suggests that both Sr and Ca from ion-leaching SrCaPs may interact to promote bone regeneration at the

bone-implant interface. In addition, the phosphate ion is a basic component in the human body and accumulates in bone minerals in the form of calcium phosphates along with calcium ions (Khoshniat et al., 2011). When phosphate ions are not present in high quantities, the generation of new bone will be blocked due to insufficient formation of hydroxyapatite (Penido and Alon, 2012).

Sr-rich CaPs consistently outperform Sr-free CaPs in *in vitro* and *in vivo* studies (Neves et al., 2017; Marx et al., 2020). In the absence of doping with other osteogenic ions, the local release of Sr is sufficient to promote bone formation, but it is difficult to ignore the effects that surface morphology, porosity, hydrophilicity, dissolution process, and local pH changes may have on osteogenesis regulation at the implantation site.

## EFFECTS ON ANGIOGENESIS

Increased angiogenesis can relieve the symptoms of ischemia and facilitate the transport nutrients, which is essential for bone repair/regeneration. Sr-CaPs have been found to increase neovascularization in many convincing experiments (Liu et al., 2011; Gu et al., 2013; Wang et al., 2014). Endothelial cells are the major seed cells that drive angiogenesis. Chen et al. (2008) observed that degradation products of Sr doped calcium polyphosphate (SCPPs) promote the proliferation, migration, and tube-like structure formation of endothelial cells (ECV304). It has been reported that *in situ* production of angiogenic growth factors, such as vascular endothelial growth factor (VEGF) and basic fibroblast growth factor (bFGF), can promote more cell infiltration, extracellular matrix synthesis, and faster angiogenesis (Saberianpour et al., 2018).

**TABLE 1 |** The osteogenic differentiation function of various Sr concentrations.

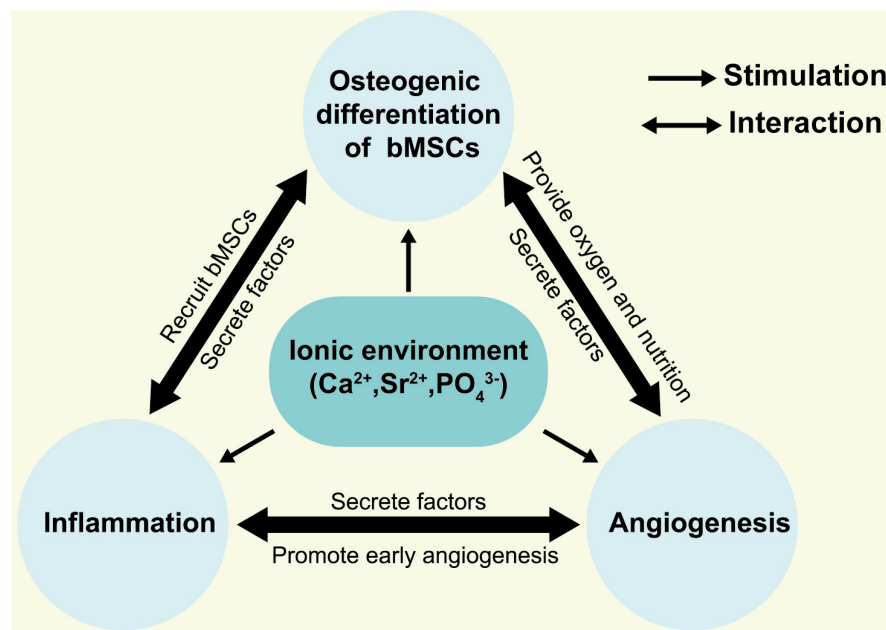
Cell lines	Strontium salt	Sr concentration	Function	References
C3H10T1/2 and mice primary bone marrow mesenchymal stem cells	Strontium chloride	1.0 mM, 3.0 mM	Promote osteogenic differentiation	Peng et al., 2009
Ovariectomy bone marrow mesenchymal stem cells	Strontium ranelate	0.25–0.5 mM	Promote osteogenic differentiation	Guo et al., 2016
Primary fetal mouse calvaria cells	Strontium ranelate	0.1–1 mM	Promote osteogenic differentiation	Bonnelye et al., 2008
Human bone marrow mesenchymal stem cells	Strontium ranelate	2.4–240 $\mu$ M	Promote osteogenic differentiation	Sila-Asna et al., 2007
Human adipose-derived stem cells	Strontium ranelate	25–500 $\mu$ M	Promote osteogenic differentiation	Aimaiti et al., 2017
	Strontium ranelate	1–3 mM	Apoptosis and inhibit osteogenic differentiation	
MC3T3-E1 pre-osteoblast cells	Strontium chloride	1 mM	Inhibit osteogenic differentiation	Xie et al., 2018
Rat bone marrow mesenchymal stem cells	Strontium ranelate	0.1 mM, 1 mM	Inhibit proliferation and promote osteogenic differentiation	Li et al., 2012
Primary rat osteoblastic cells	Strontium ranelate; strontium chloride	0.1–1 mM	Inhibit mineral deposition	Wornham et al., 2014

Liu et al. (2011) directly seeded osteoblasts (ROS17/2.8) on SCPP scaffolds with various amounts of Sr and found increased secretion of VEGF and bFGF in the culture medium. Wang et al. (2014) observed the stimulating effects of SCPP on protein secretion and mRNA expression of VEGF, bFGF, and MMP-2 from endothelial cells *in vitro*. Gu et al. (2014) co-cultured umbilical vein endothelial cells and osteoblasts on SCPP scaffolds *in vitro* and demonstrated higher VEGF and bFGF protein production compared with CPP and HAP. Furthermore, Ye et al. (2019) fabricated Sr-doped calcium phosphate/polycaprolactone/chitosan (Sr-CaP/PCL/CS) nanohybrid fibrous membranes and demonstrated high VEGF secretion from bMSCs. In the future, more experiments are needed to test the ability of SrCaPs to promote blood vessel formation *in vivo*, and more attention should be paid to the impact of the immune response.

## EFFECTS ON OSTEOIMMUNOMODULATION

Every implantable material must be biocompatible, meaning that only a limited inflammatory response occurs in the host body. Huang et al. (2020) reported the incorporation of fish-derived nano CaPs (Ca/P molar ratio was 2.35) into gelatin methacrylate generated a suitable immune microenvironment by inducing the secretion of cytokines and promoting macrophage phenotype conversion. The controllable inflammation may promote the osteogenic differentiation and angiogenesis (Zhao et al., 2018; Huang et al., 2020). The local release of Sr from implants to the adjacent host bone can reduce unfavorable

inflammatory responses (Renaudin et al., 2008), and Sr has been proven to be an anti-inflammatory agent (Wang et al., 2020). Buache et al. (2012) first verified that Sr-BCP can decrease the production of pro-inflammatory cytokines (TNF- $\alpha$  and IL-6) and the chemokine interleukin 8 in human monocytes. Braux et al. (2016) demonstrated the downregulating effect of Sr-BCP on inflammatory mediator production (MCP-1 and Gro- $\alpha$ ) by human primary osteoblasts. Sr has also been shown to suppress the expression of proinflammatory cytokines (IL-6 and TNF- $\alpha$ ) in periodontal ligament cells and macrophages (Römer et al., 2012; Gu et al., 2014). Among various immune-related cells, macrophages play a significant role in regulating bone healing after trauma or the implantation of biomaterials (Alexander et al., 2011; Batoon et al., 2019). At the early time of bone repair, resident or infiltrating monocyte-derived macrophages mostly present the pro-inflammatory M1 phenotype, and a timely switch from M1 to M2 (anti-inflammatory phenotype) leads to osteogenic cytokine release, including that of IL-10, TGF $\beta$ , and VEGF. Interestingly, prolonged M1 polarization leads to an increased release of fiber-enhancing cytokines from M2 macrophages, which leads to the formation of fibrous capsules (Chen et al., 2016). Previous evidence suggests that Sr-integrated implants elicit more pro-regenerative M2 macrophages; examples include Sr-doped calcium polyphosphate particles, Sr-substituted bioactive glass, and Ca- and Sr-modified titanium implant surfaces (Gu et al., 2014; Lee et al., 2016; Zhao et al., 2018). The incorporation of Sr into CaPs endows traditional CaPs with osteoimmunomodulation properties and positively regulates cytokine production and immune cell functions for better bone remodeling.



**FIGURE 2 |** Hypothesis for osteogenic microenvironments of SrCaPs. The local ionic environment not only stimulates the osteoblastic differentiation of bMSCs but interacts with the inflammatory cells and vascular endothelial cells.



## CONTROVERSY OVER THE EFFECTIVE STRONTIUM CONCENTRATION

Local Sr release is a critical property of SrCaPs, but the optimal concentration remains to be determined. Evidence from *in vitro* studies shows that Sr promotes osteogenic differentiation in a dose-dependent manner, and different cell lineages respond differently to Sr at similar concentrations (Table 1). Peng et al. (2009) showed that 1 mM can promote osteogenic differentiation of C3H10T1/2 or mice primary bMSCs, while Aimaiti et al. (2017) reported that 1 mM induces apoptosis and inhibits osteogenic differentiation in human primary adipose-derived stem cells. Moreover, inhibition at this concentration also occurs for primary rat osteoblastic cells and MC3T3-E1 cells (Wornham et al., 2014; Xie et al., 2018). However, the maximum circulating concentrations of Sr in the serum of patients treated with strontium ranelate (2 g per day for 3 years) is about 0.1 mM (Meunier et al., 2004). Although several *in vitro* studies have shown that Sr concentration above 1 mM can stimulate osteogenesis, it is preferable to take clinical trial results into consideration, and the effective range (below 500 mM) is recommended here (Bonnelye et al., 2008; Peng et al., 2009; Guo et al., 2016). To the best of our knowledge, no *in vivo* study on the local administration of Sr with the use of modified CaPs has reported adverse effects on bone formation and osseointegration. The reasons may be that, first, the implantation microenvironment in the body is dynamically changing, and a high dose of Sr does not last for a long time. Second, the degradation product is not only Sr, but also Ca, which matters because of the equilibrium effect of Ca to Sr (Xie et al., 2018). Third, Sr positively modulates the microenvironment via regulating immune cells and endothelial cells in the process of bone formation.

## CONCLUSION

In summary, integrating Sr ions into CaPs offers an alternative to biologics in the design of bioactive materials. It is of low

cost, has a longer shelf life, and presents low systemic risk compared to growth factors. These added benefits make Sr as therapeutic agent attractive in tissue engineering and regenerative medicine applications (Jiménez et al., 2019; Prabha et al., 2019; Mao et al., 2020). However, the mechanism of osteoinduction of SrCaPs remains complicated because inflammation and angiogenesis accompany the entire process of bone healing. In light of the previous studies, we propose a hypothesis for the osteoinduction mechanism of SrCaPs: SrCaPs build the osteoinductive microenvironment of the host bone-implant interface (Figure 2). Functional ion release and apatite layer formation on the surfaces of SrCaP ceramics allow them to interact with cells and extracellular matrices in the host system, providing bioactive bonding to bone. Specifically, a higher local concentration of Sr stimulates the osteoblastic differentiation of MSCs (1 in Figure 1), induces macrophage polarization toward a pro-regenerative M2 phenotype (2 in Figure 1), and contributes to angiogenesis (3 in Figure 1). Collectively, understanding the biological microenvironment of implant-to-tissue interactions at the bone site is important for the development of high-performance bioceramics.

## AUTHOR CONTRIBUTIONS

BW was the first author and drafted the manuscript. DC and RW designed the work. YS, JG, JC, and HW summarized related publications. All authors approved the final version to be published.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Applications of Polydopamine-Modified Scaffolds in the Peripheral Nerve Tissue Engineering

Ji Yan<sup>1,2,3†</sup>, Ruoyin Wu<sup>3,4†</sup>, Sisi Liao<sup>1,2,3</sup>, Miao Jiang<sup>1,2,3</sup> and Yun Qian<sup>1,3\*</sup>

<sup>1</sup> Department of Orthopedics, Shanghai Jiao Tong University Affiliated Sixth People's Hospital, Shanghai, China, <sup>2</sup> Shanghai Jiao Tong University School of Medicine, Shanghai, China, <sup>3</sup> Youth Science and Technology Innovation Studio, Shanghai Jiao Tong University School of Medicine, Shanghai, China, <sup>4</sup> School of Pharmacy, Shanghai Jiao Tong University, Shanghai, China

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### \*Correspondence:

Yun Qian  
lollipopcloudland@foxmail.com

<sup>†</sup> These authors have contributed  
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Peripheral nerve injury is a common and complicated traumatic disease in clinical neurosurgery. With the rapid advancement and development of medical technologies, novel tissue engineering provides alternative therapies such as nerve conduit transplantation. It has achieved significant outcomes. The scaffold surface modification is vital to the reconstruction of a pro-healing interface. Polydopamine has high chemical activity, adhesion, hydrophilicity, hygroscopicity, stability, biocompatibility, and other properties. It is often used in the surface modification of biomaterials, especially in the peripheral nerve regeneration. The present review discusses that polydopamine can promote the adhesion, proliferation, and differentiation of neural stem cells and the growth of neuronal processes. Polydopamine is widely used in the surface modification of nerve conduits and has a potential application prospect of repairing peripheral nerve injury. Polydopamine-modified scaffolds are promising in the peripheral nerve tissue engineering.

**Keywords:** polydopamine, surface modification, peripheral nerve repair, nerve conduit, tissue engineering

## INTRODUCTION

Peripheral nerve injury can lead to partial or total nerve rupture and result in paralysis, neuropathic pain, and even sensory loss. It severely impairs the patients' limb functions and reduces their qualities of life. Although autologous transplantation is the current gold standard of clinical treatment, it also has many defects, such as limited sources, mismatched diameters, and sensory dysfunction in the donor region (Samadian et al., 2020). Given these problems, the clinical application of nerve conduits is being constantly explored. To fabricate an ideal nerve conduit, the material selection and other important scaffold properties must be considered, such as pro-adhesive interface, conductivity, degradability, biocompatibility, and mechanical stability. Among them, surface modification is a common and vital tool to improve the microenvironment of the cell/scaffold interface to support an adhesive surface. Immobilized bioactive molecules play a



significant role in this process (Chen et al., 2016). Studies have shown that physical and chemical properties of the scaffold interface affect the adhesion, proliferation, and differentiation of cells on the biomaterial surface (Wozniak et al., 2004; Bettinger et al., 2009).

Hsiao et al. (2009) attached the synthetic hybrid DNA strands to the plasma membrane of living cells to allow the modification of the cell surface directly. This technique is more suitable for modifying primary cells because prolonged cell culture is not required. A study revealed that the Arg-Gly-Asp (RGD) peptide functionalized bilayers could support neural stem cells (NSCs) attachment and proliferation. The bilayers were prepared through physical absorption. However, this method was inefficient (Ananthanarayanan et al., 2010; Georgiou et al., 2013). Conducting polymers, such as polypyrrole (PPY) and poly(3,4-ethylenedioxythiophene), have been used to facilitate the growth and differentiation of neural cells, based on their electroactivity and electrical conductivity (Rim et al., 2013). However, the mechanical stability is reduced substantially. It also becomes difficult to calculate the exact density and distribution (Green et al., 2009; Gelmi et al., 2010). Besides, it was demonstrated that surface modification was necessary for Schwann cell's response and polycaprolactone (PCL) could improve biocompatibility (Luca et al., 2014).

Dopamine is a dopa derivative containing catechin groups. Catechol and amino groups in its molecular structure have high chemical activity and play an important role in the adhesion process through covalent bonds and non-covalent forces. Polydopamine (PDA), formed by oxidative self-polymerization of dopamine in weak basic Buer solution, has no special requirements on the properties of materials and can stick to almost any object surface even Teflon (Miller et al., 2017). In 2007, Lee et al. (2007) first discovered a simplified method to apply PDA for material surface modification based on the mussel adhesion protein inspiration. Since then, PDA has been widely used in surface modification of various inorganic or organic materials due to its unique properties, especially in the field of biomedicine. It mainly includes adhesion of organic templates in the biomineralization and preparation of nanocapsules for drug delivery. The biocompatibility and hydrophilicity of the materials were improved as well as the immunogenicity was reduced (Ryu et al., 2010; Cui et al., 2012). In the study of Bhang et al. (2013), it was verified that when the pheochromocytoma (PC) 12 cells were cultured on the PDA-coated surfaces, the expression of marker proteins would be enhanced in the neuronal differentiation, compared to that on the gelatin-modified surfaces. The length of the neurites extended more effectively. Compared with traditional surface modification strategies, PDA coatings have significant advantages. PDA deposition is non-specific, whereas grafting depends on specific sites on the membrane surface. The process of PDA modification is under simple and mild conditions so that it can avoid potential damage in the process of irradiation. The PDA coatings keep scaffolds moist in the plasma treatment that affects the surface permeability of the materials (Miller et al., 2017).

In recent years, the application of PDA in the nerve tissue engineering has become a research hotspot. In this review, the physicochemical properties, biocompatibility, and effects on neuronal activity of PDA as well as its applications in the peripheral nerve tissue engineering are reviewed (Figure 1).

## PHYSICOCHEMICAL PROPERTIES OF PDA

### Reactions of PDA

Properties of PDA such as adhesion are based on its chemical activity. Chemically active catechol can form hydrogen bonds with hydroxyl and chelate metal ions (Lee et al., 2007). Covalent bonds are commonly formed between catechol and many groups including sulfhydryl and amidogen (Lee et al., 2006).

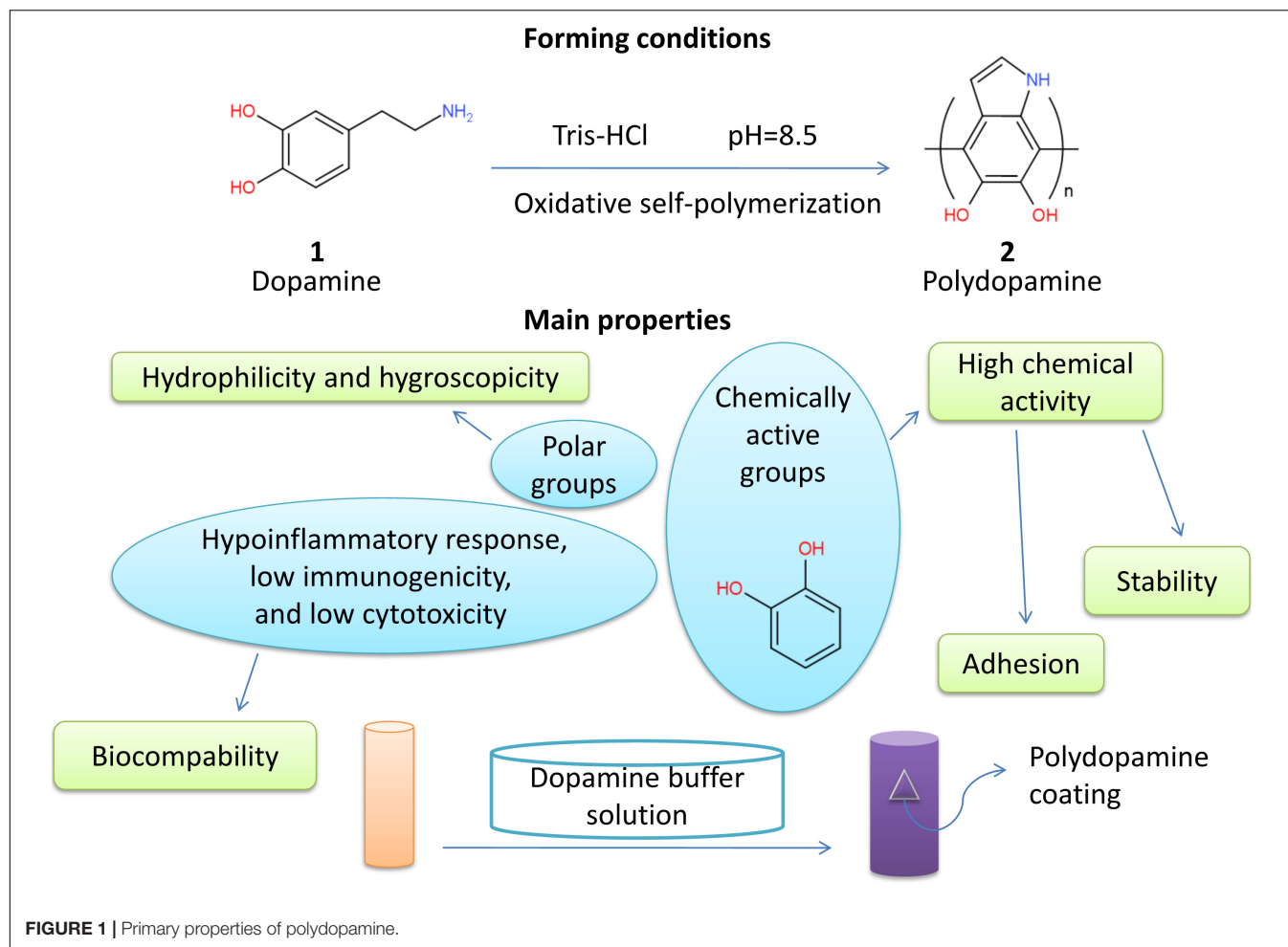
Polydopamine acts as either oxidant or reductant in a reaction because of its oxidizing quinonyl and the reductive catechol (Miller et al., 2017). Chelation reactions take place when PDA meets metal ions with every valence ( $\text{Fe}^{3+}$ ,  $\text{Zn}^{2+}$ ,  $\text{Mn}^{2+}$ ,  $\text{Cu}^{2+}$ ) (Ye et al., 2017). Noble metal ions can be reduced by PDA because the electrons that are released during the oxidation of catechol stimulate the reduction of positive ions (Ball et al., 2011).

When it comes to proteins representative of biomolecules, nucleophilic ones react with carbon on the benzene ring (Michael addition), and the primary amino groups in proteins react with PDA in the quinone form (Schiff base reactions) (Lee et al., 2009). Thiol and amino groups also react with PDA. In alkaline conditions, phenolic groups in PDA are first oxidized to homologous quinone and then react with amino groups by Schiff base reaction or Michael addition. While thiol groups often react with PDA by Michael addition, amino and imino groups in PDA make it more likely to crosslink some organic molecules on PDA (Hu and Mi, 2013; Wang et al., 2013; Liu et al., 2014). Moreover, since PDA hardly bonds to chemical reagents except for water and solutions with metal ions, it shows the advantage when it comes to deposition (Miller et al., 2017).

### Adhesion of PDA

Polydopamine deposits on organic or inorganic materials, regardless of whether the surface is hydrophilic or hydrophobic, and surface properties of composite materials are dominated (Miller et al., 2017). Although adhesion mechanisms are not clear, it is closely related to chemical compositions. There are mainly two kinds of adhesion processes: covalent bindings and non-covalent bindings. Covalent bindings involve Michael addition and Schiff base reaction, and non-covalent bindings include metal coordination, hydrogen bonds,  $\pi$  -  $\pi$  stack, and Van der Waals' force (Yu et al., 1999; Dalsin et al., 2005; Anderson et al., 2010; Li et al., 2010; Ye et al., 2011; Pop-Georgievski et al., 2012; Liu et al., 2014). The more the non-covalent bonds are, the stronger the adhesion is (Miller et al., 2017). In addition, a high concentration of  $\text{SO}_4^{2-}$ ,  $\text{NO}_3^-$ , and  $\text{Cl}^-$  interrupted deposition. It indicates that adhesion of PDA is selective (Zhang et al., 2018).

After PDA is attached to materials, it performs as a secondary platform for functional molecules (drugs or growth factors, silver



nanoparticles, and proteins) binding to the surface (Lee et al., 2009; Ball et al., 2011; Liang et al., 2019). In addition, thicker layers of PDA can bind more drugs with much longer release time (several hundred days), which endows the drugs with sustained release (Yang et al., 2018).

### Hydrophilicity and Hygroscopicity of PDA

Many phenolic and amino groups increase the hydrophilicity of PDA, so that PDA-adhered materials are more hydrophilic, and hydrophobic materials covered by PDA have even much better hydrophilicity (Miller et al., 2017). Because of the existence of strong hydrophilic groups of PDA, such as amino, imino, and catechol groups, the ability of materials to absorb water was improved (Chen et al., 2019).

### Stability of PDA

Stability of PDA depends on synthetic methods. For instance, PDA layers synthesized by air oxidation are heterogeneous and insufficient. They are unstable in polar organic solvents, and acidic and alkaline water solutions. However, PDA layers synthesized by  $\text{CuSO}_4/\text{H}_2\text{O}_2$  in alkaline conditions have better homogeneity and pH stability. They also show strong polarity in organic solvents (Zhang et al., 2016).

PDA coatings in water solutions often possess excellent mechanical strength and structural stability (Shen et al., 2015; Fang et al., 2019).

### BIOCOMPATIBILITY OF PDA

Polydopamine displays high cellular and tissue biocompatibility characteristics (Han et al., 2017). PDA coatings can reduce inflammation, immune reaction, and cytotoxicity (Hong et al., 2011). Oxidative polymerization of dopamine could promote cell adhesion and proliferation on substrates [polylactic acid, poly(lactic-co-glycolic acid) (PLGA), polyurethane(PU), PCL, hydrophobic materials (poly-ethylene, poly-tetrafluoroethylene, and poly-dimethyl-siloxane)] (Ku et al., 2010; Tsai et al., 2011; Rim et al., 2012). Hydrogel directly applied to skin can reduce inflammation if it contains PDA in synthesis (Han et al., 2017). For example, polystyrene/silver/PDA nanoparticles show no cytotoxicity at low concentration and slight cytotoxicity at high concentration, which guarantees a higher cell survival rate (Cong et al., 2014).  $\text{H}_2\text{O}_2$  can degrade PDA nanoparticles effectively, which has also been proved *in vivo* (Liu et al., 2014). Biodegradability strengthens PDA safety.



## PDA AND THE ACTIVITY OF NEURON AND NEURON-LIKE CELLS

### NSC Adhesion Enhanced by PDA

In recent years, PDA has been widely used in the modification of biomaterials due to its desirable cell adhesion. Neural stem cells have the ability of division and self-renewal. Under appropriate conditions, neural stem cells can differentiate into different types of cells, including neurons, to repair and replace damaged nerve cells. Many experiments have proved that NSCs can effectively adhere to the surface of the composites modified by PDA, indicating it is beneficial to repair injured nerves. However, the molecular mechanism of PDA promoting NSC adhesion is still unclear. The addition of PDA can enhance the hydrophilicity (Cheng et al., 2016). It was reported that enhanced hydrophilicity can promote cell adhesion and other cell behaviors (Lin and Fu, 2016). PDA contains many hydrophilic functional groups, such as hydrophilic amino group and hydroxyl group, which can provide hydrophilic group for hydrophobic surface, thus improving the hydrophilicity of the nerve conduits. In addition, vincristine, a kind of cytoskeleton protein, is closely related to local adhesion and regulates cell proliferation. The expression of vincristine in NSCs cultured with PDA was enhanced significantly, and it showed that NSCs were much more proliferative than the non-PDA treatment group (Yang et al., 2020).

### NSC Proliferation and Differentiation Improved by PDA

Postsynaptic density protein-95(PSD-95) protein exists in the postsynaptic membrane. Its content can reflect the development

and maturity of synapses. Beta III tubulin is a specific microtubulin of neurons. Glial fibrillary acidic protein (GFAP) is a specific intermediate filament protein in astrocytes. Microtubule-associated protein 2(MAP2) only exists in the skeleton of mature nerve cells. The gradual increase in PSD-95, MAP2, and beta III tubulin and decrease in GFAP can be regarded as the process of neuronal maturation and neuronal development (Hendrickson et al., 2011). Some studies have shown that the PDA modification can increase the hydrophilicity of the scaffold and then enhance the cell adhesion, leading to the increase in cell proliferation (Lin and Fu, 2016).

### Neurite Outgrowth Induced by PDA

The PDA-modified PLGA nanofiber membrane can effectively adsorb IGF-1 without damaging the growth factor activity and has an obvious slow-release effect on IGF-1. It is beneficial to maintain the long-term and stable accumulation of IGF-1 at the nerve injury site. Thus, it can maximize the function of IGF-1 (Qi et al., 2019). IGF-1 stimulates the growth of neurite, promotes the proliferation and differentiation of nerve cells, inhibits the apoptosis of nerve cells, and promotes the repair of injured nerve tissues (Table 1).

## PDA-RELATED BIOENGINEERING IN THE PERIPHERAL NERVE REGENERATION

Polydopamine has been widely applied in the field of biomedicine because of its excellent biological characteristics. Several recent studies have focused on the application of PDA in the repair

**TABLE 1 |** Functions and properties of different polydopamine-modified biomaterials.

Cell type	Adhesion	Viability	Proliferation	Differentiation	Substrate	Surface modification	Author/year
Hippocampal neurons	(+)	(+)	/	/	Electrode, insulators	PDA/PLL	Kang et al., 2011
HSCs	(+)	/	(+)	(+)	PU/PDA/ECM	/	Chen et al., 2019
RSCs	(+)	(+)	(+)	/	SG-PCL, MG-PCL	PDA/RGD	Qian et al., 2018a
Mouse C2C12 myoblasts, PC12 neuronal cells	/	(+)	(+)	(+)	Electrodes	PDA/PPY	Kim et al., 2018
PC12 cells	(+)	(+)	(+)	(+)	/	PDA	Bhang et al., 2013
HNSCs	/	/	(+)	(+)	PS, PLGA	NGF, GDNF, YIGSR, RGD-PDA	Yang et al., 2012
RSCs	(+)	/	(+)	(+)	PCL	dECM/PDA	Chen C. et al., 2018
Primary postnatal mouse cerebellar neurons	(+)	(+)	/	(+)	PNF	PDA	Sieste et al., 2018
RNSCs	(+)	(+)	(+)	(+)	PDA-PLGA	NGF	Pan et al., 2018
RSCs	(+)	(+)	/	/	PLLA	PDA/CGO/PPY	Li et al., 2020
Rat PC12 cells, primary rat DRG neurons	/	(+)	/	/	Fe <sub>3</sub> O <sub>4</sub> -Rhodamine 6G	PDA	Wang et al., 2020
Rat PC12 cells	(+)	(+)	(+)	(+)	PLGA/CNT	PDA-lam	Nazeri et al., 2020
Rat PC12 cells	(+)	/	(+)	/	CC	PDA	Chen X. et al., 2018
Rat BMSCs and SCs	(+)	(+)	(+)	(+)	PCL/gold	PDA	Qian et al., 2018b

"(+)", presentation of characteristics; PDA, polydopamine; PLL, poly-L-lysine; HSCs, human Schwann cells; PU, polyurethane; ECM, extracellular matrix; RSCs, rat Schwann cells; SG, single-layered graphene; MG, multi-layered graphene; PCL, polycaprolactone; RGD, arginyl glycyaspartic acid; PC12, pheochromocytoma 12; PPY, polypyrrole; HNSCs, human neural stem cells; PS, polystyrene; PLGA, poly(lactic-co-glycolic acid); dECM, decellularized extracellular matrix; PNF, peptide nanofiber; RNSCs, rat neural stem cells; NGF, nerve growth factor; PLLA, poly-L-lactic acid; CGO, carboxylic graphene oxide; DRG, dorsal root ganglia; CNT, carbon nanotube; lam, laminin; CC, carbonized cotton; BMSCs, bone marrow mesenchymal stem cells; SCs, Schwann cells.

of peripheral nerve injuries in order to develop suitable nerve conduit scaffolds.

Polydopamine has influenced the bioactivities of nerve conduits and the adhesion, growth, and differentiation of neural cells *in vitro*. Novel decellularized extracellular matrix (dECM) and PDA-coated 3D printed PCL-based conduits were created for nerve regeneration. The presence of PDA significantly improved the hydrophilicity and mechanical properties of conduits, as well as cellular behaviors and neuronal differentiation of Schwann cells (Chen C. et al., 2018). Similarly, the PDA coating significantly improved the hydrophilicity and cytocompatibility of fabricated carbon scaffolds. Proliferation and differentiation of nerve cells were significantly accelerated under the electrical stimulation (Chen X. et al., 2018). Nazari et al. (2020) noted longer neuronal adhesion and growth on PLGA/carbon nanotube scaffolds modified via PDA coating. Meanwhile, PDA coating has been demonstrated to facilitate the efficient immobilization of NGF and adhesion peptides onto substrates. The growth factor or peptide-immobilized substrates enhanced the proliferation and differentiation of human NSCs (Yang et al., 2012). Another study also confirmed enhanced neuronal differentiation of nerve growth factor stimulated PC12 cells on the PDA-coated scaffolds (Bhang et al., 2013). As a mussel-inspired bioactive substance, PDA shows good prospects for the modification of nerve conduits, which may be applied in further preclinical studies and clinical trials.

Not only PDA coating on the nerve conduits plays an important role, but PDA combined with other materials contributes to the cell proliferation and the property improvement of nerve conduits. The manufactured PU/PDA/ECM nerve conduits exhibited significantly enhanced hydrophilicity, biodegradability, cell proliferation, and viability (Chen et al., 2019). The joint effect of PDA and RGD was reported to benefit the adhesion and proliferation of Schwann cells and mediate the process of cell signaling for nerve repair (Qian et al., 2018a). Robust copolymerization of PDA and PPY also showed its capability of enhancing the growth and proliferation of neuronal cells and stimulating neurogenesis (Kim et al., 2018).

Aside from *in vitro* work, PDA has also been explored in some *in vivo* studies. Qian et al. (2018a) showed notable locomotor and sensory function recovery in SD rats due to implantation of graphene-based nerve conduit coated with PDA/RGD. In another study, nerve regenerative capacity of PDA-gold/PCL nanoscaffolds in improving myelin sheath growth and functional recovery was observed on SD rats with sciatic nerve defects (Qian et al., 2018b). More sensitive signals from tibia muscles were demonstrated on the rat sciatic nerve when using PDA/PPY-coated electrodes than bare or PPY-coated electrodes (Kim et al., 2018).

Although there are limitations in *in vivo* researches on PDA-based repair of peripheral nerve injury, existing studies have shown its potential in the long-term clinical effect of peripheral nerve regeneration. More pre-clinical researches are in urgent need to translate PDA-dependent peripheral nerve repair into the clinical work.

## DISCUSSION

Reliable and valid treatment with time sensitivity is of great importance. Autologous nerve transplantation has reduced its clinical outcomes due to its limitations. The immune rejection of allotransplantation also affects the normal life and recovery process of patients. It remains an urgent and unsolved problem on how to prepare nerve grafts with excellent biomimetic performance (Pfister et al., 2011; Tamaki, 2014). The substrate materials used for the preparation of nerve conduits range from macromolecular materials to nanomaterials, from non-degradable materials to degradable materials, such as chitosan, the copolymer of lactic acid-hydroxy acetic acid, and silk fibroin. Nerve conduits prepared by composite materials often include polymer materials for surface modification or strength support to promote cell adhesion and nerve repair (Ekdahl et al., 2011). Luca et al. (2014) carried out surface treatment on PCL film and improved its hydrophilicity by hydrolysis and amino hydrolysis. An *in vitro* cell test showed that the adhesion and proliferation of Schwann cells on the film was improved significantly during a short period of time, indicating the importance of surface treatment for the preparation of nerve conduits. Zhang et al. (2019) cross-linked the natural copolymer silk fibroin protein with the regenerated directional silk fibroin solution to form a high-strength mechanical scaffold. In this way, the neurons and Schwann cells of the dorsal root ganglion could migrate with the uniform positioning of the scaffold. Mecobalamin-loaded silk fibroin scaffold could promote the survival and growth of neurons which indicated its high biocompatibility. Since the surface modification method of PDA films was proposed, this polymer material has been widely used in the surface modification of metals, semiconductors, ceramics, and other materials. It gradually develops from functionalization to diversification, especially in the field of biomedical advantages (Liu et al., 2014). As is shown above, due to its unique properties, PDA can promote the adhesion, proliferation, and differentiation of nerve cells, thus supporting the promotion of peripheral nerve injury repair. Although the experimental evidence for the actual application of PDA in the surface modification of nerve conduit is not sufficient, it is believed that PDA has a broad prospect in the peripheral nerve repair.

## CONCLUSION

Polydopamine based on mussel inspiration has high hydrophilicity, durable anti-corrosion ability, strong adhesion, and high biocompatibility. It has a broad application prospect in tissue engineering and biomedicine. Catecholamine functional groups can be quickly and effectively coated on the surface of various materials for function improvement through DA oxidation and polymerization reaction triggered by alkali. PDA improves the hydrophilicity and stability of the material surface. PDA-modified biomaterials promote cell adhesion, proliferation, and diffusion. The application of PDA in the nerve tissue engineering is developing rapidly. Although the current research on peripheral nerve repair by PDA is mainly based on

*in vitro* cell experiments, abundant existing experimental results indicate that PDA plays an important role in this field. Future experimental and translational work is necessary in order to obtain more findings of PDA on the level of clinical applications.

## AUTHOR CONTRIBUTIONS

YQ, JY, and SL conceptualized and designed the manuscript. RW, JY, and MJ drafted the manuscript, designed the table, and reviewed the literature. SL designed the figure. YQ revised the manuscript. All authors approved the final version.

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# Toward Biomimetic Scaffolds for Tissue Engineering: 3D Printing Techniques in Regenerative Medicine

Justin J. Chung<sup>1†</sup>, Heejung Im<sup>1†</sup>, Soo Hyun Kim<sup>1,2</sup>, Jong Woong Park<sup>3</sup> and Youngmee Jung<sup>1,4\*</sup>

<sup>1</sup> Center for Biomaterials, Korea Institute of Science and Technology, Seoul, South Korea, <sup>2</sup> KU-KIST Graduate School of Converging Science and Technology, Korea University, Seoul, South Korea, <sup>3</sup> Department of Orthopedic Surgery, Korea University Anam Hospital, Seoul, South Korea, <sup>4</sup> School of Electrical and Electronic Engineering, Yonsei University, Seoul, South Korea

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### \*Correspondence:

Youngmee Jung  
winnie97@kist.re.kr

<sup>†</sup> These authors have contributed  
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Three-dimensional (3D) printing technology allows fabricating complex and precise structures by stacking materials layer by layer. The fabrication method has a strong potential in the regenerative medicine field to produce customizable and defect-fillable scaffolds for tissue regeneration. Plus, biocompatible materials, bioactive molecules, and cells can be printed together or separately to enhance scaffolds, which can save patients who suffer from shortage of transplantable organs. There are various 3D printing techniques that depend on the types of materials, or inks, used. Here, different types of organs (bone, cartilage, heart valve, liver, and skin) that are aided by 3D printed scaffolds and printing methods that are applied in the biomedical fields are reviewed.

**Keywords:** 3D printing, tissue engineering, bioink, scaffold, regenerative medicine

## INTRODUCTION

Three-dimensional (3D) printing, also known as additive manufacturing, is a method that can fabricate objects with complex structures by depositing materials, i.e., metals, polymers, and ceramics, layer by layer (He et al., 2014; Hung et al., 2014). A 3D object can be produced through 3D scanning technology, such as computed tomography (CT) and computer-aided design (CAD) software (Hollister, 2005; Melchels et al., 2010). After image file of an object is acquired; it is converted to an STL file format that can be sliced into layers to create a 3D model (Melchels et al., 2010; Zorlutuna et al., 2012). Charles W. Hull, the president of 3D SYSTEMS, invented the first 3D printer, which was based on a stereolithography apparatus (SLA) technique (Melchels et al., 2010; Murphy and Atala, 2014; Schubert et al., 2014). The SLA printing method obtained issued a patent in 1986 (Gross et al., 2014). In 1990, a fused deposition modeling (FDM)-type printer was developed by Scott Crump, chairman of STRATASYS (Rengier et al., 2010; Gross et al., 2014). These 3D printing techniques sent shockwaves throughout multiple industries, such as automotive, aerospace, architecture, fashion, as well as bio-medicine (Duoss et al., 2014; Gross et al., 2014), since complex 3D structures can be precisely controlled and easily produced compared to subtractive methods (He et al., 2014; Choi and Kim, 2015).

In the medical fields, 3D printing technology is a promising tool for personalized treatments (Choi and Kim, 2015; Jakus et al., 2015; Kim et al., 2016). 3D models of a patient's damaged organ can be produced to serve as a visual aid for the surgeons and to help the patient to understand his or her conditions (Melchels et al., 2010; Gross et al., 2014). Additionally, 3D printing techniques

are applied to produce scaffolds and implants for regenerative medicine (He et al., 2014; Torres-Rendon et al., 2015; Kim et al., 2018; Lee et al., 2019). For example, Mankovich et al. (1994) used additive manufacturing to fabricate calvarial bone grafts, and Morrison et al. (2015) designed 3D printed scaffolds to treat tracheobronchomalacia. There are various 3D printing techniques that are classified by the types of materials and printing methods that are used to create an object (Table 1; Sachlos and Czernuszka, 2003; Childers et al., 2015). FDM is the most common 3D printing method. Thermoplastic filaments are melted by a heating block, and then a nozzle head directs the extrusion of the melted filaments to deposit thin layers (Azari and Nikzad, 2009; Gross et al., 2014). One of the advantages of FDM is that there are wide ranges of biodegradable and biocompatible materials, or filaments, which can be printed. Additionally, toxic organic solvents are not required to dissolve the polymeric filaments for printing (Yang et al., 2002; Leong et al., 2003). For example, Hutmacher et al. (2001) fabricated poly( $\epsilon$ -caprolactone) (PCL) scaffolds with a honeycomb-like porous structure using an FDM-type printer. Fibroblasts were able to proliferate and differentiate on the scaffold. However, the high temperature applied during the melt-extrusion stage can change the inherent material properties, and high-resolution printing is challenging (Yang et al., 2002). SLA type is based on solidification of liquid resin through photo-crosslinking (Melchels et al., 2010). A stage, or a base plate, for an object is immersed in liquid resin, and then the laser beam is applied to cure the resin on the stage. After the first layer is produced, the stage moves downward and the second layer is cured to deposit on the first layer (Gross et al., 2014). A selective laser sintering (SLS) printer follows a similar process to the SLA type, but the high-powered laser is applied to sinter solid powders (Leong et al., 2003). SLA-type printers can produce objects with high resolution and design more precise structures compared to the FDM technique. However, they are limited to photo-polymerizing resins, and the resins are often toxic for biomedical applications. The SLS method does not require liquid resins or toxic organic solvents to dissolve polymers, yet the sintering process can damage materials that are biodegradable (Yang et al., 2002; Williams et al., 2005). 3D plotting is an extrusion-based technology, which expels materials from a chamber by pneumatics (Sachlos and Czernuszka, 2003). Typically, plotting pastes and viscous materials are used as printing inks (Luo et al., 2013), which are either directly printed or melted in a feeding channel before they are extruded by a pneumatic pump (Ragaert et al., 2010). Due to the mild printing conditions, various soft materials (i.e., hydrogels, biocompatible polymers, and cell spheroids) can be printed with the 3D plotting method, and it is also referred to as bioprinting when cells are printed with hydrogel inks (Murphy and Atala, 2014). Haberstroh et al. (2010) were able to fabricate 3D plotted cell-seeded scaffolds of poly(L-lactide-co-glycolide) (PLGA), tricalcium phosphate (TCP)/collagen, and TCP/collagen/chitosan successfully for bone regeneration. The bioactivity of scaffolds and bone formation in calvarial defect model were evaluated. Lee et al. (2019) developed a new type of 3D bioprinting method, which allows printing a more precise and complex organ structure using hydrogel inks. The collagen ink was printed within a thermo-reversible

support bath of gelatin microparticles to reproduce patient-specific cardiac ventricles. Inkjet printing is also widely used in regenerative medicine. It is a droplet-based extrusion printing technique, where droplets from the supplied fluid are deposited layer by layer, or patterned to desired shapes with biomolecules (Xu et al., 2013). Inkjet printing methods are cost-effective and applied in various fields from drug screening to tissue engineering (Boland et al., 2006); however, it is challenging to print viscous materials and cells (Koch et al., 2010). The laminated object manufacturing (LOM) process builds polymeric and metallic layers that are sequentially fed from a roller (Park et al., 2000). Laminates are cut with CO<sub>2</sub> laser, and then layers are bonded by a heated roller. This rapid prototyping process can fabricate large objects with low cost (Murr(ed.), 2015); however, it is challenging to make small and precise structures (Yang et al., 2002).

Every year, millions of patients are waiting organ donors and suffer from long transplant waiting lists. Tissue engineering has a potential not only to solve the current complications in organ shortage but also to improve the current level of the biomedical technology (Khademhosseini et al., 2006; Slaughter et al., 2009). Vacanti et al. (1988) established seminal work in the field of tissue engineering in the 1980s, and it is still one of the most researched fields. Conventionally, most scaffolds for tissue engineering were fabricated through a “top-down” approach, where scaffolds are designed with biocompatible polymeric materials in porous structures to biomimic the host tissue. However, for cell attachment and proliferation, the scaffolds were coated with bioactive substances, or surface modification was necessary. In contrast, the “bottom-up” approach aims to encapsulate cells in hydrogels to allow self-assembly of cell aggregation and 3D print cells directly in the form of a scaffold (Figure 1). An ideal scaffold has to possess a surface that is suitable for cell attachment and 3D inter-connected porous structures for extracellular matrix (ECM) formation and vascularization. 3D printing allows fabricating scaffolds with more controlled and precise structures (Derby, 2012; Do et al., 2015) compared to electro-spinning (Poologasundarampillai et al., 2011), foaming (Mahony et al., 2010), freeze-drying (Connell et al., 2014), and salt-leaching (Woodard et al., 2017) techniques. Here, we review 3D printing technologies for regenerative medicine, from 3D printed polymeric scaffolds to bio-artificial tissues, and promising outlooks for advanced treatments through 3D printing.

## 3D PRINTED SCAFFOLDS FOR TISSUE ENGINEERING

There have been numerous types of 3D printing techniques used and developed by researchers in the field of tissue engineering. In this section, 3D printing techniques and 3D printed biomaterials are categorized into subsections by tissues and organs that they were designed to aid. They are also summarized in Table 2.

### Bone

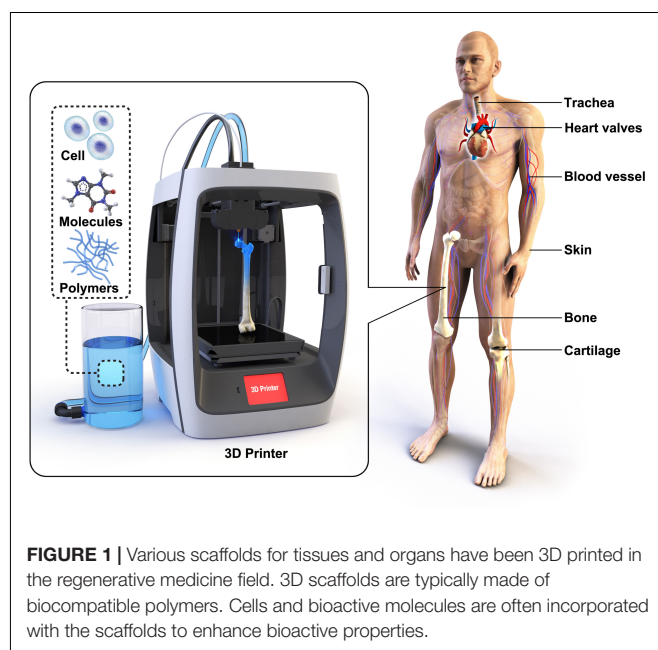
The bone regeneration process involves migration and recruitment of osteoprogenitor cells to a defect region, which will then differentiate to osteoblasts to form bone minerals or

**TABLE 1** | Different types of 3D printing techniques, and their pros and cons.

Types	Advantage	Limitations	References
FDM	Thermoplastic polymers are extruded without toxic organic solvents	Melting process can affect inherent material properties	Cornejo et al., 2000; Hutmacher et al., 2001; Yang et al., 2002; Zein et al., 2002; Leong et al., 2003; Woodfield et al., 2004; Azari and Nikzad, 2009; Gross et al., 2014; Hung et al., 2014, 2016; Kwon et al., 2015
SLA	High-resolution objects can be printed with complex structures	Printable materials are limited to liquid resins, which can be toxic	Rimell and Marquis, 2000; Leong et al., 2003; Tsang and Bhatia, 2004; Jiankang et al., 2009; Melchels et al., 2010; Gross et al., 2014; Melchiorri et al., 2016
SLS	Powdered materials are sintered through a similar process as SLA. No liquid resins are needed	Sintering can modify material properties	Yang et al., 2002; Seitz et al., 2005; Williams et al., 2005; Gbureck et al., 2007; Khalyfa et al., 2007; Morrison et al., 2015; Pei et al., 2017; Zhang et al., 2019
3D plotting	Bioceramics can be printed in mild conditions. 3D cell printing is possible by seeding cells into hydrogels	Post-sintering or curing is necessary. Constrained by temperature and complex multi-layer fabrication is challenging for bioprinting	Sachlos and Czernuszka, 2003; Wang et al., 2006; Lee et al., 2009, 2014b, 2019; Haberstroh et al., 2010; Ragaert et al., 2010; Fu et al., 2011; Wu et al., 2011; Hockaday et al., 2012; Bose et al., 2013; Gao et al., 2013; Luo et al., 2013; Mannoor et al., 2013; Duan et al., 2014; Inzana et al., 2014; Murphy and Atala, 2014; Markstedt et al., 2015; Kim et al., 2018; Nommeots-Nomm et al., 2018; Tallia et al., 2018; Li et al., 2020; Shi et al., 2020
Inkjet	Precise and controlled placement printing for small volume biological materials	Difficult to print viscose materials/cells. Large volume construct fabrication is challenging	Boland et al., 2006; Cui and Boland, 2009; Koch et al., 2010; Xu et al., 2013
LOM	Ideal for fabricating large 3D objects	Lamination coatings can be toxic and vulnerable when fabricating small constructs	Park et al., 2000; Yang et al., 2002; Murr(ed.), 2015

hydroxyapatite (HA). There are various methods to enhance bioactive properties to bone tissue engineering scaffolds, such as incorporating growth factors and gene/drug deliveries. An ideal bone scaffold should be made with biocompatible materials that act as a temporary template to withstand mechanical forces in the defect site until the host tissue is fully recovered. Specifically, biodegradation rate should be similar to the duration of bone formation process, and inter-connected porous structure for vascularization is essential for the scaffold design (Bose et al., 2012; Jones, 2013).

Synthetic biodegradable polymers, such as poly(caprolactone) (PCL), poly(glycolic acid) (PGA), poly(lactic acid) (PLA), and their copolymers, have received high attention in the biomedical field since ester linkages can be degraded by hydrolysis and their by-products are non-toxic. These biodegradable polymers have also been 3D printed to produce scaffolds for tissue engineering (**Figure 2A**). Williams et al. (2005) fabricated PCL scaffold with similar mechanical properties to that of human trabecular bone using the SLS printing technique. Bone morphogenetic protein-7 (BMP-7) was seeded to enhance the bioactivity of the scaffold, and subcutaneous implantation has shown bone formation within 4 weeks. Additionally, the scaffold was fabricated to replicate a CT-scanned minipig condyle structure, which showed a possibility of producing patient-specific scaffolds (**Figures 2B,C**). Kwon et al. (2015) synthesized PCL containing PLGC [methoxy poly(ethylene glycol)-co-L-lactide-co-glycolide-co-ε-caprolactone] copolymer scaffold through the FDM process. Human dental pulp stem cell (hDPSC)-loaded PLGC scaffold instigated bone regeneration, and the degradation rate of the scaffold was comparable to the bone formation rate.



**FIGURE 1** | Various scaffolds for tissues and organs have been 3D printed in the regenerative medicine field. 3D scaffolds are typically made of biocompatible polymers. Cells and bioactive molecules are often incorporated with the scaffolds to enhance bioactive properties.

Calcium phosphate-based ceramics have similar compositions to bone mineral; therefore, they have been widely used as bone substitutes. Since bioceramics are in powder form, the SLS method is often used to produce grid-like scaffold structures (Seitz et al., 2005; Gbureck et al., 2007; Khalyfa et al., 2007). 3D plotting is also practiced with a post-sintering process when binders and sacrificial polymers are mixed with bioceramics to produce printable inks (Pei et al., 2017; Zhang et al., 2019).

**TABLE 2 |** Summary of materials, cell types, and molecules used for 3D printed scaffolds for tissue engineering.

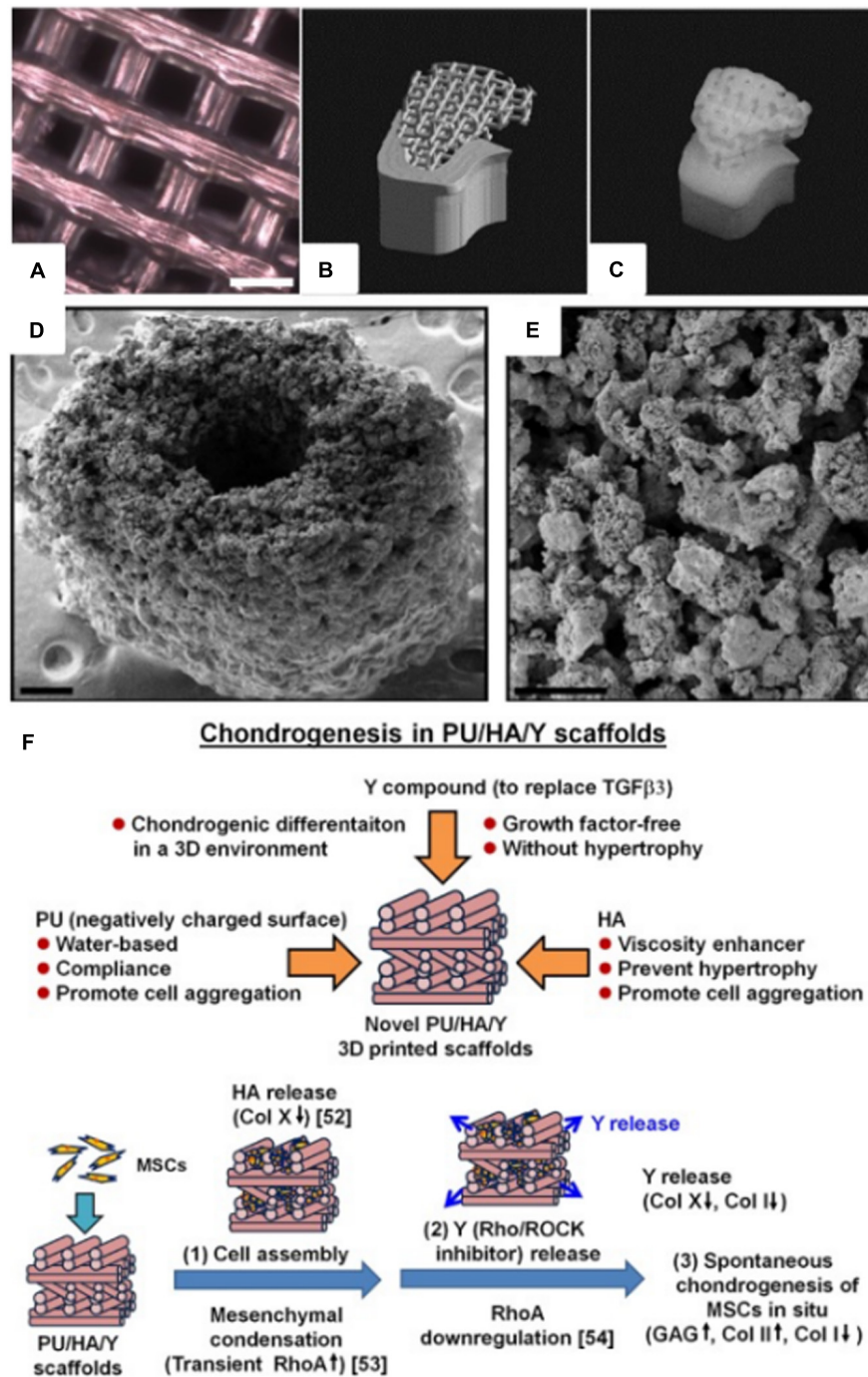
Organs	Materials	Cell type/Molecules	Printer type	References
Bone	PCL	BMP-2	SLS	Williams et al., 2005
	PLGA, TCP, Collagen, Chitosan	–	FDM and 3D plotting	Haberstroh et al., 2010
	PLGC	–	FDM	Kwon et al., 2015
	TTCP, $\beta$ -TCP, Calcium sulfate	–	SLS	Khalyfa et al., 2007
	TCP	–	SLS	Gbureck et al., 2007; Zhang et al., 2019
	HA	–	SLS	Seitz et al., 2005; Pei et al., 2017; Zhang et al., 2019
	Calcium phosphate, Type I collagen	–	3D plotting	Inzana et al., 2014
	PCL, Type I collagen, Alginate, Gelatin	DPSCs/VEGF, BMP-2	FDM and 3D plotting	Park et al., 2015
	Bioactive glasses	–	3D plotting	Fu et al., 2011; Nommeots-Nomm et al., 2018; Shi et al., 2020
Cartilage	Bioglass–gelatin hybrid	–	3D plotting	Gao et al., 2013
	PU, PEO	–	FDM	Hung et al., 2014, 2016
	PEGT/PBT block copolymer	–	FDM	Woodfield et al., 2004
	Cellulose, Alginate	Human nasoseptal chondrocytes	3D plotting	Markstedt et al., 2015
	PCL, PEG, Alginate	Chondrocytes, Adipocytes	FDM and 3D plotting	Lee et al., 2014a
	Silicon, Alginate, Silver nanoparticle	Calf articular chondrocyte	3D plotting	Mannoor et al., 2013
	Silica-Poly(tetrahydrofuran)-PCL-hybrid	–	3D plotting	Tallia et al., 2018; Li et al., 2020
Heart valve	PEG-DA, Alginate	PAVIC	3D plotting	Hockaday et al., 2012
	Me-HA, Me-Gel	HAVIC	3D plotting	Duan et al., 2014
	Collagen	Human embryonic stem cell-derived cardiomyocytes	3D plotting	Lee et al., 2019
Blood vessel	Pluronic F127-DA	–	3D plotting	Wu et al., 2011
	Poly(propylene fumarate)	–	SLA	Melchiorri et al., 2016
	Type I collagen, Gelatin	HUVECs, ECs	3D plotting	Lee et al., 2014b
	Fibrinogen, Thrombin, $\text{CaCl}_2$	HMVECs	Inkjet	Cui and Boland, 2009
Trachea	PCL, Hydroxyapatite	–	SLS	Morrison et al., 2015
	PCL	–	FDM	Chang et al., 2014
	Silk fibroin	Chondrocytes	3D plotting	Kim et al., 2018
Liver	Gelatin	Hepatocytes	3D plotting	Wang et al., 2006
	PDMS	–	SLA	Jiankang et al., 2009
	PCL, Collagen	Hepatocytes, HUVECs, HLFs	FDM and 3D plotting	Lee et al., 2016
Skin	Type I collagen	hDFB, hEKC	3D plotting	Lee et al., 2009
	Collagen	NIH-3T3, HaCaT	Inkjet	Koch et al., 2012

Although bioactive glasses are not commercially successful as bioceramics, they are known to be more bioactive, and 45S5 composition (Bioglass®) was the first artificial material that formed a chemical bond to bone (Hench, 2006; Jones, 2013; Cai et al., 2018). Various compositions of bioactive glasses were also 3D printed to biomimic the porous structure of bone (Fu et al., 2011; Nommeots-Nomm et al., 2018; Shi et al., 2020). However, inorganic scaffolds (i.e., bioceramics and bioactive glasses) are too brittle for repairing defect sites that are exposed to constant loading. Flexibility and toughness of bioactive glasses can be enhanced by introducing flexible polymers to the silica network with covalent bonding, termed inorganic–organic hybrids. This is possible since glasses can be fabricated through a sol-gel process that prevents oxidization of polymers (Sanchez et al., 2005; Jones

et al., 2006; Valliant and Jones, 2011; Chung et al., 2017). Gao et al. (2013) were able to 3D print a gelatin–bioactive glass hybrid to a grid-like structure. MC3T3-E1 osteoblast precursor cells were able to adhere and proliferate on the printed hybrid scaffold.

Bone is a nanocomposite composed of HA (50–70%) and organic matrix (20–40%), which is primarily composed of type I collagen (Clarke, 2008; Jones, 2013). Collagen is widely used as biomaterials for tissue engineering in skin (Powell et al., 2008), bone (Rodrigues et al., 2003), tendon (Young et al., 1998), and blood vessel (Zorlutuna et al., 2008) applications due to its tough mechanical properties and biocompatibility. Inzana et al. (2014) 3D printed collagen and calcium phosphate composite for bone regeneration. The scaffold was implanted in a murine femur with critical defect size, and osteoconductivity was confirmed





**FIGURE 2 |** (A) Optical microscope image of a 3D printed poly(l-lactide-co-ε-caprolactone) scaffold for adipose tissue engineering (courtesy of YJ, scale bar = 2 mm). (B) STL image of a pig condyle scaffold. (C) Front view of the 3D printed PCL scaffold (Williams et al., 2005). (D) SEM image of a 3D printed murine-sized scaffold for femoral mid-diaphysis regeneration (scale bar = 250 μm). (E) Micro-porosity of the calcium phosphate-collage composite scaffold with pore sizes of 20–50 μm (scale bar = 100 μm) (Inzana et al., 2014). (F) 3D printed PU/HA-based scaffold design, and possible mechanism of spontaneous chondrogenesis *in situ* (Hung et al., 2016) (Reproduced with permission from Williams et al., 2005; Inzana et al., 2014; Hung et al., 2016).

(Figures 2D,E). 3D bioprinting technique is one of the most recent methods of printing biomaterials; it renders 3D tissue constructs with cells embedded in hydrogels (Mironov et al., 2009; Kang et al., 2013; Levato et al., 2014). Park et al. (2015) were

able to demonstrate the multi-head bioprinting method. Vascular endothelial growth factor (VEGF) and BMP-2 were loaded to various blends of hydrogels with hDPSC. The hydrogels were 3D plotted to a PCL scaffold framework, and this fabrication method

was shown to produce large volume scaffolds, which is one of the major limitations in the tissue engineering field.

## Cartilage

The ECM of cartilage is composed of type II collagen and glycosaminoglycan (GAG), which allows to regulate expression of chondrocyte phenotype and instigate chondrogenesis (Suh and Matthew, 2000). In contrast to bone, cartilage has limited ability to naturally heal itself, since avascular structure inhibits nutrients and progenitor cells to migrate toward the defect region. Articular cartilage covers end of bones in synovial joints, which allows the bones to glide over each other; therefore, it should withstand load-bearing forces while providing low-friction surfaces. Osteoarthritis and high-impact injuries can cause articular cartilage defects, and it is one of the most challenging tissues to repair (Temenoff and Mikos, 2000; Huey et al., 2012).

Poly(ethylene glycol) terephthalate/poly(butylene terephthalate) (PEGT/PBT) block copolymer was 3D printed through the FDM technique to fabricate grid-like structured scaffold. The scaffold was seeded with bovine articular chondrocytes (bACs), which developed cartilage-like tissue *in vivo* while having mechanical properties similar to the native articular cartilage (Woodfield et al., 2004). Hung et al. (2014, 2016) were able to 3D print cartilage scaffolds with water as a printing ink solvent, which allowed incorporation of biomolecules, i.e., growth factors, with higher biocompatibility compared to inks that require organic solvents. Polyurethane (PU) particles, hyaluronic acid (HA), and TGFβ3 containing ink were 3D printed through a customized low-temperature FDM-type printer. Then, the scaffold was seeded with mesenchymal stem cells (MSCs) to improve cartilage regeneration *in vivo* (Figure 2F). Markstedt et al. (2015) developed a nanocellulose-alginate scaffold using the 3D bioprinting technique. The rheological properties of the composite bioink, which required low pressure to extrude at room temperature, allowed the production of precise 3D grid, disc, human ear, and sheep meniscus constructs. The cytotoxicity and live/dead cell-imaging assay confirmed that the scaffold was suitable for cartilage regeneration.

Other researchers have also reported 3D printing of ear-shaped structures for cartilage regeneration. Lee et al. (2014a) fabricated a human ear scaffold by printing both articular cartilage and fat tissue. Poly(ethylene glycol) (PEG) was used as a sacrificial layer since it is soluble in aqueous solutions, and PCL was printed as a main framework of the scaffold. Alginate hydrogel was used as a bioink to print chondrocyte and adipocyte. It was printed along with PCL to fabricate an ear scaffold with two distinct regions, which included a main ear part (chondrocyte) and an earlobe area (adipocyte). Additionally, co-printing of cell-seeded alginate scaffold confirmed that gene expression of both chondrocyte and adipocyte were remarkably enhanced compared to that of the control group. Mannoor et al. (2013) were able to integrate 3D biological tissue and electronic components. Alginate hydrogel with chondrocyte and silver nanoparticles was printed to structurally mimic human ears. Then, cochlea-shaped electrodes for hearing were inserted into the hydrogel construct,

which was referred to as “cyborg ears.” Cell viability of the cyborg ear was  $91.3 \pm 3.9\%$ , which is an adequate biocompatibility for the application.

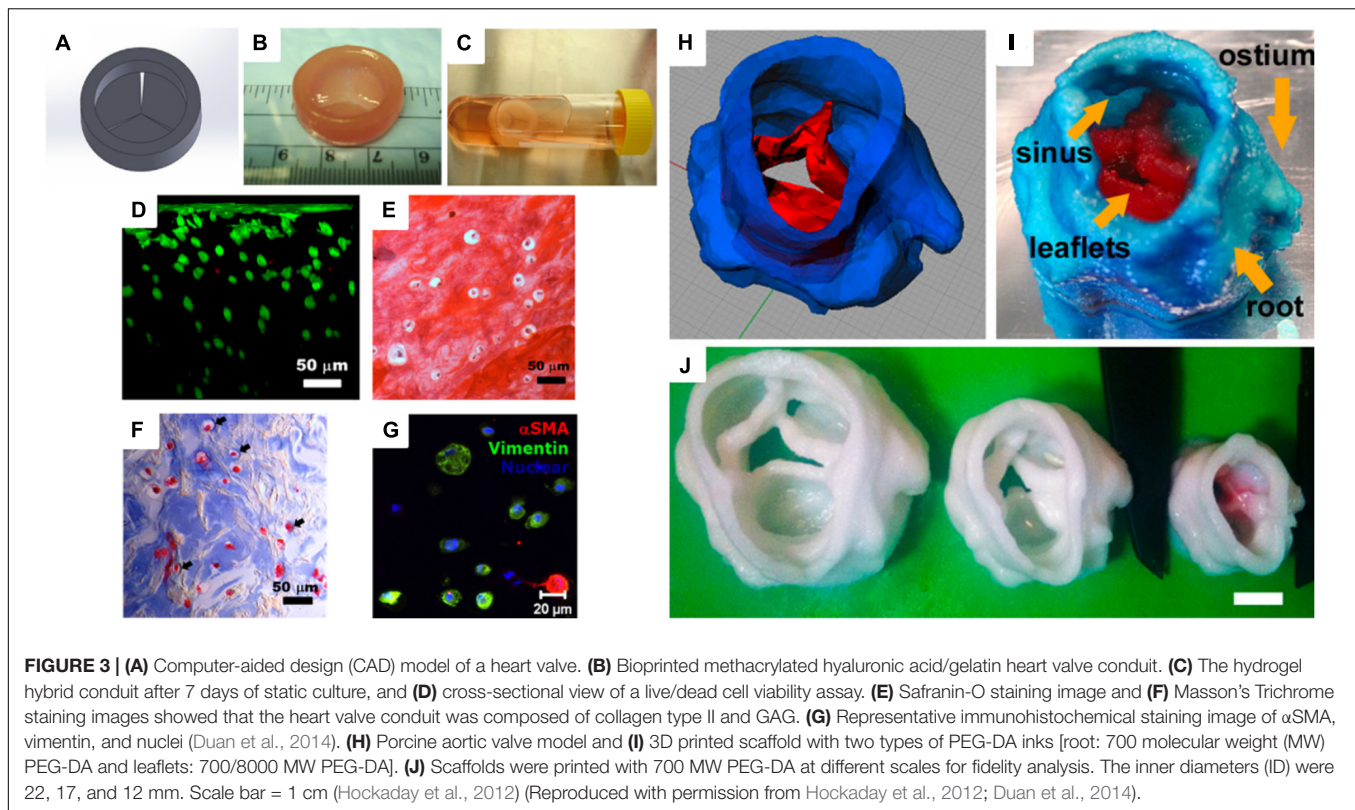
Recently, an inorganic–organic hybrid of silica-poly(tetrahydrofuran)/PCL was 3D printed to fabricate scaffolds for articular cartilage regeneration (Tallia et al., 2018; Li et al., 2020). The silica network and organic component was forming co-networks via covalent bonding, which displayed elasticity, self-healing ability, and bioactivity. The scaffold with a grid-like structure mimicked the compressive behavior of cartilage, and *in vitro* chondrogenic differentiation was observed.

## Heart Valve

Heart is one of the essential organs in human physiology, which consists of various muscles to pump blood in the circulatory system. Heart valves, two in atria and the other two in ventricle chambers, are important for the blood circulation since they prevent backward flow (Dasi et al., 2009). In 2000, the American Heart Association announced that 87,000 replacement surgeries occurred (Flanagan and Pandit, 2003). Specifically, aortic valve disease is one of the serious cardiovascular diseases that are usually treated by replacement of the valves. Many researchers have studied artificial heart valves using various polymeric materials, such as PGA, PLA, collagen, and fibrin. Similar to the scaffolds described in the previous sections, inherent structures and mechanical properties are also important for designing heart valve conduits. Therefore, a 3D printing technique has been applied to this research field for several years (Hockaday et al., 2012). For heart valve engineering, hydrogels are promising materials due to their physicochemical and mechanical stability while they are hydrated. Furthermore, hydrogels are permeable for nutrients and waste transportation. Duan et al. (2014) fabricated human aortic valvular interstitial cell (HAVIC)-encapsulated heart valve conduits with photo-crosslinkable methacrylated hyaluronic acid (Me-HA) and methacrylated gelatin (Me-Gel) hydrogels. The viscosity of the hydrogel conduits was optimized and tuned by applying different hydrogel concentrations. The 3D bioprinted hydrogel conduits confirmed cell viability and remodeling potential for initial collagen and glycosaminoglycan matrix formation (Figures 3A–G). Hockaday et al. (2012) printed heart valve scaffolds with photo-crosslinkable poly(ethylene glycol)-diacrylate (PEG-DA). The scaffold was fabricated with two types of PEG-DA with different molecular weights in order to meet the heterogeneous mechanical properties of aortic valves. The conduit had high elastic modulus and nearly 100% cell viability (Figures 3H–J).

## Blood Vessel/Trachea

In the United States, coronary artery bypass grafting surgeries are performed more than 400,000 per year. Critical drawbacks of the surgeries are graft damages during harvesting procedure, poor long-term patency, and donor morbidity. Therefore, there are high demands for the development of artificial blood vessels that can overcome the current shortcomings. An ideal artificial blood vessel should be biocompatible, anti-thrombogenic, and durable, and have comparable compliance with structural density



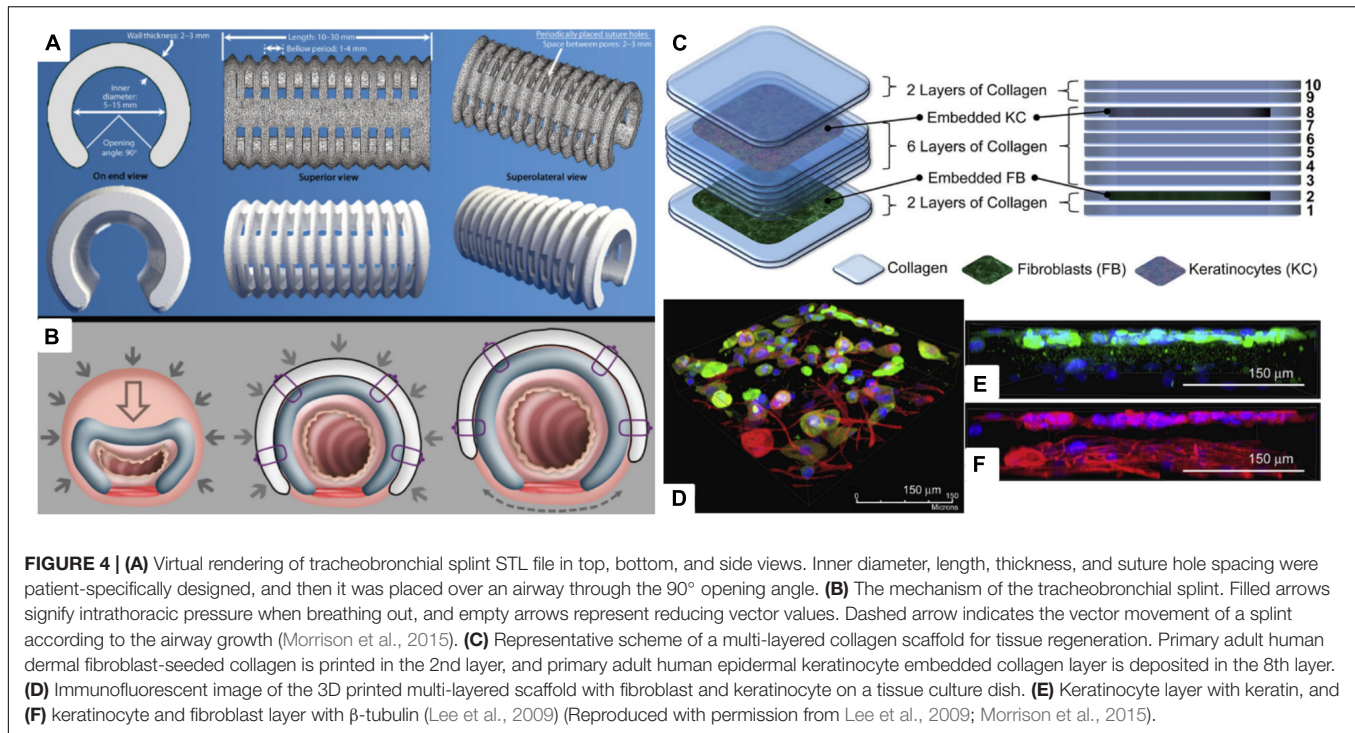
to that of the native blood vessels (Mosadegh et al., 2015). Wu et al. (2011) fabricated a biomimetic 3D microvascular network based on hydrogel matrix. The blood vessel's branching pattern was designed by omnidirectional printing of sacrificial ink in a photo-crosslinkable hydrogel matrix. The authors suggested that this technique can be applied to 3D cell culture; however, there were no cell tests performed. Similar work was performed by Lee et al. (2014b); functional vascular channels with perfused open lumen were fabricated through 3D bioprinting of collagen matrix with liquefying fugitive ink. Gelatin with endothelial cell (EC) was used as a fugitive ink, which protected from plasma protein and dextran molecule. Additionally, human umbilical vein endothelial cells (HUVECs) were cultured in the vascular channel, which was successfully aligned along the flow direction. The gene expression analysis confirmed that the 3D printed vascular channels had high potential for tissue engineering application.

Fibrin is a natural polymer formed by polymerization of fibrinogen and thrombin. It is present in human blood and involved in the wound healing process. Cui and Boland (2009) 3D printed human microvascular endothelial cell (HMVEC)-seeded bioink of thrombin and  $\text{Ca}^{2+}$  solution into a fibrinogen substrate. The scaffold was composed of fibrin channels with aligned HMVECs, and 21 days of cell culture confirmed tubular structure formation inside the channels. Poly(propylene fumarate)-based aorta graft was synthesized via digital light stereolithography technique (Melchiorri et al., 2016). The biodegradable polymer was 3D printed to an MRI/CT scanned structure, which showed

a possibility to design patient-specific aorta grafts. Additionally, the scaffold was able to confirm bioactive properties *in vivo* with comparable mechanical strength to that of human aorta.

Tracheal structure restoration and scaffold fabrications are also in great demand. Forty-three percent of pediatric patients who went through tracheostomy experience respiratory arrest due to tube occlusion (Carr et al., 2001; Berry et al., 2010). However, if the pediatric patients are supported by a temporary scaffold for 24 to 36 months, the airway growth can naturally resolve the disease. Morrison et al. (2015) were able to produce a personalized and biodegradable tracheal splint through the SLS technique. PCL powders were mixed with hydroxyapatite, which was used as a flowing agent for the laser sintering process. PCL splints were successfully implanted to pediatric patients, and they were able to expand over time with airway growth (Figures 4A,B). Chang et al. (2014) 3D printed PCL scaffolds through an FDM-type printer for tracheal regeneration. The scaffold was coated with MSC-seeded fibrin to enhance bioactivity. *In vivo* study confirmed that the scaffolds were mechanically stable and able to reconstruct trachea within 8 weeks of implantation. Kim et al. (2018) produced a ring-like cartilaginous trachea scaffold through 3D bioprinting with digital light processing technique. Chondrocytes were encapsulated in methacrylated silk fibroin, which made cross-linking possible through UV light exposure. This cell-loaded hydrogel scaffold showed homogeneously distributed cells and cartilage tissue formation *in vitro*. The chemically modified silk fibroin ink was also printed





to heart, lung, and vascular shapes, which confirmed that the bioink and printing method can be applied to various tissue engineering applications.

## Liver

The liver is a fundamental organ that is responsible for multifunctional metabolic activities. Although liver transplant has been practiced for a long time, the procedure is costly, patient survival rate is poor, and there is a shortage of organ donors (Wang et al., 2007). Wang et al. (2006) made a 3D construct of gelatin hydrogel with hepatocyte as an ECM and 2.5% glutaraldehyde as a cross-linking agent. They were able to confirm that hepatocytes in gelatin construct can survive over 2 months in *in vitro* cell culture and retain their 3D structure for a month. A chitosan–gelatin hybrid scaffold was developed to biomimic the architecture of natural liver. A highly porous and well-organized structure was fabricated by a combination of 3D printing, micro-replication, and freeze-drying techniques. The novel scaffold was also composed of intrinsic fluidic channels and hepatic chambers. Firstly, a resin mold was fabricated by the SLA technique to cast polydimethylsiloxane (PDMS) for creating a micro-replication mold. Chitosan–gelatin solution was cast in the PDMS mold followed by freeze-drying to produce a porous structure. Biodegradability and hepatocyte growth were confirmed through 7-day cell culture. More importantly, albumin secretion and urea synthesis were evident, which are representative evaluation for hepatocyte functionality (Jiankang et al., 2009). Lee et al. (2016) 3D printed PCL to a grid-like structure as a primary framework to mechanically support collagen bioinks. HUVECs, human lung fibroblasts (HLFs), and hepatocytes were encapsulated in collagen inks, printed in

between PCL struts for angiogenesis. The authors also evaluated albumin secretion and urea synthesis for confirming hepatocytes functionality and angiogenesis. Specifically, hepatocyte-, HLF-, and HUVEC-containing constructs showed highest albumin secretion and urea formation compared to that of hepatocytes only and hepatocyte- and HLF-containing groups.

## Skin

Skin is the largest organ in our body that is responsible for various functions, such as preventing loss of body fluid, acting as a barrier against pathogenic bacterium and thermotaxis, and regulating body temperature (Bonvallet et al., 2015; Ojeh et al., 2015). Severe acute and chronic wounds (i.e., burns, diabetic ulcer, pressure sores, and lesion) effect loss of dermal tissues. Skin grafts have limitations in antigenicity and shortage of transplantable tissues; therefore, there are high demands for skin regeneration (Tchemtchoua et al., 2011; Wang et al., 2013). Lee et al. (2009) used a 3D bioprinter with four-channel dispensers to print stratified skin layers. PDMS substrate was first coated with sodium bicarbonate, a pH-altering cross-linking agent. Then, collagen was printed layer by layer to fabricate a multi-layered skin construct. Specifically, among the 10 layers of skin construct, the 2nd and 8th collagen layers were embedded with cells, fibroblast and keratinocyte, respectively. Several advantages of this fabrication method include the following: the scaffold can be made on irregular surfaces, as long as cross-linking agent coating is possible, and other types of hydrogels can substitute collagen, if they are cross-linkable. Additionally, this was the first study to 3D print both keratinocyte and fibroblast for skin regeneration (**Figures 4C–F**). Koch et al. (2010, 2012) also 3D printed keratinocyte- and fibroblast-embedded collagen for



skin tissue engineering. Laser-assisted bioprinting (LaBP) with laser-induced forward transfer technique was used to produce 3D scaffolds. The LaBP method is advantageous over other bioprinting techniques since higher-resolution cell printing is possible with greater cell density. Also, various hydrogels can be printed regardless of their viscosity. The skin scaffold, which was printed in micro-scale, confirmed that each cell layer did not blend into each other. Ten days of culture confirmed the cell vitality of the cells embedded in the scaffold, and collagen layers did not intermix with each other.

## CONCLUSION AND PERSPECTIVES

Three-dimensional printing is one of the most promising technologies in tissue engineering and regenerative medicine to fabricate advanced 3D scaffolds. It allows to produce more defined and biomimetic scaffolds with bioactive factors to enhance their functionalities. Although there have been numerous studies in 3D printing for biomedical applications, there are still much room for improvement (Jakus et al., 2016; Zhu et al., 2016). Optimization of printable inks, standardization of printing methods, and higher reproducibility with mass production are major challenges. 4D printing, an integration of 3D printing with time, has emerged recently. This technology allows the printed materials to change their physical forms or functionalities when excited by an external stimulus, such as temperature, water, magnetism, and pH (Gao et al., 2016; Gladman et al., 2016). Since human body is a complex environment with various stimuli, 4D printing technology is receiving a lot of attention for medical implant surgeries. 3D bioprinting of stem cells has shown unprecedented possibilities for producing tissue constructs from bone to skin. As bioprinting technology advances, printing induced pluripotent stem (iPS) cells could take current bioactive scaffolds a step

closer to regenerate patient-specific tissues and organs. iPS cells are known to have advantages over embryonic stem cells since they can be derived from patients for autologous cell treatment. However, the reprogramming process is not fully understood so far (Yamanaka, 2009). 3D bioprinting is an encouraging technology for future regenerative medicine. It allows to deliver, or mount, cells and physicochemical factors that are essential for tissue regeneration. Furthermore, patient-specific therapies are one of the essential technologies for hospital factory, where damaged organ rendering is produced by medical imaging, and defect regenerating construct is printed with patient's cells, plasma, and tissues in the operating theater. The authors are confident that the progress in 3D printing technology will foster and enhance personalized regenerative medicine.

## AUTHOR CONTRIBUTIONS

JC and HI drafted the initial manuscript with guidance by YJ. SK, YJ, and JP made contributions and modifications according to their field of expertise. All authors reviewed and approved the manuscript.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Desktop-Stereolithography 3D Printing of a Polyporous Extracellular Matrix Bioink for Bone Defect Regeneration

Yunxiang Luo<sup>1,2†</sup>, Hao Pan<sup>3†</sup>, Jiuzhou Jiang<sup>1,2†</sup>, Chenchen Zhao<sup>1,2</sup>, Jianfeng Zhang<sup>1,2</sup>, Pengfei Chen<sup>1,2\*</sup>, Xianfeng Lin<sup>1,2\*</sup> and Shunwu Fan<sup>1,2\*</sup>

<sup>1</sup> Department of Orthopaedic Surgery, Sir Run Run Shaw Hospital, Medical College of Zhejiang University, Hangzhou, China,

<sup>2</sup> Key Laboratory of Musculoskeletal System Degeneration and Regeneration Translational Research of Zhejiang Province, Hangzhou, China, <sup>3</sup> Department of Orthopaedic, The First Affiliated Hospital of Wenzhou Medical University, Wenzhou, China

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### Edited by:

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Shanghai Jiao Tong University, China

### \*Correspondence:

Pengfei Chen  
790048914@qq.com  
Xianfeng Lin  
xianfeng\_lin@zju.edu.cn  
Shunwu Fan  
shunwu\_fan@zju.edu.cn;  
shunwu\_fan@126.com

<sup>†</sup> These authors have contributed  
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**Introduction:** Decellularized tendon extracellular matrix (tECM) perfectly provides the natural environment and holds great potential for bone regeneration in Bone tissue engineering (BTE) area. However, its densifying fiber structure leads to reduced cell permeability. Our study aimed to combine tECM with polyethylene glycol diacrylate (PEGDA) to form a biological scaffold with appropriate porosity and strength using stereolithography (SLA) technology for bone defect repair.

**Methods:** The tECM was produced and evaluated. Mesenchymal stem cell (MSC) was used to evaluate the biocompatibility of PEGDA/tECM bioink *in vitro*. Mineralization ability of the bioink was also evaluated *in vitro*. After preparing 3D printed polyporous PEGDA/tECM scaffolds (3D-pPES) via SLA, the calvarial defect generation capacity of 3D-pPES was assessed.

**Results:** The tECM was obtained and the decellularized effect was confirmed. The tECM increased the swelling ratio and porosity of PEGDA bioink, both cellular proliferation and biomineralization *in vitro* of the bioink were significantly optimized. The 3D-pPES was fabricated. Compared to the control group, increased cell migration efficiency, up-regulation of osteogenic differentiation RNA level, and better calvarial defect repair in rat of the 3D-pPES group were observed.

**Conclusion:** This study demonstrates that the 3D-pPES may be a promising strategy for bone defect treatment.

**Keywords:** 3D printing, stereolithography, tendon extracellular matrix, polyethylene glycol diacrylate, calvarial defect

## INTRODUCTION

High-energy injuries or pathological fractures, such as tumors and inflammation, are the main reasons for bone defects, which create numerous challenges in the clinical setting and require bone grafting (Campana et al., 2014; Hwang et al., 2017; Zhang H. et al., 2019). As the gold standard for bone substitution in clinical surgeries, the autogenous bone grafts application is limited by

the inadequate practicability, donor-site morbidity, and complicated surgical procedures (Sharif et al., 2016; Bez et al., 2018; Cabbad et al., 2019; Chen et al., 2019). Consequently, a more valid alternative procedure of BTE platform was proposed. BTE provides several benefits such as the rare spread of disease, lower infection or immunogenicity rate, various implantation materials, and wide availability (Hollister and Murphy, 2011). Many biomaterial substitutes have already been clinically applied due to their superior biological performance (Gibbs et al., 2016; Hassan et al., 2019).

Among all biological materials, the extracellular matrix (ECM) graft retains its natural structure and has high homology among different species, since it is derived from the biological tissue rather than the chemosynthesis materials, and it shows excellent effects in terms of the regulation of cellular adhesion, proliferation, migration, and differentiation (Pham et al., 2008). The tendon ECM (tECM) is rich in type-1 collagen, which can serve as a heterogeneous nucleation template to induce calcium and phosphorus (Ca-Pi) cluster formation (Xu et al., 2015; Thankam et al., 2018). In this biochemistry procedure, a self-assembled, pseudo-hexagonal array of collagen molecules participate and facilitate the Ca-Pi binding and nucleation (Xu et al., 2015). However, independent application of tECM to build a scaffold in BTE has several drawbacks; lack of sufficient mechanical support due to the original physical property, difficulty in forming a specific shape coinciding with the bone defect, and the lack of 3D micropore structure that is beneficial to the cell growth and differentiation (Narayanan et al., 2009).

To conquer the shortcomings that pure ECM is unable to provide, like adequate mechanical strength, the scaffold processing techniques are taken into consideration. Conventional scaffold processing techniques that fabricate various tissues, such as phase separation (Fang et al., 2019), freeze-drying (Grenier et al., 2019; Zhang L. et al., 2019), solvent casting (Ahn et al., 2018; Mao et al., 2018), gas foaming (Kaynak Bayrak et al., 2017; Catanzano et al., 2018), and electrospinning (Chan et al., 2019), cannot precisely control pore size, geometry, and interconnectivity of the scaffolds. However, 3D printing has emerged as a brand-new material processing approach, which largely overcomes these difficulties, allowing us to fabricate more bionic scaffolds for bone transplantation and to repair the bone defect in a clinical setting (Do et al., 2015). To ensure the 3D printing scaffold is built efficiently, hybridizing the natural ECM and synthetic polymer-based materials to create novel tissue-engineered scaffolds seems feasible because, in this method, the advantages, including the biocompatibility of the ECM, and the superior physical properties of the synthetic polymer-based materials, are both fully embodied (Lee et al., 2014). Though traditional 3D printing provides lots of benefits, creating a complicated 3D scaffold with natural ingredients of biological origin, such as fibrin, gelatin or hyaluronic acid, seems impractical (Knowlton et al., 2016). Stereolithography (SLA), a simple, user-friendly photo-crosslinked biomaterial printing with high resolution, gives us a brand-new solution to achieve bone regeneration (Do et al., 2015; Knowlton et al., 2016).

Here, we focus on the 3D printing-based scaffolds using the hybridization of tECM and synthetic polymer-based materials to fabricate a highly interconnected architecture bone grafts scaffold (**Scheme 1**). This 3D printed polyporous PEGDA/tECM scaffolds (3D-pPES) could promote the mesenchymal stem cell (MSCs) proliferation and migration in the defect regions. The promising bone defect regeneration could be achieved using this novel repair system.

## MATERIALS AND METHODS

All animal experiments *in vivo* were conducted following the principles and procedures of the National Institutes of Health (NIH) Guide for the Care and Use of Laboratory Animals and the guidelines for animal treatment of Sir Run Shaw Affiliated Hospital of Zhejiang University School of Medicine (Hangzhou, Zhejiang).

### Preparation of the tECM

First, grown pigs were purchased at a local slaughterhouse in Hangzhou, and ten porcine tendons were harvested from their hind legs. The tendons were exposed to three continuous freeze-thaw ( $-80$ – $37^{\circ}\text{C}$ ) cycles. Then, the tissues were immersed in 1% Triton X-100 solution for 24 h, 1% sodium dodecyl sulfate for 3 h and 200 U mL $^{-1}$  DNAase at  $37^{\circ}\text{C}$  for 12 h.

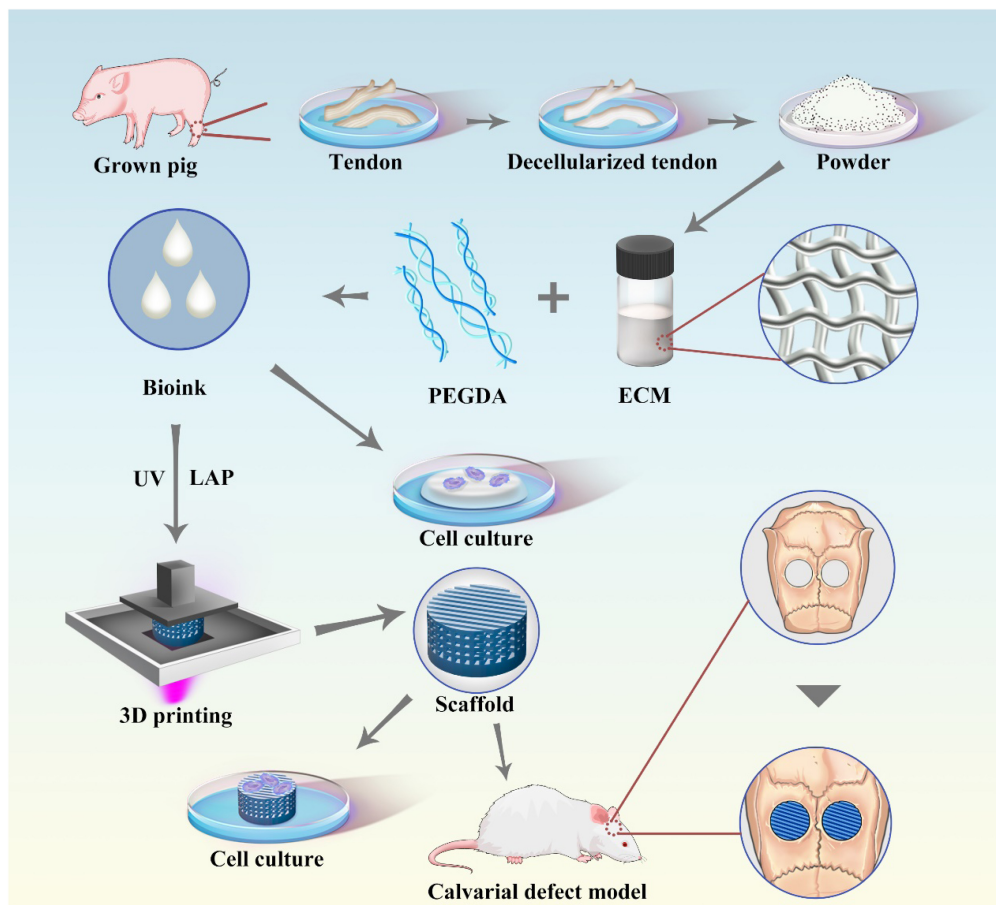
Next, 4', 6-diamidino-2-phenylindole (DAPI), hematoxylin and eosin (H&E), Masson's trichrome staining and a DNA assay were performed to evaluate the decellularized efficiency. Using a Universal Genomic DNA Kit (CW Biotech, Beijing, China) and a microplate spectrophotometer (260 nm, Thermo Fisher Scientific, Waltham, MA, United States), DNA contents were measured. Meanwhile, a hydroxyproline (Hyp) assay kit (Nanjing Keygen Biotech Co., Ltd., Nanjing, China) was used to measure the collagen content. The decellularized tECM was dehydrated, ground for digestion, and adjusted to a neutral pH (7.4).

### Physical Performance Measurement of the Hydrogels

The physical properties of bioinks with different compositions were determined. Exactly 10% (w/v) polyethylene glycol diacrylate (PEGDA) and 0.25% (w/v) lithium acylphosphinate photo-initiator (LAP) were added to the tECM hydrogel obtained previously, as described above, then the PEGDA/tECM pre-gel was crosslinked for 15 s at 375 nm of UV light exposure. Isostatic compression tests of the hydrogels were conducted in a dry state at  $25^{\circ}\text{C}$  using a universal testing system (Instron 5567, United States). The weight swelling ratio,  $Q$ , was calculated using the following equation:

$$Q = \frac{\text{Swelling mass (Ws)}}{\text{dry mass (Wd)}} \times 100\%$$

The coagulation time of several types of hydrogels with different tECM concentrations was tested on the print plane. The 3D microstructure of the lyophilized PEGDA/tECM hydrogels was observed using scanning electron microscopy (SEM).



**SCHEME 1** | Schematic illustration of the polyporous tECM bioink 3D printing scheme, included PEGDA/tECM bioink preparation, tECM hydrogel for cell culture *in vitro* and SLA-based PEGDA/tECM bioprinting that applied to calvarial defect implantation.

An Icon atomic force microscope (Dimension Icon, Bruker, Billerica, MA, United States) was used to observe the PEGDA hydrogel and scaffold.

## Cell Viability and Metabolic Activity Assays of the Hydrogels

For all the experiments on cell viability and metabolic activity assays, C57BL/6 bone marrow-derived MSCs were used, which were purchased from Cyagen Biosciences (MUBMX-90011, Santa Clara, United States). Live-Dead Cell Staining Kit (Thermo Fisher Scientific, Waltham, MA, United States) was used to assay the MSCs viability and CCK-8 kit (Cell Counting Kit-8; Dojindo Laboratories, Kumamoto, Japan) was used to test the metabolic activity of MSCs growing on the hydrogels. Using the same experimental conditions, we also carried out the cytotoxicity test of LAP.

## Spontaneous Biomineralization of the Hydrogels *In vitro*

The hydrogel specimens (pure PEGDA group and PEGDA/tECM group) were produced and merged into the modified simulated

body fluid (m-SBF;  $1.67 \times 10^{-3}$  M  $\text{CaCl}_2$ ,  $9.5 \times 10^{-3}$  M  $\text{Na}_2\text{HPO}_4$ ,  $150 \times 10^{-3}$  M  $\text{NaCl}$ , and  $100 \mu\text{g/mL}^{-1}$  polyaspartic acid) for 2 and 4 weeks. Micro-CT scanning (Siemens Inveon, Eschborn, Germany) was used following a scanning protocol of 80 kV, 500 mA, and 14.97 mm isotropic resolution. Data were obtained and analyzed using Inveon Research Workplace v. 2.2 software (Siemens, Munich, Germany).

## Biofabrication of 3D-pPES Using Dynamic Projection SLA

A digital light processing (DLP) chip (Discovery 4000; Texas Instruments, Dallas, TX, United States), a replaceable UV light source at 375 nm wavelength (OmniCure S2000; EXFO, Quebec City, QC, Canada), and XYZ stages made up our SLA printing system. Pre-designed user-defined computer-aided-design (CAD) files could be read and converted into printable programs by the DLP chip. The light from the UV light source was directed onto the print plane of the bioink through an optically-specific lens (Edmunds Optics, Barrington, NJ, United States). Catalyzed by LAP, the bioink reacted rapidly in

the projected space and quickly solidified into a 3D scaffold at certain physical strength.

## Cell Migration and Cytotoxicity Assay of 3D-pPES

CCK-8 kit (Cell Counting Kit-8; Dojindo Laboratories, Kumamoto, Japan) was used to test the metabolic activity of MSCs growing in the 3D-pPES lixivium. We conducted control studies using static cultures of MSCs in Transwell plates (Corning Inc., Lowell, MA, United States), containing porous polyester membrane inserts (0.33 cm<sup>2</sup>, 0.4 μm pores), to detect cell migration ability.

## Evaluation of the Osteoinductive Activity of the 3D-pPES *In vitro*

The scaffolds, containing 1% (w/v) tECM, were placed in 96-well plates, at  $1 \times 10^5$  cells/well MSCs. Both DMEM supplemented with 10% FBS and 100 μg/mL streptomycin were added. Osteoblastic induction medium (Sigma Aldrich Corp., St. Louis, MO, United States) was added to incubate MSCs. On day 7, real-time quantitative polymerase chain reaction (RT-qPCR) was performed to measure the expression of *ALP*, *Runx2*, *Col1a1*, *OCN*, and *OPN*. *GAPDH* was referred to as a quantitative control for RNA levels. The primer sequences are listed in **Supplementary Table S1**.

## Protein Mass Spectrometry of the tECM

Deformation and reduction of the protein sample were performed for proteomic experiments. Protein concentration was determined using BCA assay. Peptide samples were analyzed using nano-LC-MS/MS. Sequences were mapped based on gene ontology (GO) terms<sup>1</sup> to determine the biological and functional properties of all identified proteins. Meanwhile, we employed hypergeometric tests to perform GO enrichment analysis.

## Repair Assay in the Rat Critical-Sized Calvarial Defect Model

Twelve one-month-old Sprague Dawley rats were purchased and raised individually in cages. After anesthesia, bilateral full-thickness critical-sized calvarial defects (4 mm in diameter) were created. Details of each group of operation are shown in **Supplementary Table S2**. The skulls were collected and embalmed in 4% paraformaldehyde. Repair assay in the rat critical-sized calvarial defect model was implemented using micro-CT scanning. The specimens were cut along the coronal plane for H&E, Masson's trichrome, and Goldner trichrome staining.

## Statistical Analysis

Data are presented as mean ± standard deviation (SD) of at least three experiments with similar results. Experiments were run in triplicate unless stated otherwise. Either one-way ANOVA or Student's *t*-test was applied to assess the differences between the

means. \**P* < 0.05, \*\**P* < 0.01, #*P* < 0.05, and ##*P* < 0.01 were considered statistically significant.

## RESULTS

### Preparation of a Decellularized Tendon With Potential Biological Functions

The microstructure, composition and biological functions of the decellularized tendon were carefully analyzed. After decellularization, the tendon pieces were ground into powder, which was dissolved in acid to form a gel (**Figure 1A**). The tECM mainly presented a filamentous structure observed by SEM (**Figure 1A**). DAPI and H&E staining (**Figure 1B**) showed the absence of the nucleus. Well-organized fibrous structures were observed via HE staining (**Figure 1B**). Masson trichrome staining confirmed excellent collagen retention as well (**Figure 1B**). The DNA content decreased approximately by 97% (**Figures 1C,D**) after the decellularization. There was little change in collagen content before and after decellularization (**Figure 1E**).

### Physical Properties of the Hydrogels

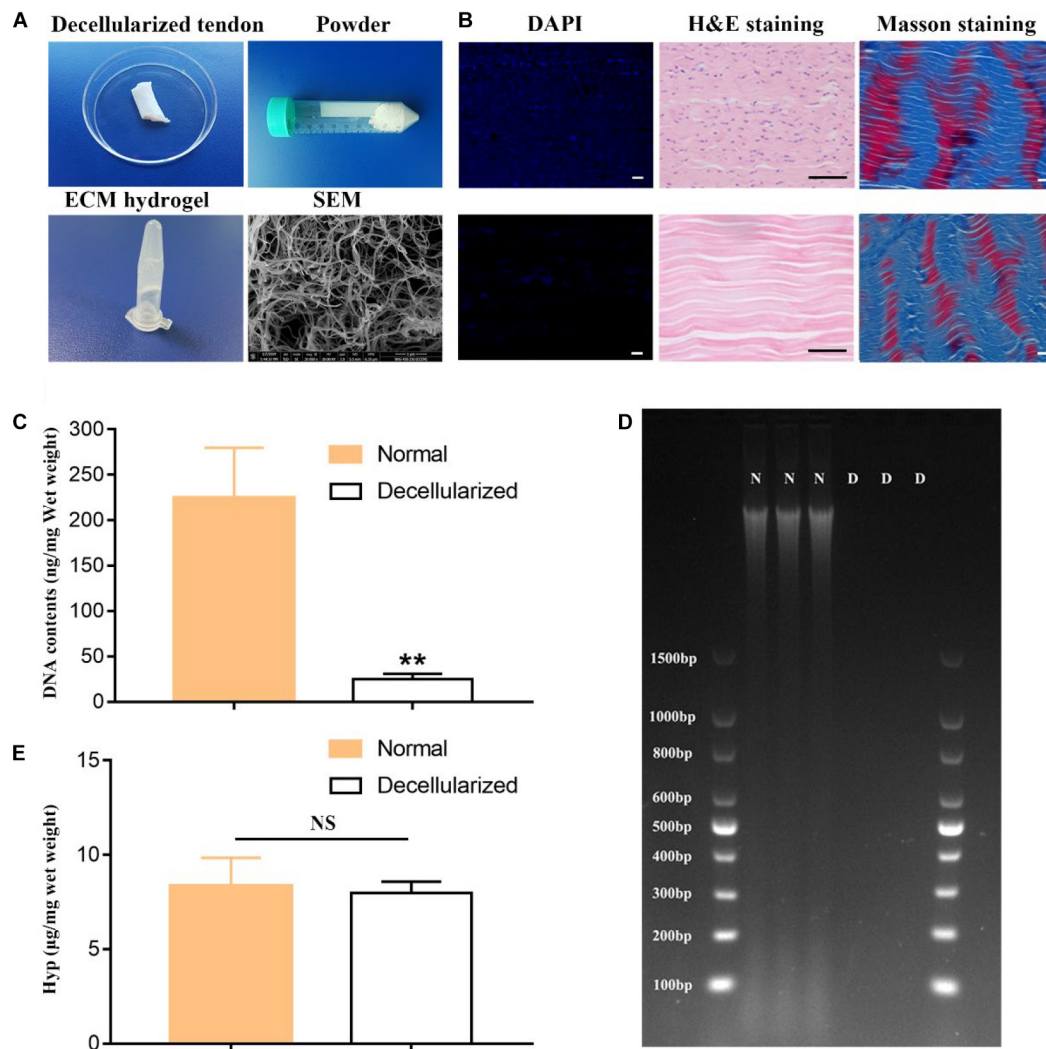
The physical properties of the bioink and hydrogels were further observed from several aspects. Both coagulation time and Young's modulus showed no visible difference when the concentration of the tECM changed from 0 to 1% (**Figures 2A,B**). The representative curve of elastic force generated by the pressure with the displacement of compression deformation is shown in **Supplementary Figure S1**. Rapid swelling and water absorption were observed within the first 100 min, then the swelling ratio verged to extremely slow after 3 h (**Figure 2C**). Moreover, 1% tECM greatly increased the swelling ratio of the PEGDA hydrogels. As the concentration of the tECM increased, an increasing number (10%) of holes appeared, observed via SEM, which contributed to the increase of the hydrogel porosity (**Figures 2D,E**). We assumed that tECM hydrogel greatly improved the physical characteristics of the hydrogel. The data above indicated that the PEGDA/tECM hydrogel was endowed with good hygroscopicity and appropriate mechanical properties.

### PEGDA/tECM Hydrogels Promote Cellular Proliferation and Biomineralization *In vitro*

The cell viability assay showed that the cell proliferation rate was significantly higher on the PEGDA/tECM hydrogel compared to the control hydrogel (**Figures 3A,B**). All groups containing tECM showed a significant difference compared to the control group; this difference was observed from day 1. Moreover, as the concentration of the tECM increased, the proliferative capacity of the cell improved (**Figures 3A,B**). To create a scaffold with a suitable cytocompatibility, we tested the cytotoxicity of LAP (**Figure 3C**). The quantitative analysis proved that high concentration (1% wt) of LAP showed more potent cytotoxicity, while the bioink, containing low concentration (0.25% wt) of LAP, met the UV crosslinking requirements of 3D printing and showed better biocompatibility (**Figure 3C**). In our following

<sup>1</sup><http://geneontology.org/>





**FIGURE 1 |** Fabrication and characterization of the hydrogel derived from decellularized tendon. **(A)** General and SEM Images of tendon decellularization and preparation of hydrogel. **(B)** H&E and DAPI staining confirmed cell removal; Masson trichrome staining confirmed collagen retention. Scale bar = 10  $\mu$ m. Quantitative detection of DNA removal retention **(C,D)** and collagen **(E)**. \*\* $P < 0.01$  compares to normal group.

experiment, the bioink that contained low concentration (0.25% wt) of LAP and high concentration of tECM (1% wt) was used uniformly. Then the PEGDA hydrogels and PEGDA/tECM hydrogels were immersed in m-SBF. Micro-CT results of the PEGDA/tECM group showed poor mineralization and relatively low BV/TV at 4 weeks, while the results at 8 weeks increased greatly ( $P < 0.05$ , **Figures 3D–F**). However, little change was observed in the PEGDA group due to the absence of tECM (**Figures 3D,E**). Hence, PEGDA/tECM hydrogels were better than PEGDA hydrogels at coordinating Ca-Pi deposition.

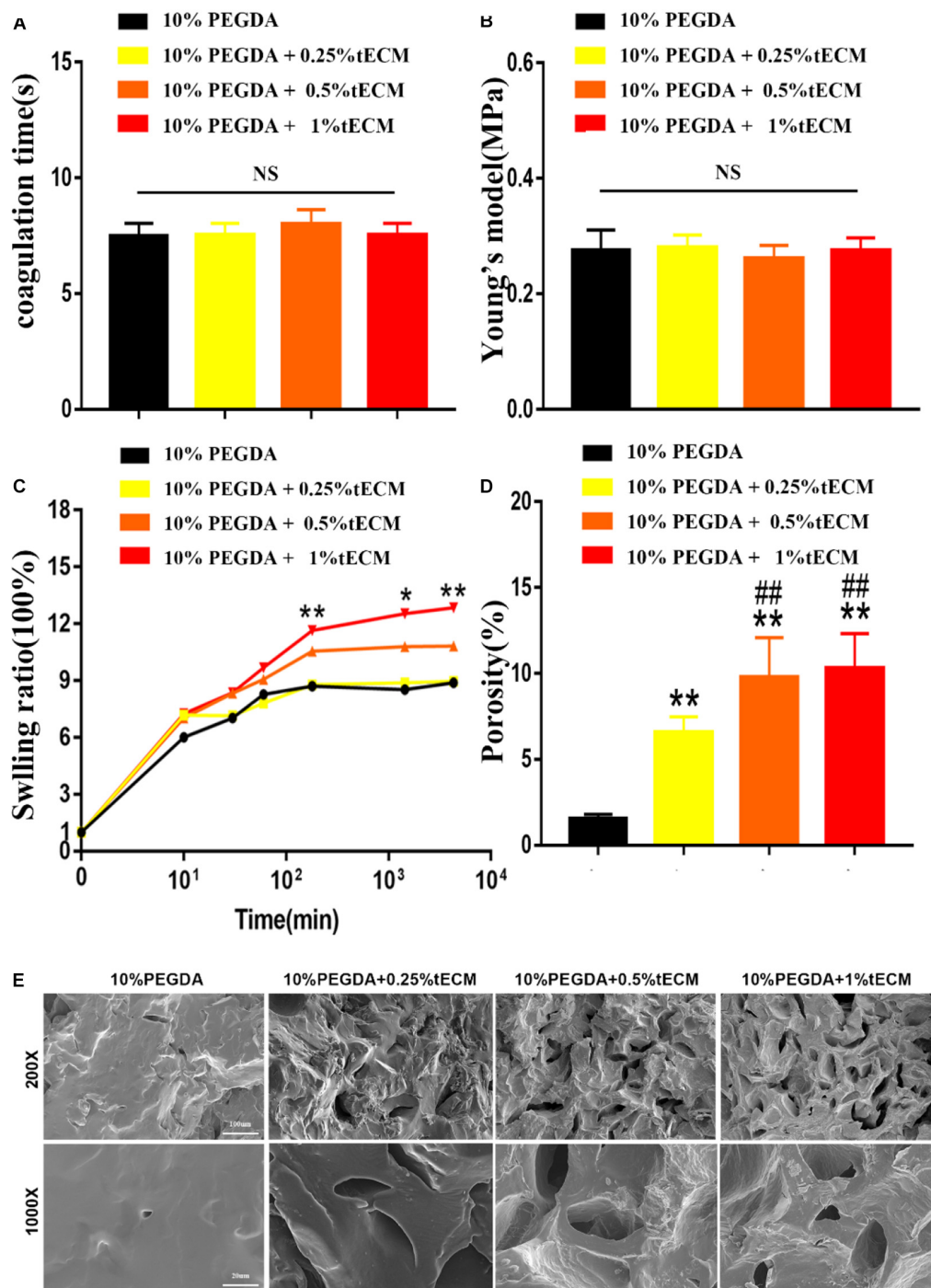
### 3D Printing of Polyporous PEGDA/tECM Bioink

Computer-aided-design files (**Figure 4A**) were used to produce the virtual mask and transmit UV light to the PEGDA/tECM bioink to print polyporous scaffolds. The structural formula of

PEGDA and LAP are shown in **Supplementary Figure S2**. The line chart of the real-time temperature of the bioink in the printing process showed that the temperature was kept below 37°C (**Supplementary Figure S3**). The SLA-based technique successfully yielded a polyporous scaffold with abundant and dense pipelines that could be observed both by the naked eye and under a microscope (**Figure 4A**). These channels were also visible by SEM after lyophilizing the scaffold for 24 h (**Figure 4A**).

### Increased Migration and Osteogenic Differentiation Triggered by the 3D-pPES

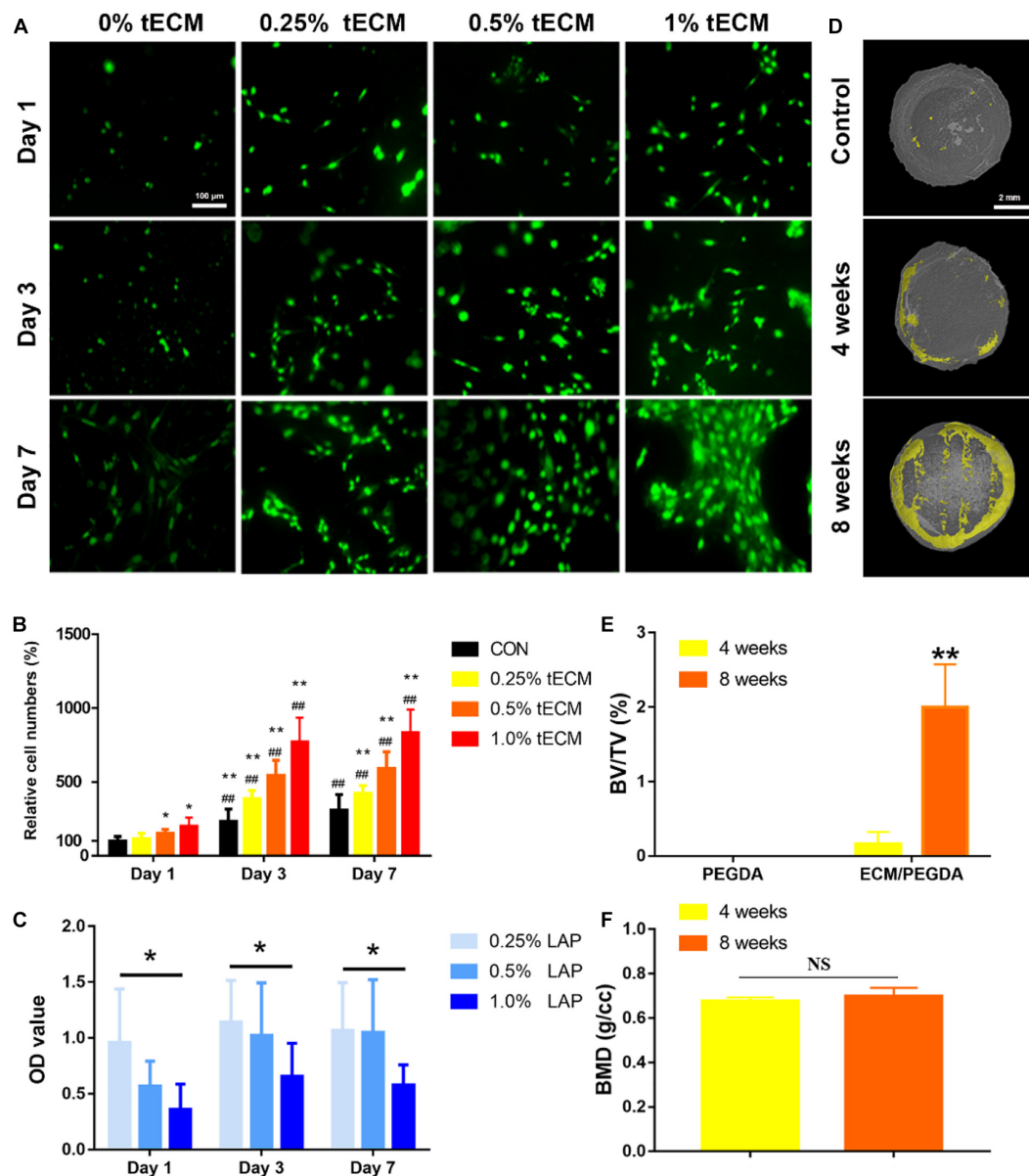
Before the MSCs were reseeded for cell differentiation, non-cytotoxicity of the 3D-pPES was confirmed (**Supplementary Figure S4**). Quantitative analysis of cell numbers showed that cell migration increased 3.6 times in the 3D-pPES group compared to the control group and PEGDA group (**Figures 4B,C**).



**FIGURE 2 |** Physical properties of the bioink and hydrogel: coagulation time (A), Young's modulus (B), swelling ratio (C) and porosity (D,E). The hydrogel was observed via SEM (E). \*\* $P < 0.01$  compares to 10% PEGDA group. ## $P < 0.01$  compares to 10% PEGDA + 0.25% tECM group. \* $P < 0.05$  compares to 10% PEGDA group.

Roughness image showed 3D-pPES group has a rougher surface than PEGDA group (Figures 4D,E). It confirmed that the addition of ECM made the surface of the scaffold rougher. Proteomic analysis of tECM was also performed to divide the identified proteins into three classes (Figures 4F–H). GO cellular

component and molecular function analysis confirmed the successful retention of various ECM structural constituents after decellularization (Figures 4F,G), which provided the bionic environment for cell growth. GO biological process analysis showed that the tECM contained proteins that function as



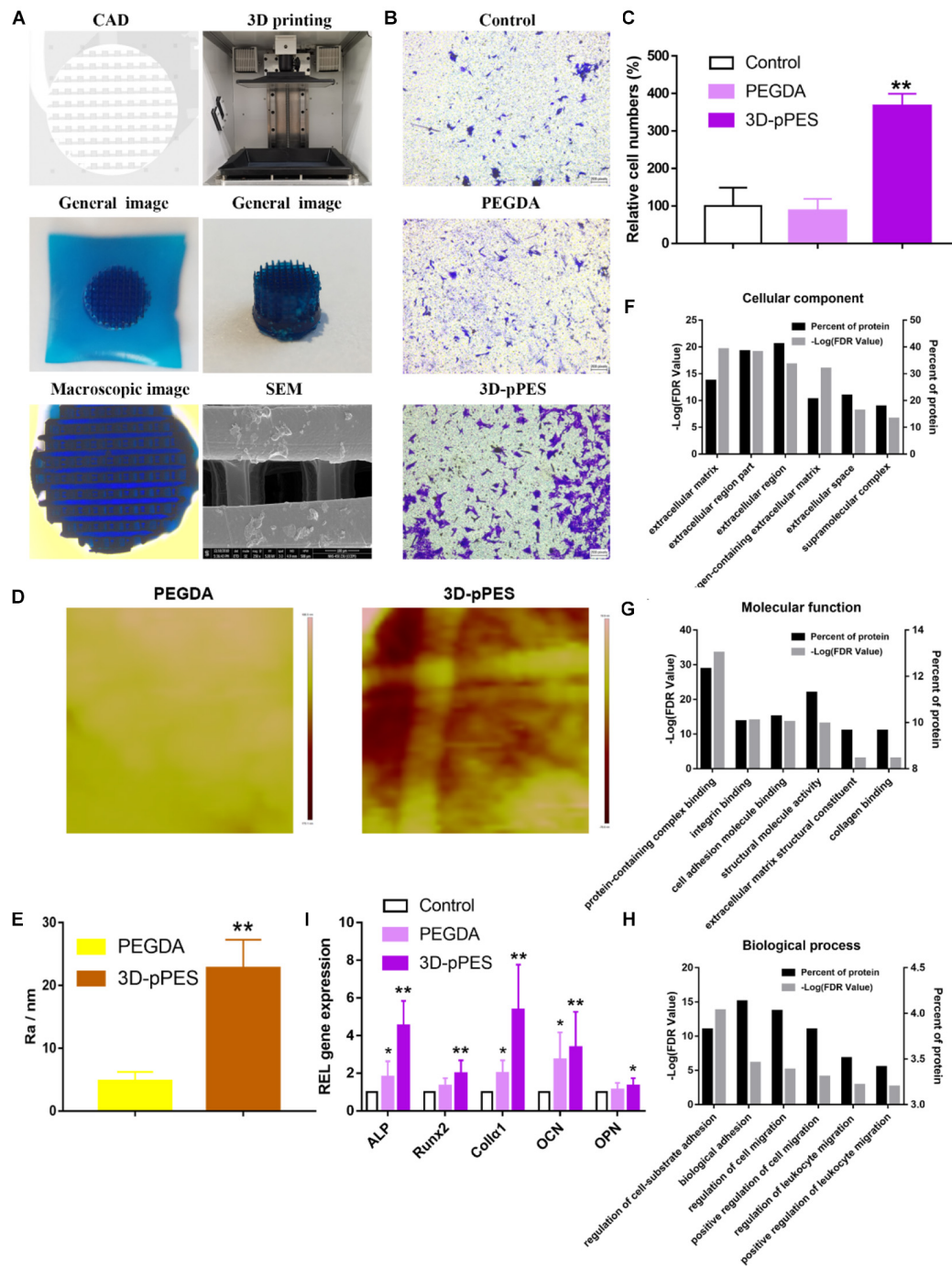
**FIGURE 3 |** Evaluation of increased cellular proliferation and biomineralization *in vitro* by the PEGDA/tECM hydrogel. **(A)** Cell proliferation on the PEGDA, 0.25% tECM/ PEGDA, 0.5% tECM/ PEGDA, 1% tECM/ PEGDA hydrogels. **(B)** Quantitative analysis of Cell proliferation on the PEGDA, 0.25% tECM/ PEGDA, 0.5% tECM/ PEGDA, 1% tECM/ PEGDA scaffolds. **(C)** Quantitative analysis of Cell proliferation at different concentrations of LAP. **(D)** Representative Micro-CT scans images of PEGDA hydrogels for 8 weeks and PEGDA/tECM hydrogels for 4 and 8 weeks. Quantitative analysis of the mineralization presented in using BV/TV **(E)** and BMD **(F)** values. \* $P < 0.05$ , \*\* $P < 0.01$  compares to control group at the same time. ### $P < 0.01$  compares to the same group on day 1. BV/TV: trabecular bone volume fraction, BMD: Bone mineral density. Scale bars are shown in the figure.

regulation of cell adhesion and migration (Figure 4H), which might have contributed to the migration in Figures 4B,C. The result of the proteomic analysis corresponded with that of the Transwell assay. Osteogenic differentiation experiment *in vitro* showed conformably upregulated expression of genes associated with osteogenesis (Runx2, ALP, Col1 $\alpha$ 1, OCN, and OPN) after 14 days of induction (Figure 4I). The rougher 3D-pPES has a more pronounced effect on osteogenic differentiation than PEGDA. These findings indicated that

the 3D-pPES had a positive effect on cell migration and osteogenic differentiation.

## Bone Regeneration Enhancement Triggered by the 3D-pPES

The relative efficacy of PEGDA and PEGDA/tECM hydrogels in promoting new bone formation was evaluated in rats with induced critical-sized calvarial bone defects (Figure 5A) at 4 and



**FIGURE 4 |** 3D printing of polyporous tECM bioink and scaffold promoting migration and osteogenic differentiation. **(A)** CAD file, printing process, and scaffold (stained with methylene blue dye) printed by the 3D printer. The 3D-pES was observed under microscopy and SEM. **(B)** Transwell migration assay of different treatments. Scale bars = 100  $\mu$ m. **(C)** Quantitative analysis of migrating cells. **(D)** Roughness image by AFM within the scope of 1  $\mu$ m<sup>2</sup>. **(E)** Quantitative analysis of roughness value Ra ( $n = 5$ ). **(F–H)** GO classification of tECM proteins. Scale bars are shown in the figure. **(I)** qPCR quantification of the relative mRNA expression of Runx2, ALP, OCN, OPN, and Col1 $\alpha$ 1 in osteoblasts cultured for 14 days on the hydrogels ( $n = 6$ ). \* $P < 0.05$  compares to control group. \*\* $P < 0.01$  compares to control group.

8 weeks after surgery. Observation and analysis of the regenerated bone were successfully conducted, applying micro-CT scanning. Newly-formed bone was observed in all groups. However, the

maximum amount of mineralized bone was measured in the PEGDA/tECM group (Figure 5B), along with the maximum value in BV/TV, Tb.Th, Tb.N, and BMD (Figures 5C–G). In



contrast, Tb.Sp value of the PEGDA/tECM group was the lowest since it was inversely proportional to the BMD value (Figure 5F). Notably, the increased value in BV/TV, Tb.Th, Tb.N, and BMD of the PEGDA group was observed. A similar conclusion could be drawn that a rigid surface of PEGDA without tECM was slightly effective in osteogenic differentiation and biomineralization.

From the H&E stained images (Figure 6A), it was clearly seen that the bone defects of the control groups and PEGDA groups were partly recovered, with the bone defect still unjoined. Little evidence supporting the new bone formation was found in the control group after 4 weeks. Denser new bone and more bone-like tissues near the border of the bone defect were seen in the PEGDA/tECM group. Markedly, the amount of the new bones observed in the PEGDA/tECM group were high compared to the other two groups (Figure 6B). Goldner's trichrome staining (Figure 6C) revealed that in PEGDA and PEGDA/tECM groups, immature woven bone and osteoid were formed in the defects, while the formation of mature lamellar bone, and even new bone marrow by the end of the observation period, were observed in the PEGDA/tECM group. All the evidence above showed a reasonable inference that the 3D-pPES had the best therapeutic effect in bone regeneration in the rat model. Meanwhile, degradation experiments *in vitro* (Supplementary Figure S5) further proved that the optimized degradation performance may further promote the function of bone repair.

## DISCUSSION

Summarizing the experimental results above, our article is the first to report this innovative composite scaffold fabricated by SLA to repair bone defect that had advantage of both the biocompatibility of ECM and mechanical strength of traditional synthetic materials. The innovative scaffold was rich in various bioactive factors, had micropore structure on the surface and interconnectivity of macrostructures, attracted cells, and promoted osteogenic differentiation. Ultimately, experimental results showed that both *in vitro* and *in vivo* experiments resulted in apparent facilitation in terms of osteogenesis.

The ECM has been reported commonly in the field of bone and other parts repair (Pham et al., 2008; Pacelli et al., 2017; Matai et al., 2020). Our team has published relevant literature before. For instance, (Qiu et al., 2020) has fabricated an injectable periosteal ECM hydrogel that dynamically integrates multiple biological functions, promotes angiogenesis and osteogenesis at the defect site. The whole dynamic process contains bone formation, remodeling and repair, which involves cell migration, ECM assembly, osteocyte embedding, and bone resorption (Shiflett et al., 2019). Our experimental results have proved the effect on cell growth, migration, and differentiation. However, the physical properties and spatial structure of this original material need to be improved (Narayanan et al., 2009).

In this study, the physical and mechanical strength of the material was significantly improved by adding PEGDA. As reported previously (Engler et al., 2006), naive MSCs differentiated toward a specific lineage and committed to phenotypes with extreme sensitivity to tissue-level elasticity,

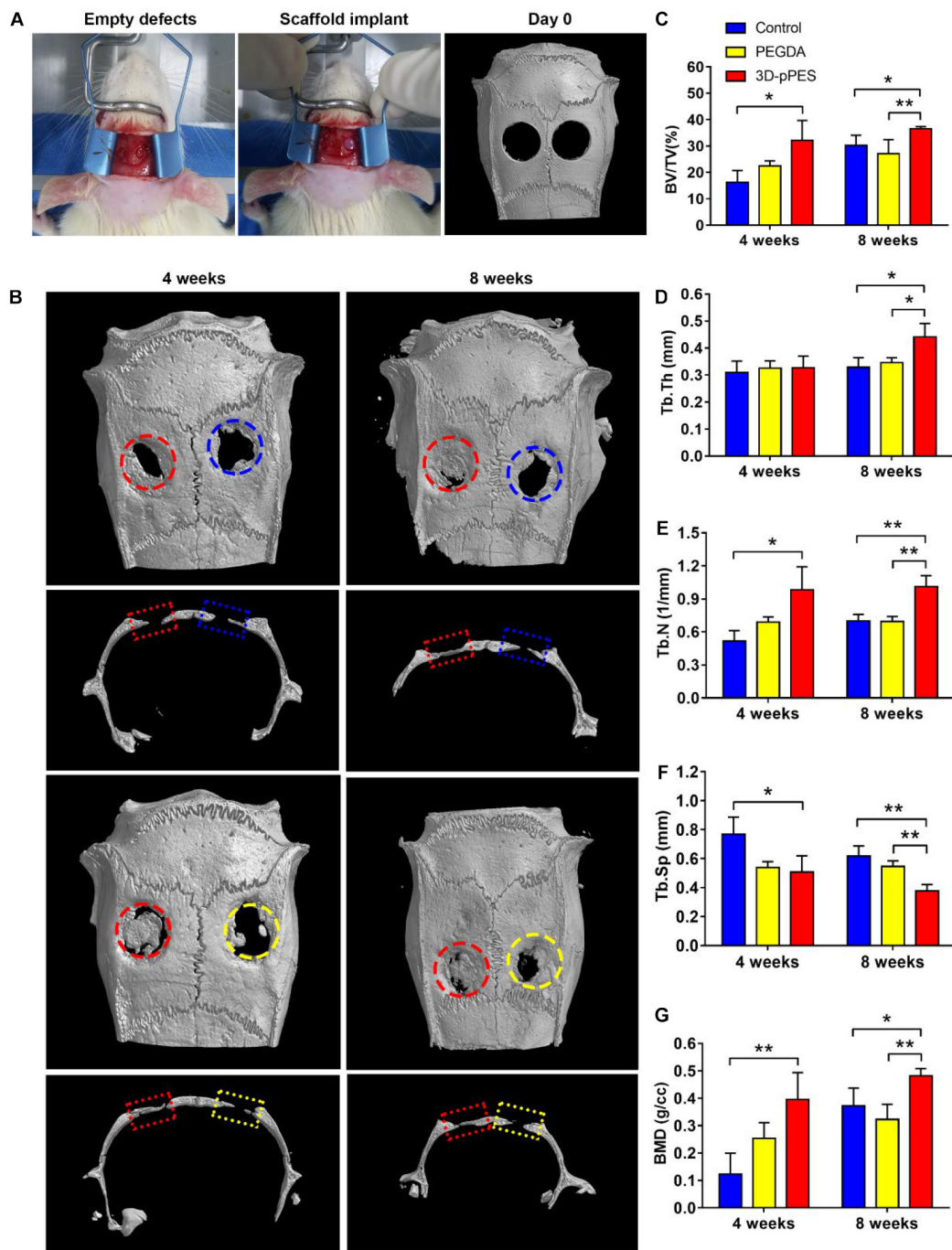
and when tissue-level elasticity of the matrices mimicked a collagenous bone, the outcome was osteogenic (Engler et al., 2006; Caliri et al., 2016). PEGDA (~0.3 MPa) improved the hardness of the material and promoted osteogenic differentiation of MSCs. In fact, in our animal experiments, there were also some weak positive results of osteogenesis in the PEGDA group.

In contrast, the addition of tECM improved the biocompatibility of traditional synthetic materials. Various parameters, including surface properties, mobility and solute diffusion, were affected by the swelling ratio, an essential material parameter in tissue engineering (Stephanopoulos et al., 2013). Microscopically, both the pore size of the polymer and the interaction between the solvent and polymer affect the value of swelling ratio (Du et al., 2008). The addition of tECM greatly improved the swelling degree of the material, and as the concentration of tECM increased, the swelling degree also increased, which was closely related to the hydrophilicity of tECM, as it contains numerous hydrophilic components. The tECM component increased the porosity of the hydrogel. The scaffold with more pores and a larger pore size creates favorable conditions for cell survival, adhesion, and migration (O'Brien et al., 2005, 2007).

To further improve the external form and internal permeability of the PEGDA/tECM hydrogel, SLA technology was applied in the fabrication of this novel scaffold. To date, A challenge that inkjet bioprinting, a traditional printing method, faces is that it is challenging to print vertical 3D structures, while SLA avoids this problem entirely (Wang et al., 2015; Bhattacharjee et al., 2018). Other drawbacks inkjet printing has include shear stress that damage cells, and print nozzle blockage (Wang et al., 2018). In contrast, the SLA printing speed is not affected by plane complexity, while the number of printing layers determines the printing time (Wang et al., 2015). Also, each printing detail may extend the printing time in traditional inkjet bioprinting. Consequently, complex 3D shapes are more likely to be created by SLA without extending the printing time (Wang et al., 2015). The printing time of a customized scaffold is extremely important since urgent surgical treatment is crucial in accidental trauma (Rauch et al., 2019).

Compared with traditional synthetic materials, 3D-pPES has been confirmed to have a rougher surface. Micro-roughness can affect the type of integrins produced by cells, promoting those subunits associated with bone proteins, such as  $\alpha_2$  and  $\beta_1$  (Olivares-Navarrete et al., 2008). Moreover, micro-rough surface evoked accelerated gene expression of the bone matrix molecules osteopontin and osteonectin and up-regulated of bone sialoprotein, collagen III and integrins (Salvi et al., 2015). In fact, both 3D micropore architectural structure and micro-rough surface of 3D-pPES accelerate the MSCs osteogenic differentiation.

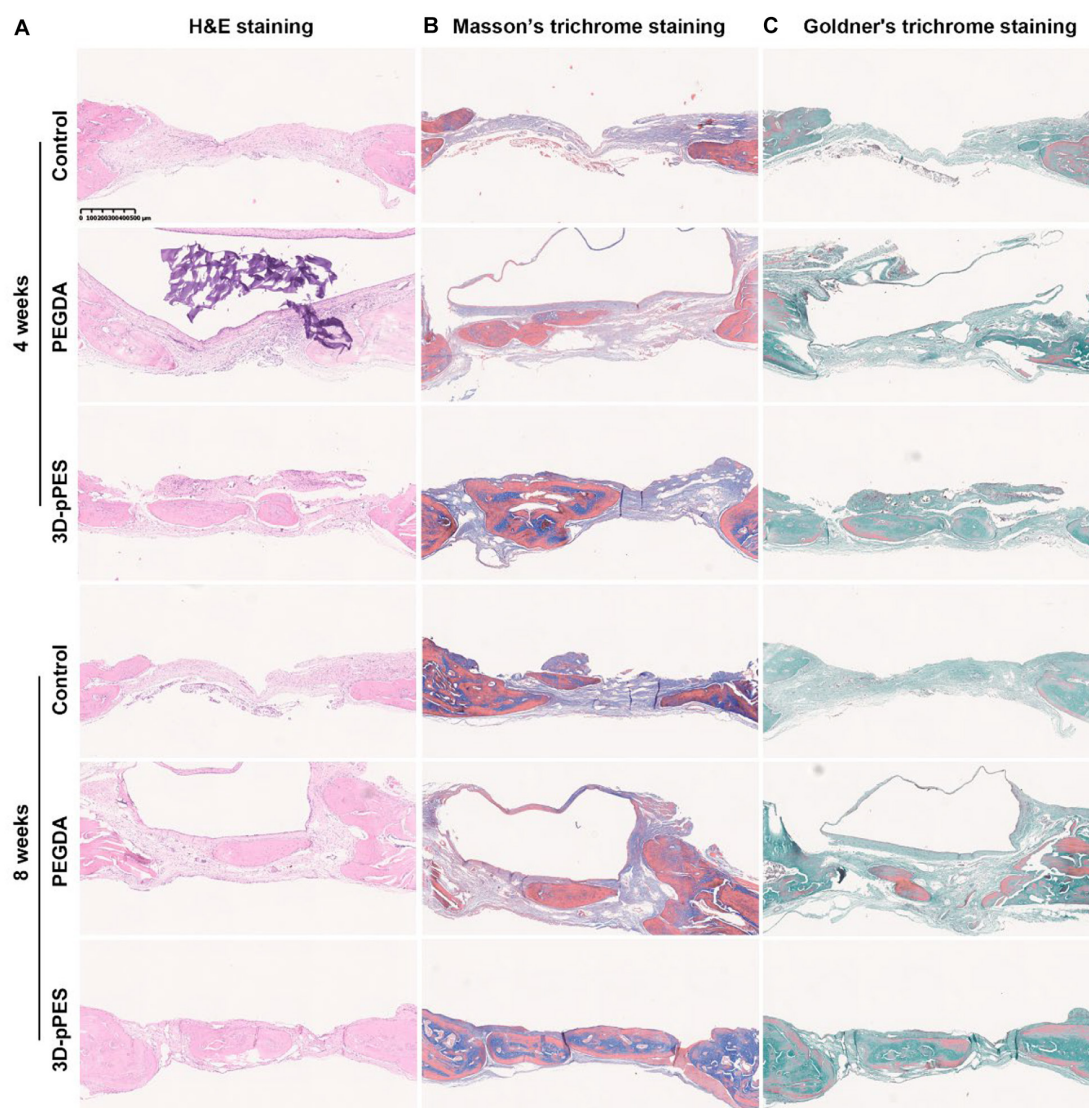
Polyethylene glycol diacrylate/tECM hydrogel showed better capability in coordinating Ca-Pi deposition, which led to high biomineralization. The cell osteogenesis induction experiment *in vitro* also proved that PEGDA/tECM hydrogels significantly and conformably upregulated the expression of genes associated with osteogenesis. There is significant evidence that endochondral (EC) ossification occurs in ECM grafts



**FIGURE 5 |** Bone regeneration in the calvarial bone defect model. **(A)** Surgical procedure and micro-CT scanning of the calvarial defect after surgery on day 0. **(B)** Mineralization of the calvarial defect was evaluated by micro-CT and bone defect healing at 4 and 8 weeks after gel implantation was showed. **(C–G)** Quantitative analysis of **(C)** BV/TV, **(D)** Tb.Th, **(E)** Tb.N, **(F)** Tb.Sp, and **(G)** BMD new bone formation area of the regenerated bone 4 and 8 weeks after implantation. BV, bone volume; TV, total volume; Tb.Th, Trabecular thickness; Tb.N, trabecular number; Tb.Sp, Trabecular separation; BMD, bone mineral density. The blue circle/rectangle represented the original defect of control group, the yellow circle/rectangle represented the original defect of PEGDA group, the red circle/rectangle represented the original defect of 3D-pPES group.

(Dennis et al., 2015) and our results agree with this conclusion. The osteogenic properties of our 3D-pPES were proved by *in vivo* experiments. The repair of critical-sized calvarial bone

defects in the 3D-pPES group showed the highest quality and the most substantial quantity of new mineralized bone formation. Meanwhile, from the results of tissue slices from animal



**FIGURE 6 |** Osteogenesis after hydrogel implantation showed by histological staining. **(A)** H&E staining of the calvarial defects at 4 and 8 weeks after hydrogel implantation. **(B)** Masson's trichrome staining of the calvarial defects at 4 and 8 weeks after hydrogel implantation, new bones were shown in dark blue. **(C)** Goldner's staining of the regenerated bone in calvarial defects, fibroblastic connective tissue was shown in light green, immature woven bone and osteoid tissue were shown in red. mature bone island (lamellar bone) was showed in dark green. Scale bar = 500  $\mu$ m.

experiments, islands of new bone formation were observed in the 3D-pPES group. Moreover, denser new bone and more bone-like tissues near the border of the bone defect were observed in the 3D-pPES group.

Degradation experiment *in vitro* proved that the 3D-pPES have the advantage of faster degradation. First, ECM has multiple biological enzymes to accelerate the degradation of scaffold (Pham et al., 2008). Second, the 3D-pPES greatly increased the surface area of the material. Moreover, it has been reported that scaffold microarchitecture profoundly influences macrophage adhesion, infiltration and differentiation (Kurpinski et al., 2010). Once invading into the scaffold microarchitecture, macrophages secrete reactive oxygen species

(ROS) and hydrolytic enzymes, contributing to oxidative and enzymatic biomaterial degradation, respectively (Wissing et al., 2019). Meanwhile, the scaffolds with larger pore size on the micron scale can promote the oxidative degradation of the synthetic scaffolds (Versaevel et al., 2012). Optimized by multiple factors, 3D-pPES showed superior degradation and repair performance.

We can reasonably speculate that a relatively biomimetic bone repair process has been achieved via intramembranous ossification (Dennis et al., 2015). MSCs proliferate intensively and differentiate into osteoblasts, forming ossification center. The ossification center acts as a primed template for subsequent osteoblast infiltration,



woven bone ossification, and bone remodeling restore healthy lamellar bone architecture (Dennis et al., 2015; Serra-Vinardell et al., 2020). Further work is needed to confirm the inference of these molecular mechanisms.

## CONCLUSION

In this study, we revealed that PEGDA/tECM hydrogel is a greatly enriched small aperture with a traditional synthetic material surface. It overcomes the drawbacks of insufficient mechanical strength of the traditional ECM and promotes cell growth. Using SLA, our 3D-pPES promote cell migration, initiate MSCs differentiation toward osteogenesis. In conclusion, promising repair efficacy of the 3D-pPES is confirmed in a bone defect model.

## DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding authors.

## ETHICS STATEMENT

The animal study was reviewed and approved by Sir Run Shaw Affiliated Hospital of Zhejiang University School of Medicine.

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## AUTHOR CONTRIBUTIONS

All authors listed have made a substantial, direct and intellectual contribution to the work, and approved it for publication.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# The Role of Paracrine Regulation of Mesenchymal Stem Cells in the Crosstalk With Macrophages in Musculoskeletal Diseases: A Systematic Review

Hongtao Xu<sup>1†</sup>, Chien-Wei Lee<sup>1,2,3†</sup>, Yu-Fan Wang<sup>1</sup>, Shuting Huang<sup>2</sup>, Lih-Ying Shin<sup>1</sup>, Yu-Hsuan Wang<sup>1</sup>, Zihao Wan<sup>1</sup>, Xiaobo Zhu<sup>1</sup>, Patrick Shu Hang Yung<sup>1</sup> and Oscar Kuang-Sheng Lee<sup>1,2,4,5\*</sup>

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Institute of Tissue Regeneration  
Engineering (ITREN), South Korea

### \*Correspondence:

Oscar Kuang-Sheng Lee  
oscarlee9203@gmail.com

<sup>†</sup>These authors have contributed  
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<sup>1</sup> Department of Orthopaedics and Traumatology, Faculty of Medicine, Prince of Wales Hospital, The Chinese University of Hong Kong, Hong Kong, China, <sup>2</sup> Institute for Tissue Engineering and Regenerative Medicine, The Chinese University of Hong Kong, Hong Kong, China, <sup>3</sup> Developmental and Regenerative Biology TRP, Faculty of Medicine, School of Biomedical Sciences, The Chinese University of Hong Kong, Hong Kong, China, <sup>4</sup> Faculty of Medicine, Li Ka Shing Institute of Health Sciences, Prince of Wales Hospital, The Chinese University of Hong Kong, Hong Kong, China, <sup>5</sup> Department of Orthopaedics, China Medical University Hospital, Taichung, Taiwan

The phenotypic change of macrophages (M $\phi$ s) plays a crucial role in the musculoskeletal homeostasis and repair process. Although mesenchymal stem cells (MSCs) have been shown as a novel approach in tissue regeneration, the therapeutic potential of MSCs mediated by the interaction between MSC-derived paracrine mediators and M $\phi$ s remains elusive. This review focused on the elucidation of paracrine crosstalk between MSCs and M $\phi$ s during musculoskeletal diseases and injury. The search method was based on the PRISMA (Preferred Reporting Items for Systematic Reviews and Meta-Analyses) and Cochrane Guidelines. The search strategies included MeSH terms and other related terms of MSC-derived mediators and M $\phi$ s. Ten studies formed the basis of this review. The current finding suggested that MSC administration promoted proliferation and activation of CD163<sup>+</sup> or CD206<sup>+</sup> M2 M $\phi$ s in parallel with reduction of proinflammatory cytokines and increase in anti-inflammatory cytokines. During such period, M $\phi$ s also induced MSCs into a motile and active phenotype via the influence of proinflammatory cytokines. Such crosstalk between M $\phi$ s and MSCs further strengthens the effect of paracrine mediators from MSCs to regulate M $\phi$ s phenotypic alteration. In conclusion, MSCs in musculoskeletal system, mediated by the interaction between MSC paracrine and M $\phi$ s, have therapeutic potential in musculoskeletal diseases.

**Keywords:** mesenchymal stem cells (MeSH ID D059630), extracellular vesicles (EVs), exosomes, macrophages, musculoskeletal

## INTRODUCTION

The inflammatory processes in response to musculoskeletal diseases and injury, such as bone fractures, osteoarthritis (OA), osteoporosis, tendon injuries, and muscle injuries, are essential for the correct restoration of structure and function to the affected area (Bosurgi et al., 2011; Mianehsaz et al., 2019; Pajarinen et al., 2019; Wang et al., 2019; Yang and Yang, 2019). However,

the dysregulation of inflammatory reactions can aggravate the tissue healing results (Saldana et al., 2019).

Macrophages (M $\phi$ s) are the critical regulators involved in initiation, propagation, and resolution of inflammatory response throughout the tissue regenerative process. M $\phi$ s have a broad spectrum of adaptive phenotypes and functional transitions that might exacerbate and resolve inflammation during tissue repair process (Saldana et al., 2019). In 2008, Mosser and Edwards analyzed the phenotypic changes of M $\phi$ s and robustly classified M $\phi$ s into two types: M1 and M2 (Mosser and Edwards, 2008). Proinflammatory M $\phi$ s are identified as the classic M1 M $\phi$ s, which are involved in the early stages of tissue repair, whereas the anti-inflammatory M $\phi$ s are identified as M2 M $\phi$ s, which dominated later stages of tissue repair (Murray et al., 2014; Spiller and Koh, 2017). Upon injury, the early presence of M1 M $\phi$ s initiates tissue repair, but the persistent of M1 activity can deteriorate the repair process (Krzyszczuk et al., 2018). On the other hand, the early presence of M2 can prevent cellular and vascular infiltration that impairs tissue development through ectopic secretion of fibrotic chemokines and cytokine (Stahl et al., 2013; Bility et al., 2014; Moore et al., 2015). Moreover, several different M2 subtypes have been identified, including M2a, M2b, M2c, and M2d (Stein et al., 1992; Donnelly et al., 1999; Anderson and Mosser, 2002). The undisciplined regulation of M $\phi$  phenotypic change impairs tissue repair, and each of the subtypes might have specific functions; therefore, further investigation is needed to identify the explicit role of M2 subtypes (O'Brien et al., 2019).

In the 1980's, Arnold Caplan and his colleagues published an isolation method of fibroblast-like stromal cells from bone marrow and first identified them as mesenchymal stem cells (MSCs) because of their multilineage differentiation potential (Caplan, 1991). A rapid expansion in the field of MSC-based therapy in immunomodulation and regenerative medicine has been acknowledged (Kingery et al., 2019). It has been reported that MSCs can switch M1 M $\phi$ s or resting M $\phi$ s into M2 M $\phi$ s (Kim and Hematti, 2009; Nemeth et al., 2009; Melief et al., 2013b). Previous studies showed that prostaglandin E<sub>2</sub> (PGE<sub>2</sub>), interleukin-4 (IL-4), IL-6, and IL-10 released from MSCs can induce the M $\phi$  polarization toward M2 to accelerate tissue regeneration (da Costa Goncalves and Paz, 2019). On the other hand, such immunomodulatory ability of MSCs is adjusted by inflammatory factors released by macrophages (Waterman et al., 2010; Carrero et al., 2012). After the stimulation, MSCs would secrete anti-inflammatory factors, such as transforming growth factor  $\beta$  (TGF- $\beta$ ) and CCL-18, to further suppress activation of lymphocytes and inhibit major histocompatibility complex (MHC) class II and CD86 in lipopolysaccharide (LPS)-stimulated M $\phi$ s (Melief et al., 2013b; Cho et al., 2014). This suggested that the interplay between MSCs and M $\phi$ s is in control of inflammation, and their crosstalk may be recommended to advocate tissue healing or repair.

Besides versatile soluble proteins, MSC-derived extracellular vesicles (MSC-EVs) have raised worldwide attention in regenerative medicine because of their immunomodulation ability (Raposo and Stoorvogel, 2013). EVs represent a heterogeneous group of cell-derived membranous vesicles,

such as microvesicles (MVs) and exosomes, which were first described in the 1970's (Vakhshiteh et al., 2019). MVs, ranging from 40 to 2,000 nm in diameter (Bruno et al., 2012; Kim et al., 2012), and exosomes, ranging from 30 to 150 nm in diameter (Li et al., 2013; Zhao et al., 2015), are critical mediators for intercellular communication via the delivery of the embedded RNAs, DNAs, and cytosolic proteins. Moreover, boosting the therapeutic potential of MSC-EVs by changing the intrinsic bioactive factors or modifying membrane by bioengineering approach is possible (Lu and Huang, 2020). Refining the culture condition of MSCs could significantly increase the production yield and improve the efficacy of MSC-EVs (Luan et al., 2017; Willis et al., 2017; Bagno et al., 2018; Cha et al., 2018; Ferguson et al., 2018). Also, MSC-derived EVs could serve as a promising drug delivery vector, owing to their high biocompatibility, high efficacy of delivery, and low immunogenicity (Clayton et al., 2003; Ridder et al., 2014; Zomer et al., 2015). EVs derived from different origins have preferential targeting cells due to their distinct membrane composition gained from their parental cells that impart differential effect on body systems (Luan et al., 2017; Ferguson et al., 2018), which could be addressed by exploiting the nature of EVs as natural carriers of miRNA or other molecules by considering them as drug delivery vehicles (Cheng et al., 2017). In addition, MSC-derived EVs in conjunction with other materials might provide substantial advances in both immunomodulation and tissue regeneration (Cosenza et al., 2018). And it has been reported that a variety of material's environment could affect cell's downstream response by cell-material interactions (Darnell et al., 2018). Taken together, bioengineered MSC-derived EVs are novel adjustable biomaterials for tissue regeneration.

The therapeutic effect of MSCs showed a clinical benefit in children suffering from osteogenesis imperfecta (Otsuru et al., 2012) and preclinical benefit in bone fracture (Li Y. et al., 2019), OA (Cosenza et al., 2017), tendon injury (Chamberlain et al., 2019), pulmonary hypertension (Lee et al., 2012), cancer (Silva et al., 2013), and infectious diseases (Cheng and Schorey, 2013). The mechanisms of these mentioned therapeutic potentials of MSCs need elucidation. This review aimed to expound the role of MSC paracrine, especially MSC-derived EVs, in the crosstalk with M $\phi$ s in musculoskeletal diseases. Moreover, a systemic understanding of MSC-EVs properties and activities will provide a solid foundation to boost MSC-EVs for regenerative medicine and will significantly facilitate the translation value of MSC-based therapy (Zhang et al., 2020).

## METHODS

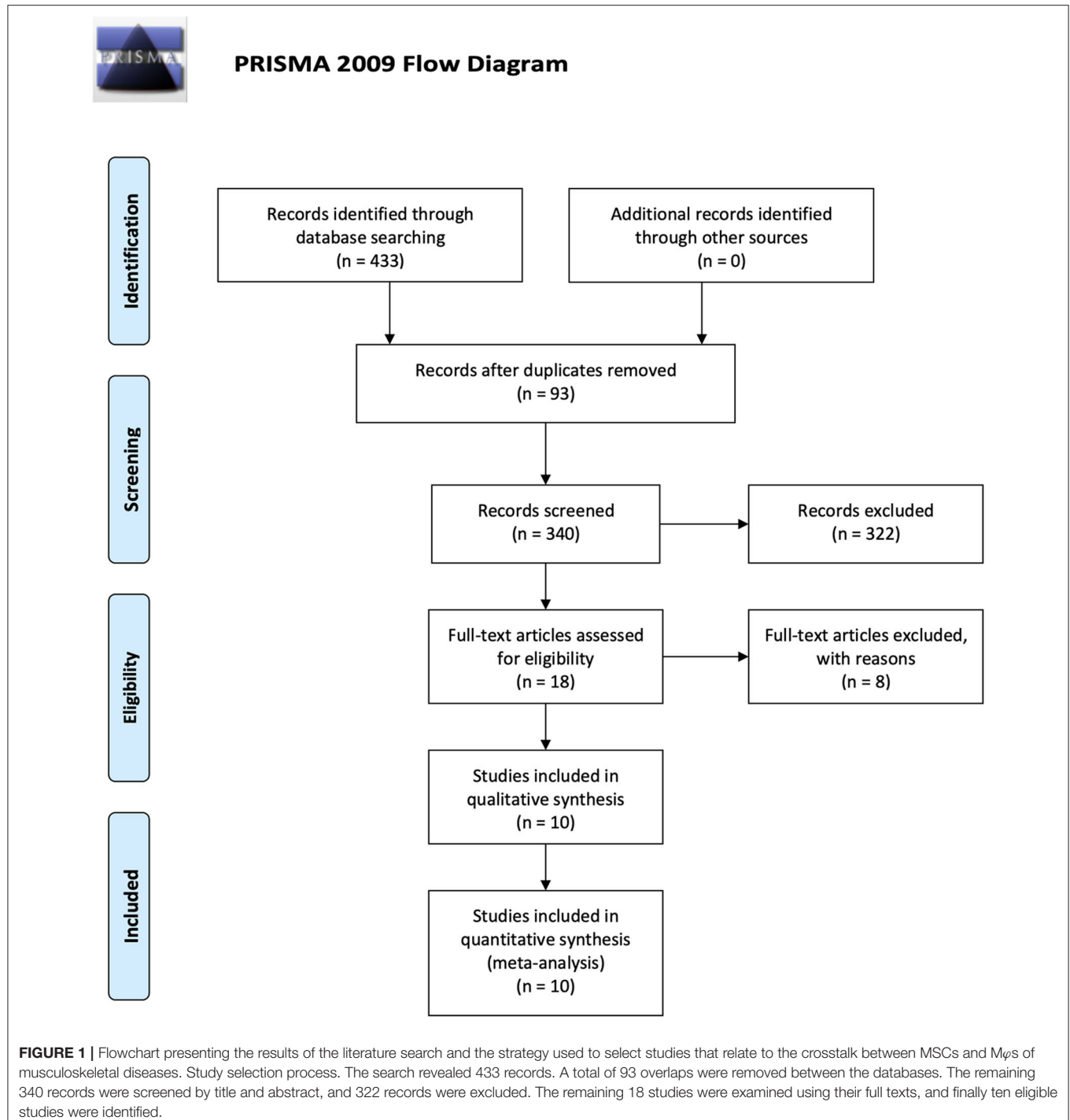
The Preferred Reporting Items for Systematic Reviews and Meta-Analyses (PRISMA) statement was used for this article. Meanwhile, the Cochrane handbook was selected as guidelines for the study protocol (Moher et al., 2009).

## Search Strategy and Study Selection

Two different investigators (H.-T.X., L.-Y.S.) conducted the customized up-to-date literature search. PubMed database and EMBASE (Excerpta Medica Database) database were selected in this study.

The title and abstract field were selected to search for MeSH terms and other related terms, which pertained to “Mesenchymal Stem Cells,” “Macrophages,” and “Paracrine Regulation,” such as “Exosomes,” “Extracellular Vesicles” and “Culture Media, Conditioned.” The details of selected search terms and searching procedures that were used in the individual database are available in Appendices 1–3 in **Supplementary Material**. Additional studies were also located by searching papers referenced in

listed articles. Those studies identified by the search outcomes were combined, and duplicates were excluded. Then, the screening procedure of titles and abstracts was performed before elaboration on the selected full-text articles. Two investigators (H.-T.X., L.-Y.S.) screened the titles and abstracts of those identified studies individually. In cases of disagreement between the two authors, a consensus was reached by discussion with a third author (C.-W.L.). After that, the full text of screened





articles were examined. Corresponding authors of the reviewed articles could be contacted with essential needs to obtain those missing data.

The search yielded 433 studies across all databases (239 studies across PubMed database and 194 studies across EMBASE database). A total of 93 duplicates were removed. According to the inclusion and exclusion criteria, all these studies were screened and reviewed by titles and abstracts first; 322 studies were excluded because they were review articles, case reports, case series, letters, chapters, or studies published more than 10 years ago. Of the 18 remaining studies, which were applied by a filter to include musculoskeletal-related studies, eight studies were excluded via reviewing full text. The 10 remaining studies underwent secondary full-text review and were confirmed as fitting the inclusion criteria. The flowchart of the selected studies selection process, which was based on the PRISMA 2009 Flow Diagram (Moher et al., 2009), is shown in **Figure 1**, and the details of inclusion and exclusion criteria are shown in **Table 1**.

Data Extraction

All relevant data were extracted by H.-T.X. and X.-B.Z.: author information, published journal, year of publication, sample source, target disease, study type (*in vivo*, *in vitro*, or both), cell management, and measurement instrument. The details of results extraction consisted of variable/control group descriptions, laboratory effects, proposed mechanisms, article conclusions, and research implications. After that, the selected articles were classified according to the type of target disease.

Methodological Quality Assessment

The methodological assessment should be used as an essential procedure, which could exclude articles with a large degree of bias or with a higher degree of potential bias, has been highlighted to readers (McElvany et al., 2015). Identified studies fitting all criteria were reviewed, and all included data were extracted and analyzed based on study heterogeneity and methodological quality. Because of the natural heterogeneity of measurement across studies, a meta-analysis could not be performed. Two independent authors (H.-T.X. and Z.-H.W.) separately assessed and graded the methodological quality of all selected studies. Disagreements between the two independent researchers were identified and resolved by discussion with a third reviewer (C.-W.L.). The selected studies were assessed with a quality scoring system raised by Wells and Julia (2009) (Appendix 4 in **Supplementary Material**). The quality assessment system was based on the following eight questions: Was the study hypothesis/aim/objective clearly described? Were the animal models for the study well-described? Were the methods well-described? Were the data collection time point clearly defined? Were the main outcome measures clearly defined? Were the experiment group well-compared with the control group? Were the results well-described? Were the articles discussed the limitation? For each question, 1 point was allocated for “yes,” and 0 point was allocated for “no.” The number of “yes” answers was counted for each selected study to give a total score out of 8. A study’s rating was considered as excellent with a score ranging from 6 to 8, good with a score ranging from 4 to 6, poor with

TABLE 1 | Inclusion and exclusion criteria.

Criteria	
Inclusion criteria	1. Full-text available;
	2. Written in English;
	3. Articles published in the last 10 years;
	4. Articles containing original data;
	5. Studies must be related to “mesenchymal stem cells” and “macrophages”;
	6. Musculoskeletal related studies.
Exclusion criteria	1. No control group;
	2. Sampling method described inconsistent;
	3. Case reports, case series and review articles, letter, chapter;
	4. Not available in the English language.

a score ranging from 2 to 4, and bad with a score ranging from 0 to 2.

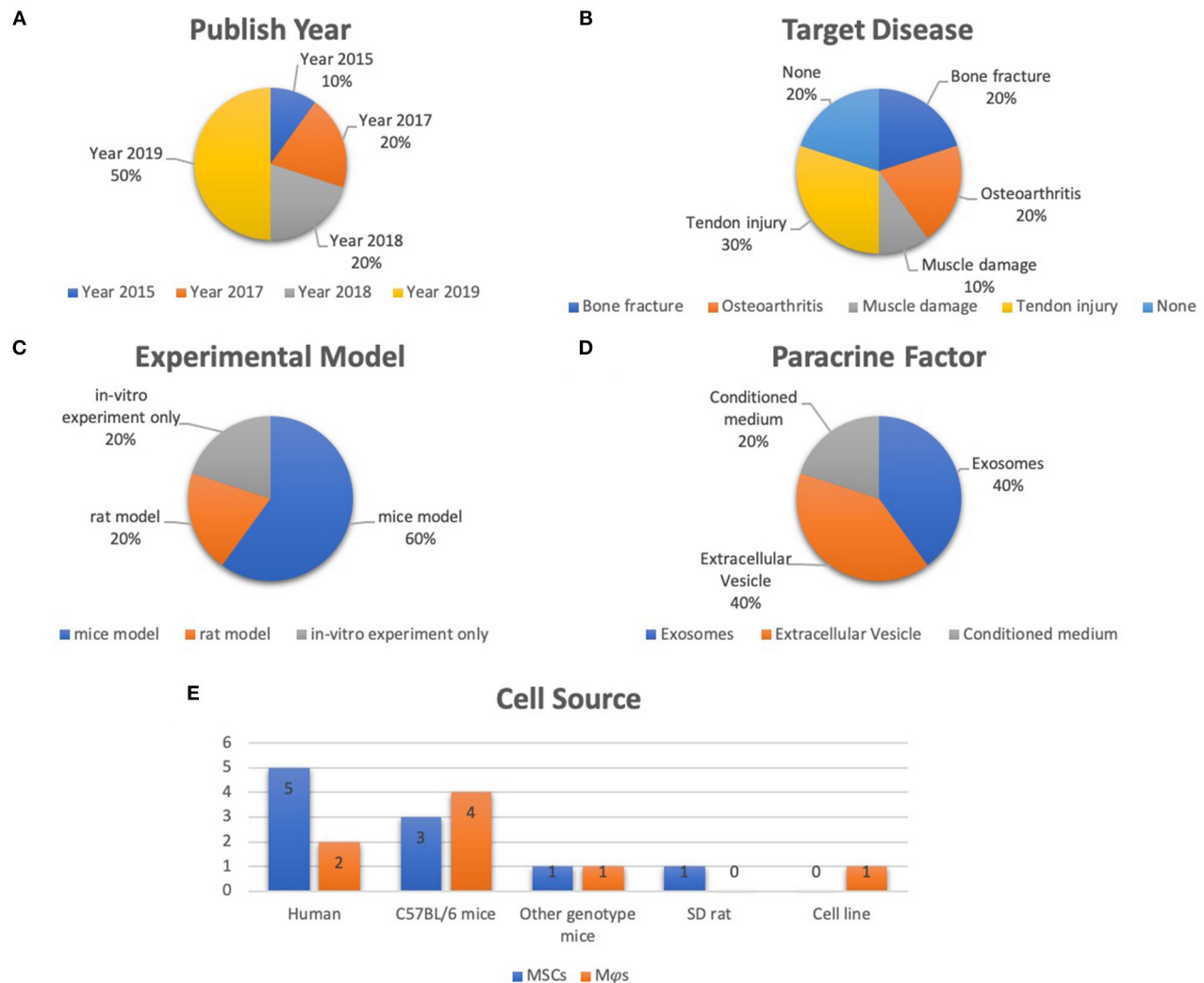
RESULTS

Study Methodology Quality Assessment

The scoring system was used to calculate all ten selected studies in Appendix 5 in **Supplementary Material**. The mean score is 6.6 (range, 5–8), including eight studies exceeding 6 points (Cosenza et al., 2017; Lo Sicco et al., 2017; Hyvarinen et al., 2018; Zhang et al., 2018; Chamberlain et al., 2019; Li Y. et al., 2019; Shi et al., 2019; Shen et al., 2020).

Study Characteristics

Of the ten selected studies, one study was published in 2015 (Chang et al., 2015), two were published in 2017 (Cosenza et al., 2017; Lo Sicco et al., 2017), two studies were published in 2018 (Hyvarinen et al., 2018; Zhang et al., 2018), four studies were published in 2019 (Chamberlain et al., 2019; Li Y. et al., 2019; Pacienza et al., 2019; Shi et al., 2019), and the final one was published in 2020 (Shen et al., 2020) (**Figure 2A**). All selected studies were related to the musculoskeletal system. For details of target diseases, two studies were bone fracture-related studies (Chang et al., 2015; Li Y. et al., 2019), two studies were OA-related studies (Cosenza et al., 2017; Zhang et al., 2018), one study was a muscle damage-related study (Lo Sicco et al., 2017), three studies were tendon injury-related studies (Chamberlain et al., 2019; Shi et al., 2019; Shen et al., 2020), and the other two studies were included without targeting disease (Hyvarinen et al., 2018; Pacienza et al., 2019) (**Figure 2B**). Six studies used a mice model (Cosenza et al., 2017; Lo Sicco et al., 2017; Chamberlain et al., 2019; Li Y. et al., 2019; Pacienza et al., 2019; Shen et al., 2020), two studies used a rat model (Zhang et al., 2018; Shi et al., 2019), and two studies performed *in vitro* experiments only (Chang et al., 2015; Hyvarinen et al., 2018) (**Figure 2C**). Three paracrine factors were extracted from the results: MSC-derived exosomes (Cosenza et al., 2017; Zhang et al., 2018; Pacienza et al., 2019; Shen et al., 2020), MSC-derived EVs (Lo Sicco et al., 2017; Hyvarinen et al., 2018; Chamberlain et al.,



**FIGURE 2 |** Representative graph of the included studies presented in the articles reviewed. **(A)** Publication year of included studies (range from 2015 to 2020). **(B)** Target disease of included studies (include bone fracture, OA, muscle damage, tendon injury). **(C)** Experimental model of included studies (including mice model, rat model, *in vitro* experiment only). **(D)** Paracrine factors (including extracellular vesicle, exosomes, CM). **(E)** Cell source of included studies (including human, C57BL/6 mice, other genotype mice, SD rat, cell line).

2019; Shi et al., 2019), and MSC-derived conditioned medium (CM) (Chang et al., 2015; Li Y. et al., 2019) (Figure 2D). Five MSC and Mφs cell sources for EVs isolation were described in these selected studies: human (Lo Sicco et al., 2017; Hyvarinen et al., 2018; Zhang et al., 2018; Chamberlain et al., 2019; Pacienza et al., 2019), cell lines (Pacienza et al., 2019), Sprague–Dawley rats (Shi et al., 2019), C57BL/6 mice (Chang et al., 2015; Cosenza et al., 2017; Lo Sicco et al., 2017; Li Y. et al., 2019), and transgenic mice [scleraxis–green fluorescent protein (GFP) tendon reporter mice, nuclear factor (NF-κB)–GFP–luciferase transgenic reporter mice, and wild-type FVB/NJ (FVB) mice] (Shen et al., 2020). The detailed data are graphically shown in Figure 2E.

## Target Diseases and Experimental Animal Models

In the bone fracture–related studies (Chang et al., 2015; Li Y. et al., 2019), Li Y. et al. (2019) established a cylindrical bone defect mice model by using an electric drill to make a defect with 1-mm diameter and 1-mm depth at the bone callus of the femur. Chang et al. (2015) studied only *in vitro* experiments, and they cocultured bone marrow–derived mesenchymal stem cells (BMSCs) with Mφs. Two studies selected OA as the target disease (Cosenza et al., 2017; Zhang et al., 2018). Cosenza et al. (2017) established a collagenase-induced arthritis mouse model by intra-articularly injecting 1U type VII collagenase (in 5 μL saline) into the knee joint of 10-week-old C57BL/6 mice at days

0 and 2. Zhang et al. (2018) generated an osteochondral defect model in an 8-week-old Sprague–Dawley rat. Osteochondral defects, with 1.5-mm diameter and 1-mm depth, were generated on the trochlear grooves of the distal femurs by a drill bit. For muscle injury study, cardiotoxin-induced muscle injury was applied in 8-week-old male C57BL/6 mice by intramuscular administration of cardiotoxin into the Tibialis Anterior (TA) muscle (Lo Sicco et al., 2017). Three tendon injury models were used in the selected studies (Chamberlain et al., 2019; Shi et al., 2019; Shen et al., 2020). Chamberlain et al. (2019) established a surgically transected Achilles tendon mouse model. After superficial digital flexor tendon was removed, the Achilles tendon was completely transected at the midpoint. Then tendon was sutured together by using 5-0 Vicryl suture. In Shen et al. (2020) study, Achilles tendon was two-thirds transected at the midpoint part between calcaneal insertion and the musculotendinous junction and then was sutured with a two-strand modified Kessler technique. Shi et al. (2019) applied the Sprague–Dawley rat patellar tendon defect model. Briefly, the central one-third of the patellar tendon was removed from the distal apex of the patellar to the insertion of the tibial tuberosity to achieve a tendon structural defect condition. In addition, Pacienza et al. (2019) established a mouse endotoxemia model by injecting LPS via tail vein, which was used to study the crosstalk between MSC-derived exosomes and M $\phi$ s. Hyvarinen et al. (2018) only cocultured BMSCs and BMSC-derived EVs with M $\phi$ s without performing animal study. The target diseases and experimental models are represented in **Table 2** in detail.

## Cell Sources and Cell Management

Five of 10 studies used human source cells (Lo Sicco et al., 2017; Hyvarinen et al., 2018; Zhang et al., 2018; Chamberlain et al., 2019; Pacienza et al., 2019). Human adipose-derived mesenchymal stem cells (ADSCs) were used for *in vitro* experiments in Lo Sicco's research; the EVs derived from ADSCs were cocultured with mouse M $\phi$ s and intralesionally injected into the mouse muscle injury model (Lo Sicco et al., 2017). In Hyvarinen et al. (2018) study, they used human BMSCs or BMSC-derived EVs to coculture with human M $\phi$ s. Zhang et al. (2018) intra-articularly injected exosomes from human embryonic stem cell–derived MSCs into rat with osteochondral defect. Chamberlain et al. (2019) also used human BMSCs and M $\phi$ s for the *in vitro* experiment. BMSCs, CD14<sup>+</sup> M $\phi$ s and exosome-educated M $\phi$ s were intralesionally injected in the mice with Achilles tendon injury. Pacienza et al. (2019) used human BMSCs and RAW 264.7 for the *in vitro* experiment. Exosomes derived from BMSCs were intravenously injected into the endotoxemia mouse model. The *in vitro* experiment cell sources were all C57BL/6 mice in Chang et al. (2015), Cosenza et al. (2017) and Li Y. et al. (2019) studies. Fibrin containing Sprague–Dawley rat BMSC-derived EVs was intralesionally injected into the patellar tendon defect rat in Shi et al. (2019) study. Shen et al. (2020) used an *in vitro* model whose cell source was isolated from specific transgenic mice. The details of cell management are summarized in **Table 2**.

## Boosting Approaches for MSC-Paracrine Mediators

Lo Sicco et al. (2017) isolated ADSC-derived EVs from normoxic and hypoxic culture condition. Li Y. et al. (2019) used a Transwell coculture system of ADSCs and M $\phi$ s without extra management of MSCs paracrine mediators. The other eight studies collected and purified EVs or other paracrine mediators from MSC-derived CM by differential centrifugation approaches (Chang et al., 2015; Cosenza et al., 2017; Lo Sicco et al., 2017; Hyvarinen et al., 2018; Zhang et al., 2018; Chamberlain et al., 2019; Pacienza et al., 2019; Shi et al., 2019; Shen et al., 2020). The detailed methods are described in **Table 2**.

## Measurement Instruments

For classifications on the results, the measurement instruments, including immunohistochemistry, reverse transcriptase–polymerase chain reaction analysis, Western blot analysis, enzyme-linked immunosorbent assay, flow cytometry analysis, multiplex cytokine assay, confocal laser scanning, micro-computed tomography, and mechanical testing, were used. Also, fractal analysis (Chamberlain et al., 2019) and bone parameter analysis (Cosenza et al., 2017) were applied. Detailed results are listed in **Table 2**.

## Experimental Variables and Controls

For two bone fracture–related studies, Li Y. et al. (2019) locally injected a total of  $1 \times 10^6$  ADSCs at the injury site of their cylindrical bone defect model; the therapeutic efficacy of ADSCs was compared to the non-injected group. In the *in vitro* experiment, the bone marrow–derived M $\phi$ s were cocultured with  $1 \times 10^5$  ADSCs in the upper chamber of Transwell under with high-glucose Dulbecco modified eagle medium containing 10% fetal bovine serum (FBS). Chang et al. (2015) planted  $1 \times 10^5$  bone marrow–derived M $\phi$ s into six-well plates and cultured 24 h. Then,  $1 \times 10^5$  BMSCs or apoptotic BMSCs (exposed to UV light treatment for 30 min) were placed on the M $\phi$ s in  $\alpha$ -minimum essential medium containing 10% FBS. For each time point, M $\phi$ s cultured alone, apoptotic BMSCs cultured alone, and BMSCs cultured alone served as control groups.

For the OA-related studies, Cosenza et al. (2017) applied the *in vitro* OA-like chondrocytes model to investigate the effect of microparticles, BMSC-derived exosomes, BMSC-derived CM, and BMSCs on M $\phi$ s in a Transwell system. For their *in vivo* experiment, BMSCs, microparticles, and exosomes were intra-articularly injected into the arthritis model. Those OA mice without treatment served as a control group. In Zhang et al. (2018) study, exosomes and phosphate-buffered saline (PBS) (used as control) were intra-articularly injected into the rat osteochondral defect model after surgery, respectively.

For three tendon injury–related studies, two of them (Chamberlain et al., 2019; Shen et al., 2020) reported the coculture results of EVs and M $\phi$ s, and the another one study reported *in vivo* experiment (Shi et al., 2019). Chamberlain et al. (2019) reported that exosome-educated M $\phi$ s could be generated by MSC-derived EVs, and M $\phi$ s treated with PBS served as control groups in their *in vitro* experiment. To compare the therapeutic potential on tendon healing, the authors intralesionally injected

**TABLE 2 |** Study characteristics and outcomes.

Author	Journal	Year	Cell source	Target disease	Study type	Cells management	Bioengineering method for MSCs paracrine mediators	Measurement instrument
Chang J	Bone Res	2015	C57BL/6 mice	Bone fracture	<i>In-vitro</i>	BMSC CM and M $\phi$ s cell contract co-culture	BMSCs intrinsic bio-activation: the supernatants from BMSCs cultures were collected and stored at $-80^{\circ}\text{C}$ until used as conditioned medium.	Scratch assay, BMSCs migration assay, IL-6 ELISA assay, cell growth assay, Gene expression by RT-PCR, Western blot
Cosenza et al., 2017	Sci Rep	2017	C57BL/6 mice	Osteoarthritis	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : BMSC Exos and M $\phi$ s <i>In-vivo</i> (arthritis model): IA injections of BM-MSCs, MPs or Exos.	BMSCs intrinsic bio-activation: BMSC-CM was centrifuged at 300 g for 10 min to eliminate cells and 2,500 g for 25 min to remove debris and apoptotic bodies. For MP isolation, CM was centrifuged at 18,000 g for 1 h in polyallomer tubes; the pellet was then suspended in PBS and submitted to a second round of centrifugation. For Exos, supernatant from MP fraction was filtered on 0.22 $\mu\text{m}$ porous membrane and centrifuged at 100,000 g for 2 h.	Flow cytometry analysis, Bone parameter analyses, Confocal laser scanning microscopy, Histological analysis
Lo Sicco	Stem Cells Transl Med	2017	Human-ADSCs; C57BL/6 mice-M $\phi$ s	Muscle damage	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : ADSC EVs and M $\phi$ s <i>In-vivo</i> (muscle injury model): IL injection of ADSCs EVs	ADSCs extrinsic modification: ADSC EVs were isolated from normoxic- and hypoxic-conditioned media by differential centrifugation at 300 g for 10 min, 2,000 g for 20 min, 10,000 g for 30 min at $4^{\circ}\text{C}$ to eliminate cells and debris	Protein quantification and immunoblot analysis, Flow cytometry analysis, qRT-PCR, Immunofluorescence analysis, Histology and morphometric analysis
HyvÄärinen K	Front Immunol	2018	Human	/	<i>In-vitro</i>	BMSCs or BMSC-EVs and M $\phi$ s co-culture	BMSCs intrinsic bio-activation: BMSCs media were collected and centrifuged at 2,000 g for 10 min to remove cell debris. The supernatant was ultracentrifuged with Optima <sup>TM</sup> MAX-XP Ultracentrifuge (Beckman Coulter) at 100,000 g 1.5 h $+4^{\circ}\text{C}$ with MLA-50 rotor (k-factor = 92, Beckman Coulter), and the pelleted EVs were combined. For the second EV collection, the cell starvation was continued in 200 ml $\alpha$ -MEM at $37^{\circ}\text{C}$ , 5% $\text{CO}_2$ for 2 days followed by replication of EV centrifugation steps.	Flow cytometry analysis, cytokine (IL-10, IL-23, IL-22) and LMs measurements
Zhang S	Biomaterials	2018	Human embryonic stem cell derived MSCs	Osteoarthritis	<i>In-vivo</i>	Sprague-Dawley rat osteochondral defect model: IA injection of MSCs-Exos	BMSCs intrinsic bio-activation: MSCs were grown in a chemically defined medium for 3 days and exosomes were purified from the CM.	Histology and immunohistochemistry, Multiplex cytokine assay (IL-1 $\beta$ , IL-6, TNF- $\beta$ )
Chamberlain CS	Stem Cells	2019	Human	Tendon injury	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : BMSC-EVs and M $\phi$ s; <i>In-vivo</i> (Foxn1nu mouse model of Achilles tendon injury): IL injection of BMSCs, CD14+ M $\phi$ s or EEMs	BMSCs intrinsic bio-activation: BMSCs CM was centrifuged using a Beckman Coulter Allegra X-15R centrifuge at 2,000 g at $4^{\circ}\text{C}$ for 20 min. Clarified supernatant CM was then centrifuged in a Beckman Coulter Optima L-80 XP Ultracentrifuge at 100,000 g at $4^{\circ}\text{C}$ for 2 h with a swinging bucket SW 28 rotor to pellet EVs.	Flow cytometry analysis, IHC/Immunofluorescence/Histology; Fractal analysis; Mechanical testing

(Continued)



TABLE 2 | Continued

Author	Journal	Year	Cell source	Target disease	Study type	Cells management	Bioengineering method for MSCs paracrine mediators	Measurement instrument
Pacienza N	Mol Ther Methods Clin Dev	2019	Human-BMSCs; RAW 264.7-M $\phi$ s cell	/	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : LPS in combination with Exos and M $\phi$ s; <i>In-vivo</i> (endotoxemia mouse model): IV injection of Exos	BMSCs intrinsic bio-activation: BMSCs CM was applied directly at room temperature to a column containing the anion exchange resin (Q Sepharose Fast Flow, GE Healthcare, Chicago, IL, USA) that had been equilibrated with 50 mM NaCl in 50 mM phosphate buffer (pH 7.5). The column resin was washed with 100 mM NaCl in 50 mM phosphate buffer (pH 7.5) and then eluted with 500 mM NaCl in 50 mM phosphate buffer (pH 7.5).	qRT-PCR, Quantitation of TNF- $\alpha$ , IL-1 $\beta$ , and IL-6 by ELISA
Shi Z	J Transl Med	2019	Sprague-Dawley rats-BMSCs	Tendon injury	<i>In-vivo</i>	Sprague-Dawley rat patellar tendon defect model: injection of fibrin containing EVs	BMSCs CM was centrifuged sequentially at 300 g for 10 min followed by 2,000 g for 10 min to remove cellular debris. The supernatants were then ultracentrifuged at 100,000 g for 2 h to obtain a pellet containing the EVs, which was resuspended in 200 $\mu$ L of PBS. EVs-enriched fraction was centrifuged at 1,500 g, 30 min with 100-kDa molecular weight cut off (MWCO) hollow fiber membrane (Millipore, Billerica, MA, USA). Then, EVs were passed through a 0.22- $\mu$ m filter.	Histology and immunohistochemistry, Gene expression, Histological analysis
Li Y	J Cell Biochem	2019	C57BL/6 mice	Bone fracture	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : ADSCs and M $\phi$ s cell contract co-culture; <i>In-vivo</i> (a cylindrical bone defect model): IL injection of ADSCs	/	Immunohistochemistry, Western-blot analysis, RT-PCR, Enzyme-linked immunosorbent assay, micro-CT
Shen H	J Orthop Res	2020	Scleraxis-GFP tendon reporter mice or NF- $\kappa$ B-GFP-luciferase transgenic reporter mice-ADSCs Wild type FVB/NJ (FVB) -M $\phi$ s	Tendon injury	<i>In-vitro</i> and <i>in-vivo</i>	<i>In-vitro</i> : Exos and M $\phi$ s co-culture; <i>In-vivo</i> (NGL mouse model of Achilles tendon injury and repair): collagen sheet loaded with EVs from na $\bar{A}$ ve ASCs or IFN $\gamma$ -primed ASCs	ADSCs intrinsic bio-activation: ADSCs were cultured in an EV collection medium (2% EV-free FBS in $\alpha$ -MEM) for 48 h. CM from ASC culture (150 ml from $\sim$ 2.5 E+07 cells per isolation) with or without IFN $\gamma$ pre-treatment was collected and centrifuged at 500 g for 10 min and 10,000 g for 30 min at 4°C to remove large vesicles. After passing through a 0.22 $\mu$ m filter, the medium was further centrifuged at 100,000 g for 90 min at 4°C.	M $\phi$ s assays, NF- $\kappa$ B-Luciferase Imaging <i>in-vivo</i> , RT-PCR, Histology

Both the methodology employed and the results obtained by each article are represented in this table. BMSCs, bone marrow stem cells; ADSCs, adipose tissue derived stem cells; M $\phi$ s, macrophages; Exos, exosomes; MPs, microparticles; EVs, extracellular vesicles; CM, conditioned medium; IA, intra-articular; IL, intralesional; IV, intravenous; LMs, lipid mediators; EEMs, exosome-educated macrophages; LPS, lipopolysaccharide.

$1 \times 10^6$  human BMSCs,  $1 \times 10^6$  CD14<sup>+</sup> M $\phi$ s,  $1 \times 10^6$  exosome-educated M $\phi$ s, or Hanks balanced saline solution (used as the injured control) to the surgical sites; the contralateral intact Achilles tendon without any treatment was used as a control (Chamberlain et al., 2019). In Shen et al. (2020) study, EVs produced by IFN- $\gamma$ -primed and non-primed mouse ADSC were used to treat with M $\phi$ s to evaluate the impact of EVs on the M $\phi$ s inflammatory response. And EV collection medium and EV-free CM were used as controls in the *in vitro* experiment. In Shen et al. (2020) *in vivo* experiment, collagen sheet loaded with EVs was applied around the repair sites, and the collagen sheet only was set as a control group. Shi et al. (2019) placed fibrin glue containing BMSC-EVs in the window defect of patellar tendon. The fibrin glue alone and non-surgical rats were designed as control groups.

In Lo Sicco et al. (2017) study, ADSC-derived EVs and PBS were intramuscularly injected into an injured TA muscle as experimental and control groups, respectively.

No target disease was mentioned in Hyvarinen et al. (2018) and Pacienza et al. (2019) studies. M $\phi$ s cocultured with MSCs and MSC-EVs served as experimental variables in Hyvarinen et al. (2018) study, and M $\phi$ s served as the control group (Hyvarinen et al., 2018). In Pacienza et al. (2019) study, LPS-stimulated M $\phi$ s treated with BMSC-derived exosomes served as the experimental group, and the three control groups were M $\phi$ s treated with complete medium alone, LPS, and LPS plus dexamethasone. In Pacienza et al. (2019) *in vivo* experiment, LPS alone or in combination with exosomes ( $\sim 5 \times 10^9$  vesicles) was injected into mice through tail vein, respectively. Control group was injected with saline (Pacienza et al., 2019). Detailed results are listed in Table 3.

## Laboratory Effects and Proposed Mechanisms

The bone fracture-related studies showed that femoral bone formation and bone volume all increased after ADSC injection. The increase in osteoblasts after ADSC injection was accompanied by increases in CD206<sup>+</sup>, CD68<sup>+</sup>, CD11b<sup>+</sup>, and F4/80<sup>+</sup> cells, which represented a typical M2 surface phenotype. Meanwhile, the transcript and protein expression of Arg-1, M2 marker genes, were also up-regulated (Pacienza et al., 2019). Two OA-related studies reported that bone volume, cartilage thickness, hyaline cartilage formation, migration and proliferation of chondrocytes, and matrix synthesis were increased by BMSC-derived exosomes (Cosenza et al., 2017; Zhang et al., 2018). The therapeutic effect of anti-inflammation on OA was in line with a decrease in M1 M $\phi$ s (F4/80<sup>+</sup>, CD86<sup>+</sup>, MHCII<sup>+</sup>, CD40<sup>+</sup>), IL-1 $\beta$ , and tumor necrosis factor  $\alpha$  (TNF- $\alpha$ ), as well as increases in chondrocytes, M2 M $\phi$ s (CD163<sup>+</sup>), and IL-10 (Cosenza et al., 2017; Zhang et al., 2018). Moreover, BMSC-derived exosomes increased type II collagen deposition, and decreased type I collagen in cartilage was reported by Zhang et al. (2018). Such protective effects on chondrocytes might be attributed to the increase in Survivin, Bcl-2, and FGF-2 by BMSC-derived exosomes administration.

Lo Sicco et al. (2017) reported that ADSC-derived EVs were internalized in M $\phi$ s at the muscle injury site, which could

promote myofiber regeneration. Interestingly, a dynamic change of M $\phi$ s subpopulation was observed after transplantation of ADSC-derived EVs. ADSC-derived EVs increased M1 M $\phi$ s (Ly6C<sup>+</sup>, CD11b<sup>+</sup>, CD40<sup>+</sup>, CD86<sup>+</sup>) at 24 h post-treatment and increased M2 M $\phi$ s (CD206<sup>+</sup>, CD51<sup>+</sup>, CD36<sup>+</sup>) at 72 h post-treatment.

Extended results could be concluded from three tendon injury-related studies (Chamberlain et al., 2019; Shi et al., 2019; Shen et al., 2020). The shift of M $\phi$ s from M1 to M2 showed consistent variation to the increase in ultimate stress, Young's modulus (Chamberlain et al., 2019), NF- $\kappa$ B activity (Shen et al., 2020), fiber alignment score, tendon matrix formation, tenogenesis, and tendon cell proliferation (Shi et al., 2019) and the decrease in type I/type III collagen ratio (Chamberlain et al., 2019), gap-rupture rate (Shen et al., 2020), and tendon cell apoptosis (Shi et al., 2019) by BMSC-derived or ADSC-derived exosome administration.

## DISCUSSION

Nowadays, it has been emphatically contended that all newly presented clinical interventions should start with and end with a systematic review, or even meta-analysis (Clarke et al., 2007). Although the therapeutic effects of MSCs via paracrine factors have been demonstrated in many pre-clinical investigations, the substantial clinical evidence in beneficial effects of MSCs in musculoskeletal diseases is not adequately investigated (Regenberg et al., 2009; Lukomska et al., 2019). This systematic review aims to analyze the impact of MSC and MSC-derived EVs on M $\phi$ s in the treatment of musculoskeletal diseases. According to the outcomes of the included studies, MSCs could promote musculoskeletal tissue repair or healing via their paracrine regulation on M $\phi$ s.

## Interactions of Predominant M $\phi$ s With MSC-EVs

M $\phi$ s are predominant myeloid cells that chronologically accumulate in musculoskeletal tissue at the onset of injury-induced inflammation and exhibit regulatory activity at all stages of the healing process (Tidball, 2011; Varol et al., 2015). Therefore, M $\phi$ s are potent triggers for tissue healing processes, including cell recruitment, proliferation, and remodeling (Artlett, 2013).

Growing evidence has demonstrated that the phenotypic switch of M $\phi$ s is critical in MSC-mediated tissue regeneration, which is presented in Figure 3. Moreover, M1 M $\phi$ -released proinflammatory cytokines enhanced the migratory capacity of MSCs that facilitates the accessibility of exogenous MSCs toward the injured site. Then, the attracted MSCs would regulate M $\phi$  phenotypes into M2 via paracrine effect to facilitate the tissue remodeling at a later stage (Le Blanc and Mougiakakos, 2012; Maxson et al., 2012; Shi et al., 2012; Bernardo and Fibbe, 2013; Ma et al., 2014). Interestingly, the secretomes of MSCs, such as EVs or exosomes, were also altered by M1 M $\phi$  that boosts the therapeutic effect of MSCs, so-called primed MSCs (Aktas et al., 2017; Saldana et al., 2019).

**TABLE 3 |** Evaluations and results list of selected studies in which the therapeutic potential of the administration of MSCs for the treatment of musculoskeletal diseases.

Target disease	References	Variables	Controls	Laboratory effects	Proposed mechanisms	Conclusions	Research implications
Bone fracture	Chang et al., 2015	Seeding BM Mφs first. Then placing primary or apoptotic BMSCs. Additional validation: IL-6 KO mice.	Mφs cultured alone, apoptotic BMSCs cultured alone, and BMSCs cultured alone.	↑ BMSCs migration ↑ BMSCs proliferation	↑ IL-6 proteins and mRNA ! IL-6/gp130/STAT3 pathway	BMSCs are the main contributing cells of juxtacrine IL-6 production. Juxtacrine cultures accelerated BMSCs migration and numbers.	Increase the understanding of Mφs in bone formation.
	Li Y. et al., 2019	ADSCs and BM Mφs co-culture system; ADSCs locally injection.	Untreated control mice	↑ femoral bone formation ↑ femoral bone volume	↑ osteoblasts ↑ CD206+ cells ↑ CD68+ cells ↓ iNOS+ cells ↑ CD11b+ F4/80+ cells ↑ IL-1 $\alpha$ proteins ↓ TNF- $\alpha$ proteins and mRNA ↓ iNOS proteins and mRNA ↑ Arg-1 proteins and mRNA ↑ MR proteins and mRNA ↑ Runx-2 and osterix and OPG and ALP genes ↓ RANKL genes	ADSCs and Mφs can synergistically contribute to bone repair through mutual regulation of their differentiation and cytokine secretion.	The interactions between ADSCs and BM Mφs could be a promising therapeutic strategy in the rehabilitation of bone damage.
Osteoarthritis	Cosenza et al., 2017	Mediums of OA like chondrocytes model were replaced by medium containing MPs, Exos, BMSCs-CM or BMSCs (transwell); BMSCs, MPs or Exos intra-articularly injection.	OA control mice	Restore the anabolic equilibrium ↓ apoptotic chondrocytes ↓ macrophage activation ↑ bone volume, cartilage degradation (surface/volume ratio) and thickness	↓ MMP-13, ADAMTS5, iNOS genes ↑ ACAN, COL2B, COL1 genes ↓ F4/80+ cells ↓ CD86, MHCII or CD40 markers ↓ TNF- $\alpha$ proteins ↑ IL-10 proteins	Exos were more efficient than MPs and BMSCs in chondroprotective and anti-inflammatory function.	MPs and Exos help to promote re-establish chondrocyte homeostatic state.
	Zhang et al., 2018	Intra-articular injection of Exos.	Intra-articular injection of PBS	↑ neotissue formation and ECM deposition of s-GAG ↑ Wakitani score ↑ surface regularity ↑ hyaline cartilage formation ↑ percentage areal deposition of type II collagen ↓ percentage areal deposition of type I collagen ↑ chondrocyte migration, proliferation and matrix synthesis ↑ metabolic activity	↑ chondrocytic cells ↓ PCNA+ cells ↑ CD163+ cells ↓ CD86+ cells ↓ IL-1 $\beta$ and TNF- $\alpha$ ↑ Survivin, Bcl-2, FGF-2 mRNA ! AKT and ERK pathways	Exos mediate cartilage repair by mounting a coordinated, multi-faceted response of enhancing proliferation, migration and matrix synthesis, attenuating apoptosis and modulating immune reactivity.	Exo could be provided as a cell-free MSC therapeutic.
Muscle damage	Lo Sicco et al., 2017	EVs-Mφs co-culture; EVs were intramuscularly administered into muscle.	PBS were intramuscularly administered into muscle	↑ internalization of EVs in Mφs ↑ Mφs proliferation ↓ M1/M2 ↓ mononucleated myoblasts ↑ fibers containing nuclei $\geq 2$	↑ Ly6C, CD11b, CD40, CD86 (post-treat 24h) ↑ CD206, CD51, CD36 (post-treat 72h) ↓ CD11b, CD86 (post-treat 72h) ↓ IL-6/IL-10 ↑ Arg-1, Ym-1 markers ↓ Nos-2 markers ↑ CD206+ cells ↓ Ly6C+ cells ↑ MCP-1 ↑ Pax-7, MyoD genes (activated satellite cells); eMyhc gene (regenerating fibers)	EVs co-cultured with responding BM-derived Mφs, shifting the balance toward a M2 phenotype.	Establish an alternative cell-free approach of EVs for the induction of regenerative processes.

(Continued)

TABLE 3 | Continued

Target disease	References	Variables	Controls	Laboratory effects	Proposed mechanisms	Conclusions	Research implications
Tendon injury	Chamberlain et al., 2019	EVs-Mφs co-culture; BMSCs, CD14+ Mφs or EEMs locally injection.	Mφs controls were treated with PBS; Contralateral controls	↑ EVs educated Mφs ↑ ultimate stress and Young's modulus ↓ M1/M2 ratio ↓ type I collagen ↓ type I/type III collagen ratio	↑ CD206 mean fluorescence intensity and cells ↑ PD-L1 mean fluorescence intensity and cells ↑ M2 Mφs ↓ M1 Mφs ↑ endothelial cells	EVs-educated Mφs treatments improve mechanical properties for tendon function as shown by reduce endogenous M1/M2 ratio indicating less inflammation.	EEMs treatment provides a novel strategy in musculoskeletal injuries.
	Shen et al., 2020	EVs-Mφs co-culture; Collagen sheet loaded with EVs was applied around the repair site.	EVs collection medium and EV-free conditioned medium controls; Collagen sheet only control. IL-1 (EVs from IFNγ-primed ASCs)	↑ NF-κB activity ↑/↓ Matrix gene expression ↓ gap-rupture rate	↓ IL-1 gene (only EVs from IFNγ-primed ASCs) ↑ IFNγ gene (only EVs from IFNγ-primed ASCs) ↑ MMP-1, Sox-9 genes ↑ Col-1α1, Col-2α1 and Col-3α1 genes (only EVs from IFNγ-primed ASCs) ↑ collagen staining (only EVs from IFNγ-primed ASCs)	EVs from ASCs can shift the Mφs phenotypic response to tendon injury from a default M1 to a M2 phenotype.	EVs could be a new cell-free therapy, for tendon repair with the potential for improved therapeutic efficacy and safety.
	Shi et al., 2019	Fibrin glue containing EVs was placed in the window defect	Fibrin glue alone and untreated controls	↑ Fiber alignment score ↑ anti-inflammatory response ↑ tendon matrix formation ↑ tenogenesis ↓ tendon cell apoptosis ↑ tendon cell proliferation	↑ CD163 marker ↑ IL-4, IL-10 mRNA and + cells ↓ IFNγ, IL-1β, IL-6, CCR7 + cells ↑ SCX, TNMD, Col-1α1, Col-3α1 genes ↑ CD146+ cells ↓ cleaved caspase 3 signals	EVs derived from BMSCs can help to improve the quality of tendon healing by promoting an anti-inflammatory environment.	These findings provide a basis for the potential clinical use of BMSC-EVs in tendon repair.
None	Hyvarinen et al., 2018	MSCs-Mφs coculture; MSC-EVs-Mφs coculture	Mφs only control	↓ FRI of CD163 ↓ cytokine levels of Mreg-CM ↑ LMs level and pathway markers ↑ phagocytic ability	↓ CD163+ cells ↓ IL-10, IL-22, IL-23, TNF-α proteins ↑ arachidonic acid-derived PGE2, 15-HETE, docosahexaenoic acid-derived 17-HDHA ↑ CD206 Mregs and receptors	Both MSCs and MSC-EVs decrease IL-23 and IL-22 while increasing PGE2 production.	MSC-EVs may potentiate tolerance-promoting proresolving phenotype of human Mregs.
	Pacienza et al., 2019	LPS in combination with Exos-Mφs coculture; Tail vein injection of LPS	FBS medium alone control, containing LPS control, and LPS plus dexamethasone control; Tail vein injection of saline control	↑ anti-inflammatory activity ↑ predictive efficacy	↓ IL-6, IL-1β levels ↓ iNOS mRNA	Exos could suppress LPS-induced inflammation.	<i>In-vitro</i> Mφs assay predicts the <i>in-vivo</i> anti-inflammatory potential of Exos

Both the methodology employed and the results obtained by each article are represented in this table. Apoptotic BMSCs (exposed to UV light treatment for 30 min); !, activation; KO, knockout; TNF-α, tumor necrosis factor-α; RANKL, receptor activator of nuclear factor κB ligand; iNOS, inducible nitric oxide synthase; MR, mannose receptor; Arg-1, Antibodies against arginase 1; Runx-2, runt-related transcription factor 2; OPG, osteoprotegerin; ALP, alkaline phosphatase; Mregs, regulatory macrophages; MPs, microparticles; Exos, exosomes; EVs, extracellular vesicles; EEMs, exosome-educated macrophages; OA, osteoarthritis; ECM, extracellular matrix; FRI, fluorescence intensity; LMs, lipid mediators; LPS, lipopolysaccharide.

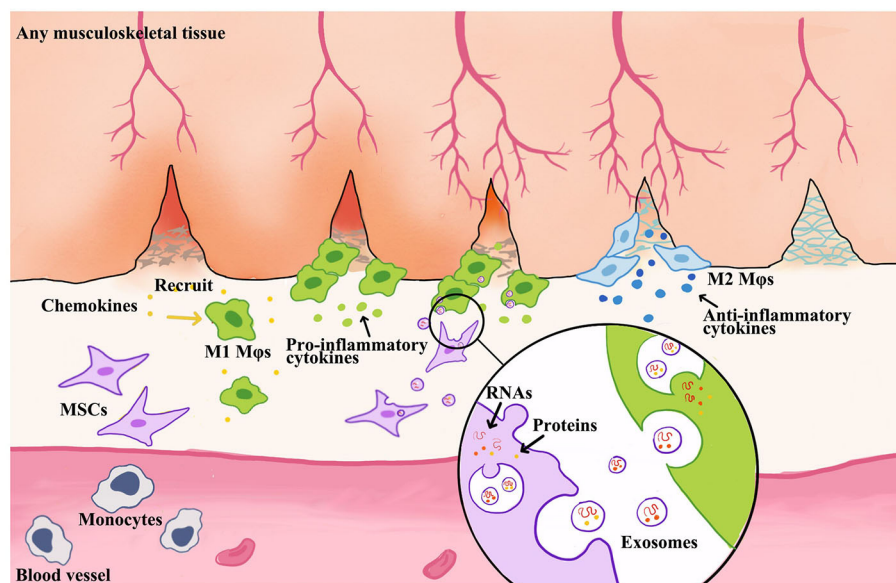


The cross-talk between M $\phi$ s and MSC-derived EVs regulates a shift in M $\phi$ s subtypes from M1 into M2 (Galli et al., 2011). An increasing number of musculoskeletal tissue injury models showed that M $\phi$ s phenotypic alteration mediated MSC-based therapy (Lo Sicco et al., 2017; Chu et al., 2019). M1 M $\phi$ s promote recruitment of inflammatory immune cells and release extracellular matrix (ECM) degrading proteins to allow quick migration to inflamed sites. As the M $\phi$ s shift to M2 subtypes, the release of proinflammatory cytokines is inhibited, angiogenesis is stimulated, and fibroblasts are activated to produce and restore more ECM. Also, M $\phi$ s induced MSCs into a motile phenotype with increased secretion of IL-6 and IL-10, which benefit MSCs to migrate to injury site (Anton et al., 2012; Wolfe et al., 2016). Typically, MSC-EVs impact the maturation of M $\phi$ s by decreasing the expression levels of IL-12 and TNF- $\alpha$  and increasing IL-6 and IL-10 in M $\phi$ s (Kim and Hematti, 2009; Cosenza et al., 2017; Lo Sicco et al., 2017; Hyvarinen et al., 2018; Shi et al., 2019). PGE<sub>2</sub> and TNF- $\alpha$  inducible gene 6 (TSG-6) embedded in MSC-EVs can promote M2 polarization in inflammatory microenvironment (Nemeth et al., 2009; Maggini et al., 2010; Melief et al., 2013b). EVs, which were released from proinflammatory cytokines-activated MSCs, could enhance anti-inflammatory properties to suppress MHC class II and CD86 signaling in LPS-stimulated M $\phi$ s (Melief et al., 2013a,b; Cho et al., 2014). Lo Sicco et al. (2017) and Chamberlain et al. (2019) reported the MSC-EVs change the M1-to-M2 M $\phi$ s ratio. And Lo Sicco et al. (2017) demonstrated that MSC-EVs could be efficiently internalized by responding cells, inducing an increase in their proliferation rate, and shifting the balance

toward an alternatively anti-inflammatory M2 phenotype. M2 M $\phi$ s phenotype could be commonly associated with the secretion of the anti-inflammatory cytokine IL-10 and scavenger receptors CD206 and CD163 (Sica and Mantovani, 2012; Mantovani et al., 2013).

In this review, we found consistent results suggesting that an M2 phenotype could be induced from M1 M $\phi$ s upon coculture with MSCs, MSC-EVs, and CM. The enrichment of M1 M $\phi$ s appears at early phases (1–3 days) during bone remodeling (Chang et al., 2015; Li Y. et al., 2019), muscle regeneration (Lo Sicco et al., 2017), and tendon healing (Chamberlain et al., 2019; Harvey et al., 2019; Shi et al., 2019; Shen et al., 2020) and is later replaced by M2 M $\phi$ s (4–7 days). As M $\phi$  polarization and tissue repair by MSC-EVs are highly associated, MSC-EVs-mediated M $\phi$ s phenotypic transformation must play a significant role during tissue healing. However, the molecular action of MSC-EVs in such M $\phi$  polarization and tissue regeneration needs further investigation (Chen et al., 2008; Rodero and Khosrotehrani, 2010).

Although studies have provided evidence of immunosuppressive effects of MSCs in clinical trials for graft-vs.-host disease and Crohn disease (Godoy et al., 2019), the transplanted MSCs have not been proven in the persistence after injection and the contribution in tissue regeneration (Pittenger, 2009; Parekkadan and Milwid, 2010; Caplan and Correa, 2011). It is likely that the main immunosuppressive effects of MSCs resulted from paracrine regulation through secreted mediators, including EVs (Raposo and Stoorvogel, 2013). The immune modulation effect of MSC has been generalized as



**FIGURE 3 |** Schematic illustration of MSC-derived exosome-guided macrophage reprogramming. MSC-derived exosomes can induce a conversion of M1 to M2 M $\phi$ s and accelerate musculoskeletal tissue healing. M $\phi$ s could be activated by inflammatory chemokines and then to produce proinflammatory factors. This creates a feedback loop whereby proinflammatory cytokines produced by M $\phi$ s stimulate MSC to produce immune modulators, such as exosomes or EVs. Therefore, the formation of exosomes begins with membrane invagination in the form of endosomes, leading to the development of the early endosomes. Upon maturation, the endosomes become multivesicular endosomes, which release their contents in the form of exosomes.

the inhibition of both innate and adaptive immunity and also derives inflammatory, autoimmune, and infectious disease pathology (Le Blanc and Mougiakakos, 2012; English, 2013). However, increasing analysis in animal models of inflammation demonstrated that MSC-EVs suppressed immune response through the transfer of RNAs and protein (Cantaluppi et al., 2012; Arslan et al., 2013). The inflammatory signals are essential to initiate and maintain the MSC- or EV-mediated tissue repair process; afterward, MSC-derived EVs could manipulate the niche by switching M $\phi$  phenotype to facilitate tissue repair.

## Functional MSC-EVs Cargo and Boosting Approaches

Recently, the use of MSC-based therapies has emerged. The therapeutic benefits of MSC transplantation have been attributed into two types, EVs and soluble factors. Soluble components include a wide variety of secreted chemokines, growth factors, and hormones with immunomodulatory activity. For example, PGE<sub>2</sub> and TGF- $\beta$  are vital mediators of anti-inflammatory in therapeutic therapy of MSCs (Yoo et al., 2013). And lots of anti-inflammatory proteins also represent tissue protection, such as TSG-6, which has been reported with healing ability by reducing the influx of neutrophils to the tissue injury site (Oh et al., 2010). The studies regarding MSC-derived EVs have grown exponentially since it has been recognized that EVs containing mRNA, miRNA, and protein could exchange intracellular information and act as sophisticated mediators of recipient cell behavior, particularly in immunomodulation (Wolfers et al., 2001; Valadi et al., 2007; Skog et al., 2008). MSC-derived EVs contain not only more than 200 mRNA and 60 miRNA, but also more than 800 proteins (Bruno et al., 2009; Chen et al., 2010; Lai et al., 2012). These EV-derived proteins have been reported with an integral role in activating anti-inflammatory responses and regulating the cascade of tissue healing process in various injury models (Lee et al., 2012; Zhang et al., 2018; Li et al., 2019).

It has been demonstrated that genetically modified MSCs possess great pro-survival, proangiogenic, and anti-inflammatory properties not only by the altering release of soluble proteins, but also EVs (Huang et al., 2013; Wang et al., 2016; Lou et al., 2017; Ma et al., 2017; Qu et al., 2017; Tao et al., 2017). MSC-derived EVs are engineered at the cellular level under natural conditions, and it further highlighted the unique advantage of EV-based nanoplateforms for cargo delivery (Luan et al., 2017). For clinical applications, some advantages of MSC-derived EVs include easier injection, reduced M $\phi$ s phagocytosis and vascular occlusion (EL Andaloussi et al., 2013), innate biocompatibility, high physicochemical stability, high penetrability, and long-distance communication (Clayton et al., 2003; Ridder et al., 2014; Zomer et al., 2015).

Many strategies have been applied to modify EVs, including cell modification and direct EV modification (Armstrong and Perriman, 2016). Because EVs are secreted from cells, they can intrinsically express some lipids or cell adhesion molecules and ligands that naturally target certain types of recipient cells

(Luan et al., 2017). Undoubtedly, it was inevitable that genetic engineering has been used to modify EVs. Overexpression of mRNA or miRNA in cells could be assembled into EVs, which could be fused to target cells to introduce or inhibit gene expression (Kosaka et al., 2010; Akao et al., 2011; Ridder et al., 2014; Zomer et al., 2015). For a significant amount of time, researchers have explored non-native biomaterials to cells to augment therapeutic function (Armstrong et al., 2015; Correia Carreira et al., 2016). Therefore, biomaterials delivered to the membrane could also naturally be incorporated into budding EVs, while internalized material may be packaged into exosomes for secretion (Armstrong et al., 2017). Taking advantage of these methods allows cellular processes and cell engineering techniques to be specifically adapted to EV functionalization.

Because only a small fraction of material could be packaged into EVs, the efficiency issue results in a low-yielding ending. And besides the widely explored cancer cells, the yield of exosomes derived from MSCs is one of the major factors that limits the expansion of cell-free therapeutic productions (Phan et al., 2018). Functional EVs could be derived from native ECM (Huleihel et al., 2016; Du et al., 2017), and three-dimensional environment for cell attachment and growth has been particularly attracted because it can mimic the ECM structure and function (Phan et al., 2018). Such structure has been reported with the effects of influencing EV secretion. For example, Tao et al. reported that Avitene Ultrafoam collagen hemostat caused the BMSCs to release 2-fold of exosomes compared to the plastic surface culture based on protein assay (Tao et al., 2017). Besides, a bioactive artificial ECM that was modified by adding molecules has been used to imitate native ECM mimicking structures and to improve yield of MSC exosomes by presenting specific functional ligands (Hao et al., 2017).

Moreover, an essential and urgent solution can provide an approach to improve the yield without sacrificing the functionality or with enhancing the efficacy simultaneously. Therefore, researchers tried to purify EVs, which could ensure that all modified sites or encapsulated species could be localized at the vesicle (Armstrong et al., 2017). As non-living entities, EVs have a major advantage over cells when they received membrane surface modification. It has been reported that excessive pressures, temperature, chemical induction, or hypoxia environment exposure could cause membrane disruption, vesicle aggregation, and surface protein denaturation (Smyth et al., 2014; Armstrong et al., 2017; Lo Sicco et al., 2017). Also, multivalent electrostatic interactions, receptor-ligand binding, and hydrophobic insertion have been commonly applied as methods of biological membrane modifications (Nakase and Futaki, 2015; Correia Carreira et al., 2016; Lee et al., 2016). In addition, electroporation, which is an alternative approach to EV active loading strategies, has been reported to transiently permeabilize the EV membrane to enhance the absorptivity of small molecules (Tian et al., 2014; Fuhrmann et al., 2015). Taken together, the engineered EVs will open up exciting opportunities in EV-based therapies

by boosting therapeutic capability, which is beyond their native functions.

## STRENGTHS AND LIMITATIONS

This systematic review poses several advantages compared to other attempts to summarize the experimental results of MSC-derived paracrine mediators for musculoskeletal diseases. First, the up-to-date literature search has been yielded by two widely used databases: PubMed and EMBASE. The search strategies included MeSH terms and other related terms. Therefore, we could identify a number of eligible studies, which might remain relatively unnoticed. Second, the quality of included studies and risk of bias, including publication bias, were assessed. Third, study heterogeneity had been explored to point out potential explanatory variables. Fourth, to standardize the spontaneous recovery in the control groups, we analyzed the variables and controls specifically. And studies without scientific controls have been excluded during full-text screening procedure.

There were several limitations to this study. The first limitation is the small number of included studies. The second limitation is that we could not standardize the effectiveness of M $\phi$ s depletion among different studies. Moreover, the depletion is not permanent and, once subsided, could result in a reactive increase in M $\phi$ s numbers with unknown consequences. Third, these studies utilized different cell sources and delivery methods. Different types of animals were selected for *in vivo* studies. Fourth, the assessment methods also widely varied among studies; therefore, it was impossible to perform a quantitative analysis or a meta-analysis with the included studies.

## CONCLUSION

This review demonstrated that MSC and MSC-EVs are authentic biomaterials to treat musculoskeletal problems. The broad therapeutic effect of MSC and MSC-EVs attribute to the

management of M $\phi$  polarization, at least in part. A further understanding in the molecular mechanism of how MSCs regulate M $\phi$  polarization will facilitate the development of bioengineering approach to boost the therapeutic capacity of MSCs and their clinical application. However, more pre-clinical studies are needed to understand how EVs and their subcomponent play a role in musculoskeletal tissue healing process.

## DATA AVAILABILITY STATEMENT

All datasets generated for this study are included in the article/**Supplementary Material**.

## AUTHOR CONTRIBUTIONS

HX: conception and design, manuscript writing, and data analysis and interpretation. C-WL: conception and design, financial support, and manuscript writing. Y-FW: manuscript writing. L-YS, Y-HW, ZW, and XZ: data analysis and interpretation. PY and OL: administrative support and final approval of manuscript. SH: revisions to scientific content of manuscripts. All authors contributed to the article and approved the submitted version.

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## SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fbioe.2020.587052/full#supplementary-material>

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Cell-Derived Extracellular Matrix for Tissue Engineering and Regenerative Medicine

Marisa Assunção<sup>1,2</sup>, Dorsa Dehghan-Baniani<sup>2</sup>, Chi Him Kendrick Yiu<sup>1,2</sup>, Thomas Später<sup>3</sup>, Sebastian Beyer<sup>4</sup> and Anna Blocki<sup>1,2,5\*</sup>

<sup>1</sup> School of Biomedical Sciences, Faculty of Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>2</sup> Institute for Tissue Engineering and Regenerative Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>3</sup> Institute for Clinical and Experimental Surgery, University of Saarland, Saarbrücken, Germany, <sup>4</sup> Department of Biomedical Engineering, Faculty of Engineering, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>5</sup> Department of Orthopaedics and Traumatology, Faculty of Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong

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### \*Correspondence:

Anna Blocki  
anna.blocki@cuhk.edu.hk

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Cell-derived extracellular matrices (CD-ECMs) captured increasing attention since the first studies in the 1980s. The biological resemblance of CD-ECMs to their *in vivo* counterparts and natural complexity provide them with a prevailing bioactivity. CD-ECMs offer the opportunity to produce microenvironments with customizable biological and biophysical properties in a controlled setting. As a result, CD-ECMs can improve cellular functions such as stemness or be employed as a platform to study cellular niches in health and disease. Either on their own or integrated with other materials, CD-ECMs can also be utilized as biomaterials to engineer tissues *de novo* or facilitate endogenous healing and regeneration. This review provides a brief overview over the methodologies used to facilitate CD-ECM deposition and manufacturing. It explores the versatile uses of CD-ECM in fundamental research and therapeutic approaches, while highlighting innovative strategies. Furthermore, current challenges are identified and it is accentuated that advancements in methodologies, as well as innovative interdisciplinary approaches are needed to take CD-ECM-based research to the next level.

**Keywords:** cell-derived extracellular matrix, stem cell niche, cell differentiation, tissue engineering, regenerative medicine, skeletal repair, cardiovascular repair, cell-extracellular matrix interactions

## INTRODUCTION

The extracellular matrix (ECM) is the non-cellular component present in all connective tissues and has a composition specific for each tissue. It is comprised of a complex and highly organized three-dimensional macromolecular network of biomolecules. These include fibrous proteins (such as collagens) and glycosaminoglycan (GAG)-based components. Fibrous ECM components form the backbone of the polymer network, thereby providing shape/stability and tensile strength to tissues. They also regulate cell adhesion and support cell migration. GAG-based components fill the interstitial space, ensuring hydration and lubrication of tissues, and acting as a reservoir and modulator of cytokine signaling (Theocharis et al., 2016; Yong et al., 2020).

ECM-driven communication arises from a complex combination of biochemical, topological and biomechanical cues, facilitating a reciprocal dialogue with cells, which can respond via remodeling of the ECM. This multi-dimensional signaling enables the ECM to guide intricate cellular and tissue processes such as homeostasis, healing and regeneration (Kaukonen et al., 2017).



## ECM AS A BIOMATERIAL

The ECM is a biomaterial designed by nature that underwent over 600 million years of material optimization (Ozbek et al., 2010). It serves as a blueprint for many man-made biomimetic biomaterials. Nonetheless, these materials represent oversimplified versions of the ECM that are not able to replicate its complex bioactivity (Kaukonen et al., 2017). As a result, ECM derived from decellularized tissues, remains one of the most successful biomaterials in clinics (Hussey et al., 2018).

Unfortunately, tissue-derived ECM faces various challenges to its clinical application. The limited availability of human cadaveric tissue leads to the use of animal tissue-derived ECM as an alternative source. Especially the incomplete decellularization of tissue carries the risk of disease transmission and immunological rejection. Some ECMs are plainly not available, since some specific tissues are hard to isolate (e.g., stem cell niches). Further, tissue-derived ECM is set in its composition, therefore cannot be customized in its bioactivity toward a specific application (Aamodt and Grainger, 2016).

As cell-derived ECM (CD-ECM) partially recapitulates the complex biological machinery of native tissue (Ahlfors and Billiar, 2007), it can address many of the tissue-derived ECM's limitations. It can be derived from human cell cultures by gentle decellularization to remove immunogenic components, while preserving its bioactivity. ECM-synthesizing cells can be standardized and pre-screened (Sharma et al., 2020), minimizing the risk of disease transmission. Deriving ECM *in vitro* provides the opportunity to select appropriate ECM-producing cell types, further modify them (e.g., genetically) and expose them to specific stimuli, thus enabling the creation of ECM with desired properties (Maia et al., 2020). CD-ECM is therefore an incredibly versatile material to be used in physiological studies and therapeutic approaches.

## METHODOLOGIES TO GENERATE CD-ECM

Stromal cell-derived ECMs are rich in collagens (Antebi et al., 2015), while endothelial/epithelial CD-ECMs contain a laminin-rich basement membrane-like ECM (Davis and Senger, 2005). CD-ECM can be generated by culturing cells scaffold-free in 2D and 3D cultures (Serebriiskii et al., 2008; Hoshiba et al., 2016; Sharma et al., 2020). Alternatively, cells can also be seeded within hydrogels or scaffolds, forming hybrid CD-ECM-based materials (Sart et al., 2014; Suhaeri et al., 2018).

### Facilitating ECM Deposition *in vitro*

Slow ECM assembling kinetics *in vitro* necessitate long cell culture periods up to several weeks to harvest sufficient CD-ECM amounts for the desired application (Bourget et al., 2012). This can be improved by adjusting culture conditions (Hoshiba, 2017).

The most essential supplement for robust ECM deposition is ascorbate, a cofactor of lysyl hydroxylase and prolyl hydroxylase, essential enzymes in collagen fibrillogenesis (Pinnell, 1985). Collagen type I is the most prominent ECM component and its

deposition increases the overall yield of CD-ECM and improves its mechanical properties. Nonetheless, rapid degradation of ascorbate (Grinnell et al., 1989) calls for frequent media changes, thereby discarding the not-yet deposited ECM components. A stable form of ascorbate (2-phospho-L-ascorbate) can reduce the frequency of medium replacements (Chen et al., 2011).

The yield of deposited ECM can be amplified by introducing macromolecules, which emulate the crowded conditions present *in vivo*. The biophysical principle of macromolecular crowding (MMC) relies on macromolecules occupying space, thereby increasing the effective concentration of other molecules and the thermodynamic activity of the system. This has profound effects on protein folding, molecular interactions and enzyme kinetics (Chen et al., 2011). In particular, under MMC more ECM can be deposited within 1 week than after several weeks under non-crowded conditions. Most commonly used “crowders” are Ficoll, carrageenan, polyvinylpyrrolidone and dextran sulfate (Lareu et al., 2007; Lu et al., 2011; Blocki et al., 2015; Gaspar et al., 2019), albeit dextran sulfate was recently found to act as a precipitating agent, independent of MMC (Assunção et al., 2020).

Culturing cells with low serum concentration (<1% v/v) was also beneficial, as serum carries exogenous matrix metalloproteases that degrade ECM and imbalance the ECM's natural remodeling rate (Satyam et al., 2014; Kumar et al., 2015). Furthermore, hypoxia was shown to induce synthesis of ECM richer in collagenous proteins and angiogenic factors, as seen in fibroblasts (Distler et al., 2007; Kumar et al., 2018) and mesenchymal stem cells (MSCs) (Cigognini et al., 2016; Du et al., 2017).

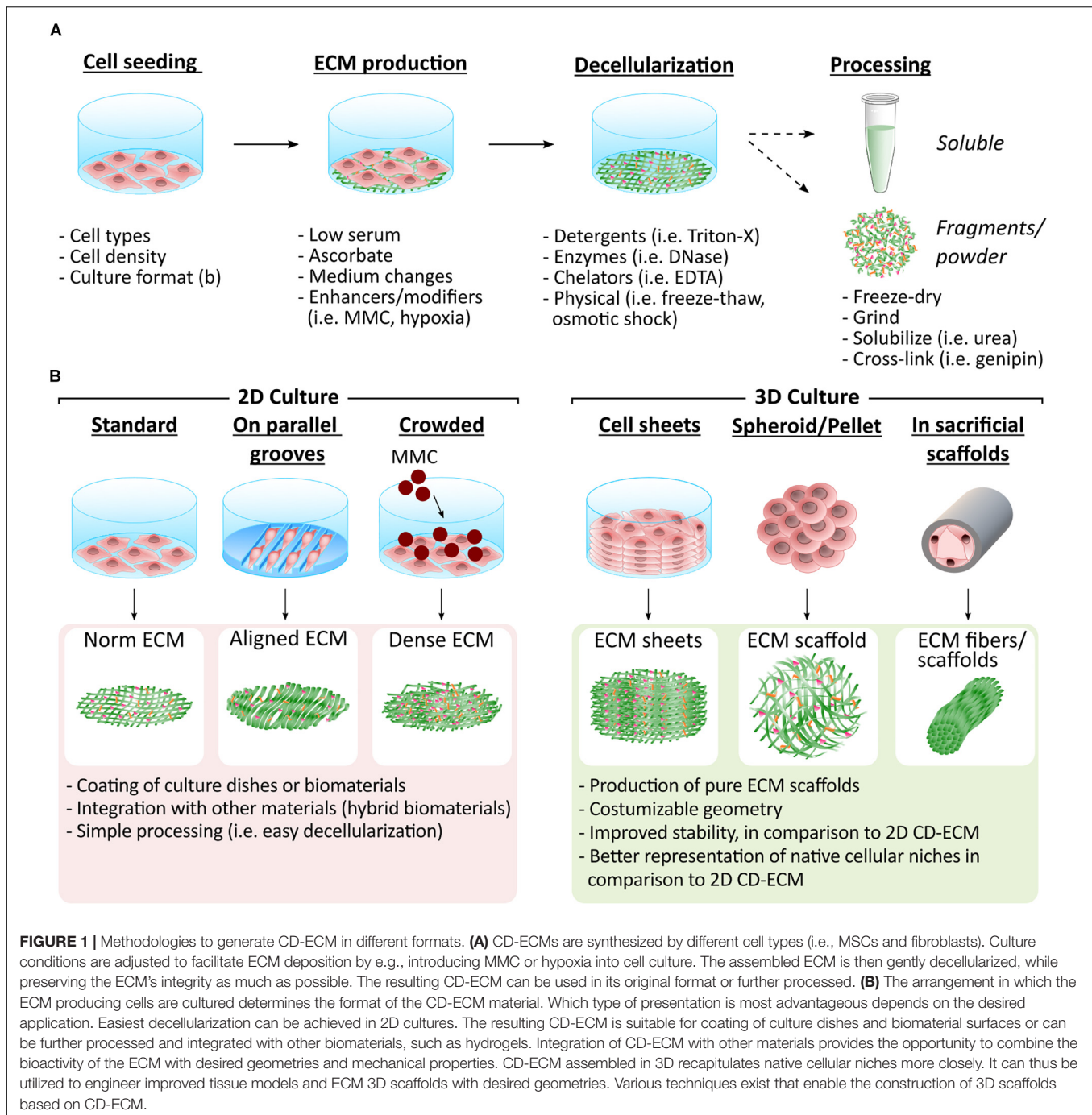
## Decellularization and Processing of CD-ECM

CD-ECMs are usually generated in a small format, permitting gentle decellularization methods with focus on maintaining architecture and bioactivity. Most methods use detergents, enzymes, chelating agents, mechanical approaches and combinations thereof (Figure 1A; Woods and Gratzner, 2005; Faulk et al., 2014; Levorson et al., 2014; Gilpin and Yang, 2017). Complete decellularization is further achieved by removing genetic material with nucleases to prevent host immune reaction, as can be observed in tissue-derived ECMs (Crapo et al., 2011).

Decellularized CD-ECMs can then be used in their original format, fragmented (Carvalho et al., 2019b), grinded (Wei et al., 2015) or solubilized (Decaris et al., 2012). These formats give rise to 2D ECM layers or more complex 3D structures comprising 3D scaffolds (McAllister et al., 2009), spheroids (Cheng et al., 2009), fibers (Roberts et al., 2017), and sheets (Sharma et al., 2020; Figure 1B).

## APPLICATIONS OF CD-ECM

Numerous applications have been explored for CD-ECMs including the improvement of cellular functions, seen in tailored cellular niches, the study of ECM in a physiological



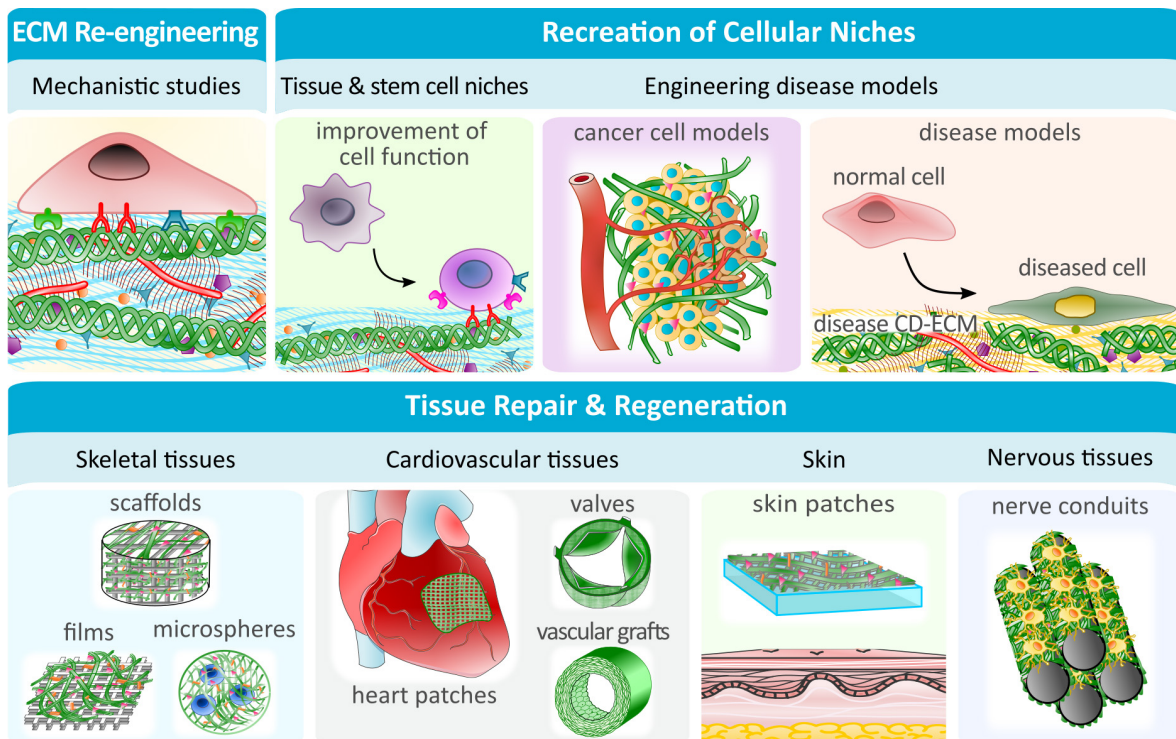
and pathophysiological context, and the application in tissue engineering and regenerative medicine (TERM) (Figure 2).

## Recreation of Cellular Niches

### Stem Cell Niches

The emulation of the native cellular microenvironment in culture is a prerequisite to maintain the cells' phenotype and function. This is especially true for sensitive cell types, such as stem cells, which are known to undergo senescence and lose their stemness *ex vivo* (Hoshiba et al., 2016).

Various studies demonstrated that MSC-derived ECM can recapitulate the stem cell niche sufficiently to protect reseeded MSCs from oxidative stress, promote their proliferation, and conserve their stemness (Chen et al., 2007; Lai et al., 2010; Liu et al., 2016; Xing et al., 2020). CD-ECMs were also shown to maintain the native phenotype of neural progenitor cells (Yang et al., 2017; Hoshiba et al., 2018), embryonic stem cells (ESCs) (Klimanskaya et al., 2005), periodontal ligament stem cells (Xiong et al., 2019) and hematopoietic stem cells (Prewitz et al., 2013). Furthermore, ECMs derived from younger MSCs were shown



**FIGURE 2 |** CD-ECM applications in fundamental research, pathophysiological studies and regenerative medicine. The ease with which CD-ECM can be modified, makes it the ideal platform to study detailed ECM mechanisms or the role of cellular niches under physiological and pathophysiological conditions. Specialized engineered cellular niches can be further utilized to improve cellular functions *in vitro*, such as stemness. In TERM, CD-ECMs can be created with specific mechanical and biological properties to be used on their own (i.e., as vascular grafts) or to enhance the performance of (semi-) synthetic biomaterials.

to rejuvenate *in vitro*-aged and chronologically-aged MSCs (Pei et al., 2011; Sun et al., 2011; Lin et al., 2012). These effects were tightly linked to the biological profile of the ECM (reviewed in Sart et al., 2020).

### Tissue-Specific Niches

Similar to MSC-derived ECM supporting stemness, ECMs derived from adipogenically or osteogenically induced MSCs promoted the respective lineage commitment of reseeded MSCs via integrated structural and regulatory proteins (Ang et al., 2014; Jeon et al., 2018; Carvalho et al., 2019a). Chondrogenic differentiation was best supported by chondrogenic ECM deposited in 3D (Cheng et al., 2009; Lu et al., 2011). Synovial MSC-derived ECM also protected chondrocytes from pro-inflammatory stimuli (Yan et al., 2020).

CD-ECMs from stromal, endothelial and epithelial cells could improve the function of specialized cell types, such as podocytes (Satyam et al., 2020), chondrocytes (Wei et al., 2015; Yang et al., 2018; Zhang et al., 2019), hepatocytes (Grant et al., 2017; Guo et al., 2017), Schwann cells (Xiao et al., 2016), as well as promote natural killer cell differentiation (Lee et al., 2020).

Similarly to adult stem cells, CD-ECMs synthesized by differentiating ESCs were able to promote early differentiation of ESCs, even without external factors (Goh et al., 2013). ECM produced by an endoderm-inducing cell line and ECM from liver progenitor cells promoted differentiation of pluripotent cells into

insulin-expressing pancreatic  $\beta$ -cells (Higuchi et al., 2010) and hepatic cells (Kanninen et al., 2016), respectively.

Hence, CD-ECMs can be utilized to tailor cell and tissue-specific niches to promote cellular functions and study cell-niche interactions in detail.

### Engineering ECM in Disease

The ECM has a long-implicated role in disease development and progression, although the exact mechanisms often remain elusive. While the CD-ECM platform provides the opportunity to manipulate ECM and study it in detail, few studies utilized CD-ECM to study ECM mechanisms in disease (Raghunathan et al., 2018), most of them related to cancer.

It is currently well accepted that the tumor microenvironment plays a pivotal role in cancer cell behavior, including proliferation, invasiveness, metastasis and drug resistance (Serebriiskii et al., 2008). CD-ECMs provide the prospect to improve cancer models by recreating the cancer microenvironment using standard 2D, 3D cultures or more complex, organ-on-a-chip strategies (Gioiella et al., 2016; Kaukonen et al., 2017; Hoshiba, 2018). Indeed, culture of cancer cells on tumor CD-ECMs led to more physiologically relevant cancer cell phenotypes, as observed in various carcinoma (Serebriiskii et al., 2008; Eberle et al., 2011; Kaukonen et al., 2017), breast (Castelló-Cros et al., 2009; Hoshiba and Tanaka, 2013), and colon (Hoshiba, 2018) cancer



models. Increased malignancy and drug resistance of cells was observed on invasive cancer CD-ECMs, in comparison to non-invasive cancer CD-ECMs (Hoshiba and Tanaka, 2013; Hoshiba, 2018). In contrast, upon culture on MSC-derived ECM, cancer cells proliferated less (Marinkovic et al., 2016) and showed reduced tumorigenicity upon implantation (Sun et al., 2010). Differences in cancer cell behavior were attributed not only to the biochemical composition of the tumor-associated ECM, but also to changes in stiffness (Kaukonen et al., 2016; Hoshiba, 2018) and a decreased cell adhesion (Hoshiba and Tanaka, 2013).

## Engineering and Characterization of CD-ECM to Study ECM Physiology

The ease of manipulating CD-ECM *in vitro* provides the opportunity to examine the reciprocal relationship between cells and their ECM.

Biochemical ECM re-engineering could be achieved through direct addition of functional groups (Xing et al., 2015) or exogenous factors (Sart et al., 2020), genetic modification (Higuchi et al., 2010) or growth factor stimulation of ECM-synthesizing cells (Wolchok and Tresco, 2012). Other changes in culture conditions, such as hypoxic cultures, were also shown to affect ECM properties and bioactivity (Hielscher, 2013).

Mechano-physical re-engineering could be achieved by culturing ECM-secreting cells in 3D sacrificial hydrogels (Yuan et al., 2018), on micro-molds (Schell et al., 2016), and micro- and nano-grooves (Ozguldez et al., 2018; Almici et al., 2020; Yang et al., 2020), forcing cell reorganization and leading to ECM assemblies with unique architectures (i.e., parallel fiber alignment). ECM postprocessing, such as cross-linking, could further alter ECM stiffness (Subbiah et al., 2016) or the overall presentation of CD-ECM. In particular, cross-linking of pepsin-solubilized CD-ECM with genipin resulted in the formation of hydrogels (Nyambat et al., 2020). Changes in biochemical and mechano-physical properties of the ECM led to changes in gene expression and behavior of reseeded cells (Kim et al., 2015; Ozguldez et al., 2018; Sart et al., 2020).

CD-ECM characterization and correlation with specific bioactivities can contribute to the mechanistic understanding of the ECM. ECM ultrastructure can be generally studied by scanning electron microscopy or atomic force microscopy (Kaukonen et al., 2016; Raghunathan et al., 2018). The latter method can also be used for biomechanical characterization (Prewitz et al., 2013; Assunção et al., 2020). Identification of proteins of interest is best performed by antibody-based assays such as immunocytochemistry or western blotting (Sart et al., 2020). Proteomic analysis based on mass spectroscopy enables the simultaneous identification of many components, however also faces challenges based on the insolubility and high complexity of the ECM (Ragelle et al., 2017; Senthebane et al., 2018; Silva et al., 2019). Furthermore, additional methods, such as Raman microscopy, can be used for biochemical characterization (Brauchle and Schenke-Layland, 2013).

## CD-ECM Applications in TERM

CD-ECM uses for TERM have been increasingly explored, either with CD-ECM alone or integrated in biomaterials. 3D scaffolds purely composed of CD-ECM were produced by decellularizing stacked cell sheets (McAllister et al., 2009) and pellets (Zwolinski et al., 2011), or depositing ECM in sacrificial materials, such as hollow tubes (ECM fibers) (Roberts et al., 2017) and foams (ECM porous scaffolds) (Wolchok and Tresco, 2010; **Figure 1B**).

For applications that require specific mechanical properties of the biomaterials, CD-ECM was integrated with synthetic materials, forming hybrid scaffolds (Schenke-Layland et al., 2009; Carvalho et al., 2019b; Sart et al., 2020). Hybrid materials met mechanical requirements, while providing adequate biochemical stimuli, thus facilitating implant integration and functionality (Silva et al., 2020). Commonly, CD-ECM was utilized as a coating by simply decellularizing cells on the biomaterial surface (Kumar et al., 2016; Junka et al., 2020), although solubilized CD-ECM was also used as a coating (Decaris et al., 2012). A more sophisticated approach introduced azide-modified monosaccharides into culture media, which subsequently were incorporated into the ECM. The CD-ECM could then be covalently “clicked” to material surfaces (Ruff et al., 2017). Alternative approaches directly incorporated CD-ECM into the biomaterial during synthesis (e.g., electro-spinning) (Schenke-Layland et al., 2009; Carvalho et al., 2019b).

CD-ECMs based biomaterials were mainly investigated for skeletal and cardiovascular repair, although other applications such as in skin (Suhaeri et al., 2018) and peripheral nerve repair (Gu et al., 2017) were also explored.

### CD-ECM for Skeletal Repair

Most approaches to engineer CD-ECM-carrying bone implants utilized inorganic materials (reviewed in Zhang et al., 2016), such as meshes and scaffolds (Kang et al., 2012; Antebi et al., 2015; Kim et al., 2015; Jeon et al., 2016; Kumar et al., 2016; Noh et al., 2016; Junka et al., 2020; Silva et al., 2020). These were coated with ECM assembled by collagen I-overexpressing epithelial cells (Noh et al., 2016), fibroblasts (Kim et al., 2015), MSCs (Kang et al., 2011; Silva et al., 2020), endothelial cells (Kang et al., 2012), osteoblasts (Jeon et al., 2016; Kumar et al., 2016) and combinations thereof (Junka et al., 2020). CD-ECM coated scaffolds promoted attachment, proliferation, and bone-like tissue formation *in vitro*. In a more advanced approach, Kim et al. (2015) enhanced an PLGA/PLA-based mesh scaffold coated with CD-ECM, by covalently conjugating heparin to the ECM. The heparin then acted as a growth factor reservoir for bone morphogenic protein-2 (BMP2), thereby promoting bone healing *in vivo* (Kim et al., 2015). CD-ECM was also used to increase retention of osteogenically precommitted MSCs on biomaterial surfaces after implantation. This revitalized ECM successfully repaired mouse calvaria defects (Zeitouni et al., 2012).

Therapeutic approaches targeting cartilage repair mainly utilized 3D scaffolds purely composed of CD-ECM (Jin et al., 2007; Tang et al., 2013, 2014) or CD-ECM-loaded hydrogels (Yuan et al., 2013). Indeed, 3D scaffolds of chondrocyte- and MSC-derived ECM reseeded with chondrocytes induced ectopic



hyaline-like cartilage formation *in vivo* (Jin et al., 2007; Tang et al., 2013). When applied to an osteochondral defect together with bone marrow stimulation, autologous MSC-derived ECM could enhance cartilage repair (Tang et al., 2014). In another study, a protective effect on the degenerating cartilage could be demonstrated, when collagen I microspheres containing nucleus pulposus CD-ECM and MSCs were injected into a rabbit degenerative disc model (Yuan et al., 2013).

### CD-ECM for Cardiovascular Tissue Engineering and Repair

CD-ECMs have been explored as cardiac patches for cell-delivery (Schmuck et al., 2014; Kim et al., 2019), as well as to engineer heart valves replacements (Weber et al., 2013) and blood vessel grafts (McAllister et al., 2009; Syedain et al., 2014; Xing et al., 2017).

Cardiac patches were composed of fibroblast ECM alone (Schmuck et al., 2014) or combined with a polyvinyl alcohol sheet, resulting in a stretchable scaffold for cell delivery. Application of the latter in a rat myocardial infarct model resulted in improved cardiac remodeling (Kim et al., 2019).

A cardiac valve prototype containing vein-derived fibroblast ECM was implanted in a non-human primate. Albeit valve functionality was reduced, there was a significant improvement in repopulation by host cells, when compared to decellularized human heart valves (Weber et al., 2013).

McAllister et al. (2009) utilized partially devitalized autologous fibroblast/endothelial CD-ECM sheets to form vascular access grafts for dialysis patients. Complete remodeling and repopulation of CD-ECM occurred, although diffuse dilation of the graft was observed (McAllister et al., 2009).

In order to improve this low graft resistance, Syedain et al. (2014) stimulated tubular fibroblast cultures in a pulsed-flow-stretch bioreactor. Upon implantation of the decellularized graft into the femoral artery of sheep, no dilation was observed. Once completely recellularized, the grafts resembled native vessels in terms of cellular composition, ECM architecture and mechanical properties (Syedain et al., 2014).

## CONCLUSION AND OUTLOOK

Although CD-ECM was continuously explored for over three decades and many safety concerns associated with tissue-derived products can be addressed, relatively slow advancements were made over the years. This can be partially attributed to the low

amounts of CD-ECM that can be harvested *in vitro*, indicating that strategies for upscaling processes as well as manufacturing of larger 3D constructs need to be developed.

In addition, most TERM approaches used unmodified ECM from MSCs or tissue-specific cell types to induce cellular responses *in vitro* and *in vivo*. And although various approaches on how to re-engineer the CD-ECM are proposed, relatively few are applied to address scientific questions or to manufacture biomaterials with enhanced desired bioactivities. The reason for the limited progress can be partially attributed to our restricted fundamental understanding of the ECM. Hence, functional studies in combination with CD-ECM characterization will have to be adopted. Another reason is that re-engineering approaches are mainly focused on biological manipulation. Research at the interface to other disciplines such as materials science is indeed required to enable further evolvement of the CD-ECM research field. Future applications could focus on bio-inks with tailor-made bioactivities for 3D bioprinting or improved biomimetic cell niches in organ-on-a-chip approaches.

In conclusion, CD-ECM based research is far from its full potential. Advancements in methodologies as well as innovative interdisciplinary approaches are needed to pave the way for an exciting next generation of CD-ECMs for basic research and therapeutic approaches.

## AUTHOR CONTRIBUTIONS

All authors contributed to the elaboration of this review.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Primary Extracellular Matrix Enables Long-Term Cultivation of Human Tumor Oral Mucosa Models

Leonie Gronbach<sup>1</sup>, Philipp Jurmeister<sup>2,3</sup>, Monika Schäfer-Korting<sup>1</sup>, Ulrich Keilholz<sup>4</sup>, Ingeborg Tinhofer<sup>3,5</sup> and Christian Zoschke<sup>1\*</sup>

<sup>1</sup> Institute of Pharmacy (Pharmacology and Toxicology), Freie Universität Berlin, Berlin, Germany, <sup>2</sup> Institute of Pathology, Berlin Institute of Health, Humboldt-Universität zu Berlin, Corporate Member of Freie Universität Berlin, Charité – Universitätsmedizin Berlin, Berlin, Germany, <sup>3</sup> Heidelberg and German Cancer Consortium Partner Site Berlin, German Cancer Research Center, Berlin, Germany, <sup>4</sup> Comprehensive Cancer Center, Berlin Institute of Health, Humboldt-Universität zu Berlin, Corporate Member of Freie Universität Berlin, Charité – Universitätsmedizin Berlin, Berlin, Germany, <sup>5</sup> Department of Radiooncology and Radiotherapy, Berlin Institute of Health, Humboldt-Universität zu Berlin, Corporate Member of Freie Universität Berlin, Charité – Universitätsmedizin Berlin, Berlin, Germany

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### \*Correspondence:

Christian Zoschke  
christian.zoschke@fu-berlin.de

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3D tumor models clearly outperform 2D cell cultures in recapitulating tissue architecture and drug response. However, their potential in understanding treatment efficacy and resistance development should be better exploited if also long-term effects of treatment could be assessed *in vitro*. The main disadvantages of the matrices commonly used for *in vitro* culture are their limited cultivation time and the low comparability with patient-specific matrix properties. Extended cultivation periods are feasible when primary human cells produce the extracellular matrix *in situ*. Herein, we adapted the hyalograft-3D approach from reconstructed human skin to normal and tumor oral mucosa models and compared the results to bovine collagen-based models. The hyalograft models showed similar morphology and cell proliferation after 7 weeks compared to collagen-based models after 2 weeks of cultivation. Tumor thickness and VEGF expression increased in hyalograft-based tumor models, whereas expression of laminin-332, tenascin C, and hypoxia-inducible factor 1 $\alpha$  was lower than in collagen-based models. Taken together, the *in situ* produced extracellular matrix better confined tumor invasion in the first part of the cultivation period, with continuous tumor proliferation and increasing invasion later on. This proof-of-concept study showed the successful transfer of the hyalograft approach to tumor oral mucosa models and lays the foundation for the assessment of long-term drug treatment effects. Moreover, the use of an animal-derived extracellular matrix is avoided.

**Keywords:** extracellular matrix, head and neck cancer, oral mucosa, personalized medicine, tissue engineering, tumor microenvironment, long-term cultivation, Hyalograft 3D

## INTRODUCTION

Stromal, endothelial, and immune cells create a unique environment for each individual tumor with altered paracrine signaling compared to the normal tissue (Zheng and Gao, 2019). This cellular tumor microenvironment can promote tumor growth, invasion, and dissemination (Varol, 2019) as well as treatment resistance (Jo et al., 2018). The impact of the extracellular matrix

(ECM) as the major component of the tumor microenvironment in these biological processes remains contradictory or unexplored (Pickup et al., 2014; Saggiaro et al., 2020). Commonly, tumors dysregulate the composition and structure of the surrounding normal tissue toward an inflamed, hypoxic, and desmoplastic tumor microenvironment (Zheng and Gao, 2019). The effect of the tumor environment on the biology of tumors of the oral cavity remains to be investigated.

Patient-specific tumor ECMs are rarely recapitulated *ex vivo*. Tumor cells are either cultivated in scaffold-free ultra-low attachment plates or embedded in collagen of animal origin, e.g., Matrigel (Langhans, 2018). Furthermore, non-human matrices like cellulose are used as scaffolds for *ex vivo* tumor models (Nath and Devi, 2016). Major drawbacks of these approaches include poor stability, limited lifespan, and underrepresentation of patient-specific tumor microenvironment components. Initially designed to better reconstruct human skin, the hyalograft-3D is a biodegradable, non-immunogenic scaffold, which consists of esterified hyaluronic acid fibers. It is certified for medical use and allows the fibroblasts to produce and assemble their own ECM (Campoccia et al., 1998). Thereby, hyalograft-based skin models extended the life by six times, compared to collagen-based skin models (Stark et al., 2006).

Recently, we developed normal and tumor oral mucosa models emulating head and neck cancer, with a collagen scaffold (Gronbach et al., 2020) to improve non-clinical drug evaluation. The 3D model showed large similarities in morphology, grading, and protein expression profiles to patient's tumors. Moreover, the tumor models recapitulated docetaxel and cetuximab effects in line with clinical observations of head and neck-cancer. However, the cultivation of the collagen-based tumor models for a maximum of 2 weeks enabled only the investigation of short-term drug effects. This represents a major limitation for studies investigating the impact of genetic heterogeneity and therapy-driven clonal evolution in acquired drug resistance in the tumor (Magdeldin et al., 2014; Braig et al., 2017).

Herein, we assessed whether by using the hyalograft-3D approach human tumor oral mucosa models could be maintained in *ex vivo* cultures for up to 7 weeks, without major changes in tumor cell viability and proliferative activity. In addition, the impact of the ECM on tumor growth and invasion in hyalograft-based tumor oral mucosa models was compared with their collagen-based counterparts.

## MATERIALS AND METHODS

### Materials

Collagen G, DMEM 10× and HEPES buffer were purchased from Merck (Darmstadt, Germany). Hyalograft-3D was purchased from Anika Therapeutics (Bedford, MA, United States). The thrombin-fibrinogen-solution tisseel® was purchased from Baxter (Deerfield, IL, United States).

Human oral keratinocytes and human oral fibroblasts, as well as the respective cell culture media were purchased from ScienCell (Carlsbad, CA, United States). The tumor-cell line SCC-25 from the tongue (RRID: CVCL\_1682, Rheinwald and

Beckett, 1981) was a generous gift from Howard Green, Dana-Farber Cancer Institute (Boston, MA, United States). The detailed composition and origin of the construct growth and construct differentiation media were described elsewhere (Gronbach et al., 2020). Here, these media were supplemented with the transforming growth factor (TGF)-β1 and aprotinin, obtained from ThermoFisher Scientific (Waltham, MA, United States) and Merck. 12-well plates and 12-well inserts (0.4 μm pore size) were obtained from Greiner bio-one (Leipzig, Germany).

Hematoxylin, eosin, rothistol, and rothistokit were purchased from Carl Roth (Karlsruhe, Germany). Periodic acid was from Sigma-Aldrich and Schiff's reagent was obtained from Merck. Primary antibodies were purchased from abcam (Cambridge, United Kingdom): hypoxia-inducible factor 1α (1:200; RRID: AB\_880418), Ki-67 (1:100; RRID: AB\_302459), laminin-332 (1:500; RRID: AB\_1566368), Tenascin C (1:1000; RRID: AB\_2043021), vascular endothelial growth factor (1:200; RRID: AB\_299738). Cytokeratin Pan Plus KL1 antibody (1:100; RRID: AB\_2864507) was from Zytomed (Berlin, Germany). Anti-mouse and anti-rabbit IgGs (H + L), with F(ab')<sub>2</sub> Fragment (Alexa Fluor® 488 and 594 Conjugate; RRIDs: AB\_1904025, AB\_2714182) were obtained from Cell Signaling Technology (Danvers, MA, United States). DAPI (4',6-Diamidin-2-phenylindol) mounting medium was purchased from dianova (Hamburg, Germany). The *in situ* cell death detection kit (TUNEL assay) was purchased from Sigma-Aldrich (Munich, Germany).

### Cell Culture

Human oral keratinocytes and human oral fibroblasts (ScienCell) were cultured in oral keratinocyte and fibroblast medium, respectively, at 37°C with 5% CO<sub>2</sub>. The SCC-25 tumor-cell line was grown in DMEM/F-12 Ham medium, supplemented with 9% fetal calf serum, 0.9% L-glutamine, and penicillin/streptomycin. The medium was changed three times a week and the cells were passaged after reaching confluency of 80%. The cell line was tested for mycoplasma and regularly checked by single nucleotide polymorphism authentication (Multiplexion; Heidelberg, Germany). Cell culture was performed according to standard operating procedures and referred to good cell culture practice.

### Multilayered Oral Mucosa Model Building

The multi-layered oral mucosa models (Figure 1A) were constructed as a lamina propria growing underneath an epithelium. All cultures were kept at 37°C and 5% CO<sub>2</sub> in a humidified atmosphere. The building of collagen-based oral mucosa models was described previously (Gronbach et al., 2020). Briefly, 1 × 10<sup>5</sup> human oral fibroblasts per model were mixed with a buffered solution and added to collagen. After solidification of the matrix, construct growth medium was added to the model and changed three times until day 7. Thereafter, either 1 × 10<sup>6</sup> human oral keratinocytes or 1 × 10<sup>6</sup> SCC-25 cells were seeded onto the lamina propria compartment for normal or tumor oral mucosa models, respectively. From day 14, the construct surface was kept medium-free to expose the epithelium to the air and the construct growth medium was supplemented

with 0.25 mmol/l ascorbic acid acting as construct differentiation medium. On day 21, the models were snap frozen and stored at  $-80^{\circ}\text{C}$ .

The generation of hyalograft-3D was described previously (Stark et al., 2006). In brief, the hyalograft-3D is a fleece-like matrix, composed of recombinant human hyaluronic acid fibers, esterified with benzylic alcohol to retard its degradation. Here, hyalograft-3D was cut into disks of 10 mm in diameter to fit the size of 12-well cell culture inserts. Next,  $1 \times 10^5$  human oral fibroblasts per model were resuspended in a thrombin solution (10 international units/ml), mixed with a fibrinogen solution (8 mg/ml) and subsequently added to the pre-cut hyalograft-3D pieces. During the following 7 days, the fibroblasts were allowed to replace the fibrin by *in situ* produced ECM components (Stark et al., 2006). Thereafter, either human oral keratinocytes or SCC-25 cells were seeded onto the lamina propria compartment as described above for the collagen model. The construct growth medium was supplemented with 1 ng/ml transforming growth factor- $\beta 1$  and 500 international units/ml aprotinin. TGF- $\beta 1$  reduces keratinocyte differentiation and growth (Dahler et al., 2001). Aprotinin, a serin-protease inhibitor was used to limit fibrinolysis and thus premature model degradation. Medium was changed three times per week. From day 14, the construct surface was kept medium-free and aprotinin was reduced to 200 international units/ml in the construct differentiation medium. At the end of the cultivation period, the models were snap frozen and stored at  $-80^{\circ}\text{C}$ .

## Morphology and Protein Expression

The models were cut into 7  $\mu\text{m}$  thick slices using a cryotome (Leica CM 1510S; Leica, Wetzlar, Germany) and fixed with 4% paraformaldehyde. The cryosections were subjected to either hematoxylin and eosin (H&E), periodic acid-Schiff (PAS), immunofluorescence staining or immunohistochemistry (IHC). For the H&E staining, slides were successively submerged into hematoxylin (5 min), water (5 min), eosin (30 s), 70 and 99.9% ethanol (2 min) and rotihistol (2 min). Finally, the slides were fixed with the rotihistokit and a cover slide. PAS staining was performed on a Tissue-Tek Prisma Plus Automated Slide Stainer (Sakura Finetech, Staufen, Germany). Slides were incubated with periodic acid for 10 min, followed by staining with Schiff's reagent for 10 min and hematoxylin for 7 min. For immunofluorescence staining, the samples were permeabilized for 5 min by a 0.5% triton solution, blocked for 30 min with 5% goat serum and incubated over night with the primary antibody at  $4^{\circ}\text{C}$ . Afterward the slides were incubated for 1 h with the secondary antibody. In the end, DAPI mounting medium was added to stain cell nuclei and fixed the samples. IHC staining was done on a BOND MAX Automated Slide Stainer (Leica) using the HP1 program and the BOND polymer Refine Detection System (Leica). Images were taken with a fluorescence microscope (BZ-8000; Keyence, Neu-Isenburg, Germany) and analyzed using the ImageJ software (Schneider et al., 2012).

## Apoptosis Quantification

For apoptosis measurements, the *in situ* cell death detection kit was used according to the manufacturer's instructions. The kit

detects DNA fragments in apoptotic cells based on TdT-mediated dUTP-biotin nick end labeling (TUNEL).

## Data Analysis

Data are presented as the mean + standard deviation (SD) obtained from up to three independent experiments. Due to the explorative data analysis, a level of  $p \leq 0.05$ , calculated using non-parametric Kruskal-Wallis tests and subsequent Dunn's *Post hoc*-tests, was considered to indicate a statistically significant difference.

## RESULTS

### Morphological Analysis

We extended the culture period from 2 weeks of collagen-based normal oral mucosa models (c-NOM) and tumor oral mucosa models (c-TOM) to 7 weeks in hyalograft-based h-NOM and h-TOM models. To evaluate the impact of the scaffold, we cultured also h-NOM and h-TOM for 2 weeks (Figure 1A).

The epithelium of c-NOM models consisted of a basal layer with rounded cells and multiple layers of spinous cells, as found in non-keratinized oral mucosa (Figure 1B). All TOM models depicted an unstructured, hyperproliferative, and thickened epithelial layer with atypical, enlarged, irregular tumor cells and hyperchromatic nuclei. The tumor morphology appeared desmoplastic in particular in h-TOM models after 7 weeks of culture (Figure 1B, inserts).

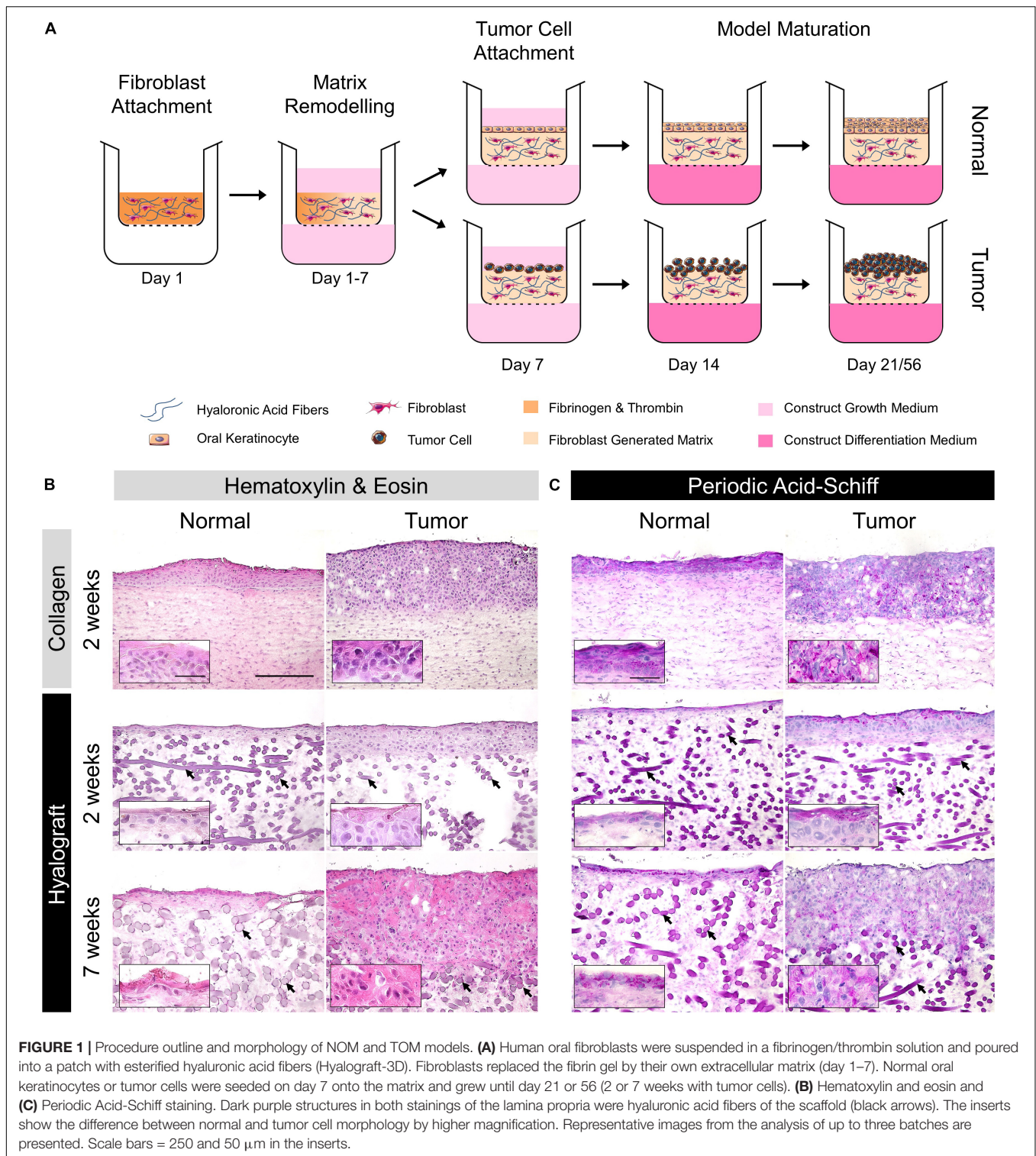
The glycogen distribution was confined to the upper epithelial layers of the h-NOM model, while glycogen was found in all epithelial layers of c-NOM models (Figure 1C). A similar pattern was observed in TOM models after 2 weeks of culture (Figure 1C, inserts). Only after 7 weeks of culture the glycogen distribution became also patchy in h-TOM models. Concurrently, cytokeratin-positive tumor cells penetrated the hyalograft-3D matrix only slightly as tumor nests, but massively invaded the lamina propria compartment as single cells (Figure 2A). The final tumor thickness in h-TOM models exceeded tumor thickness of c-TOM models, but the difference was not statistically significant (Figure 2B).

The large structures in the lamina propria of hyalograft-based models were hyaluronic ester fibers, which were unspecifically stained by hematoxylin and eosin, periodic acid-Schiff as well as DAPI. The unspecific staining might be explained by the large three-dimensional structure of the fibers and their negative charge, which prevented the washout of stains as well as monoclonal antibodies, and led to the intercalation of DAPI into the fibers.

### Protein Expression

The basement membrane protein laminin-332 was expressed in particular between the epithelial layer and the lamina propria in both h-NOM and c-NOM models (Figure 2C). In contrast, the expression of laminin-332 was more heterogeneous in TOM models with the highest levels in h-TOM models after 2 weeks of culture (Figure 2D), in particular observed in the subepithelial zone in h-TOM models.

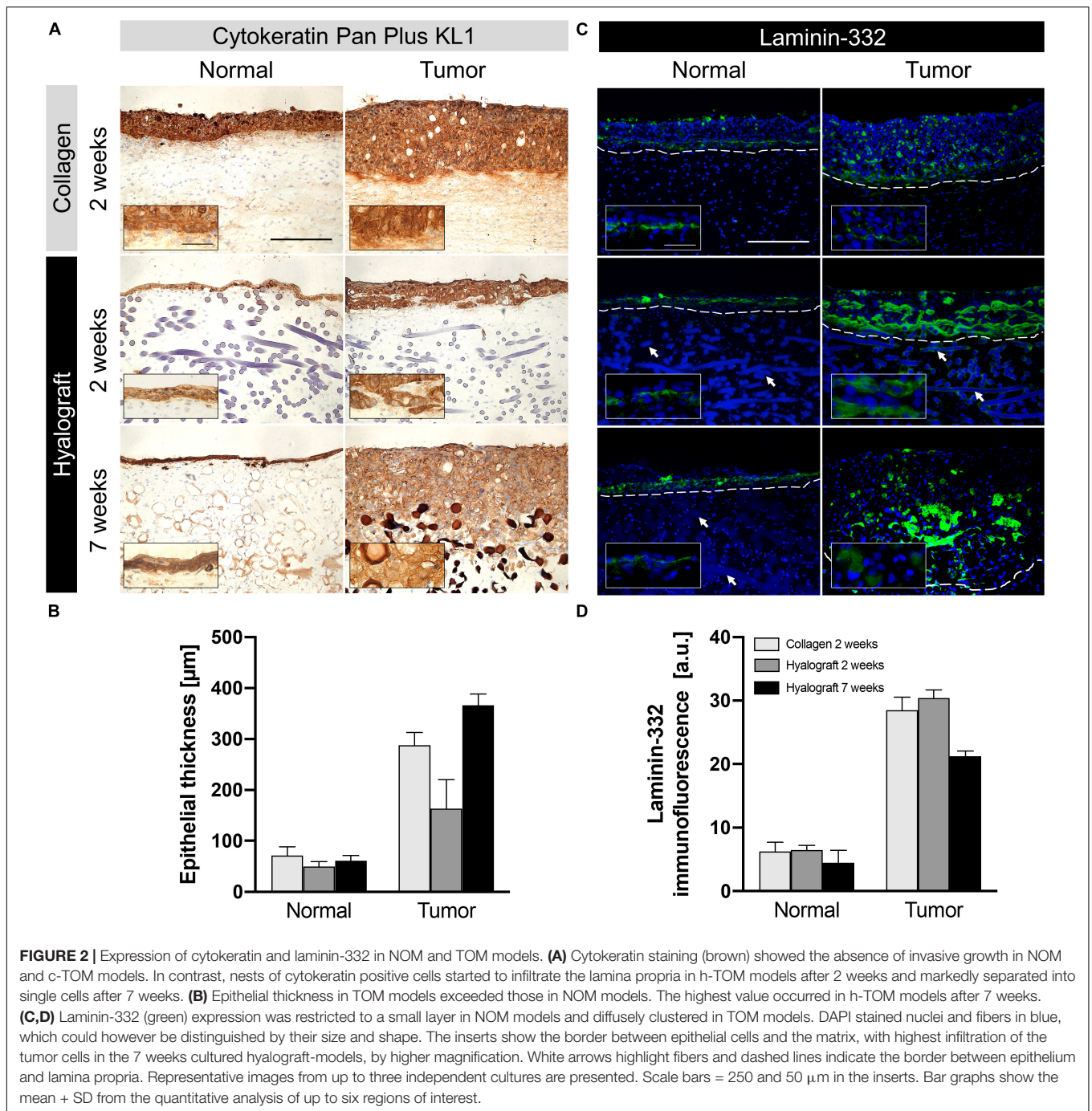




The extracellular matrix protein tenascin C was most abundant in collagen-based models with no difference between NOM and TOM models (Figure 3A). Tenascin C expression markedly decreased in hyalograft-based models already after 2 weeks of cultivation and further

declined to 33% ( $p > 0.05$ ) after 7 weeks. Again, no relevant difference between NOM and TOM models was observed (Figure 3B).

The hypoxia-inducible factor (HIF)-1 $\alpha$  was detected in the entire tumor mass of c-TOM models, and particularly in



central tumor areas in h-TOM models (Figure 3C). Very low levels of HIF-1 $\alpha$  were detected in both c-NOM and h-NOM models (Figure 3D).

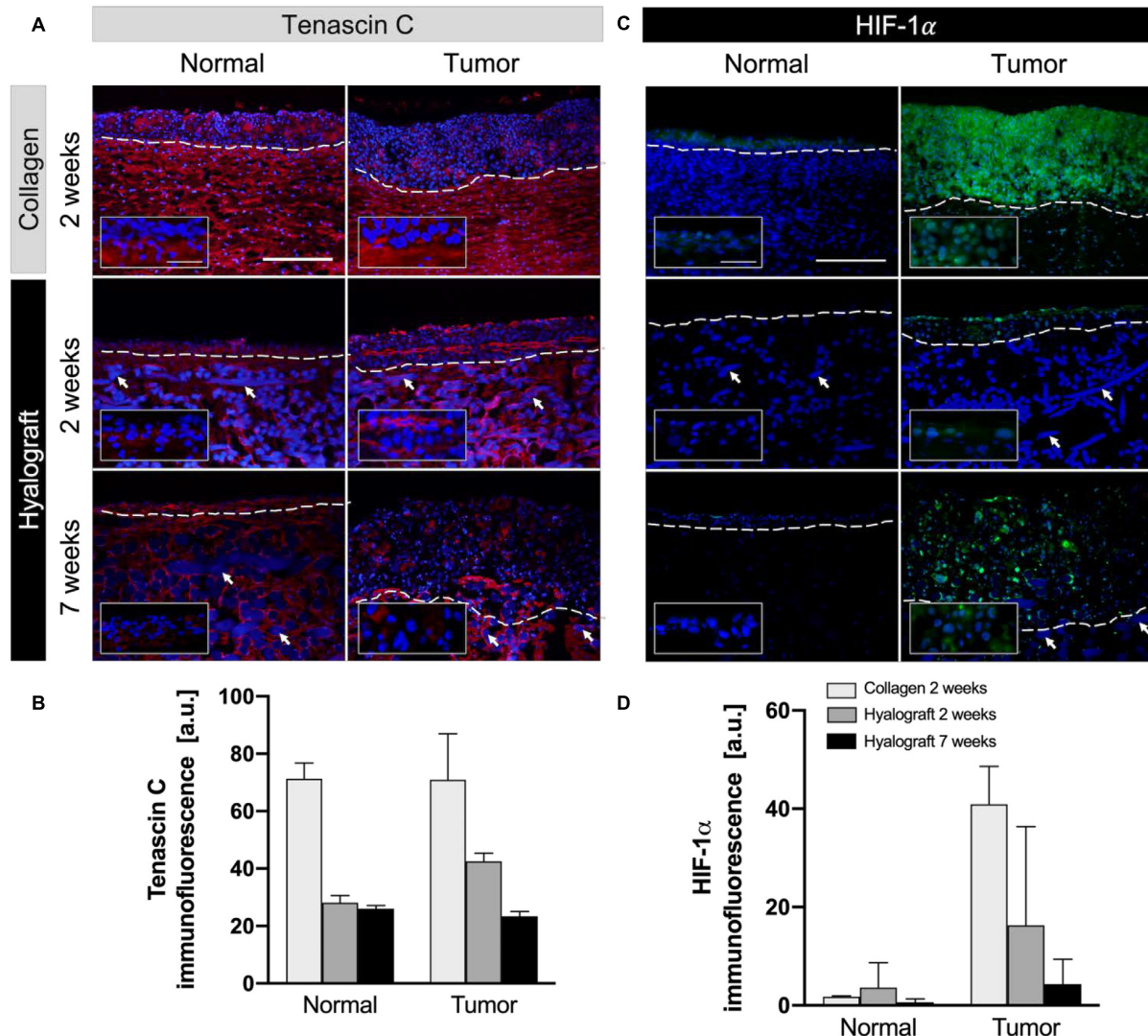
Overall, vascular endothelial growth factor (VEGF) was expressed at similar levels in the c-TOM and h-TOM models; however, the type of matrix interfered with its localization. While VEGF was detected in the entire tumor areas of c-TOM models, it was restricted to the border between the tumor layer and the lamina propria in h-TOM models (Figures 4A,B). VEGF expression further increased after 7 weeks both in

NOM and TOM models ( $p > 0.05$ ). Increased VEGF levels were particularly observed close to hyaluronic acid fibers in h-NOM models.

## Proliferation and Apoptosis

Proliferation was higher in tumor compared to normal models, irrespective of the used matrix (Figures 4C,D). Importantly, tumor cells continued to proliferate excessively in h-TOM models until the end of the 7-week culture within all regions of the culture.





**FIGURE 3 |** Expression of tenascin c and HIF-1 $\alpha$  in NOM and TOM models. **(A,B)** Tenascin c (red) was less expressed in hyalograph-based models than collagen-based models and further decreased during cultivation. **(C,D)** HIF-1 $\alpha$  (green) showed increased expression in TOM models, with matrix-dependent differences. DAPI stained nuclei and fibers in blue. The inserts show the border between epithelial cells and the matrix for the tenascin C staining and highlights of the HIF-1 $\alpha$  staining by higher magnification. White arrows highlight fibers and dashed lines indicate the border between epithelium and lamina propria. Representative images from up to three independent cultures are presented. Scale bar = 250 and 50  $\mu$ m in the inserts. Bar graphs show the mean + SD from the quantitative analysis of up to six regions of interest of interest.

In c-NOM and the 7 weeks cultured h-NOM models, only few apoptotic cells could be detected (mean = 2.5%), while in the 2 weeks h-NOM models apoptotic cells made up to 20% of the epithelial cells (**Supplementary Figure 1**).

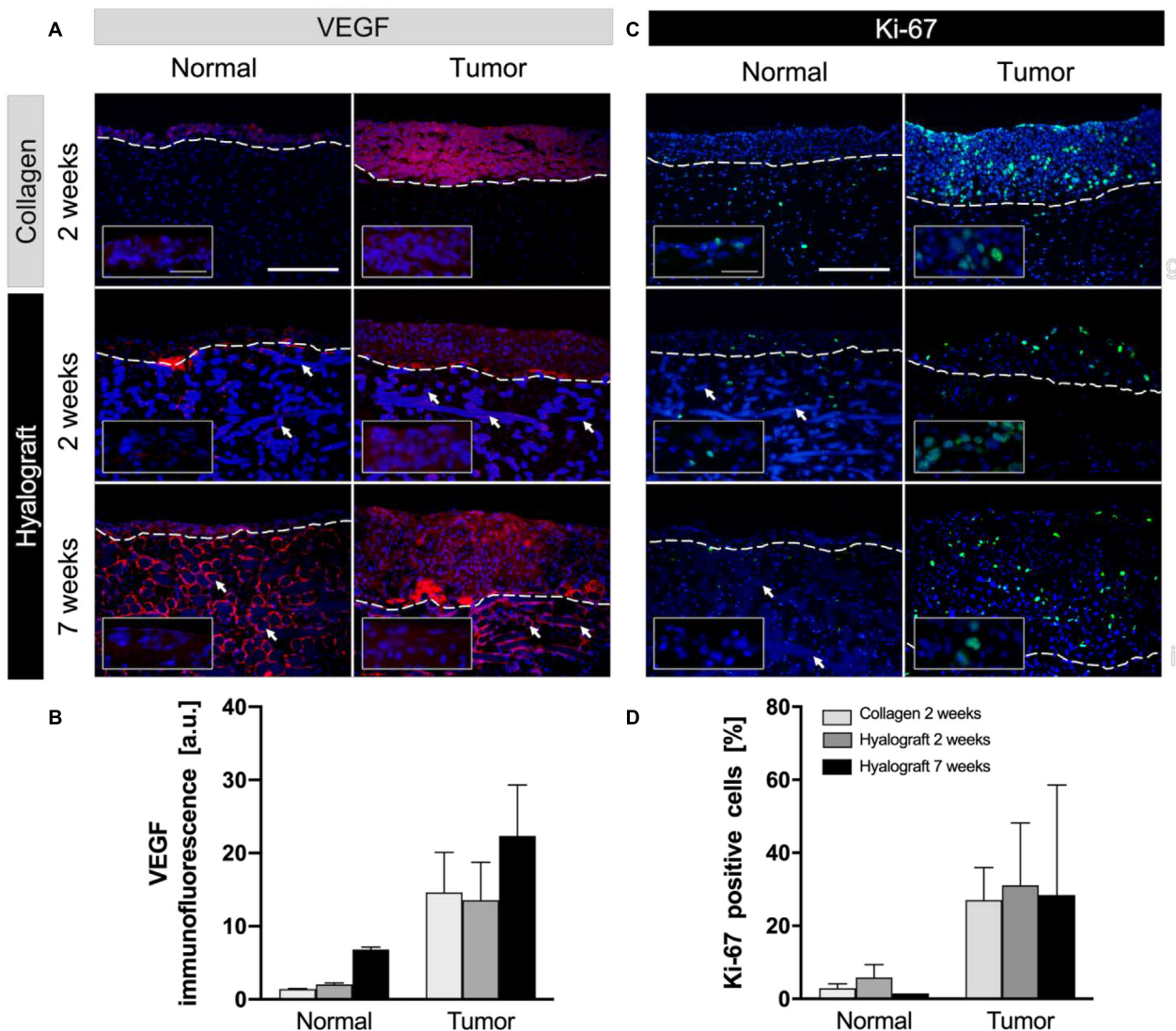
In TOM models, both 2 weeks cultured c-TOM and h-TOM models depicted less than 5% apoptotic cells, but the 7 weeks cultured h-TOM models showed up to 30% apoptotic cells.

## DISCUSSION

We here showed that normal and tumor oral mucosa models can be successfully cultured in a hyalograph-based scaffold, allowing

extended *ex vivo* cultivation. Our data corroborate previous findings showing that hyaluronic acid and its derivatives provide a well-defined and tunable scaffold for *ex vivo* tumor models (Fong et al., 2014). Moreover, hyalograph-based models are not affected by the poor adhesion of epithelial layers and the tendency to shrink of collagen-based models (Stark et al., 2004). In contrast, nylon-meshes and collagen-chitosan-sponges, which have been tested for elongated cultivation periods have the disadvantage of requiring long pre-cultivation and displaying considerable stiffness, thus complicating tissue sectioning and analysis (Michel et al., 1999; Stark et al., 2006).

Hyalograph-based tumor models contained high numbers of proliferative cells and recapitulated hallmarks of oral cancer



**FIGURE 4 |** Expression of vascular endothelial growth factor (VEGF) and proliferation (Ki-67) in NOM and TOM models. **(A,B)** Highest expression of VEGF (red) was observed in h-TOM models after 7 weeks. **(C,D)** The number of proliferative cells (Ki-67 positive, green) was increased in TOM models until the end of the cultivation period compared to NOM models. The inserts highlight detected VEGF and Ki-67 in the epithelial layers by higher magnification. Representative images from up to three independent cultures are presented. Scale bars = 250 and 50  $\mu\text{m}$  in the inserts. Bar graphs show the mean + SD from the quantitative analysis of up to six regions of interest.

even after a cultivation period of 7 weeks. In particular, increased epithelial thickness, abundant cellular pleomorphism, and the altered laminin-332 expression, very well reflected the histopathological characteristics of patient tumors (Miyazaki, 2006; Bernstein et al., 2013; Jerjes et al., 2019). Thus, h-TOM models should be suitable to monitor long-term tumor progression as well as the effects of anti-proliferative drugs and the potential tumor re-growth after an initial treatment cycle. An improved understanding of the re-growth kinetics after drug treatment would help to overcome drug resistance, which is currently the major cause of treatment failure (Vasan et al., 2019).

Beside large similarities in protein expression patterns of hyalograft- and collagen-based models, there was a significant

difference in tenascin C expression. The increased expression of tenascin C in collagen-based models might explain the faster growth in the epithelial layers of both c-NOM and c-TOM models, since tenascin C is known as a provisional matrix for keratinocyte growth (Pellegrini et al., 1999). Moreover, the expression of the extracellular matrix proteins tenascin c and fibronectin discriminates low- and high-risk tongue cancers (Sundquist et al., 2017). Low tenascin C expression in the h-TOM model established from SCC-25 cells is in line with the previously described poorly invasive phenotype of this cell line model (Ramos et al., 1997).

Normal oral fibroblasts better confined tumor invasion in hyalograft- than in collagen-based models after 2 weeks.



This difference might be related to paracrine signaling between fibroblasts and tumor cells which has been shown to depend on the composition of the ECM (Barcellos-Hoff and Bissell, 1989; Boudreau and Bissell, 1998). Laminin-332 appears to play a key part in the invasion process, in line with its higher expression in h-TOM compared to c-TOM models. While a well-defined laminin-332 expression is typical for normal tissues, clustered laminin-332 expression is known to promote cell survival and tumorigenesis, especially in squamous cell carcinoma (Marinkovich, 2007). In addition, the occurrence of desmoplasia in h-TOM models might contribute to delayed invasive growth and reduced hypoxia compared to c-TOM models. These differences in the ECM of collagen- and hyalograft-based models need to be considered in evaluating drug effects since hypoxia reduces the clinical efficacy of anticancer drugs (Brennan et al., 2005; Johnstone and Logan, 2006).

Although this proof-of-concept study shows the suitability of the hyalograft scaffold for the *ex vivo* cultivation of TOM models, future studies need to elucidate the scaffold effects on patient-derived tumor cells and compare these results to *in vivo* tumors. One limitation of the current h-TOM model is the relative high percentage of apoptotic tumor cells in long-term cultures. Further approaches for model improvement in the future might thus include also testing of additional supplements to the construct growth medium. Moreover, future studies will show whether the hyalograft approach better recapitulates the interaction of immune and tumor cells in an immunocompetent model of oral mucosa tumor, which seems very likely since the scaffold is non-immunogenic (Galassi et al., 2000). Given their close correlation to the individual tumor, long-term cultivation of human TOM models offer the opportunity to study tumor re-growth and alterations in the tumor stroma after initial treatment and thus will help to better understand drug resistance mechanisms.

## CONCLUSION

The hyalograft-3D approach recapitulated key features of human oral squamous cell carcinoma in multi-layered *ex vivo* tumor models for up to 7 weeks. The long-term cultivation provides

the basis for studying tumor re-growth and stromal alterations following an initial anti-cancer drug therapy. Moreover, the well-defined and tunable hyaluronic acid derivatives might help to better culture patient-derived cells. Finally, hyalograft-based models can be extended by the addition of further tumor stroma components and relinquish the use of animal-based scaffolds.

## DATA AVAILABILITY STATEMENT

The raw data supporting the conclusions of this article will be made available by the authors on reasonable request.

## AUTHOR CONTRIBUTIONS

LG and CZ: conceptualization and design. LG, PJ, and CZ: investigation. CZ: project administration. MS-K, UK, and CZ: supervision. LG and CZ: visualization. LG and CZ: writing-original draft preparation. LG, MS-K, IT, and CZ: writing-review and editing. All authors contributed to the article and approved the submitted version.

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The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fbioe.2020.579896/full#supplementary-material>

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Cartilage Extracellular Matrix Scaffold With Kartogenin-Encapsulated PLGA Microspheres for Cartilage Regeneration

Yanhong Zhao<sup>1,2\*</sup>, Xige Zhao<sup>1,2†</sup>, Rui Zhang<sup>1,2</sup>, Ying Huang<sup>1,2</sup>, Yunjie Li<sup>1,2</sup>, Minhui Shan<sup>1,2</sup>, Xintong Zhong<sup>1,2</sup>, Yi Xing<sup>1,2</sup>, Min Wang<sup>1,2</sup>, Yang Zhang<sup>3</sup> and Yanmei Zhao<sup>4\*</sup>

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### \*Correspondence:

Yanhong Zhao  
leafzh@126.com  
Yanmei Zhao  
zhaoyanmei@126.com

<sup>†</sup> These authors have contributed  
equally to this work

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<sup>1</sup> Stomatological Hospital of Tianjin Medical University, Tianjin, China, <sup>2</sup> Tianjin Medical University, Tianjin, China, <sup>3</sup> Tianjin Hospital, Tianjin, China, <sup>4</sup> Institute of Disaster Medicine, Tianjin University, Tianjin, China

Repair of articular cartilage defects is a challenging aspect of clinical treatment. Kartogenin (KGN), a small molecular compound, can induce the differentiation of bone marrow-derived mesenchymal stem cells (BMSCs) into chondrocytes. Here, we constructed a scaffold based on chondrocyte extracellular matrix (CECM) and poly(lactic-co-glycolic acid) (PLGA) microspheres (MP), which can slowly release KGN, thus enhancing its efficiency. Cell adhesion, live/dead staining, and CCK-8 results indicated that the PLGA(KGN)/CECM scaffold exhibited good biocompatibility. Histological staining and quantitative analysis demonstrated the ability of the PLGA(KGN)/CECM composite scaffold to promote the differentiation of BMSCs. Macroscopic observations, histological tests, and specific marker analysis showed that the regenerated tissues possessed characteristics similar to those of normal hyaline cartilage in a rabbit model. Use of the PLGA(KGN)/CECM scaffold may mimic the regenerative microenvironment, thereby promoting chondrogenic differentiation of BMSCs *in vitro* and *in vivo*. Therefore, this innovative composite scaffold may represent a promising approach for acellular cartilage tissue engineering.

**Keywords:** cartilage tissue engineering, decellularized cartilage extracellular matrix, poly(lactic-co-glycolic acid), composite scaffold, kartogenin

## INTRODUCTION

Because of its avascular, alymphatic, and aneural properties, articular cartilage has poor spontaneous repair and regeneration capabilities (Malfait, 2016). The widespread articular cartilage injury caused by trauma, infection, osteoarthritis, tumors, and other diseases is a challenging aspect of clinical treatment (Huang et al., 2016). Traditional treatment methods (e.g., mosaicplasty, autologous chondrocyte transplantation, and microfracture surgery) have a number of limitations, including poor repair effect and additional trauma (Loeser et al., 2016). In the past decade, tissue engineering, especially cell-free tissue engineering, which involves scaffold implantation and microfracture creation to promote the recruitment of endogenous cells has been studied to overcome these limitations (Zhen et al., 2013; Trounson and McDonald, 2015).

Stimulating factors are among the most basic elements for cell-free tissue engineering. Previous studies have focused on cartilage stimulation methods that support the capacity for self-repair. Signaling molecules, such as transforming growth factor beta-3 (TGF- $\beta$ 3), have been widely used as chondrogenic modulators (Mi et al., 2003; van der Kraan and van den Berg, 2007; Cals et al., 2012; Almeida et al., 2014). Kartogenin (KGN), a non-toxic and stable small molecule first described by Johnson et al. (2012) is sufficiently small to avoid immune responses; it demonstrates good stability, cost-effectiveness, and ease of storage and transportation, compared to signaling molecules (e.g., TGF- $\beta$ 3). During mesenchymal stem cell (MSC) differentiation, KGN frees CBF $\beta$ , which binds to the transcription factor RUNX1. The resultant CBF $\beta$ -RUNX1 complex plays a critical role in the activation of cartilage matrix transcription; therefore, KGN has great potential for use in cartilage regeneration (Wu et al., 2014). Furthermore, KGN has protective effects with respect to existing cartilage tissue and reduces the decomposition of chondrogenic proteins; thus, it slows the progression of cartilage injury (Cai et al., 2019). Although KGN has been investigated as a cartilage-promoting molecule, two common models of KGN application remain problematic: (1) direct injection into the joint cavity, (Xu et al., 2015; Kwon et al., 2018) which is ineffective because most KGN is absorbed by the circulatory system and repeated injections increase both pain and risk of infection; (2) combination of KGN with drug delivery system: such as hydrogel (Wang S.J. et al., 2019), platelet-rich plasma (PRP) (Liu et al., 2019), nanoparticles, or microspheres (Kang et al., 2014). Although several drug delivery systems have been used to achieve sustained release of KGN, the potential application of KGN in cartilage tissue engineering is seldom investigated, and additional mesenchymal stem cells implantation are needed (Li et al., 2016; Wang J. et al., 2019). Therefore, methods for application of KGN would need to be further studied. Previous studies have shown that scaffolds loaded with drug-encapsulated microspheres can achieve sustained drug release and ensure that cells within the scaffold maintain high levels of activity (Gupta et al., 2017; Zhang et al., 2017). Among the most attractive polymer candidates, poly(lactic-co-glycolic acid) (PLGA)—a typical compound used in fabrication of biodegradable microspheres or nanoparticles, which has been approved for use by the Food and Drug Administration—is widely used as a vehicle for delivery of pharmaceutically relevant payloads (Danhier et al., 2012; Fernando et al., 2018). Furthermore, the amino groups of PLGA can bind covalently to the carboxyl groups of KGN, thus enhancing the permeability and retention of KGN and promoting its therapeutic effects (Kang et al., 2014; Fan et al., 2018; Chen et al., 2020).

Another critical component of cell-free tissue engineering comprises the biomaterial scaffold. Decellularized extracellular matrix (ECM) derived from cartilage (CECM) can preserve natural ECM components and avoid immune rejection. Therefore, it has been used as a scaffold for cartilage regeneration (Sutherland et al., 2015; Monibi and Cook, 2017). CECM scaffold can provide a suitable microenvironment for the cells it contains; moreover, it exhibits natural adhesion to the cells and growth factors. Notably, CECM is biodegradable, which allows

the scaffold to participate in the construction of new cartilage tissue. All of these characteristics promote cell proliferation and functional expression (Yang et al., 2008; Ravindran et al., 2015; Oh et al., 2018; Wang et al., 2018). However, its rapid degradation rate and weak mechanical performance limit the application of ECM scaffold in tissue engineering (Oh et al., 2018). Recently, the introduction of composite scaffolds based on ECM and polymers has provided mechanically and chemically stable properties, while retaining biocompatibility and biodegradability; this offers a new strategy for application of the CECM scaffold. In our previous study, we designed a three-layer biomimetic porous scaffold based on CECM, nano-hydroxyapatite, and silk fibroin. PLGA microspheres encapsulating growth factor rhBMP-2 and TGF- $\beta$ 3 were loaded into the three-layer scaffold to form a slow-release biomimetic osteochondral scaffold. This composite scaffold was able to effectively repair osteochondral defects (Dong et al., 2020).

Here, we fabricated cartilage-derived ECM scaffolds that contained PLGA microspheres or nanoparticles for use in controlled delivery of KGN; the constructs were designed to promote cartilage repair by means of simple procedures. Without the requirement for cell transplantation, this system could recruit endogenous BMSCs from the bone marrow cavity through microfracture apertures. This study was performed to: (1) characterize the ability of the PLGA(KGN)/CECM composite scaffold to support sustained release and chondrogenic differentiation ability *in vitro*, and (2) to evaluate the effects of implantation of PLGA(KGN)/CECM composite scaffold on articular cartilage defects *in vivo*.

## MATERIALS AND METHODS

### Materials and Reagents

CECM was obtained from fresh porcine articular cartilage by decellularization, in accordance with methods described previously (Yang et al., 2008). Other materials, including biochemical and chemical reagents, are listed in the supporting information.

### Preparation of KGN-Encapsulating PLGA Microspheres and Nanoparticles

PLGA microspheres encapsulating KGN were prepared by the solid-oil-water double solvent evaporation technique. Briefly, PLGA (MedChemexpress, NJ, United States) was dissolved in dichloromethane to approach a final mass fraction of 5%. KGN was dissolved in dimethyl sulfoxide and then mixed with 5% PLGA. The mixture was then added dropwise into 1% polyvinyl alcohol (PVA) solution at an oil phase:water phase volume ratio of 1:20 and emulsified at 16,000 rpm for 3 min, using a high-shear homogenizer in an ice bath. The suspension was then transferred to a magnetic stirrer and stirred overnight at room temperature. Finally, the suspension was centrifuged and the sediments were washed with deionizer water several times. KGN-PLGA microspheres were obtained after the sediments had been freeze-dried for 24 h and stored at  $-20^{\circ}\text{C}$ . PLGA nanospheres were prepared by ultrasonication. First, 2.5% PLGA solution



containing KGN was added in a dropwise manner to 2.5% PVA solution at an oil phase: water phase volume ratio of 1:5; it was then emulsified by ultrasonication (500 W, 3 min) to prepare an oil/water emulsion. Subsequently, the ethyl acetate was extracted by adding six volumes of 0.5% PVA aqueous solution to the emulsion. Subsequent steps were performed as described for the preparation of PLGA microspheres.

### Morphology Observations and Size Distribution Measurements of Microspheres and Nanoparticles

The freeze-dried PLGA microspheres were uniformly coated on the surface of the conductive adhesive of the sample table. The shape and surface morphology of PLGA microspheres were observed by scanning electron microscope (SEM; Quanta 200; FEI, Hillsboro, OR, United States). To determine the morphology of the PLGA nanoparticles, the nanoparticles were resuspended by deionized water and dripped on a copper mesh and then visualized by transmission electron microscopy (TEM; HT7700 Exalens, Hitachi, Japan). The sizes of the microspheres and nanoparticles were measured by Zetasizer Nano ZS90 (Malvern Instruments, Worcestershire, United Kingdom).

### Extraction of Cartilage-Derived ECM

The cartilage used in the ECM-derived scaffold was separated with a scalpel from the femoral condyle and patellar groove. The cartilage was cut into small slices (1 mm<sup>3</sup>) using eye scissors, then washed several times with phosphate-buffered saline (PBS). The cleaned cartilage pieces were then placed in an appropriate amount of Tris-HCl; they were disrupted and decellularized using a tissue homogenizer to create a cartilage slurry. The homogenized tissue was centrifuged at 4°C and the supernatant was removed. Next, the sediments were incubated with 1% Triton X-100 in hypotonic Tris-HCl (Solarbio, Beijing, China) with gentle agitation at 4°C for 24 h; they were rinsed several times with deionized water, then incubated for 12 h in DNase (Sigma-Aldrich, NJ, United States) and RNase (Sigma-Aldrich, NJ, United States) with agitation at 37°C. Finally, the homogenate was centrifuged and washed as described above. The resulting sediment was freeze-dried and stored at -20°C for further use.

### Fabrication of KGN-PLGA Microspheres and CECM Scaffold Composite System

Freeze-dried CECM was resuspended in deionized water to form a 3% (w/v) suspension. An adequate amount of KGN (MedChemexpress, Rocky Hill, NJ, United States) was then added to the suspension. The mixture was poured into a cylindrical mould, frozen at -80°C for 72 h, and freeze-dried for 48 h. The resulting scaffolds were sterilized and trimmed to an appropriate size. Next, cross-linking was performed in anhydrous ethanol solution containing 50 mM 1-ethyl-3-(3-dimethylaminopropyl) carbodiimide (EDC) and 20 mM *N*-hydroxysuccinimide (NHS) for 24 h at 4°C; cross-linked scaffolds were rinsed several times with PBS. The process produced PLGA microsphere-embedded CECM composite scaffolds with diameter of approximately 5 mm and depth of approximately 3 mm. Scaffolds loaded with TGF-β3

were obtained by immersion in 400 ng/mL TGF-β3 (Peprotech, Rocky Hill, NJ, United States) solution (24 h at 4°C) and then freeze-dried.

### Characterization of Scaffolds

The morphology and structure of scaffolds were characterized by scanning SEM. Briefly, the scaffold specimens were cut into thin sections and fixed on the platform. After they had been sputter-coated with gold, all sections were examined by SEM. Pore diameter distributions were determined using ImageJ software (NIH, Bethesda, MD, United States).

### Determination of Encapsulation Efficiency and Drug Loading of Microspheres

First, several concentrations of KGN standard solutions were prepared and the absorption peak area was determined by high-performance liquid chromatography (HPLC). A standard curve was obtained with the concentration as the x-axis and the peak area as the y-axis. Subsequently, the KGN-PLGA microspheres were submerged in 0.1 M NaOH solution at 1 mg/mL and shaken gently at 37°C for 24 h. The peak area was determined by HPLC and the KGN content in the sample (W<sub>1</sub>) was calculated by the standard curve formula. The encapsulation rate and drug loading amount of KGN in PLGA microspheres were calculated using the following formulae:

$$\text{Encapsulation efficiency} = W_1/W_k \times 100\%$$

and

$$\text{Drug loading} = W_1/(W_p + W_1) \times 100\%,$$

where W<sub>k</sub> represents the total volume of KGN and W<sub>p</sub> represents the total volume of PLGA.

### KGN Release From Scaffolds

First, the two scaffolds (KGN-CECM and PLGA(KGN)-CECM) were soaked in 5 mL of sterile PBS and shaken at 37°C. PBS was collected daily and replaced with 5 mL of fresh PBS. The concentration of KGN in the collected PBS was determined by HPLC, as described above. Thus, sustained-release curves of KGN-CECM and PLGA(KGN)/CECM were obtained.

### Isolation and Culture of Bone Marrow Mesenchymal Stem Cells

BMSCs were isolated from 4-month-old New Zealand White rabbits. All experimental protocols were approved by the Animal Experimental Ethics Committee of Tianjin Hospital. Following injection of 30% urethane (2.2 mL/kg) into the ear vein, BMSCs were obtained from the rabbits by bone marrow aspiration. After the operative region had been shaved and disinfected, a 2-mL aliquot of bone marrow was aspirated from the femur of each rabbit and placed in a sterile centrifuge tube. Following the addition of 5 volumes of Hanks balanced salt solution, the tubes were centrifuged for 7 min at 1,000 rpm; the mixtures were then divided into three layers. The top two layers were transferred to a new tube and resuspended in 5 mL of Dulbecco's Modified Eagle Medium supplemented with penicillin and 10% fetal bovine

serum (Gibco, Grand Island, NY, United States). BMSCs were then cultured at  $5 \times 10^6$  cells/cm<sup>2</sup> in 75-mL culture flasks under an atmosphere of 5% CO<sub>2</sub> at 37°C. Half of the volume of culture medium was changed after 3 days; the entire volume of culture medium was changed within 1 week. Third-generation (P3) cells were stored for further use.

### Chondrogenic Induction *in vitro*

First, four scaffolds were prepared: blank CECM, CECM loaded with PLGA(KGN), CECM loaded with TGF-β3, and CECM loaded with PLGA(KGN) and TGF-β3. BMSCs were suspended at  $1 \times 10^7$  cells/mL, then dropped into each of the scaffolds to construct cell-scaffold complexes. The cell-scaffold complexes were then incubated in chondrogenic induction media (α-Minimum Essential Medium supplemented with 10% fetal bovine serum, 100 nM dexamethasone, 50 μg/mL vitamin C, 1% ITS, 100 μg/mL sodium pyruvate, 40 μg/mL proline, 30 μg/mL L-glutamate, and 1% antibiotic-antimycotic) for 14 or 28 days under an atmosphere of 5% CO<sub>2</sub> at 37°C. The cell-scaffold complexes were fixed with 4% paraformaldehyde; dehydrated and embedded in paraffin wax using a tissue dehydrator; processed for histological analysis; and stained with haematoxylin and eosin (H&E), safranin O, and toluidine blue. All procedures were carried out in accordance with the instructions provided by the reagent manufacturers. For quantitative evaluation of aggrecan and type II collagen in the ECM, the cell-scaffold complexes were removed from the incubator, transferred to liquid nitrogen for several minutes, and then stored at -80°C. Aggrecan and type II collagen enzyme-linked immunosorbent assay (ELISA) kits (R&D Systems, Minneapolis, MN, United States) were used in accordance with the manufacturer's instructions.

### Cytotoxicity Assay

The cytotoxicity of the scaffold resulting from the residual reagents and/or processing was determined by assessment of cell proliferation using cell counting kit-8 (CCK-8; Dojindo, Rockville, MD, United States). Briefly, the four freeze-dried scaffolds were placed in 24-well plates and pre-wetted with culture solution in an incubator for at least 12 h before cell seeding. Then, P3 BMSCs ( $1 \times 10^7$  cells/mL) were seeded onto each scaffold and incubated at 37°C under an atmosphere of 5% CO<sub>2</sub> for 7 days. CCK-8 solution was added to each well at 1, 3, 5, and 7 days, in accordance with the manufacturer's instructions. The absorbances at 450 nm (A<sub>450</sub>) of the samples were recorded using a microplate reader. Six replicates of each scaffold type were studied.

### Cell Adhesion and Viability Assessment

Cell adhesion and morphology of BMSCs grown on PLGA(KGN)/CECM scaffold were assessed by SEM after 48 h of cell culture. The viability of adhesive cells on the scaffold was measured using a live/dead cell viability assay kit at 1 and 3 days, in accordance with the manufacturer's instructions. Briefly, the cell-scaffold samples were incubated in a solution containing calcein ( $2 \times 10^6$  M) and ethidium homodimer-1 ( $4 \times 10^6$  M) for 15–20 min at 37°C, under an atmosphere of 5% CO<sub>2</sub>; they were then rinsed two to three times with sterile

PBS. Cell viability was visualized using laser confocal microscopy (Leica). The numbers of live (green fluorescence) and dead (red fluorescence) cells were calculated using ImageJ.

### Chondrogenic Induction *in vivo*

A full-layer cartilage defect model was used to evaluate the ability of the PLGA(KGN)/CECM scaffold to promote chondrogenic induction. This study was approved by the Institutional Research Ethics Committee of Tianjin Medical University. Twenty-four female New Zealand White rabbits (4–5 months old, 2.5–3.0 kg) were randomly divided into four groups and acclimatized for 1 week at Tianjin Institute of Orthopedics and Sports Medicine before the operation. Rabbits were injected with 30% urethane (2.2 mL/kg) through the ear vein. A lateral para-patellar skin incision, approximately 3 cm in length, was made in both knee joints of each rabbit. Following layer-by-layer incision, the joint capsule was opened. The femoral condyle was exposed following patellar dislocation. A corneal trephine was used to drill a full-thickness cartilage defect (diameter of 4 mm and depth of 1 mm). An acne needle was used to prepare a microfracture on the subchondral bone layer. Defect sites were treated with one of the four scaffolds: (1) CECM, (2) CECM-TGFβ3, (3) PLGA(KGN)/CECM, and (4) PLGA(KGN)/CECM-TGFβ3. The incisions were then sutured in a layer-by-layer manner. Each rabbit was intramuscularly injected with 800,000 units of penicillin once per day for 1 week postoperatively. At 4 and 12 weeks postoperatively, all rabbits were sacrificed by anesthesia overdose and whole knees were collected. Finally, the effects on chondrogenic induction *in vivo* and cartilage defect repair were evaluated by general observation, H&E staining, toluidine blue staining, safranin O staining (Solarbio, Beijing, China), and type II collagen and aggrecan Western blotting.

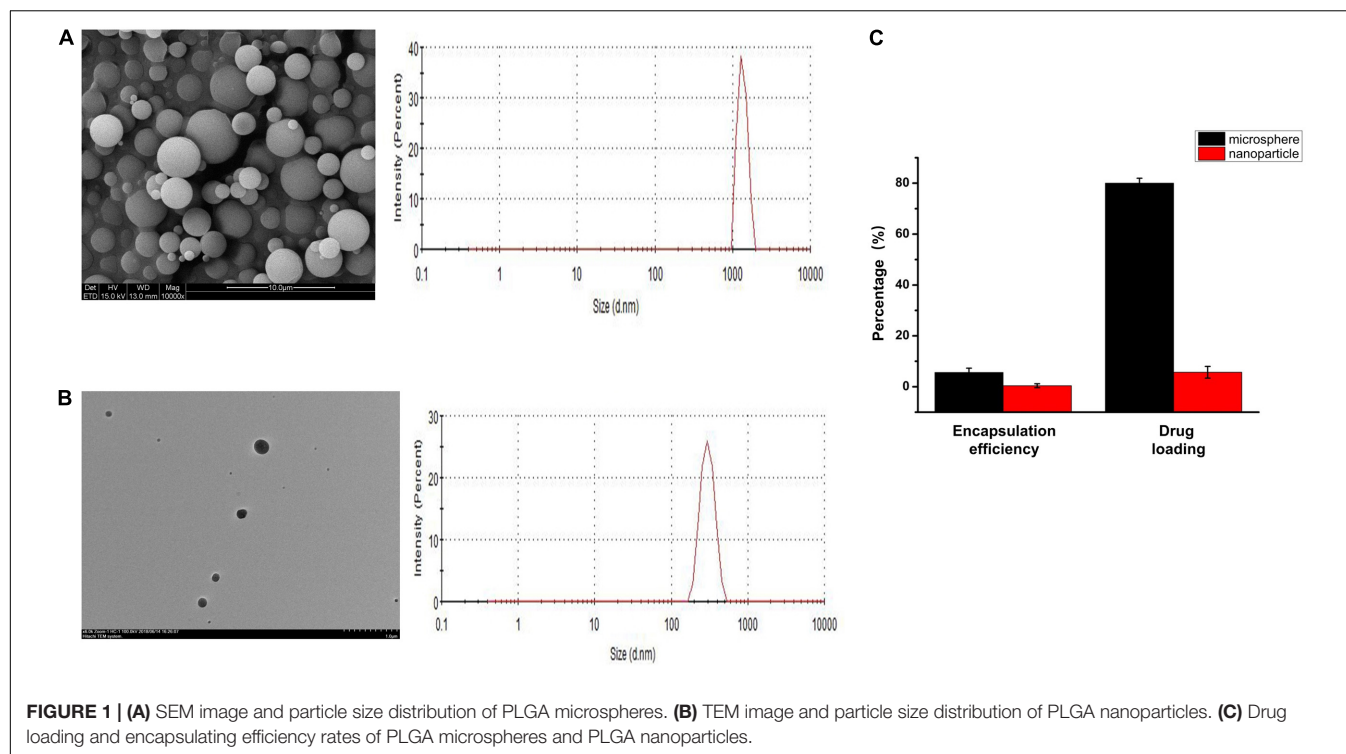
### Statistical Analysis

Data analysis was performed with IBM SPSS Statistics, version 19.0 (IBM Corp., Armonk, NY, United States). Data are presented as the means ± standard deviations. One-way analysis of variance was performed to determine statistical significance, followed by Tukey's *post hoc* test or Dunnett's T3 *post hoc* test. For all analyses,  $P < 0.05$  was considered to indicate statistical significance.

## RESULTS

### Characterization of PLGA(KGN) Microspheres and Microsphere-Embedded CECM-Derived Scaffolds

The prepared PLGA microspheres and nanoparticles loaded with KGN showed a spherical morphology with a smooth external surface; incorporation of KGN did not produce any meaningful changes in the PLGA scaffold structure (Figures 1A,B). NanoZS90 analysis indicated that the mean diameters of PLGA microspheres and nanoparticles were 2.186 μm and 288.8 nm, respectively. A standard curve was constructed to determine KGN concentration. The trapping efficiency and drug loading



rate of PLGA microspheres were 80 and 5.6%, respectively; for PLGA nanoparticles, those parameters were 5.7 and 0.4%, respectively (**Figure 1C**). Therefore, PLGA microspheres were selected for subsequent experiments. The morphology of the PLGA(KGN)/CECM composite scaffold is presented in **Figure 2A**. The scaffold demonstrated alveolar construction; the interconnected pores of the scaffold exhibited a uniform distribution and the apertures were generally uniform in size. Quantitative analysis revealed that the mean pore diameter was 140  $\mu\text{m}$  (**Figure 2A**), which met the requirements for use in cartilage tissue engineering. This scaffold was presumed to provide sufficient space for cell adhesion and proliferation, and was considered suitable for the transport of nutritive materials and metabolic waste.

### ***In vitro* Release of KGN From Microspheres and Scaffold System**

The release of KGN from the CECM/PLGA system (CKP) was monitored for 30 days and compared with the direct release of KGN from the CECM (CK) scaffold. The cumulative release curve (**Figure 2B**) clearly showed two distinct release patterns of CECM/PLGA and CECM. The release curve of CK reached a plateau after initial rapid release in the first 7 days; cumulative release during that period reached 70% of the gross release. The release curve of CKP demonstrated that there was almost no KGN release in the first 5 days and cumulative release reached approximately 45% of the gross release during the first 10 days; thereafter, it entered a sustained slow-release phase cumulative release reached approximately 45% of the gross release during the first 10 days. The results showed that the CKP

displayed controlled and sustained release behavior. Therefore, it was considered suitable for long-term treatment in promoting cartilage regeneration.

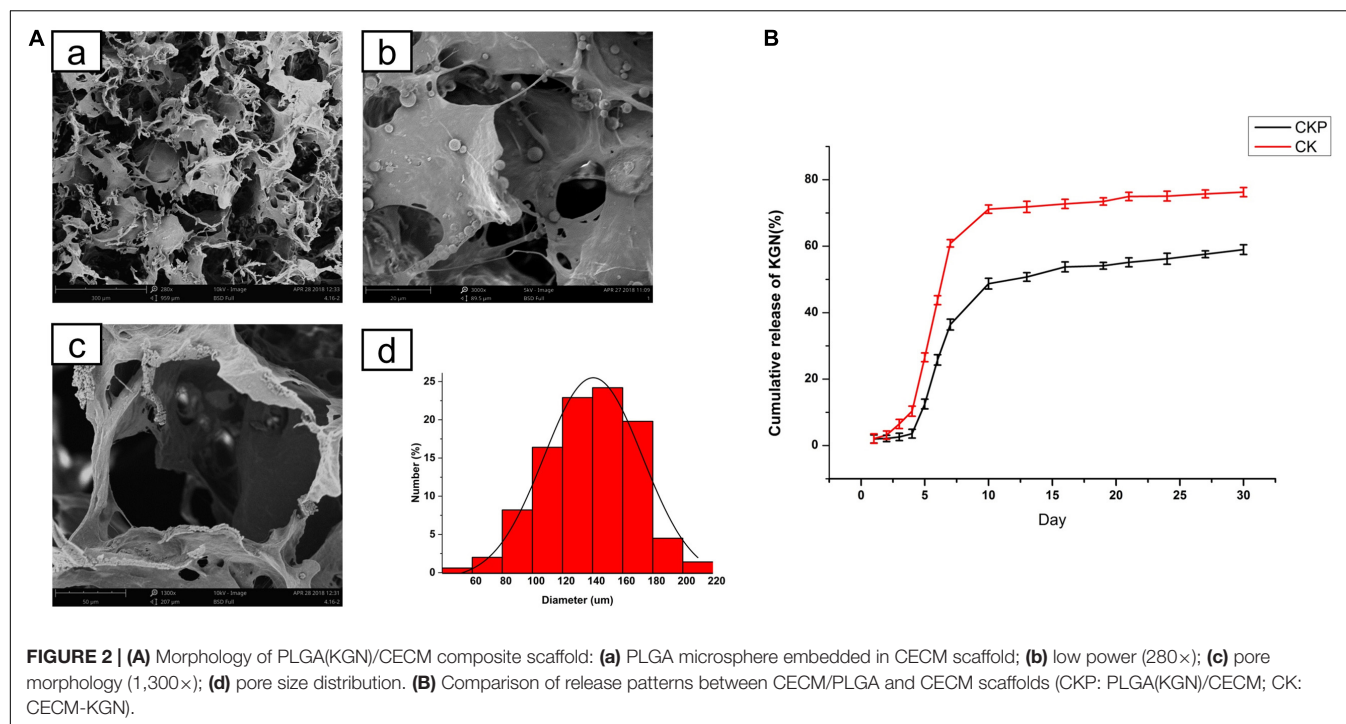
### **Cell Viability and Proliferation Inside the CECM/PLGA Composite Scaffold**

To assess the biocompatibility of PLGA(KGN)/CECM scaffolds, BMSCs were seeded in the scaffolds. The biocompatibility of scaffolds was evaluated by cell viability, cell proliferation, and cell adhesion assays. Cell viability was assessed by live/dead staining. As shown in **Figure 3A**, there were few dead cells on the scaffold, indicating that BMSCs on the scaffold exhibited good cell viability. Cell proliferation, as determined by CCK-8 assay, is shown in **Figure 3B**. The number of cells increased over time in each group; the three induced groups showed higher rates of increase than the control group, indicating that the scaffold remained safe and non-toxic after embedding of KGN-encapsulated PLGA microspheres. SEM images of cell adhesion in the PLGA(KGN)/CECM composite scaffold are shown in **Figure 3C**. BMSCs with rough morphology were attached to the inner wall surface of the scaffold; some extended pseudopods and wound through the scaffold. These results further validated that the PLGA(KGN)/CECM scaffold promoted cell adhesion, which could be beneficial for use in cartilage regeneration after *in vivo* implantation.

### **Isolation and Chondrogenic Differentiation of the BMSCs**

BMSCs were successfully extracted and cultured to passage 3. Following chondrogenesis culture of BMSCs within the four





scaffolds for 4 weeks, histological staining (H&E, safranin O, and toluidine blue) and quantitative ELISA analysis of glycosaminoglycan (GAG) and type II collagen were used to compare the chondrogenesis capability of BMSCs among the four scaffolds. As shown in **Figure 4A**, H&E, toluidine blue, and safranin O staining results indicated that KGN and TGF- $\beta$ 3 both promoted proliferation and chondrogenic differentiation of BMSCs; the cell-scaffold complex exhibited consistent histological findings among the three staining methods. With the exception of the control group, which showed no intact cells or nuclei within CECM scaffolds, the numbers of cells in the other three induced groups (TGF- $\beta$ 3, KGN, and TGF- $\beta$ 3 + KGN) significantly increased over time. The nucleus and cytoplasm were visible by H&E staining, indicating that the cells were in good overall condition; the cell morphology changed from long fusiform to round or oval, indicating that the MSCs had differentiated into chondrocytes. Toluidine blue and safranin O staining revealed that most cells and their secretions in the three induced groups were stained, indicating the presence of type II collagen and GAG components after 28 days of culture.

Longitudinal comparison of the staining conditions of the four groups after 14 and 28 days of culture indicated no significant differences among the TGF- $\beta$ 3, KGN, and TGF- $\beta$ 3 + KGN groups. We presumed that KGN and TGF- $\beta$ 3 did not exhibit a synergistic effect with respect to inducing BMSC differentiation into chondrocytes. These findings were consistent with the results of quantitative analysis of two representative components of hyaline cartilage: GAG and type II collagen (**Figure 4B**). After 14 days of culture, the total amounts of type II collagen and GAG in the three induced groups did not significantly differ from the total amounts in the control group. After 28 days

of culture, the total amounts of type II collagen and GAG were significantly higher in the three induced groups (TGF- $\beta$ 3, KGN, TGF- $\beta$ 3 + KGN) than the control group; however, the total amounts did not significantly differ among the three induced groups. After 14 days of culture, there were no significant differences in the enhancement of type II collagen level among the four groups, while the enhancement of GAG level in the three induced groups significantly differed from the enhancement in the control group. After 28 days of culture, the enhancement of type II collagen level in the three induced groups also significantly differed from the enhancement in the control group.

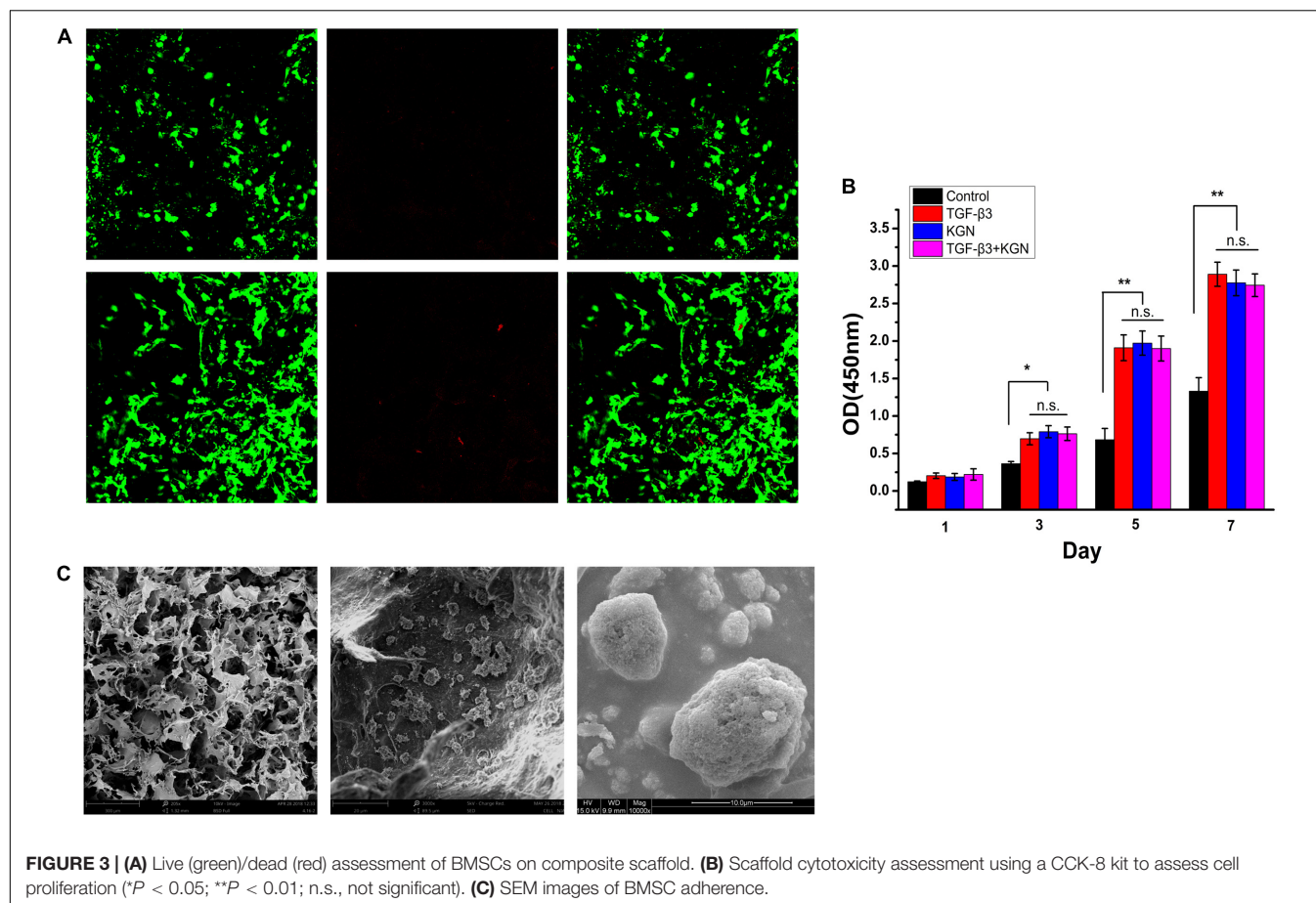
## Cartilage Repair Assessment in the Rabbit Model

A cartilage defect model was successfully established, as described in the Methods section. At predetermined time points of 8 and 12 weeks, the rabbits were sacrificed and their femoral condyles were harvested. Subsequent assessments were performed to estimate the cartilage regeneration efficiency *in vivo*.

## Macroscopic Observation

Macroscopic observation of repaired cartilage was performed, as shown in **Figure 5**. At 4 weeks postoperatively, the vestiges of the articular cartilage defect remained, but new tissue began to appear in the three induction groups, whereas it did not in the blank scaffold group (treatment with CECM scaffold alone). At 12 weeks postoperatively, the control group exhibited broad areas of cartilage destruction and surface denudation, while constructs induced with either KGN or TGF- $\beta$ 3 generally showed an intact surface. The boundary with surrounding natural cartilage was





less noticeable in the KGN-induced group, compared to the TGF-β3-induced group. Combined induction with both factors did not enhance the repair effect, compared to either KGN or TGF-β3 alone.

### Histological Staining Analysis

H&E staining of regenerated cartilage in the four groups is depicted in **Figure 6A**. At 12 weeks postoperatively, H&E staining revealed cartilage defects in the blank scaffold group. In contrast, cartilage defect areas in the KGN, TGF-β3, and TGF-β3 + KGN groups were mostly covered with new cartilage tissue; the thickness of the new tissue was similar to that of the surrounding normal cartilage layer. The chondrocyte concentration and ECM deposition were significantly enhanced, while the number of inflammatory cells was reduced, compared with 4 weeks postoperatively. Blue dye was evident in toluidine blue staining assays (**Figure 6A**), further confirming the changes in cartilage matrix content. Compared with the other two groups, staining intensity was higher in the KGN and TGF-β3 + KGN groups that contained higher glucosamine polysaccharide content in the cartilage ECM.

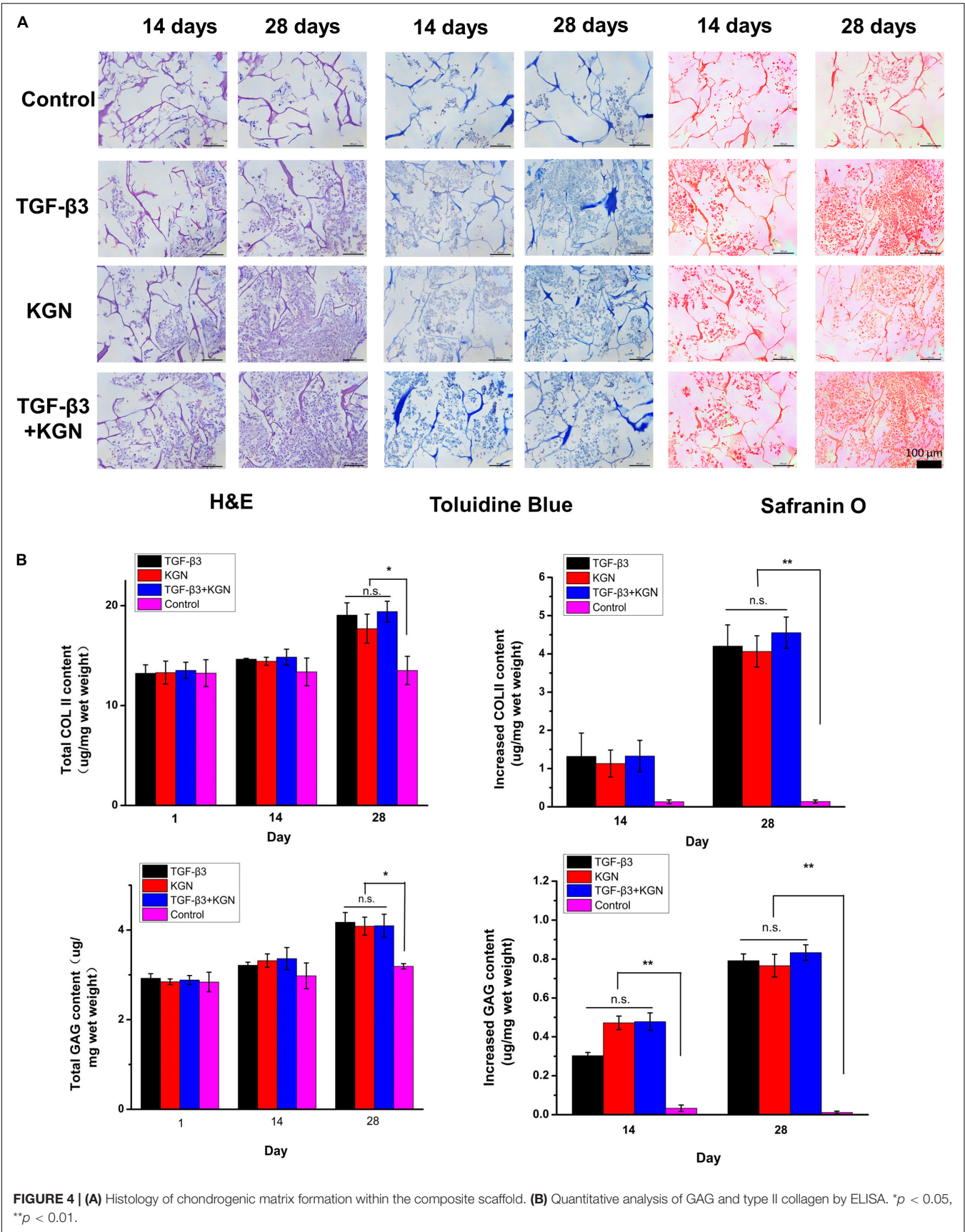
### Western Blotting Analysis

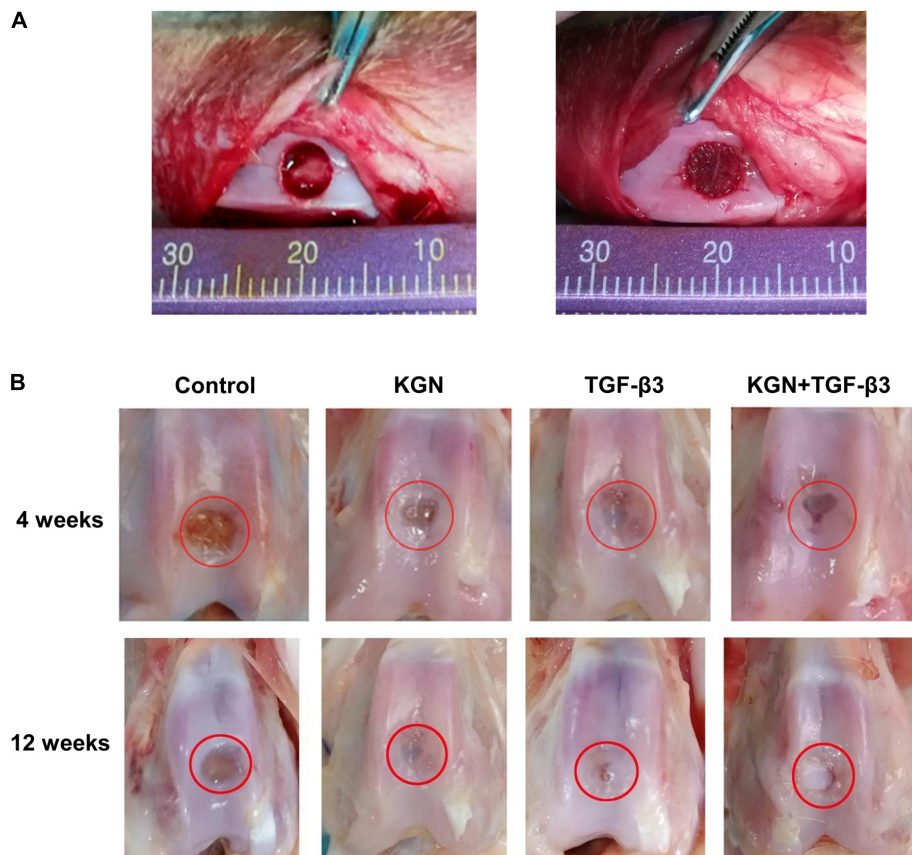
Type II collagen and aggrecan protein expression were further assessed by Western blotting analysis (**Figures 6B,C**). At both 4

and 8 weeks postoperatively, the three induced groups displayed much stronger bands of type II collagen and aggrecan, compared with the blank scaffold group. However, there were no visible differences among the three induced groups.

### DISCUSSION

Tissue engineering offers novel and effective approaches for repairing articular cartilage defects. While numerous cell-based techniques have been applied to repair and regenerate cartilage defects, these techniques have multiple limitations, including cellular apoptosis and increased risk of disease transmission, resulting in low repair efficiency (Trounson and McDonald, 2015; Huang et al., 2016). Cell-free strategies have been proposed to repair and regenerate the injured cartilage by recruitment of endogenous stem cells, thus avoiding many of the limitations and pitfalls of cell-based strategies and allowing broader clinical application. The present study demonstrated that a cell-free scaffold based on CECM and PLGA microspheres loaded with KGN provided an optimal *in vitro* environment for the attachment and chondrogenesis of BMSCs. Following implantation into a cartilage defect in the rabbit knee joint, the scaffold supported chondrogenic differentiation of endogenous BMSCs, thus promoting the regeneration of hyaline cartilage.



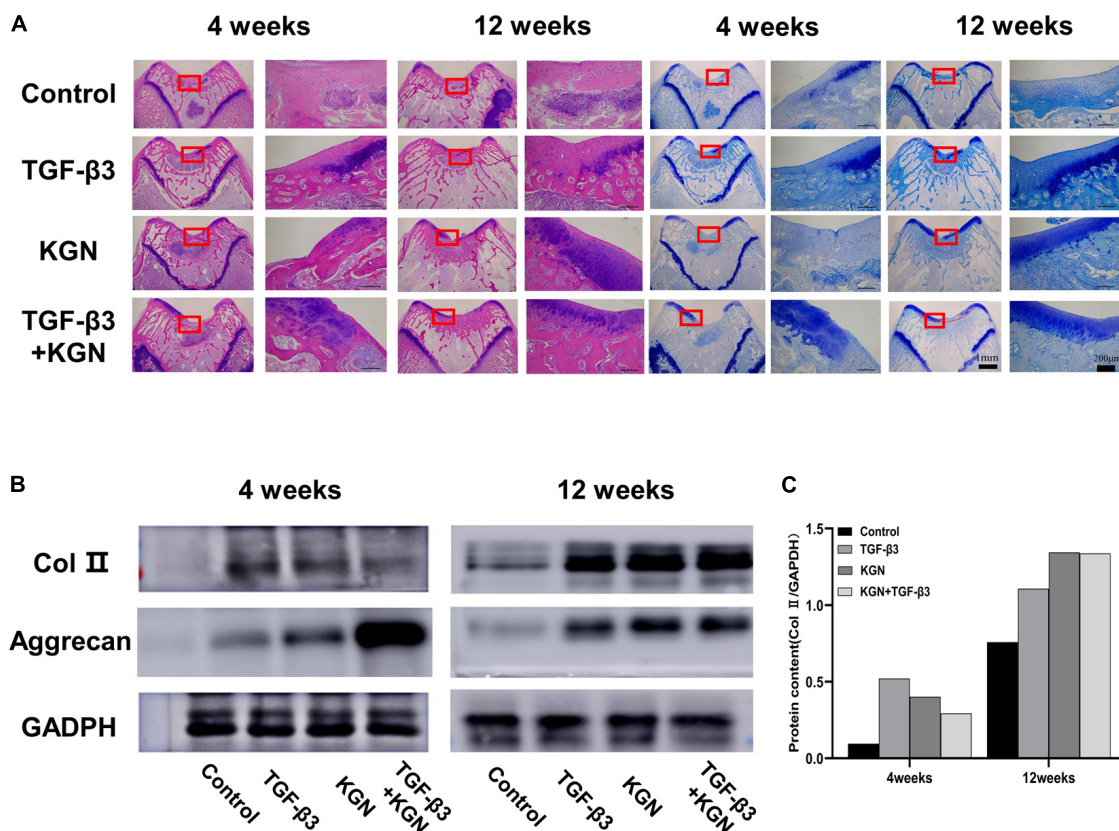


**FIGURE 5 | (A)** Full-thickness cartilage defect model of rabbit femoral trochlea. **(B)** Gross observation at different time points.

The fabrication of biomimetic scaffolds to recreate the structural features of native articular cartilage is crucial for successful cartilage regeneration. In our previous studies, we developed and characterized a structurally biomimetic oriented scaffold derived from CECM, which could closely mimic the natural chondrogenesis microenvironment (Yang et al., 2008, 2017). The CECM scaffold preserves multiple structural and functional proteins (e.g., fibronectin, hyaluronic acid, and GAG), which could modify the secretion of GAG and type II collagen to further induce specific cellular responses and direct new tissue formation (Sutherland et al., 2015; Ng et al., 2016). In the present study, we combined PLGA and CECM to produce composite scaffolds, which revealed the complementary benefits of the two materials. Notably, the presence of CECM provided good scaffold biocompatibility; additionally, the presence of PLGA conferred adequate mechanical support and relatively long degradation time, thereby matching the neocartilage formation rate (Lih et al., 2016). We showed that the composite scaffold with appropriate pore structure and pore diameter served as a guide for BMSC attachment, distribution, and proliferation; thus facilitating chondrogenic differentiation (Figure 2A). These findings indicated that we had fabricated the physiological structure of native cartilage.

The PLGA microsphere-embedded CECM scaffold can serve to mimic some features of the native ECM; it also enables sustained and localized delivery of bioactive molecules to achieve prolonged exposure of BMSCs to bioactive molecules. Previous studies have shown that the delivery effect of PLGA is related to multiple factors (Wu et al., 2013; Zhu et al., 2019) including the grain size of PLGA spheres; this parameter mainly affects the specific surface area, distribution of hydrophobic drugs in spheres, and polymer degradation rate (Chen et al., 2015; Almeida et al., 2020). Therefore, it is necessary to identify a suitable particle size range that can preserve resources and effectively control KGN release. In the present study, we fabricated PLGA microspheres and nanoparticles for encapsulation of KGN. Following multiple analyses, we chose PLGA microspheres for subsequent experiments because of their considerable advantages in terms of drug loading and encapsulating efficiency (Figure 1C). To evaluate the controlled release effect of the composite scaffold, we monitored the release of KGN for 30 days. The release curve of KGN loaded in the CKP group demonstrated that there was almost no KGN release in the first 5 days and cumulative release reached approximately 45% of the gross release during the first 10 days; thereafter, it entered a sustained slow-release phase (Figure 2B). We speculate that this is due to the protection of CECM scaffolds





**FIGURE 6 | (A)** Histology of defects at 4 and 12 weeks. **(B)** Western blotting analysis of type II collagen and aggrecan at 4 and 12 weeks. **(C)** Gray value analysis of type II collagen bands.

on PLGA microspheres, which prevents the degradation of PLGA microspheres caused by rapid water entry. Therefore, in the first 5 days, only the drugs on the surface of the microspheres were released. The microspheres began to degrade in 5–10 days, which accelerated the drug release inside the microspheres. The CKP group demonstrated more robust sustained-release capability, compared to the CK group with more obvious sudden release; this implied that the composite scaffold had a double-sustained effect on KGN release, such that it may be suitable for long-acting administration of KGN in clinical applications. Notably, the PLGA microsphere-embedded scaffolds improved the diffusion path because KGN was confined to the core region of the PLGA microspheres and initially dispersed through the shell polymer before departure from the CECM. In addition, the amine groups of PLGA and the carboxyl groups of KGN exhibited strong covalent linkages, which aided in controlled release.

The bioactivity of released KGN was assessed by measuring the ability of the KGN-loaded scaffold to promote BMSC growth and chondrogenesis. Our assessment indicated that this KGN-releasing system promoted BMSC viability and proliferation (Figure 3); it also enhanced proteoglycan deposition and type II collagen content compared to the control group, potentially promoting the regeneration of cartilage *in vitro* (Figure 4). These findings were consistent with previous research demonstrating

the chondrogenic effect of KGN supplementation on BMSC growth (Johnson et al., 2012). Furthermore, *in vivo* evidence indicated nearly complete repair of cartilage defects at 12 weeks after implantation of the KGN-releasing composite scaffold. Notably, the neocartilage was integrated with its surrounding tissue and subchondral bone (Figure 6A). The levels of type II collagen and GAG expression were elevated in the three induced groups, compared to the control group (Figures 6B,C), thereby improving the quantity and quality of regenerated tissue at the chondral interface. A previous study demonstrated similar improvements in proliferation and chondrogenesis of human adipose-derived MSCs or synovium-derived MSCs by treatment with KGN (Shi et al., 2016; Zhu et al., 2017). Therefore, we speculated that the cartilage repair process could be continued by recruiting other sources of MSCs when BMSCs were no longer enriched after microfracture healing. Further studies are required to examine this aspect of cartilage repair treatment.

In the present study, KGN loaded in CECM/PLGA was able to induce differentiation of BMSCs in a manner similar to that of TGF- $\beta$ 3 loaded in CECM, but combined treatment did not show synergistic effects. Our observations were consistent with the findings by Zhu et al., who demonstrated that KGN was a suitable replacement for TGF- $\beta$ 3 in terms of type II collagen induction and aggrecan deposition, but that it did not



promote or enhance the effects of TGF- $\beta$ 3 (Zhu et al., 2017). However, Jia et al. reported that the combination of TGF- $\beta$ 3 and KGN synergistically promoted chondrogenic differentiation of synovial fluid-derived MSCs via the Smad 2/3 pathway (Jia et al., 2019). Recently, Jing and colleagues also reported that KGN preconditioning could exert dually beneficial effects on TGF- $\beta$ 3-induced chondrogenic differentiation of human umbilical cord MSCs (Jing et al., 2019). Therefore, we speculate that the synergistic effect of combined KGN and TGF- $\beta$ 3 treatment may be based on sequential application consisting of KGN preconditioning with subsequent TGF- $\beta$ 3 induction; the synergism disappeared when the sequential application was disrupted in this study. Another possibility is that the synergistic effect may only be observed for some specific type(s) of MSCs (e.g., adipose-derived MSCs). Further studies are required to explore the mechanisms underlying these effects.

To sum up, although the CECM scaffolds and PLGA microspheres have been widely performed in the cartilage tissue engineering, the combined application of these two materials is rarely investigated. In this study, we updated the application approach of KGN through loading KGN into the CECM/PLGA scaffold. The porous CECM/PLGA scaffolds can not only cover the advantages of CECM porous scaffolds and PLGA microspheres, but also sustain the KGN release. Besides, the scaffold in the present study showed the similar cartilage regeneration efficiency with TGF- $\beta$ 3, hence, we hope that KGN can be a suitable replacement for TGF- $\beta$ 3 in cartilage tissue engineering to avoid some adverse effects caused by excessive TGF- $\beta$ 3 (Mi et al., 2003).

This study had some limitations. First, the biomechanical properties of the scaffold were not fully analyzed. Second, the present results were obtained in a rabbit model. Therefore, before human clinical trials, the scaffold should be examined in large-animal models that are immunologically similar to humans, which can provide evidence of its safety and efficiency.

## CONCLUSION

We successfully prepared a composite scaffold by blending KGN-encapsulated PLGA microspheres and CECM. *In vitro*

assessments demonstrated that the PLGA(KGN)/CECM composite scaffold prolonged the activity of KGN and supported the attachment, proliferation, and differentiation of BMSCs. Initial *in vivo* analysis indicated that the PLGA(KGN)/CECM induced superior hyaline-like neocartilage repair; the neocartilage successfully integrated with its surrounding tissue in a rabbit model. Moreover, the PLGA(KGN)/CECM system avoids difficult cell manipulation during preparation, providing a new small molecule-based strategy for cartilage tissue engineering.

## DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/supplementary material, further inquiries can be directed to the corresponding author/s.

## ETHICS STATEMENT

The animal study was reviewed and approved by Tianjin Medical University.

## AUTHOR CONTRIBUTIONS

YHZ and YMZ designed and conceptualized the work. XGZ and RZ carried out the study and collected the crucial background information. YH, YL, and XTZ collected the data. MS analyzed and interpreted the data. XGZ and YHZ wrote the manuscript. YX, MW, and YZ revised the manuscript critically. All authors have read and approved the final manuscript.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Bioconjugation of a Collagen-Mimicking Peptide Onto Poly(vinyl alcohol) Encourages Endothelialization While Minimizing Thrombosis

Novella M. Bates<sup>1</sup>, Heather E. Heidenreich<sup>1</sup>, Meghan E. Fallon<sup>1</sup>, Yuan Yao<sup>2</sup>, Evelyn K. F. Yim<sup>2</sup>, Monica T. Hinds<sup>1</sup> and Deirdre E. J. Anderson<sup>1\*</sup>

<sup>1</sup> Department of Biomedical Engineering, Oregon Health & Science University, Portland, OR, United States, <sup>2</sup> Department of Chemical Engineering, University of Waterloo, Waterloo, ON, Canada

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### \*Correspondence:

Deirdre E. J. Anderson  
anderdei@ohsu.edu

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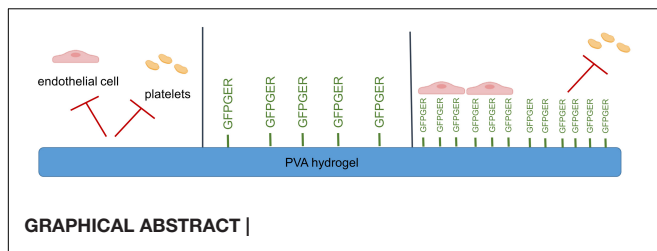
Poly(vinyl alcohol) hydrogel, PVA, is a suitable material for small-diameter vascular grafting. However, the bioinert properties of the material do not allow for *in situ* endothelialization, which is needed to combat common graft failure mechanisms, such as intimal hyperplasia and thrombosis. In this work, the surface of planar and tubular PVA was covalently modified with a collagen-mimicking peptide, GFPGER. The surface of modified PVA was characterized by measuring contact angle and x-ray photoelectron spectroscopy. Endothelial cell attachment to GFPGER-modified PVA was quantified and qualitatively examined using immunohistochemical staining. Then, *in vitro* hemocompatibility testing was performed by quantifying platelet attachment, coagulation factor XII activation, and initiation of fibrin formation. Finally, an established *ex vivo*, non-human primate model was employed to examine platelet attachment and fibrin formation under non-anticoagulated, whole blood flow conditions. GFPGER-modified PVA supported increased EC attachment. *In vitro* initiation of fibrin formation on the modified material was significantly delayed. *Ex vivo* thrombosis assessment showed a reduction in platelet attachment and fibrin formation on GFPGER-modified PVA. Overall, GFPGER-modified PVA encouraged cell attachment while maintaining the material's hemocompatibility. This work is a significant step toward the development and characterization of a modified-hydrogel surface to improve endothelialization while reducing platelet attachment.

**Keywords:** poly(vinyl alcohol), GFPGER, carbonyldiimidazole, hemocompatibility, vascular graft, platelets, fibrin, endothelial cells

## INTRODUCTION

Development of novel, synthetic, small-diameter vascular grafts for treatment of life-threatening cardiovascular diseases has been limited by the challenge of mitigating common failure mechanisms, such as hyperplasia and thrombosis. Currently available and widely used synthetic materials include Dacron® and expanded polytetrafluoroethylene (ePTFE), which are primarily used in bypass surgeries for large diameter (>6 mm) vessels (Nakayama et al., 2004; Kumar et al., 2011; Fayol et al., 2013). Current synthetic materials fail at small diameters, likely due to a mechanical mismatch with native tissues and a lack of an endothelium to promote a healthy, blood-contacting surface. Alternatively, autologous tissues can be used for small-diameter





prostheses; however, patients in need of treatment are often critically ill and may not have suitable vessels for this application. Thus, there remains an unmet clinical need for synthetic biomaterials which are hemocompatible and capable of supporting an endothelium.

One synthetic cardiovascular biomaterial under development for vascular tissue engineering applications is poly(vinyl alcohol), PVA, a hydrogel that is amenable to surface modification and has tunable mechanical properties. Previous work thoroughly characterized the mechanical properties of the PVA grafts and found that PVA hydrogels have an elastic modulus ranging from 250 to 500 kPa (Cutiongco et al., 2016a,b), which is similar to the rabbit femoral artery that has a modulus of 230 kPa (Uchida et al., 1989). These hydrogels have a modifiable circumferential compliance, which can mimic native vasculature. PVA compliance ranged between 3 and 7% (Cutiongco et al., 2016a,b), which is comparable to the rabbit femoral artery of 6% (Uchida et al., 1989). Despite PVA having suitable mechanical properties for cardiovascular applications, the bioinert surface, critical for preventing thrombosis, also prevents the material from being endothelialized *in situ* after implantation (Ino et al., 2013). The native vascular endothelium regulates thrombosis and prevents intimal hyperplasia by secreting inhibitory factors, such as nitric oxide and prostacyclin (Liu et al., 2014). Therefore, a widely accepted approach for attaining long term success of vascular grafts is to improve hemocompatibility of synthetic graft surfaces by encouraging *in situ* endothelialization. Chemical modification of bioinert devices like PVA hydrogels, can encourage endothelialization.

To generate a biomimetic tissue engineered surface, the addition of extracellular matrix (ECM)-derived peptides has been shown to increase endothelial cell (EC) attachment. However, a major drawback is that many of these peptides also support unwanted platelet attachment. For decades, work in this area focused on immobilizing peptides based on the Arg-Gly-Asp (RGD) motif in fibronectin; however, there have been mixed reports on platelet adhesion to these sequences (Schmedlen et al., 2002; Li et al., 2008; Sivaraman and Latour, 2010; Gabriel et al., 2011; Cutiongco et al., 2015a; Anderson et al., 2019). Similarly, the hexapeptide, Gly-Phe-Pro-Gly-Glu-Arg (GFPGER), based on the GFOGER motif within collagen, where the O represents hydroxyproline (Xu et al., 2000; Seo et al., 2010), has been studied for its ability to support EC attachment. Both of these sequences specifically recognize  $\alpha 1\beta 1$  and  $\alpha 2\beta 1$  integrins, which are known to bind ECs and both bind and upregulate platelet activation. The incorporation of GFPGER into materials has been shown to promote cell growth and, while in solution, the GFPGER sequence bound platelets but did not cause activation

of the platelets (Munoz-Pinto et al., 2015). The effect on platelet attachment and aggregation due to a covalently linked GFPGER surface is unknown.

Previous work from our group examined the biomimetic effects of mixing ECM proteins and peptides, including GFPGER, into a hydrogel network to enhance EC attachment (Anderson et al., 2019). We found that the GFPGER peptide minimized platelet attachment *in vitro* when compared to PVA-collagen hydrogels. However, antiplatelet monotherapy was required to minimize platelet adhesion under flow in our established *ex vivo*, non-human primate, whole blood, shunt model. This was likely due to the random presentation of amino acids on the hydrogel surface. The possibility of the binding domain being buried within the hydrogel network also suggested the need for a more specific and well-controlled surface binding approach. We hypothesized that covalent attachment of GFPGER to the surface of PVA hydrogels would encourage endothelialization of the material, while mitigating thrombotic responses. The current work presents a systematic examination of this biomimetic tissue-engineered surface for cell attachment and thrombotic potential of PVA hydrogels with GFPGER at various concentrations conjugated to the surface using a 1,1'-carbonyldiimidazole (CDI) linker. This work is a significant step toward the development of hemocompatible off-the-shelf devices for the treatment of cardiovascular disease.

## MATERIALS AND METHODS

### PVA Film and Tube Manufacturing

Planar PVA films and tubes with an inner diameter (ID) of 4 mm were manufactured, as previously described (Cutiongco et al., 2015b, 2016b). In brief, PVA films were manufactured by adding 15% sodium trimetaphosphate (STMP, Sigma-Aldrich, St. Louis, MO, United States) to aqueous 10% PVA (Sigma-Aldrich) followed by 30% sodium hydroxide. The solution was cast into tissue culture-treated well plates at a 1.5:1 volume to surface area ratio ( $\mu\text{L}/\text{mm}^2$ ). Films were left to crosslink in a sterile incubator at room temperature with 95% humidity followed by drying in a biosafety cabinet. Films were rehydrated under sterile conditions in  $10\times$  phosphate buffered saline (PBS, Fisher Scientific, Waltham, MA, United States) followed by  $1\times$  PBS and subsequent deionized water. Prior to modification, samples were dried in an oven overnight. PVA films were then covalently modified with GFPGER (Sigma-Aldrich) peptides. The surface of the PVA films was activated using 100 mg/mL CDI (Sigma-Aldrich) in dimethyl sulfoxide (DMSO, Sigma-Aldrich) for 1 h then briefly rinsed three times with  $1\times$  PBS before adding GFPGER (0, 15, 30, 60, and 120  $\mu\text{g}/\text{mL}$ ) in PBS solutions at  $37^\circ\text{C}$  overnight (GFPGER<sub>0</sub>, GFPGER<sub>15</sub>, GFPGER<sub>30</sub>, GFPGER<sub>60</sub>, and GFPGER<sub>120</sub>, respectively). Modified films were rinsed with  $1\times$  PBS three times before use.

For the fabrication of tubular PVA samples, a cylindrical mold with an outer diameter of 3.75 mm was coated with a thin PDMS film (Dow Corning, Midland, MI, United States). The mold was treated with air plasma and then was immediately immersed in a solution of crosslinking PVA. Dip casting was performed repeatedly for 12 dips, with a drying duration of

15–30 min between each dip. The tubes were then cured at 18–20°C for 3 days before rehydration in PBS and DI water, as described above, then thoroughly rinsed in DI water overnight. For GFPGER modification, rehydrated and dried PVA tubes were immersed in a 100 mg/mL CDI solution followed by 0, 15, or 120 µg/mL GFPGER solution, as described for the films. Samples were briefly rinsed three times with PBS in between immersions. PVA films were used for surface characterization, cell attachment, and *in vitro* platelet attachment and fibrin formation assays. Tubular samples were used for *ex vivo* thrombosis testing in whole blood.

## Surface Characterization

### Quantification of Conjugated GFPGER Peptide on the Surface of PVA

Poly(vinyl alcohol) films in a 96-well plate were modified with GFPGER as described above (GFPGER<sub>0</sub>, GFPGER<sub>15</sub>, GFPGER<sub>30</sub>, GFPGER<sub>60</sub>, and GFPGER<sub>120</sub>) with the inclusion of a five percent supplementation of each peptide concentration with fluorescent labeled GFPGER [GFPGER{Lys(5-FAM), GenScript United States, Piscataway, NJ, United States}]. The modified films were rinsed with 1× PBS three times before being placed in an Infinite M200 spectrophotometer (Tecan, Männedorf, Switzerland). Fluorescent peptides conjugated to the surface of the PVA films were excited at 488 nm and emission was read at 525 nm. A standard curve of 5% fluorescent-labeled GFPGER at a range of peptide concentrations was used to quantify the peptide on the surface.

### Captive Bubble Contact Angle

Static contact angle was measured by using a captive bubble method with an optical contact angle system (OCA20, Future Digital Scientific Corp). GFPGER-modified samples were immersed in DI water, and an air bubble (4 µL) was injected into the water with a syringe. The bubble was allowed to attach to the surface of the samples and imaged by a camera. The contact angle on each surface was then calculated using SCA20 software (DataPhysics Instruments United States Corp., Charlotte, NC, United States).

### X-Ray Photoelectron Spectroscopy

X-ray photoelectron spectroscopy (XPS) was used to obtain the elemental composition on the surfaces of the PVA samples. XPS survey spectra were measured using a VG ESCALab 250 with a monochromatic Al K-alpha X-ray source (1486.6 eV), and high resolution spectra of C1s were obtained using a standard magnesium X-ray source (1253.6 eV) at Waterloo Advanced Technology Laboratory (University of Waterloo, Waterloo, ON, United States).

## Endothelialization of Surfaces

### Cell Isolation

Carotid arteries were harvested from baboons, and primary endothelial cells (ECs) were isolated, as previously described (Anderson and Hinds, 2012). Artery lumens were filled and incubated for 5 min with 600 U/mL collagenase type II (Worthington Biochemical Corp., Lakewood, NJ, United States)

then massaged and flushed into collagen type I (Corning Inc., Corning, NY, United States)-coated tissue culture treated well plates containing Vasculife basal medium (Lifeline Cell Technology, Frederick, MD, United States) supplemented with 18% fetal bovine serum (FBS, Hyclone, Logan, UT, United States). Cell cultures were grown to confluence before sorting with Dynabeads (Invitrogen, Carlsbad, CA, United States) for CD31+ cells per the manufacturer's protocol. Sorted cells were expanded then stored in freezing media (50% Vasculife basal medium, 40% FBS, and 10% DMSO). Cells thawed for experimentation were maintained in 10% FBS Vasculife basal medium and used at the fourth or fifth passage.

### Cell Seeding and Quantification

Rehydrated peptide-modified PVA films were incubated in FBS for 1 h at 37°C before seeding ECs at a density of 0.2 M cells/mL in 10% FBS Vasculife basal medium. For cell quantification, cells were cultured for 48 h, rinsed, and frozen dry at –20°C before further lysing with 0.02% sodium dodecyl sulfate in sodium citrate buffer. DNA in the lysate was quantified using a PicoGreen assay per the manufacturer's protocol (Invitrogen, Carlsbad, CA, United States). Cells were similarly seeded onto tissue culture treated plastic wells to serve as a positive control to the PVA films. Treatment group cell quantities were calculated as a percent confluence from this control.

### Immunostaining

Cells were incubated on PVA samples for 48 h then stained, as previously described (Jurney et al., 2018). In brief, cells adhered to samples were fixed using 3.7% paraformaldehyde (PFA) then stained for VE-cadherin [primary antibody, Santa Cruz Biotechnology, mouse IgG1 monoclonal, 1:100, 1 h and IgG1 Alexa-488 as a secondary (Invitrogen, goat polyclonal, 1:500, 30 min)], and DAPI (Invitrogen, 1:10000, 5 min) as a nuclear stain. Samples were imaged with a Zeiss LSM 880 inverted confocal microscope system. Low-magnification immunofluorescent (IF) images were collected using a 10× PlanApo objective NA = 0.45. Z-stacks are presented in figures as maximum intensity Z-projections and post-processing of all images was performed using FIJI (SciJava) (Schindelin et al., 2012; Hagen and Hinds, 2019).

## Thrombogenicity of Surfaces

### Washed Platelet and Platelet-Rich Plasma Preparation

Human venous whole blood was drawn from healthy donors into sodium citrate, as previously described (McCarty et al., 2005). In brief, whole blood was centrifuged at 200 g for 20 min to obtain platelet-rich plasma (PRP). In select studies, purified platelets were isolated from PRP by further centrifugation at 1,000 g for 10 min in the presence of 0.1 µg/mL prostacyclin. The pellet was resuspended in HEPES/Tyrod's buffer (129 mM NaCl, 0.34 mM Na<sub>2</sub>HPO<sub>4</sub>, 2.9 mM KCl, 12 mM NaHCO<sub>3</sub>, 20 mM HEPES, 5 mM glucose, 1 mM MgCl<sub>2</sub>; pH 7.3) containing 0.1 µg/mL prostacyclin. The platelets were washed once via centrifugation (1,000 g for 10 min) and resuspended in HEPES-Tyrod's buffer.

## Static Platelet Adhesion Quantification

Static platelet adhesion was quantified using a method modified from Vaničková et al. (2006) to incorporate PVA films. Purified washed platelets suspensions ( $5.0 \times 10^8$  platelets/mL) were incubated on peptide-modified PVA films, in the absence of ECs, in a microtiter plate for 1 h at room temperature. Samples were then rinsed three times with PBS to remove all non-adherent platelets. The amount of platelets attached to the surface of each material was quantified by measuring platelet acid phosphatase activity using a calibration curve of platelet solutions ( $0$ – $5.0 \times 10^8$  platelets/mL).

## Coagulation Quantification

Pooled platelet-poor plasma (PPP, ISTH SSC Lot 4) was incubated with peptide-modified PVA, in the absence of ECs, films followed by the addition of  $\text{CaCl}_2$  (8 mM). Absorbance at 405 nm was documented over time. Fibrin generation lag times were defined by a 5% increase over baseline absorbance. Rates of fibrin generation were defined by maximum slopes.

Activation of coagulation factor XII (FXII, 200 nM, Haematologic Technologies, Inc., Essex Junction, VT, United States) in the presence of coagulation enzymes generated on GFPGER-modified samples, in the absence of ECs, was measured in a purified system using a chromogenic substrate, Spectrozyme-FXIIa (American Diagnostica, Inc., Stamford, CT, United States), as previously described (Bates et al., 2020).

## Whole Blood Platelet and Fibrin Adhesion Under Flow Quantification

Rehydrated GFPGER-modified tubes, in the absence of ECs, (prepared from 15 or 120  $\mu\text{g/mL}$  GFPGER solutions) were incorporated into a chronic, femoral arteriovenous (AV) shunt via connective silicone tubing in an established non-anticoagulated, non-human primate model (Cutiongco et al., 2015a; Anderson et al., 2019). Juvenile male baboons (9–12 kg) underwent a minor surgical procedure, cannulation of the superficial femoral artery and vein, for the placement of a chronic AV shunt. The AV shunts were composed of implanted sterilized silicone rubber tubing (3.0 mm ID, SIL-TEC, Technical Products, Inc, Lawrenceville, GA, United States). Single devices were connected to the implant with Silastic tubing (4 mm ID, Dow Chemical, Midland, MI, United States) (**Supplementary Figure 1**). Upstream of the device was a flow probe (Transonic, Ithaca, NY, United States) to measure blood velocity, and downstream of the device was a tubing clamp to control the flow at a steady rate. Autologous platelets and homologous fibrin were labeled with indium-111 ( $^{111}\text{In}$ ) and iodine-125 ( $^{125}\text{I}$ ), respectively. Samples were subjected to whole blood flow at 100 mL/min and platelet accumulation onto the samples was quantified in real-time by measurement of  $^{111}\text{In}$  radiation using a Brivo NM615 nuclear imaging camera (General Electric, Boston, MA, United States). The amount of fibrin was quantified as an end-point measurement of  $^{125}\text{I}$  radiation approximately 30 days post experiment, once the  $^{111}\text{In}$  decayed, using a 1480 Wizard gamma counter (PerkinElmer, Waltham, MA, United States). CDI-active PVA without peptide (GFPGER<sub>0</sub>) and plain PVA were used as negative controls, collagen-coated ePTFE (Collagen, 1 mg/mL equine collagen type I, Chrono-log Corp.) was used

as a positive control and clinical grade ePTFE (IMPRA®, Bard Peripheral Vascular, Inc., Tempe, AZ, United States) with an ID of 4mm was used as a clinical control.

After whole blood testing, the tubes were rinsed with PBS and fixed with 3.7% PFA for 48–72 h before additional rinsing and storage. Some of the samples used in the shunt studies were examined with micro-computed tomography (MicroCT). The samples were prepared as described previously (Gupta et al., 2020). Briefly, PVA samples were filled with Microfil® polymer to render the lumens radiopaque. Since the Microfil® polymer is incompatible with ePTFE, ePTFE samples were soaked in Lugol's (Sigma-Aldrich) solution to render the thrombus radiopaque.

Shunt studies were performed in a male *Papio anubus* baboon (10.5 kg), which was under the care of the Oregon National Primate Research Center (ONPRC) staff. All studies abided by guidelines provided by the National Research Council and the Committee on Care and Use of Laboratory Animals of the Institute of Laboratory Animal Resources. All studies were approved by the ONPRC Institutional Animal Care and Use Committee.

## Micro-Computed Tomography Analysis

Micro-computed tomography (Inveon, Siemens) imaging was used to analyze thrombus area over the length of the samples (volume) in the small-diameter vascular grafts formed under flow by whole blood in *ex vivo* shunts, as described previously (Gupta et al., 2020). Imaging was performed with  $2 \times 2$  binning, 220 projections, at an exposure of 660 ms/projection. Thrombus physical characteristics were determined with the Amira® software package (FEI, version 5.2.2) by a trained operator blinded to specific sample treatments. The software also generated three-dimensional (3D) volume renderings of either the thrombus for ePTFE grafts or the lumen for collagen-coated and PVA grafts. The software-calculated volumes and lengths of each sample were used for subsequent analysis. Amira® was utilized, as previously described (Gupta et al., 2020), to segment specific regions of interest (ROIs) in each cross-sectional image. For PVA samples, the ROI was the luminal area of the graft. ePTFE samples were segmented for the thrombi as the ROI, and collagen-coated ePTFE samples contained ROIs of both the luminal and thrombus areas. 3D representations of either the lumen (PVA and collagen-coated ePTFE) or thrombus (ePTFE) of the grafts were generated from ROI summations, and the surface areas of each cross-sectional slice were subsequently measured. These cross-sectional data were then utilized in analyzing average thrombus generation over the entire graft surface. The cross-sectional areas obtained from each slice were used to generate topography maps of the thrombus surface for each individual sample. This allows for characterization of the thrombus surface and quantification of the intrasample variability of the thrombus over the length of a single sample.

## Statistical Methods

Data are presented as mean  $\pm$  standard deviation (SD) for all studies. Probability values of  $p < 0.05$  were considered to be statistically significant. To determine statistical significance of captive bubble contact angle ( $n = 3$ ), XPS ( $n = 3$ ), peptide quantification ( $n = 5$ ), lag time and rate of fibrin generation



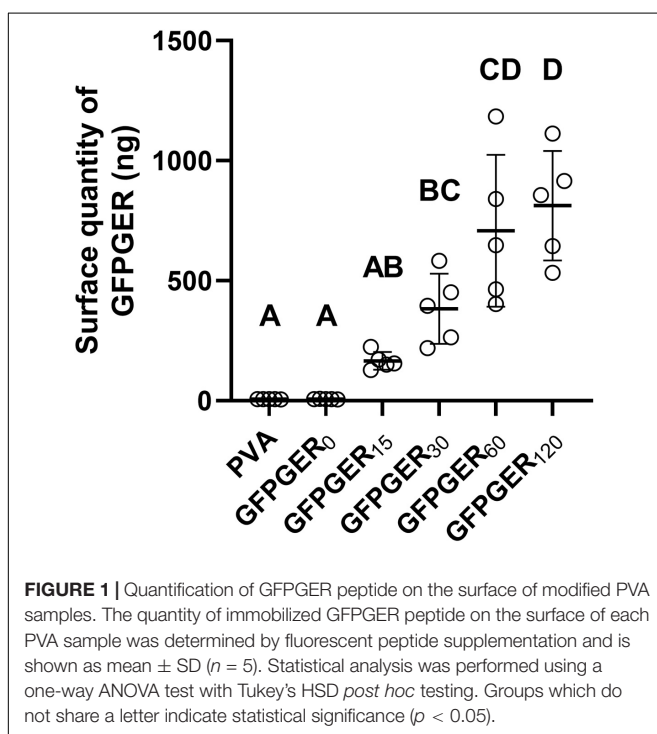
( $n = 3$ ), static *in vitro* platelet adhesion ( $n = 3$ ), and *ex vivo* end-point fibrin accumulation data ( $n = 6$ ), one-way analysis of variance (ANOVA) tests were performed. To determine statistical significance for *ex vivo* platelet data, a repeated measures ANOVA was performed on surface modified PVA samples ( $n = 6$ ). Platelet reactivity of collagen-ePTFE, ePTFE, and unmodified PVA controls samples ( $n = 2$ ) were confirmed and compared to historic data, but these controls were not included in the statistical analysis. All studies were tested for ANOVA assumptions using Levene's and qqplots and, when necessary, data were natural log transformed. A subset of the *ex vivo* samples were tested with microCT ( $n = 4$  for GFPGER<sub>0</sub>, GFPGER<sub>15</sub>, and GFPGER<sub>120</sub>) using a one-way ANOVA between the average luminal cross-sectional area (volume/length) for each test group. SDs of the volume data generated over the entire graft length for each single sample (generated from per slice area data) were also compared with a one-way ANOVA to determine if intrasample variability was significantly different between test groups.

## RESULTS

### Surface Characterization

#### Peptide Quantification

Gly-Phe-Pro-Gly-Glu-Arg peptides were supplemented with 5% fluorescent-labeled peptide to quantify the amount of GFPGER immobilized onto the surface of PVA at each concentration. A significant increase in peptide quantity was found on GFPGER<sub>30</sub>, GFPGER<sub>60</sub>, and GFPGER<sub>120</sub> when compared to unmodified samples (Figure 1). Additionally, GFPGER<sub>60</sub> and GFPGER<sub>120</sub> were significantly increased from GFPGER<sub>15</sub>, and GFPGER<sub>120</sub> was significantly increased from GFPGER<sub>30</sub>.

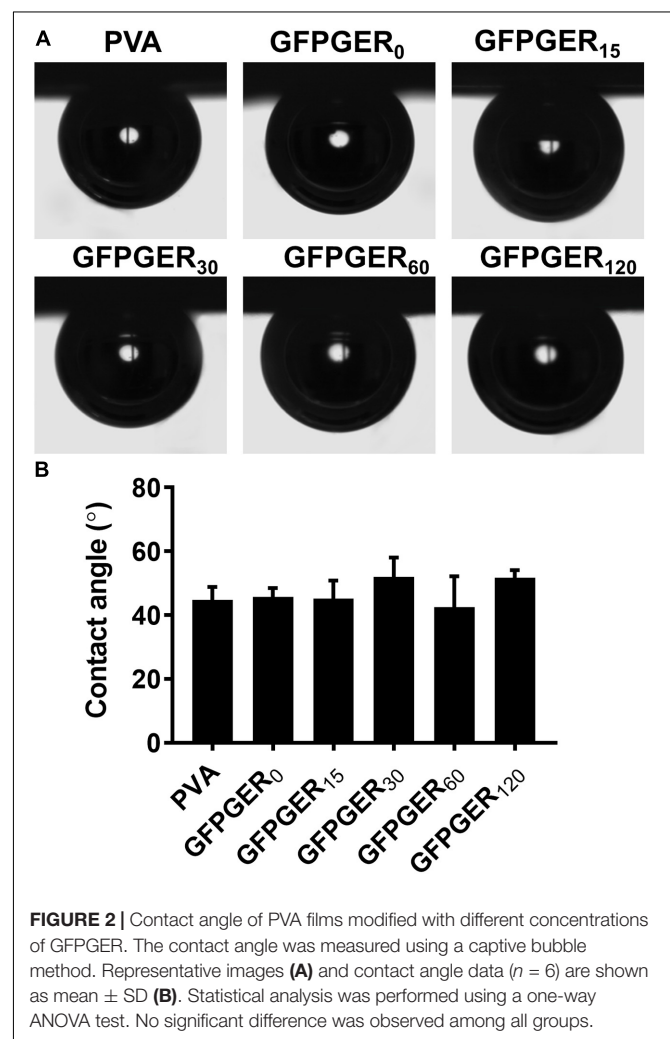


### Contact Angle

Contact angle did not exhibit a significant difference among all test groups (Figure 2). The surfaces of PVA and GFPGER modified PVA samples all remained hydrophilic, with contact angles ranging from 44.8 to 51.7°.

### XPS

X-ray photoelectron spectroscopy survey spectra of PVA and GFPGER modified PVA are shown in **Supplementary Figure 1A**. GFPGER and CDI contain nitrogen and, thus, all modified PVA groups showed significant N1s peaks. **Table 1** and **Figure 3** showed the atomic percentage of C, O, and N on different samples. GFPGER<sub>120</sub> had significantly higher C% than GFPGER<sub>30</sub> and GFPGER<sub>0</sub>, and significantly lower O% than all the other groups. GFPGER modified PVA samples had significantly higher N% than GFPGER<sub>0</sub>. **Supplementary Figures 2B–D** shows the high-resolution spectra of carbon on PVA, GFPGER<sub>0</sub>, and GFPGER<sub>120</sub> samples. High-resolution spectrum of carbon on PVA showed characteristic peaks of the PVA structure at 282, 284, and 286 eV from C–C/C–H, C–O, and C=O, respectively. In addition to the characteristic PVA peaks, GFPGER<sub>0</sub> also showed a peak at 288 eV from the –O–C=O





**TABLE 1** | XPS results of PVA films modified with different concentration of GFPGER.

Atomic %	C	O	N
PVA	76.8 ± 2.0	23.1 ± 2.0	ND
GFPGER <sub>0</sub>	73.9 ± 1.9	22.7 ± 2.1	3.3 ± 0.3
GFPGER <sub>15</sub>	71.4 ± 1.0	23.9 ± 0.8	4.6 ± 0.2
GFPGER <sub>30</sub>	74.0 ± 1.6	21.8 ± 1.6	4.11 ± 0.1
GFPGER <sub>60</sub>	72.2 ± 1.7	23.2 ± 1.9	4.6 ± 0.3
GFPGER <sub>120</sub>	78.6 ± 2.2	17.3 ± 1.9	4.17 ± 0.3

Atomic percentage of carbon, nitrogen, and oxygen. Data ( $n = 3$ ) are shown as mean ± SD (ND = not detected).

bond in the imidazole carbamate intermediate groups. GFPGER contains NH-C(=O) groups, and the peak at 288.7eV observed from GFPGER<sub>120</sub> samples further confirmed the presence of GFPGER on the samples.

## Endothelialization of Surfaces

We examined the ability of PVA to attach ECs after the hydrogel was modified with GFPGER peptide. ECs that adhered to samples were quantified using PicoGreen® as a percentage of cells adhered to tissue culture plastic and observed qualitatively using fluorescent staining. Only the GFPGER-modified samples (15, 60, and 120 µg/mL) showed a significant increase in EC attachment when compared to GFPGER<sub>0</sub> (Figure 4A). These results were also reflected qualitatively in the obtained stained images (Figure 4B). GFPGER<sub>120</sub> showed particularly robust VE-cadherin staining suggestive of strong cell-cell interactions. Adherent ECs were not detected nor observed on plain PVA samples, similar to results from our work (Journey et al., 2018; Anderson et al., 2019).

## Thrombogenicity of Surfaces

### Static *in vitro* Platelet Adhesion Quantification

We studied the extent to which peptide-modified PVA samples supported static platelet adhesion using a solution of purified

human platelets *in vitro*. There were no statistical differences found (Figure 5).

### Fibrin Clotting Time, Rate of Fibrin Generation *in vitro*, and FXII Activation

We quantified the time to fibrin clot formation in PPP and the rate of fibrin formation on peptide-modified PVA samples. GFPGER-modified samples all had prolonged times to initiation of fibrin generation when compared to CDI only-modified samples (Figure 6A). There were no significant differences observed in the rate of fibrin generation in the presence of GFPGER-modified PVA films when compared to the baseline (Figure 6B).

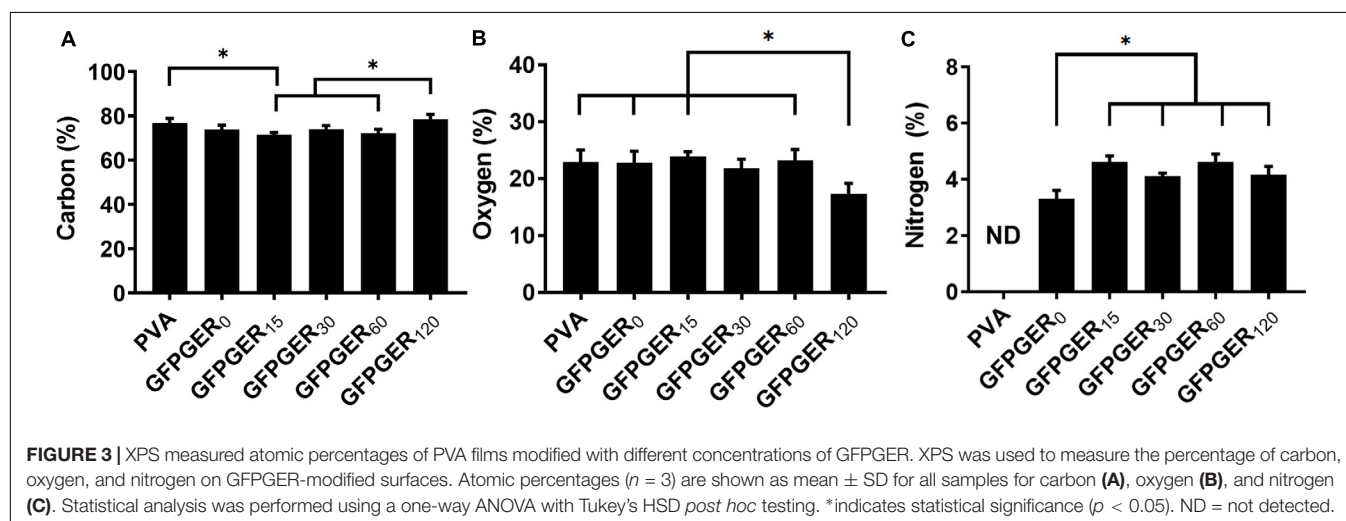
To determine if the contact pathway was playing a role in the difference of time to initiation of fibrin formation, activation of FXII was measured (Figure 6C). There were no significant differences observed between unmodified and GFPGER-modified PVA samples.

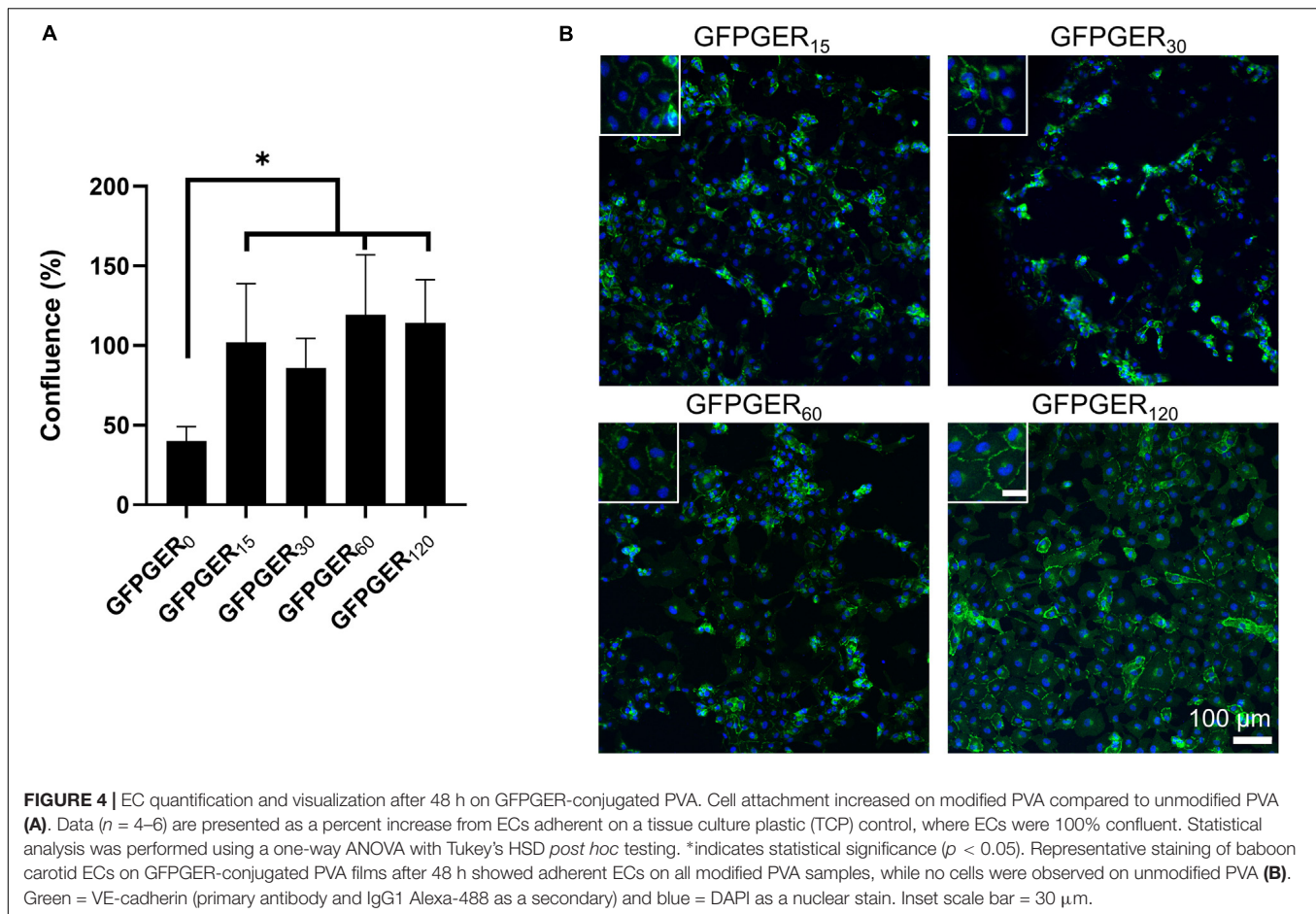
### Whole Blood Platelet and Fibrin Adhesion Under Flow Quantification

We then examined platelet and fibrin deposition onto GFPGER-modified tubes from whole, flowing blood with *ex vivo* thrombosis testing. We observed a significant decrease in platelet attachment onto GFPGER-modified samples when compared to the CDI only-modified samples for both concentrations of tested conjugated peptide, low (15 µg/mL) and high (120 µg/mL) (Figure 7A). Unmodified PVA samples were also quite low compared to the positive and clinical controls and consistent with both concentrations of GFPGER modifications. We observed a similar significant decrease in fibrin formation on GFPGER<sub>15</sub> and GFPGER<sub>120</sub> when compared to the CDI-only samples (Figure 7B).

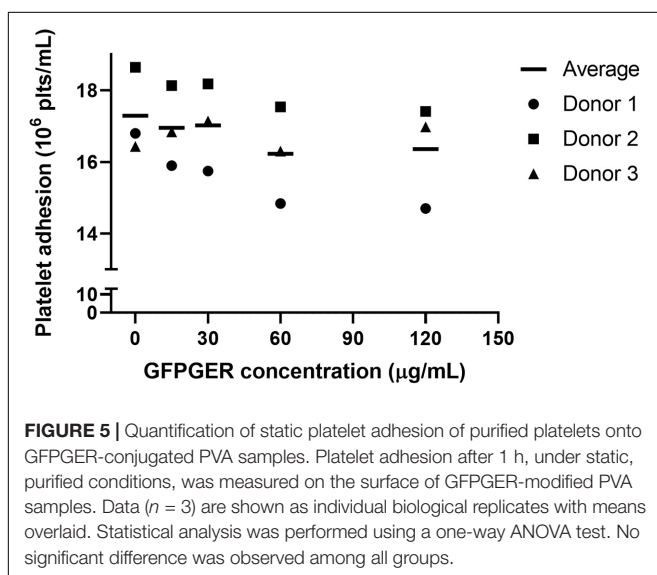
### MicroCT

Three-dimensional renderings of either the lumen (PVA and collagen-coated ePTFE) or thrombus (ePTFE) area along the length of the samples (volume) were generated based on cross-sectional ROI segmentation of each graft type (Figure 8).





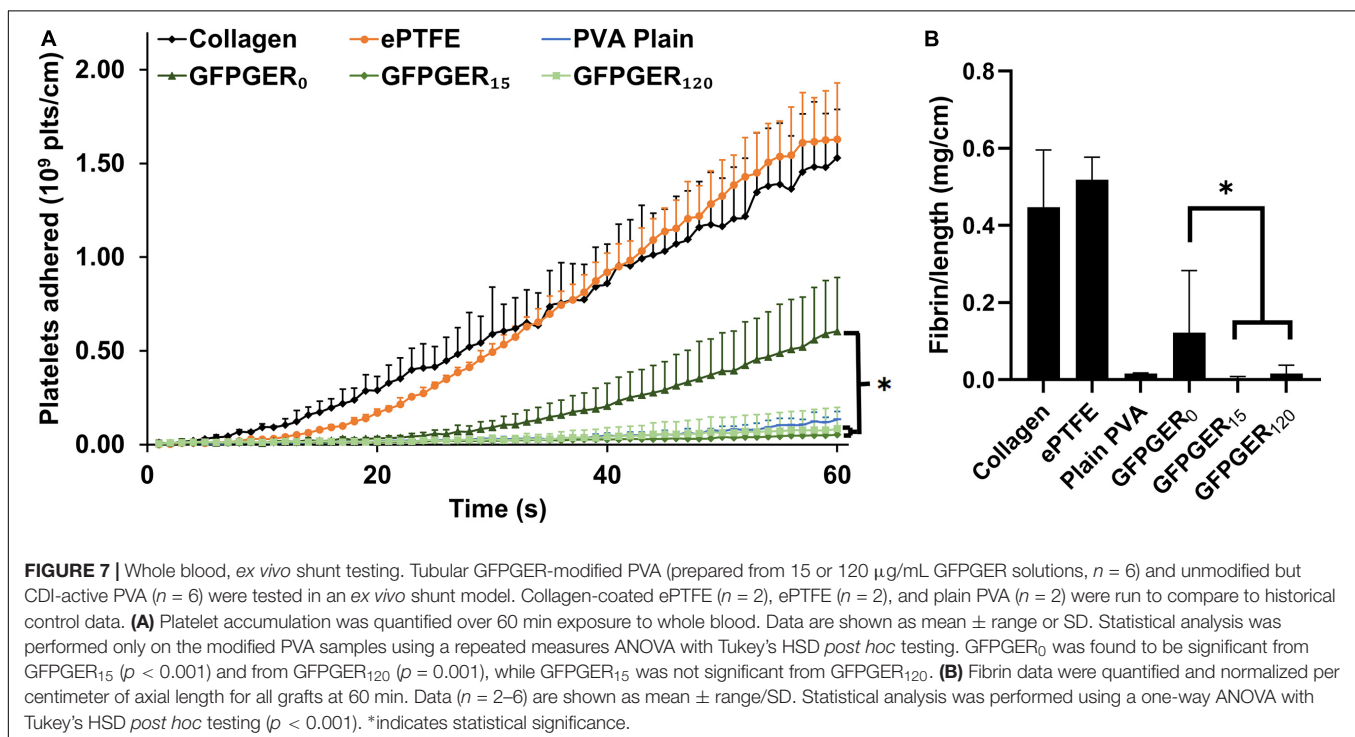
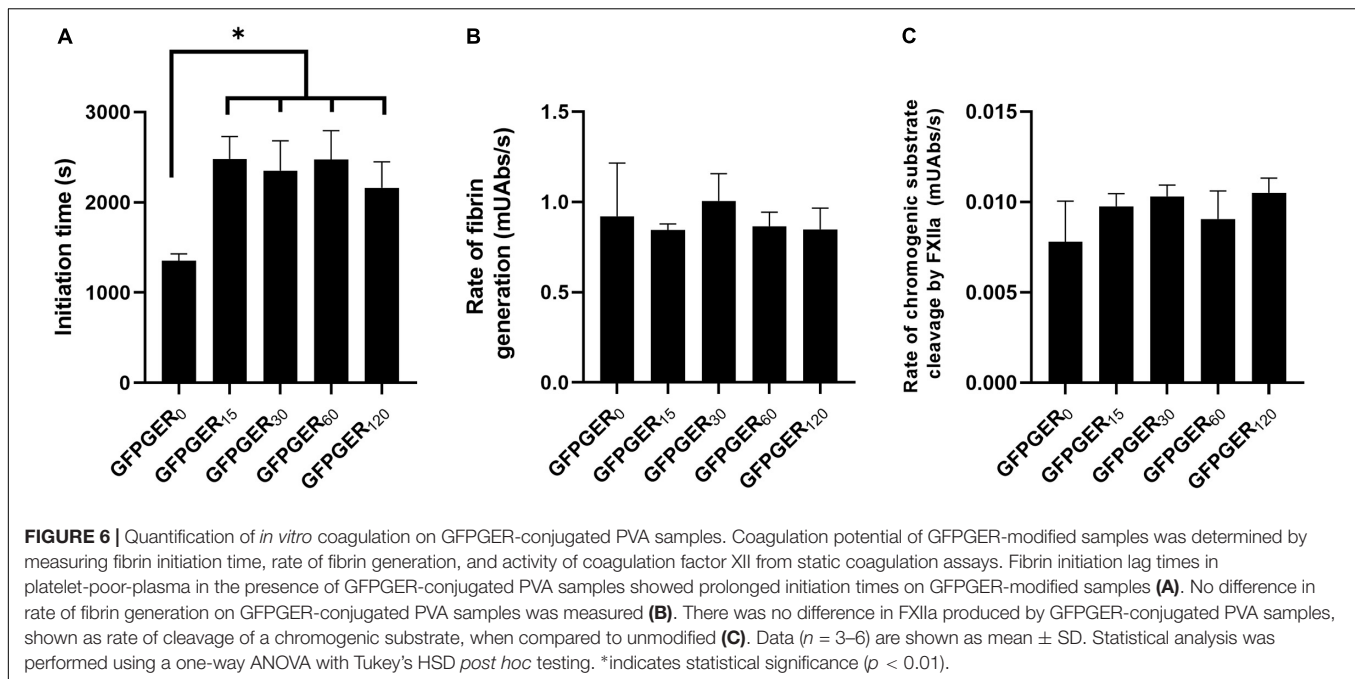
The measurement of thrombus area along the length of each individual graft as a percentage of the maximal luminal area was evaluated and representative samples are shown in **Figure 9**.



This representative GFPGER<sub>0</sub> sample (**Figure 9A**) had an average thrombus area percentage of  $12.10 \pm 3.99\%$  of the maximal luminal area, the representative GFPGER<sub>15</sub> sample (**Figure 9B**) had an overall thrombus area percentage of  $3.66 \pm 1.11\%$ , and the representative GFPGER<sub>120</sub> sample (**Figure 9C**) had an overall thrombus area of  $2.68 \pm 1.41\%$ . By capturing the area per sample slice, this method of analysis allows for the study of variability of an individual thrombus. The intrasample variability was quantified as the standard deviation of the individual thrombus area as a percent of the maximal luminal area of each sample type. These standard deviations were averaged and compared between all the PVA samples ( $n = 4$ ) resulting in variabilities of GFPGER<sub>0</sub> ( $4.17 \pm 0.87\%$ ), GFPGER<sub>15</sub> ( $3.10 \pm 2.84\%$ ), and GFPGER<sub>120</sub> ( $4.26 \pm 2.39\%$ ). These variability data were not statistically significant. The average Amira<sup>®</sup> luminal area for each type of PVA modifications ( $n = 4$ ) were also compared and showed no statistical significance between the three GFPGER-modified groups (**Figure 10**).

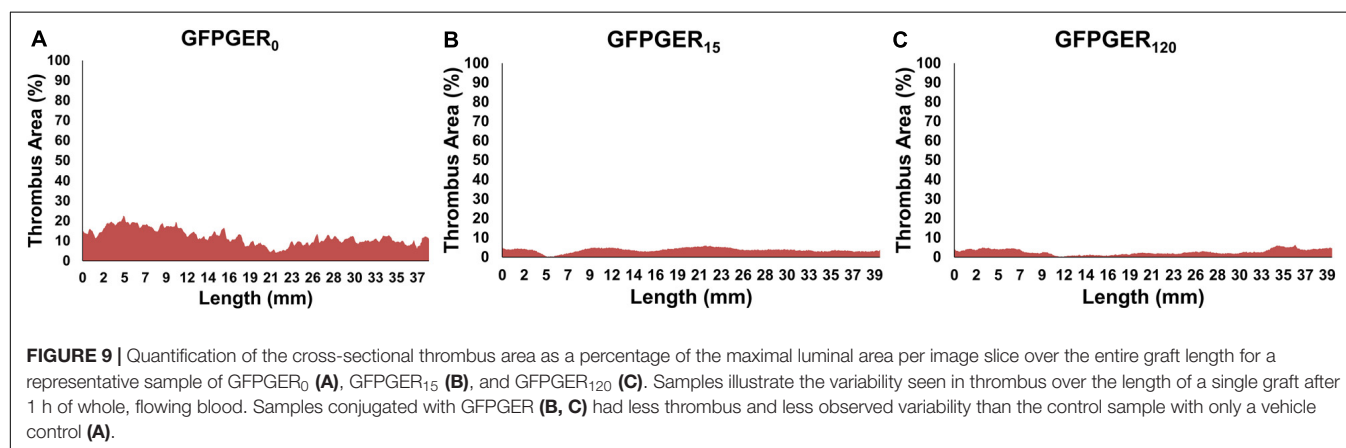
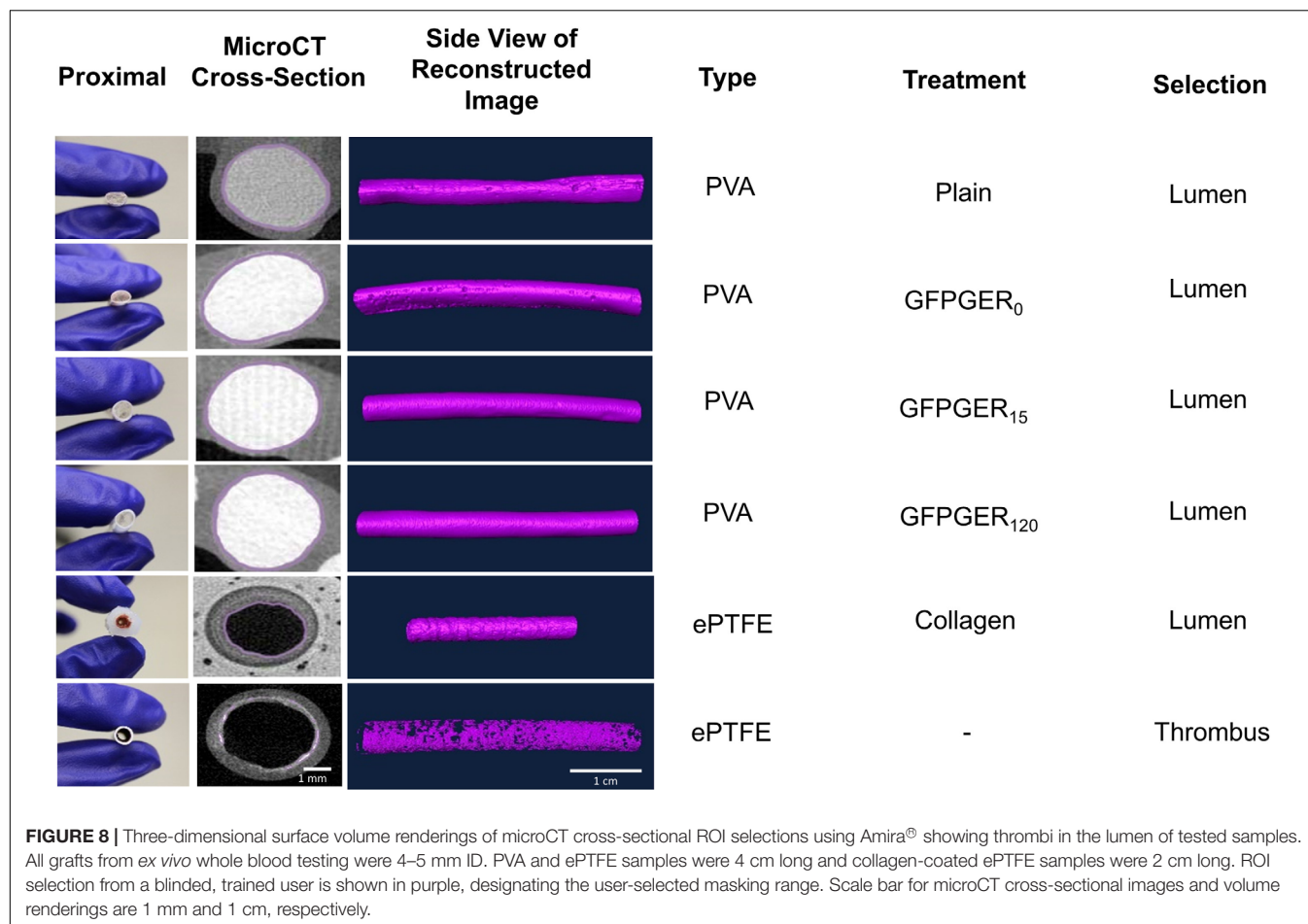
## DISCUSSION

Successful integration of synthetic small-diameter vascular grafts *in vivo* is impeded by a lack of *in situ* endothelialization and material-vessel compliance mismatch leading to neointimal



hyperplasia and thrombosis and potential loss of patency. Studies of compliant PVA biomaterials suggest that the low thrombogenicity of the material makes it a promising vascular graft material (Chaouat et al., 2008; Cutiongco et al., 2015a,b; Journey et al., 2018; Anderson et al., 2019; Rizwan et al., 2020; Yao et al., 2020). However, in the absence of surface modifications, PVA is unable to support EC attachment and growth. We hypothesized that the covalent attachment of a

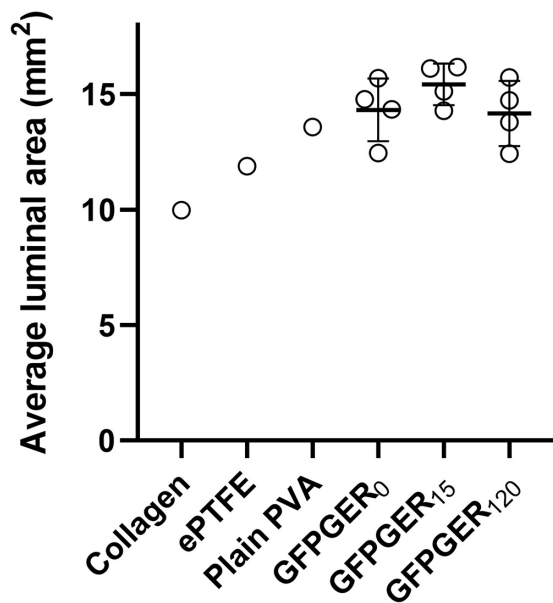
collagen-mimicking peptide to PVA would enhance EC adhesion, without compromising hemocompatibility of the material. Thus, we modified PVA hydrogel surfaces using CDI chemistry with the collagen-mimetic peptide, GFPGER, a low-cost, commercially available peptide, which specifically recognizes  $\alpha 1\beta 1$  and  $\alpha 2\beta 1$  integrins. We performed characterization of this modified surface as well as thorough *in vitro* and *ex vivo* hemocompatibility testing to assess the thrombotic potential of the material.



Using CDI as a crosslinker enabled a concentration dependent increase in GFPGER covalently bound to the PVA surface. Upon examination of the surface chemistries present on the modified hydrogels, we found an increase in C and N%, which indicated the introduction of amine groups onto the hydrogel, while the contact angle remained unchanged when compared to unmodified PVA. CDI has been used as a chemical crosslinker to modify the surface of polymers and hydrogels, including PVA,

with full length proteins (Nuttelman et al., 2001; Rizwan et al., 2020; Yao et al., 2020). CDI reacts with hydroxyl groups to form highly reactive, anchoring carbamate groups, which are then available to react with proteins or peptides for immobilization. Nuttelman et al. (2001) have previously used CDI to covalently attach fibronectin, an ECM protein found to play a role in cell adhesion and proliferation, to PVA. In their work, an 11-carbon spacer, bromoundecanoic acid, was added to the PVA





**FIGURE 10 |** Average luminal area of grafts after exposure to whole flowing blood in the *ex vivo* shunt. Collagen ( $n = 1$ ), ePTFE ( $n = 1$ ), plain PVA ( $n = 1$ ), GFPGER<sub>0</sub>, GFPGER<sub>15</sub>, and GFPGER<sub>120</sub> (all covalently modified GFPGER PVA samples were  $n = 4$ ) grafts were analyzed for the average cross-sectional area per slice of the lumen. Data are presented as the mean  $\pm$  SD of the average luminal area. Statistical analysis was performed only on experimental (GFPGER) samples using a one-way ANOVA. No statistical differences were observed.

hydrogel in an intermediate step before CDI activation, followed by the addition of fibronectin. Using this chemical scheme, they found that fibroblast cell attachment, proliferation, and migration were all significantly increased on fibronectin-conjugated PVA when compared to the unmodified material. However, the additional acid wash step is cumbersome and unnecessary, as shown by Rizwan et al. (2020), in which CDI was used to covalently attach gelatin, another cell-adhesive protein, directly to PVA. In this prior study, CDI alone was used to amine the surface of PVA before protein modification and the presence of gelatin on the PVA surface after the covalent modification was confirmed using Fourier transform infrared spectroscopy and XPS. Their surface characterization results showed a significant increase in nitrogen on the surface, which, like the results in the current study, indicate the presence of amine groups on the material surface. While some peptides, such as REDV (Lei et al., 2012), require spacers in order to retain their bioactivity, we successfully grafted GFPGER to PVA using CDI and the peptide remained active. Additionally, it is unlikely that using CDI chemistry to immobilize peptides to the surface of PVA would alter the mechanical properties of the material as was shown by previous work from our group (Rizwan et al., 2020), whereas there is more potential for physical alteration of the material when mixing peptides into the hydrogel network. To our knowledge, the work presented here is the first time that CDI has been used to covalently attach an ECM-mimicking peptide to the surface of PVA.

In our cell studies, we found that GFPGER-conjugated PVA supported the formation of an endothelial layer. This result is similar to cell attachment results obtained in our previous GFPGER-mixed PVA studies, where we found greater cell adhesion to GFPGER-mixed PVA samples when compared to PVA modified with collagen and other proteins and peptides (Anderson et al., 2019). In the current study, EC adhesion to most concentrations of GFPGER-modified PVA was significantly increased when compared to the CDI-only PVA baseline (GFPGER<sub>0</sub>), and a confluent layer of ECs was seen at the lowest and highest concentrations of GFPGER-conjugated samples. Despite the surface characterization which suggested that the GFPGER had not saturated the surface at the lower concentrations, it was interesting to see that all the examined concentrations had substantially increased cell adhesion that was not significantly different from other GFPGER concentrations. Plain PVA is biologically inert and does not support cell attachment alone (Cutiongco et al., 2015a, 2016b; Rizwan et al., 2020). The EC attachment attained is likely through the  $\alpha 1 \beta 1$  and  $\alpha 2 \beta 1$  integrins, which are well known to interact with GFPGER (Seo et al., 2010). This cell attachment is in agreement with the work of several groups, including ours (Chaouat et al., 2008; Munoz-Pinto et al., 2015; Anderson et al., 2019). GFPGER has been shown to support cell attachment and spreading on several surfaces, including a collagen-mimetic protein with GFPGER incorporated, Scl2<sub>GFPGER</sub>, coated onto tissue culture plastic (Seo et al., 2010) and a Scl2<sub>GFPGER</sub> incorporated PEG hydrogel (Cereceres et al., 2015). However, we were able to successfully conjugate GFPGER alone directly to a hydrogel surface via a simple chemical scheme, eliminating the need for site-directed mutagenesis or in-house protein synthesis.

Endothelial cells have been shown to have a distinct cobblestone appearance when cultured *in vitro* (Glynn and Hinds, 2014). Despite being cultured under identical conditions, the morphology of ECs on the different densities of GFPGER-conjugated PVA varied. ECs on GFPGER<sub>15</sub> were less spread and dominated by their nucleus whereas cells on GFPGER<sub>120</sub> appear to have spread more with a larger cytoplasm and the characteristic VE-Cadherin expression of *in vitro* cultured confluent ECs. The difference in peptide density was also demonstrated by the distinct cell wall borders observed around cells on hydrogels with the highest concentration of GFPGER. The increase in confluency and altered morphology on GFPGER<sub>120</sub> indicates that higher peptide concentrations may lead to phenotypic alteration of ECs. Hydrogels with a higher density of peptide have more binding sites for EC integrins, which encourages EC spreading to a greater degree than lower densities of EC adhesive peptide. However, this phenomenon was not observed in GFPGER-mixed PVA studies (Anderson et al., 2019); thus, more studies would need to be performed to understand the relationship between conjugated-peptide density and EC morphology and phenotype.

Importantly, the CDI-GFPGER modification of PVA did not increase platelet attachment and aggregation nor did the modification initiate clotting via the coagulation cascade. In our evaluation of GFPGER-conjugated PVA hydrogels, we found that all GFPGER-conjugated samples delayed the initiation time of fibrin clotting of PPP and decreased the fibrin formation

*ex vivo* when compared to the baseline material (GFPGER<sub>0</sub>). The lack of a significance difference found in the analysis of FXII activation on PVA samples suggested that contact pathway was not a potential coagulation initiation mechanism. However, more studies will have to be done to determine the mechanism of fibrin formation inhibition on GFPGER-modified samples. *In vitro* and *ex vivo* assessment of platelet attachment onto GFPGER-conjugated samples indicated no alteration of the PVA hemocompatibility. MicroCT analysis of thrombi formed on PVA samples after *ex vivo* testing showed minimal thrombus formation. These results contrast what is known about collagen-platelet interactions (Alberio and Dale, 1998; Farndale, 2006; Ruggeri and Mendolicchio, 2007; Harrison et al., 2011) and our previous GFPGER-mixed PVA studies (Anderson et al., 2019), where we found significantly higher platelet attachment to GFPGER-mixed samples when compared to unmodified PVA. The GFPGER binding site to  $\alpha 2\beta 1$  integrin on platelets is known to contribute to platelet activation and aggregation (Cossemans et al., 2008; Munnix et al., 2008). The divalent cation in the  $\alpha 2$  I-domain of the integrin coordinates with the glutamate in collagen or collagen-mimicking peptides, leading to a conformational change in the receptor followed by activation. Rich et al. (1999) predicted that the ligand-binding site in the  $\alpha 1$  subunit, which is found on ECs, is longer and more flexible than the  $\alpha 2$  subunit (Xu et al., 2000), which is present on the platelet surface. Previous studies found that GFPGER has slightly less affinity for the  $\alpha 2$  subunit, when compared to GFOGER, the native integrin binding sequence in collagen (Sipilä et al., 2018). This decreased affinity may be enough to minimally bind platelets and not activate key pathways leading to aggregation. Additionally, when GFPGER is directly bonded to the surface of PVA, as with the CDI linker, the glutamate that participates in integrin coordination is always available, leading to a controlled interaction between the material and biological components. Mixing the peptide into the PVA, as was done previously (Anderson et al., 2019), buries some of the key groups needed for integrin recognition, which likely led to non-specific binding of both cells and platelets (Anderson et al., 2019). Thus, CDI-mediated binding of GFPGER is a promising surface modification of blood contacting materials, which does not activate platelets.

Ultimately, the work presented herein supported our hypothesis that covalent attachment of GFPGER to the surface of PVA hydrogels would encourage endothelialization of the material, while attenuating thrombosis. We presented a systemic characterization and evaluation of GFPGER-conjugated PVA hydrogels using *in vitro* and *ex vivo* experimentation. We have shown that covalently modifying PVA with a low cost, widely available commercial peptide, GFPGER, for use as a synthetic vascular graft material may prevent thrombosis, while promoting *in situ* endothelialization of the material. A limitation of this work is the short time duration in which thrombogenesis is studied. Future work will include longer shunt study times and implantation studies using established rabbit (small animal) carotid bypass grafting (Cutiongco et al., 2016b) and non-human primate (large animal) aortoiliac bypass grafting (Anderson et al., 2018) implant models both with an end-to-side methodology to increase pre-clinical understanding

of material integration. These models provide the best approach for recapitulating clinical flow dynamics, which is frequently neglected in other vascular graft animal models (Zilla et al., 2007). Additionally, further *in vitro* work will be done to investigate specific EC phenotype and SMC proliferation on GFPGER-conjugated PVA. The promising results reported here with our novel synthetic biomaterial support the potential of this covalently linked surface modification either to coat off-the-shelf cardiovascular materials or for tissue-engineered constructs, supporting endothelialization without aggregating platelets.

## DATA AVAILABILITY STATEMENT

The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation.

## ETHICS STATEMENT

The animal study was reviewed and approved by the ONPRC Institutional Animal Care and Use Committee.

## AUTHOR CONTRIBUTIONS

NB performed *in vitro* thrombogenicity studies, peptide quantification studies, analyzed data, wrote and prepared the manuscript/figures for publication. HH performed *in vitro* thrombogenicity studies and endothelial cell attachment studies. MF performed microCT studies, associated analysis, and contributed to writing and preparing the manuscript/figures. YY performed surface characterization work, prepared tubular samples for arteriovenous shunt studies, analyzed data, and contributed to writing and preparing the manuscript/figures. EY and MH contributed to the original idea, experimental design, scientific guidance, discussions, laboratory space, and funding. DA contributed to the original idea, experimental design, scientific guidance, discussions, funding, *in vitro* thrombogenicity studies, endothelial cell attachment studies, and contributed to writing and preparing the manuscript/figures. All authors revised the manuscript.

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## SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fbioe.2020.621768/full#supplementary-material>

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Engineering Extracellular Matrix Proteins to Enhance Cardiac Regeneration After Myocardial Infarction

Hamid Esmaeili<sup>1</sup>, Chaoyang Li<sup>2</sup>, Xing Fu<sup>2\*</sup> and Jangwook P. Jung<sup>1\*</sup>

<sup>1</sup> Department of Biological Engineering, Louisiana State University, Baton Rouge, LA, United States, <sup>2</sup> School of Animal Sciences, Louisiana State University AgCenter, Baton Rouge, LA, United States

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### \*Correspondence:

Jangwook P. Jung  
jjung1@lsu.edu  
Xing Fu  
xfu1@agcenter.lsu.edu

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Engineering microenvironments for accelerated myocardial repair is a challenging goal. Cell therapy has evolved over a few decades to engraft therapeutic cells to replenish lost cardiomyocytes in the left ventricle. However, compelling evidence supports that tailoring specific signals to endogenous cells rather than the direct integration of therapeutic cells could be an attractive strategy for better clinical outcomes. Of many possible routes to instruct endogenous cells, we reviewed recent cases that extracellular matrix (ECM) proteins contribute to enhanced cardiomyocyte proliferation from neonates to adults. In addition, the presence of ECM proteins exerts biophysical regulation in tissue, leading to the control of microenvironments and adaptation for enhanced cardiomyocyte proliferation. Finally, we also summarized recent clinical trials exclusively using ECM proteins, further supporting the notion that engineering ECM proteins would be a critical strategy to enhance myocardial repair without taking any risks or complications of applying therapeutic cardiac cells.

**Keywords:** extracellular matrix, cardiomyocyte proliferation, myocardial infarction, cardiac repair, acellular therapeutics

## INTRODUCTION

Post-natal cardiomyocytes (CMs) are terminally differentiated cells in the heart and lack proliferative capacity. A withdrawal from the cell-cycle correlates with multinucleated and polyploid CMs (Derks and Bergmann, 2020). At the time of birth, the neonatal human heart comprises primarily mononucleated CMs and ~30% binucleated CMs, and this proportion of mononucleated and binucleated CMs does not change significantly after birth. The DNA of most nuclei is duplicated to become mononucleated tetraploid in childhood when the cells undergo hypertrophy (Bergmann et al., 2009). The overall arrest in CM division is due to the downregulation of cell cycle regulators (Walsh et al., 2010; Zebrowski et al., 2015).

After myocardial infarction (MI), several strategies to replenish lost CMs have been proposed. Engraftment of exogenous cells to restore the damaged myocardium is an attractive remedy to mitigate the progression of cardiac fibrosis. Exogenous sources include stem cell-derived CMs or cardiac progenitor cells. The first-generation cell therapy employed mesenchymal stem cells (MSCs) or similar derivatives (Perin et al., 2015), and the second-generation cell therapy (Cambria et al., 2017) utilized pluripotent stem cell (PSC)-derived CM (Chong et al., 2014; Shiba et al., 2016; Liu et al., 2018), which showed a certain extent of success for cardiac repair with stem

cells. However, a major problem associated with cell therapy includes relatively low retention and integration of delivered cells, where only 10–15% are retained regardless of the source (Hou et al., 2005), and only about 1% of injected cells remained after 1 month (Nguyen et al., 2016). In addition, significant arrhythmia, excessive immunosuppression, and potential teratoma formation are the major roadblocks of applying therapeutic cells toward clinical application (Berry et al., 2019; Bolli and Wysoczynski, 2019). Alternatively, cell therapy with endogenous resources is a preferred strategy. However, the contribution of endogenous cardiac stem cells is proven to be minimal, and 31 associated publications from the Anversa laboratory were retracted (Chien et al., 2019), leading to a pause on a clinical trial involving c-kit<sup>+</sup> cardiac stem cells by the National Institutes of Health [National Heart, Lung, and Blood Institute (NHLBI)] as of October 2018. Interestingly, these results can lead to the longstanding notion that it may be possible to achieve cardiac regeneration without physically presenting cells into the injured heart (French and Holmes, 2019). The consistent mismatch between insignificant cellular engraftment and significant functional improvement has led to our assertion that the functional benefits might well be derived from paracrine actions of the transplanted cells (French and Holmes, 2019), which initiated next-generation therapies with *cell-free* approaches (Cambria et al., 2017). Here, we succinctly summarize extracellular matrix (ECM) proteins directly relevant to or instructing CM proliferation to establish better strategies for enhanced cardiac repair.

## STIMULATION SIGNALS FOR CARDIOMYOCYTE PROLIFERATION

### Extracellular Matrices Associated With Cardiomyocyte Proliferation

From many recent studies (Frangogiannis, 2019), the composition and mechanical properties of the ECM (Chaudhuri et al., 2020) may play a critical role in inducing the regeneration of the myocardium (Yahalom-Ronen et al., 2015). The following studies are recent investigations on ECM proteins and their roles in CM proliferation.

#### Fibronectin

Fibronectin is a multidomain, high-molecular-weight glycoprotein, present at low levels in the ECM of the healthy heart. Proteomics analysis of ECM compositions with developmental ages showed that collagen I and III and laminins increase gradually from fetal to adult, while fibronectin decreased with development (Williams et al., 2014). *In vivo*, fibronectin is strongly upregulated in the heart after MI (Konstandin et al., 2013a). Thus, short-term induction of fibronectin following myocardial injury may be tied to a beneficial role in cardiac repair by CM proliferation. The same group proved that fibronectin promotes CM hypertrophy by nuclear factor of activated T cell (NFAT) *in vivo* and *in vitro*, while fibronectin attenuates the activation of physiological growth *in vitro* (Konstandin et al., 2013b). Co-culture of mammalian embryonic

cardiac fibroblast (cFB) and CM can promote CM proliferation, and fibronectin secreted by embryonic mouse cFB plays a pro-proliferative role in this process (Ieda et al., 2009). The mechanism of CM proliferation is only partially attributed to fibronectin or collagen type III that promoted CM proliferation by activating heparin-binding EGF-like GFs via  $\beta 1$  integrin signaling (Ieda et al., 2009). In zebrafish heart regeneration, loss-of-function approaches indicated that high expression of fibronectin does not remuscularize the heart (Wang et al., 2013), but fibronectin is necessary for functional regeneration by mobilizing and integrating CMs into the injured region. As such, cFB and fibronectin need additional players to proliferate CMs for cardiac repair.

#### Periostin

Periostin is a multimodular protein composed of a signal peptide necessary for secretion, a small cysteine-rich module for the formation of multimers via disulfide bonds, four FAS1 (fascin-1) domains interacting with integrins, and a hydrophilic C-terminal region known to interact with other ECM proteins (Kii et al., 2010). Periostin is primarily expressed in the developing heart, but not in healthy adult ventricular myocardium (Snider et al., 2008; Hortells et al., 2020). After acute MI, periostin is re-expressed in the infarct border zones by activated cFBs (Kanisicak et al., 2016). It was reported earlier that periostin can switch differentiated mononucleated CMs into the cell cycle and induce cardiac regeneration with improved myocardial function, as evidenced by an increase in DNA synthesis, aurora B kinase detection, and CM cytokinesis (Kühn et al., 2007). Recently, a report investigated the impact of genetic ablation of periostin in neonatal mice following MI and showed that periostin mediates PI3K/GSK3 $\beta$ /cyclin D1 signaling pathway for myocardial regeneration (Chen et al., 2017). However, this conclusion still remains controversial in adult mice with inducible expression of full-length periostin in that periostin is abundant in the infarcted mouse myocardium in the absence of regeneration (Lorts et al., 2009). Moreover, the periostin-induced cell cycle reentry is mediated in an integrin-dependent manner, which may impact other non-CMs expressing integrins (Kühn et al., 2007).

#### Agrin

Agrin is a heparan sulfate proteoglycan. It harbors three laminin-globular (LG) domains within its C-terminal region, and the first two LG domains (LG1 and LG2) are sufficient for binding to  $\alpha$ -dystroglycan ( $\alpha$ DG). Agrin promotes cell cycle reentry in both neonatal and adult mice (Bassat et al., 2017), and a separate study showed that Hippo/Yap signaling is a key signaling mechanism to mediate endogenous CM dedifferentiation and proliferation (Morikawa et al., 2017). *In vitro* administration of C-terminal agrin from post-natal day 1 (P1) increased CM proliferation (Bassat et al., 2017). The injection of recombinant agrin to the myocardium after MI in juvenile and adult mice also induced CM cell cycle reentry in the healthy myocardium adjacent to the infarcted regions, resulting in reduced scar size and improved cardiac function (Bassat et al., 2017). Although a single administration

of agrin promotes cardiac regeneration in adult mice after MI, the degree of CM proliferation observed in this model suggests that additional therapeutic mechanisms are required for functional regeneration of the myocardium (Bassat et al., 2017). In a preclinical porcine model of ischemia reperfusion, local (antegrade) delivery of a single dose of recombinant agrin into the infarcted heart resulted in significant improvement in heart function, infarct size, improved angiogenesis, suppressed inflammatory response, and cell cycle reentry (Baehr et al., 2020). Recent studies reported that binding of agrin to  $\alpha$ DG could contribute to enhanced CM proliferation. To expedite such changes, modulating the stiffness of microenvironments could synergistically initiate CM proliferation via dedifferentiation–proliferation–redifferentiation of CMs (Yahalom-Ronen et al., 2015; Wang et al., 2017; Judd et al., 2019).

### Slit-2 and Nephronectin

Slit-2 is a neuronal protein and is the only binding partner of  $\alpha$ DG with a single LG domain, while two other homologous Slit-1 and Slit-3 are not yet reported to bind to  $\alpha$ DG (Wright et al., 2012). Nephronectin is expressed in CMs throughout the heart and is secreted into the cardiac jelly (Patra et al., 2011). From embryonic cFB-derived ECMs, Slit-2 and nephronectin promote CM cytokinesis both *in vitro* and *in vivo* (Wu et al., 2020), but not cell cycle entry of post-natal CMs. The authors postulated that Slit-2 and nephronectin may act directly on CM and activate intracellular signaling pathways, such as RhoA (Backer et al., 2018).

### Decellularized Extracellular Matrix

Instead of a single ECM component, decellularized zebrafish cardiac ECM (zECM) was intramyocardially injected to treat adult mice after MI (Chen et al., 2016). Given the high regenerative capacity of adult zebrafish hearts, decellularized zECM made from normal or healing hearts can induce mammalian heart regeneration. In a mouse model of acute MI, although a single injection of both normal and healing zECM improved cardiac functional recovery and repair, the healing zECM induced better improvements on heart function. Groups treated with zECM exhibited proliferation of the remaining CMs and multiple cardiac precursor cell populations and reactivation of ErbB2 expression in CMs. NRG1, a mitogen of CMs and a ligand of ErbB2/ErbB4 complex, was detected in zECM but only minimally in murine ECM. The presence of NRG1 in zECM and the reactivation of its receptor ErbB2 in zECM-treated hearts are consistent with the observed proliferation of CMs and improvement of cardiac function. In addition, decellularized porcine myocardial-derived ECM hydrogels were developed (Seif-Naraghi et al., 2013) and showed increases in cardiac muscle and improvements in cardiac function following an injection into the infarct (Christman, 2019). Application of decellularized porcine cardiac ECM to cardiac explant (post-natal day 1) with simultaneous modulation of stiffness using BAPN (3-aminopropionitrile) and ribose (stiffening) (Wang et al., 2020) presented a case that both ECM proteins and mechanical properties of microenvironments are important modulators for cardiac regeneration. Thus, these cases indicate

that cardiac ECM-based therapeutics needs to combine with biomechanical modification.

### Microenvironmental Contribution to Cardiomyocyte Proliferation

In addition to CM proliferation via cell cycle reentry, heart regeneration requires dedifferentiation, which indirectly initiates proliferation, and the migration of CM to the injured sites, followed by redifferentiation. Clinically, mechanical unloading of diseased hearts can improve adverse remodeling and improve metabolism (Uriel et al., 2018). The stiffness of the ECM in the myocardium increases progressively, which is correlated with CM cell cycle arrest. By modulating the stiffness of polydimethylsiloxane (PDMS) substrates, compliant (5 kPa) substrates promoted dedifferentiation and proliferation of neonatal CMs including a disorganized sarcomere network and conspicuous cell cycle reentry (Yahalom-Ronen et al., 2015). In contrast, rigid (2 MPa) substrates facilitated karyokinesis (nuclear division) leading to binucleation. Thus, the compliant microenvironment could facilitate CM dedifferentiation and proliferation via its effect on the organization of the cytoskeleton (Yahalom-Ronen et al., 2015). In addition to the first report of neonatal (up to 7 days post-partum) cardiac regeneration (Porrello et al., 2011), another recent investigation found that neonatal regeneration sharply declines within 48 h, with hearts of 2-days-old mice responding to amputation with fibrosis, rather than regeneration (Notari et al., 2018). By comparing the global transcriptomes of mouse hearts at P1 and P2, the authors reported that most differentially expressed transcripts encode ECM proteins and structural constituents of the cytoskeleton. Pharmacological inhibition of the cross-linked enzyme LOX (lysyl oxidase) using BAPN rescued the ability of heart regeneration after apical amputation in P3 neonatal mice. On the other hand, stiffer substrates (10 to 50 kPa) were shown to increase CM proliferation and Yap activity in cultures of  $\beta$ -catenin double-knockout CMs ( $\alpha$ E-catenin and  $\alpha$ T-catenin), indicating that stabilizing cytoskeleton stimulates the nuclear translocation of Yap (Vite et al., 2018). The differences between published works may be attributed to varying experimental techniques including dimensionality, tissue vs. culture conditions, and stiffness range.

### CLINICAL APPLICATION OF EXTRACELLULAR MATRIX-BASED BIOMATERIALS FOR CARDIAC REPAIR

Here are a few examples for ECM-based biomaterials specifically for cardiac repair. More acellular injectable biomaterials for treating MI are reviewed elsewhere (Hernandez and Christman, 2017; Christman, 2019). In addition, commercially available ECM-based scaffolds for cardiac repair are discussed in the recent reviews therein (Swinehart and Badyalak, 2016; Pattar et al., 2019).

CorMatrix is a scaffold derived from small intestinal submucosa (SIS) and is the most widely used SIS-ECM product in cardiovascular surgery, which also recently received Food and

Drug Administration (FDA) approval (Mosala Nezhad et al., 2016). CorMatrix ECM cardiac patches were tested in clinical trials (ClinicalTrials.gov identifier: NCT02887768), claiming to promote endogenous cardiac regeneration. However, a study utilizing CorMatrix patches in infants with congenital heart disease did not show evidence of native cardiac tissue ingrowth within 21 months (Nelson et al., 2016). Further complications were reported from other clinical trials with a CorMatrix patch, including patch dehiscence after atrioventricular continuity reconstruction following massive posterior annulus decalcification and mitral valve replacement for mitral stenosis due to dystrophic calcification (Poulin et al., 2013). These results suggest that CorMatrix may elicit eosinophilic inflammation in human patients after implantation, perhaps via  $\alpha$ -gal (galactose- $\alpha$ -1,3-galactose) present in the porcine intestine (Mosala Nezhad et al., 2016), which probably supports the notion that completely defined therapeutics would be beneficial to avoid adverse reactions in human patients.

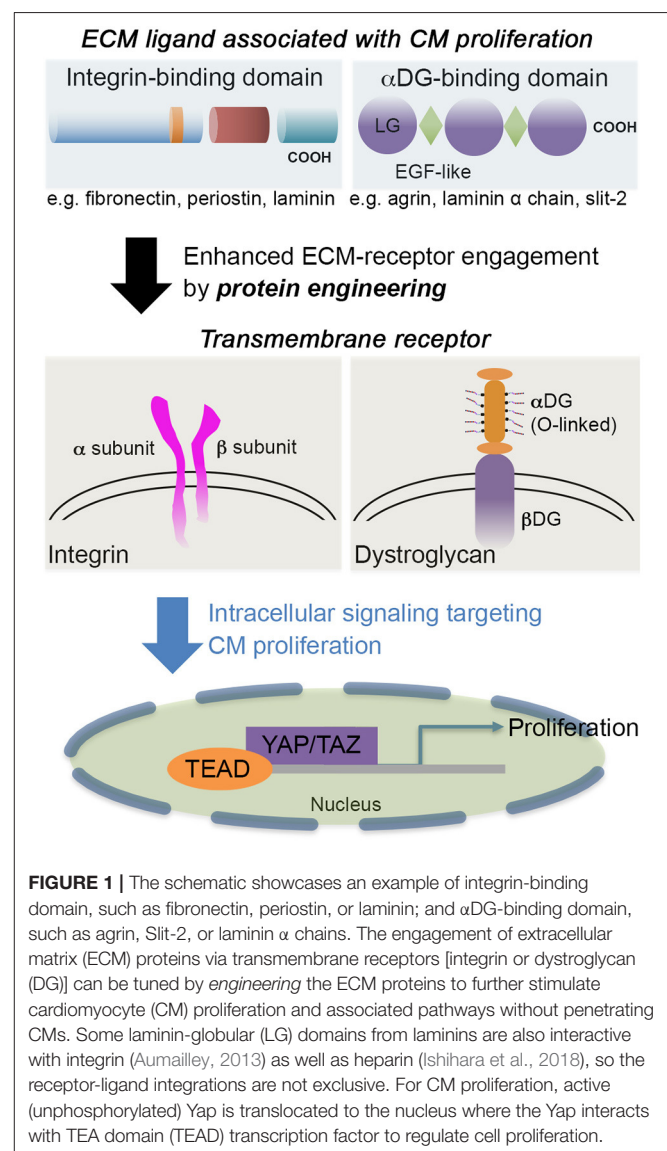
VentriGel is an ECM hydrogel derived from decellularized porcine myocardium (Singelyn et al., 2012; Seif-Naraghi et al., 2013; Hernandez and Christman, 2017) examined in a recently published clinical trial (ClinicalTrials.gov identifier: NCT02305602). The outcomes of the first-in-man trial highlighted the safety and efficacy of the treatment over 6 months (Traverse et al., 2019). VentriGel is a relatively weak hydrogel (Johnson et al., 2011), exhibiting two orders of magnitude lower stiffness than the stiffness (13 Pa of storage modulus at 8 mg/mL) of healthy, normal adult myocardium (around 10–15 kPa; Pandey et al., 2018). While there is yet sufficient evidence for the capability of decellularized hydrogel to promote endogenous cardiac regeneration, the ECM signals of normal healthy myocardium can prove a promising strategy for engineering biomaterials for cardiac repair.

## CONCLUSIONS AND OUTLOOK

From earlier studies treating p38 mitogen-activated protein (MAP) kinase inhibitor (SB203580) for CM mitosis (Engel et al., 2005, 2006), a number of stimulation signals have been identified for CM to reenter cell cycle and to promote [cyclin A2 (Shapiro et al., 2014); a cocktail of CDK1, CDK4, cyclin D1, and B1 (Mohamed et al., 2018); Tbx20 (Xiang et al., 2016); and hypoxia-inducible factor 1 $\alpha$  (HIF1 $\alpha$ ) (Guimarães-Camboa et al., 2015)] or inhibit [Meis1 (Mahmoud et al., 2013) and thyroid hormone (Hirose et al., 2019)] preexisting CM proliferation. Hippo (Heallen et al., 2011; Leach et al., 2017) and NRG1/ErbB4 (D'uva et al., 2015) pathways could be a molecular strategy to promote adult CM proliferation. The Hippo-DGC (dystrophin-glycoprotein complex)-agrin studies identified that viral delivery or a direct injection of CM proliferation agonist could be a viable cardiac repair strategy (Morikawa et al., 2017). More recently, ERBB2-ERK (extracellular signal-regulated kinase)-YAP mechanotransduction signaling was shown to trigger CM mitosis and epithelial-to-mesenchymal (EMT)-like transition toward phenotypic plasticity (Aharonov et al., 2020). Another important consideration is to exploit the metabolic switch from

mitochondrial oxidative phosphorylation to glycolysis to induce CM proliferation. Recent studies reported that CM proliferation can be enhanced by inhibiting fatty-acid utilization with deletion of pyruvate dehydrogenase kinase-4 (PDK4) (Cardoso et al., 2020), activating Nrg1/ErbB2 signaling (Honkoop et al., 2019), and activating PPAR $\delta$ /PDK1/p308Akt/GSK3 $\beta$ / $\beta$ -catenin-pathway (Magadum et al., 2017).

Translation of technologies to augment the stimulation signals requires a thoughtful examination, especially considering the oncogenic potential of activating growth pathways (Heallen et al., 2019). Another promising strategy of activating CM proliferation is to deliver an intrinsic extracellular factor (e.g., FSTL-1) via an engineered patch to stimulate endogenous repair (Wei et al., 2015). This acellular approach reduces the laborious effort to prepare therapeutic cells, while avoiding potential tumor formation and adverse immune rejection from the





patient. However, such an extracellular factor can also potentially stimulate non-CMs. Specific ligands that only allow engagement with CMs are needed to avoid adverse activation of the expansion of cFBs and their differentiation into myofibroblasts (specifically associated with fibrosis) (Fu et al., 2018).

Laminin  $\alpha$  chains and several proteoglycans harbor a few tandem arrays of LG domains. Despite the structural similarity between agrin and laminin, binding affinity to  $\alpha$ DG and the configuration of a tandem array of LG domains are distinct (Dempsey et al., 2019). This different feature may have conferred the different roles of ECM proteins containing LG domains in CM proliferation and differentiation. Thus, the therapeutic application of LG domain containing ECMs (agrin, laminin, and Slit-2, as depicted in **Figure 1**) needs to have further specification in their molecular nature and the receptors exclusively expressed in CM (for recent reviews on LG domain containing molecules, see Hohenester, 2019a,b; Yap et al., 2019).

ECM proteins that were applied for myocardial regeneration augmented the additional modification to enhance longevity and contribution for remuscularization (e.g., CorMatrix and VentriGel). ECM proteins binding to  $\alpha$ DG, agrin, or Slit-2 contributed to CM proliferation (Bigotti et al., 2020), but further mechanistic understanding requires establishing a better strategy for CM proliferation. In addition to providing novel engineering strategies for cardiac repair, more critical analysis of CM proliferation assays and induction of CM proliferation by microRNA, metabolic switch, or small molecule is necessary to inform the field to efficiently reach the goal of myocardial regeneration (Leone and Engel, 2019). A recent phase I clinical trial (ESCORT, NCT02057900) with fibrin and Matrigel

composites incorporating human embryonic stem cell (hESC)-derived cardiac progenitor cells (Isl1<sup>+</sup>) proved the safety of the approach (Menasche et al., 2018; Menasché, 2020), which showed a synergistic contribution to the regeneration of the myocardium without apparent integration of the delivered cardiac cells. Thus, a protein engineering strategy could be a starting point to target a specific receptor and well-defined signal pathways (e.g., Hippo/Yap) (Bassat et al., 2017; Morikawa et al., 2017) (**Figure 1**). Then, engineering multifactorial, acellular biomaterials with a simple deployment strategy could be a therapeutic goal (Christman, 2019). Ideally, both mechanical compensation and biochemical definition would be necessary. In addition, what levels of complexity we have to address are an important question to answer (Ogle et al., 2016) since a small increase in the ejection fraction of 5–10% in the function of the left ventricle would be a meaningful resolution to mitigate heart failure for patients suffering from post-injury.

## AUTHOR CONTRIBUTIONS

JJ and XF conceived the overall topics of discussion. All authors wrote, read, and approved the final manuscript.

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# Influence of the Mechanical Environment on the Regeneration of Osteochondral Defects

Sarah Davis<sup>1\*</sup>, Marta Roldo<sup>1</sup>, Gordon Blunn<sup>1</sup>, Gianluca Tozzi<sup>2</sup> and Tosca Roncada<sup>1\*</sup>

<sup>1</sup> School of Pharmacy and Biomedical Sciences, University of Portsmouth, Portsmouth, United Kingdom, <sup>2</sup> Zeiss Global Centre, School of Mechanical and Design Engineering, University of Portsmouth, Portsmouth, United Kingdom

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### \*Correspondence:

Sarah Davis  
sarah.davis@port.ac.uk  
Tosca Roncada  
tosca.roncada@port.ac.uk

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Articular cartilage is a highly specialised connective tissue of diarthrodial joints which provides a smooth, lubricated surface for joint articulation and plays a crucial role in the transmission of loads. *In vivo* cartilage is subjected to mechanical stimuli that are essential for cartilage development and the maintenance of a chondrocytic phenotype. Cartilage damage caused by traumatic injuries, ageing, or degradative diseases leads to impaired loading resistance and progressive degeneration of both the articular cartilage and the underlying subchondral bone. Since the tissue has limited self-repairing capacity due to its avascular nature, restoration of its mechanical properties is still a major challenge. Tissue engineering techniques have the potential to heal osteochondral defects using a combination of stem cells, growth factors, and biomaterials that could produce a biomechanically functional tissue, representative of native hyaline cartilage. However, current clinical approaches fail to repair full-thickness defects that include the underlying subchondral bone. Moreover, when tested *in vivo*, current tissue-engineered grafts show limited capacity to regenerate the damaged tissue due to poor integration with host cartilage and the failure to retain structural integrity after insertion, resulting in reduced mechanical function. The aim of this review is to examine the optimal characteristics of osteochondral scaffolds. Additionally, an overview on the latest biomaterials potentially able to replicate the natural mechanical environment of articular cartilage and their role in maintaining mechanical cues to drive chondrogenesis will be detailed, as well as the overall mechanical performance of grafts engineered using different technologies.

**Keywords:** osteochondral defects, tissue engineering, biomaterials, articular cartilage, mechanobiology, stem cells, mechanical testing

## OSTEOCHONDRAL DEFECTS

Osteochondral defects are areas of damage that involve both the articular cartilage and the underlying subchondral bone and can be caused by ageing, diseases (such as osteoarthritis and osteochondritis dissecans) or trauma. Osteoarthritis (OA) is a degenerative joint disease that affects over 250 million people worldwide (Hunter et al., 2014). Prevalence of the disease is increasing due to an ageing population and, in the US alone, 70 million people over the age of 65 are at risk of developing OA by the year 2030 (Bhatia et al., 2013). OA, originally thought to be a disease primarily affecting articular cartilage, is now considered to affect all tissues in the diarthrodial joint, including subchondral bone, ligaments, menisci, joint capsule, and synovial



membrane (Torres et al., 2006; Hunter et al., 2009; Lo et al., 2009; Krasnokutsky et al., 2011). As well as changes to the chondrocytes and the cartilage matrix, osteoarthritis is characterised by structural changes such as joint space narrowing, osteophyte formation and subchondral sclerosis that cause pain and joint immobilisation. Subtle changes to subchondral bone can be observed early, though the precise chronology of how these changes affect the OA process remains to be uncovered and the role of the subchondral bone in initiating and advancing disease progression is now receiving greater attention (Li et al., 2013). The crosstalk between subchondral bone and articular cartilage is complex and can induce biomechanical and biochemical changes in the overlying cartilage (Hu et al., 2020). The more obvious effect of changes to subchondral bone can be seen in conditions such as osteonecrosis, osteosclerosis, and Osteochondritis Dissecans (OCD). In osteonecrosis, and osteosclerosis imbalances in the bone remodelling process causes changes in bone turnover, mineralization, and subchondral bone volume, reducing overall bone density. This alters the biomechanical environment of the osteochondral unit and causes strain changes in the overlying cartilage during loading that may lead to pathological changes. OCD is a pathologic condition that affects subchondral bone formation resulting in subchondral bone fragments that disrupt the overlying articular surface. The exact causes of OCD are still unknown, yet repetitive microtrauma, abnormal endochondral ossification, and genetic factors are thought to play a role in its development (Grimm et al., 2014). Primarily, repetitive overloading or trauma is thought to disrupt the blood supply resulting in osteonecrosis. This in turn, may induce microcracks in the subchondral bone plate and underlying bone, resulting in fragmentation of bone and cartilage, causing inflammation, and joint pain. Another example is where cartilage loss adjacent to subchondral bone marrow lesions is common and is probably associated with changes in the modulation of this crosstalk (Hunter et al., 2009).

Articular cartilage is a viscoelastic tissue that provides a smooth and lubricated surface for joint movement, which also plays a key role in the absorption and dissipation of loads to the underlying subchondral bone. Healthy articular cartilage is an avascular, a-neural and a-lymphatic tissue, composed mainly of a proteoglycan rich extracellular matrix (ECM), type II collagen and chondrocytes. Mechanical properties of articular cartilage largely depend on ECM composition and organisation, however, mechanical stimulation is essential for cartilage development as well as maintaining cartilage homeostasis (Sanchez-Adams et al., 2014; Prein et al., 2016). Nevertheless, it has been demonstrated that excessive loading, either as single acute event or repetitive stresses, induces the expression of degradative enzymes such as metalloproteinase with a thrombospondin type 1 motif 5 (ADAMTS5) and matrix metalloproteinase-13 (MMP13), affecting matrix composition and hence playing a pivotal role in pathogenesis (Nakagawa et al., 2012; Buckwalter et al., 2013; Houard et al., 2013; Chang et al., 2019). Both OA (particularly post-traumatic osteoarthritis: PTOA) and OCD are associated with high-impact sports and abnormal loading/ joint injury, and therefore tend to affect highly stressed joints such as the knee and elbow. Since mechanical loading plays such a vital

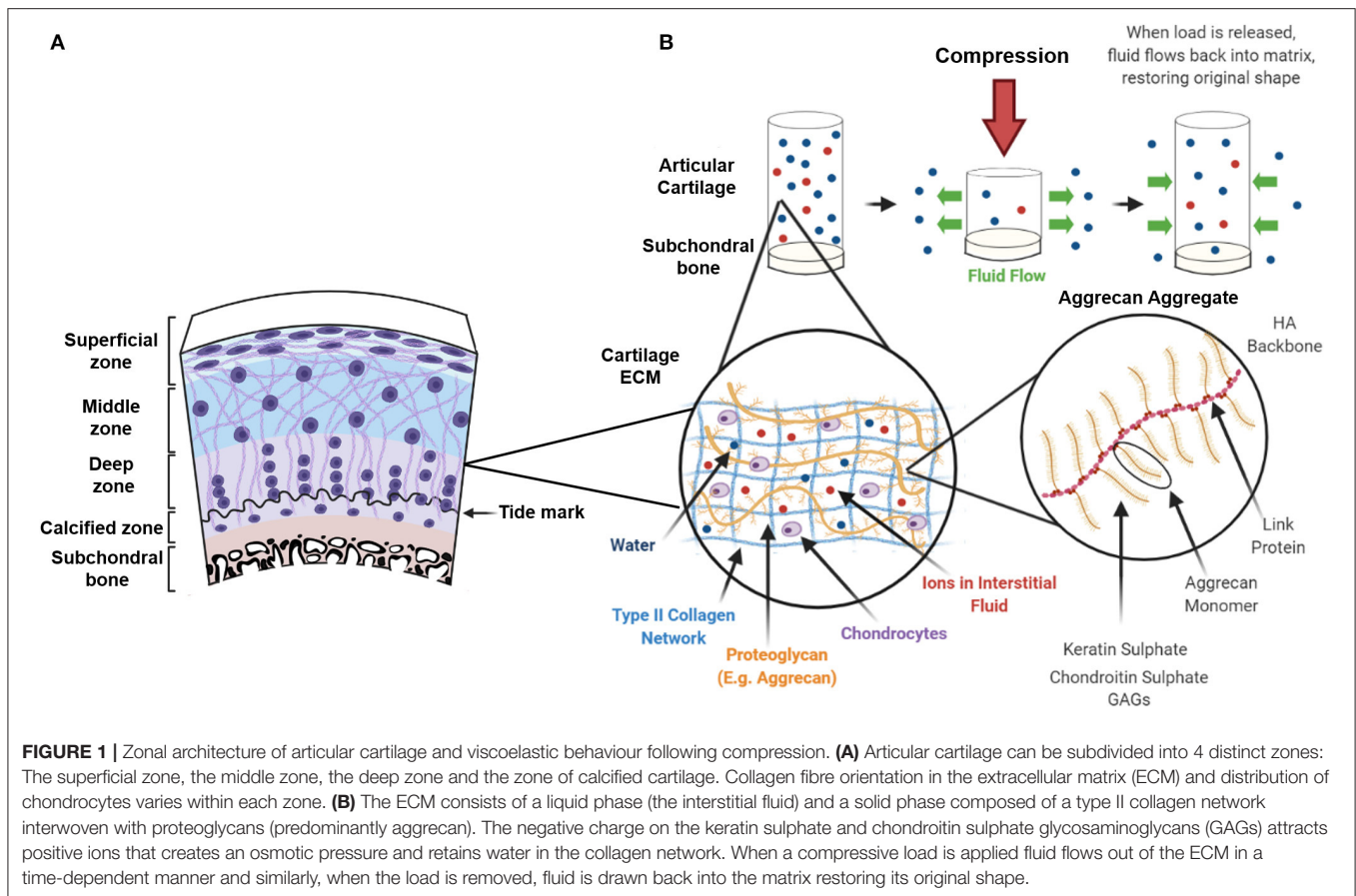
role in the initiation and progression of osteochondral defects and associated conditions, a deeper understanding of cartilage-bone mechanics is essential for developing better diagnosis and treatment methods.

This review will focus on the biomaterials able to replicate the natural mechanical environment of articular cartilage and the effect of mechanical cues resulting from the use of these scaffolds in directing and enhancing chondrogenesis. Importantly, osteochondral implants must be able to withstand the mechanical environment in the joint, which is responsible for these mechanical cues and that they are tested during their initial development with this environment in mind.

## THE MECHANICAL ENVIRONMENT OF NATURAL CARTILAGE

Articular cartilage can be subdivided into four distinct zones: the superficial zone, the middle zone, the deep zone and the calcified zone (**Figure 1A**). Each zone exhibits a particular arrangement and organisation of chondrocytes and ECM proteins, mainly collagen type II (Col II) and proteoglycans, determining the tensile strength, flexibility and load-bearing ability of cartilage (Baumann et al., 2019). Since articular cartilage is a non-uniform structure, it presents challenges when trying to determine strain patterns and relative stiffness. This is due to variation in the orientation of collagen fibres, proteoglycan distribution, and molecular/ion content throughout the depth of native cartilage, which is a function of the anatomical location within the joint, and the type of loading applied.

The superficial zone represents the 10–20% of articular cartilage and contains flattened chondrocytes. In the superficial zone, thin collagen fibres (mainly collagen type II and IX) are tightly packed and aligned parallel to the articular surface to protect deeper layers from shear stress (Sophia Fox et al., 2009; Correa and Lietman, 2017). Moreover, the parallel arrangement of collagen provides tensile stiffness and strength providing the tissue with high mechanical stability. This thin layer acts as a barrier regulating not only the diffusion transport of nutrients and oxygen to the underlying cartilage structures but also the ingress and egress of large biomolecules (Leddy et al., 2008). Lubricin, which is responsible for reducing surface friction, is produced by chondrocytes only in this zone (Correa and Lietman, 2017). The middle zone represents 40–60% of the articular cartilage and is characterised by sparsely distributed rounded chondrocytes and a proteoglycan rich ECM (consisting mainly of aggrecan) (Fisher et al., 2019). Within the middle zone, collagen type II has thicker fibres and is obliquely distributed. The deep zone is characterised by the highest proteoglycan content and the lowest water concentration. Collagen fibres are thick and run perpendicular to the articular surface (Baumann et al., 2019). Chondrocytes are parallel to the collagen fibres and arranged in columns. Due to the high content of negatively charged proteoglycans, the deep zone is responsible for providing the greatest compressive resistance to articular cartilage (Gilbert and Blain, 2018). The deep zone is reported to have the highest stiffness, along with the superficial zone, corresponding to



locations where the collagen fibre density is greatest, whereas the middle zone is characteristically softer (Schinagl et al., 1997; Bellucci and Seedhom, 2001). Articular cartilage stiffness has been reported to range from 0.1 to 6.2 MPa, with variabilities among studies that depend on sample type and testing setting (Boschetti et al., 2004; Robinson et al., 2016; Patel et al., 2019; Zheng et al., 2019; Guimarães et al., 2020). Within the middle and the deep zone, each chondrocyte is surrounded by a 2–4  $\mu\text{m}$  thick collagen type VI rich pericellular matrix (PCM), which forms the chondron. The PCM seems to play a functional role in initiating signal transduction within the cartilage during load-bearing (Leddy et al., 2008). A study by McLeod et al. (2013) showed depth-dependent mechanical inhomogeneity of the elastic moduli of the ECM throughout the cartilage zones, yet zonal uniformity of the PCM elastic moduli in comparison. Cartilage stiffness has also shown to decrease with increasing severity of OA (Kleemann et al., 2005). The calcified zone is characterised by hypertrophic chondrocytes and has a high content of collagen type X (Col X). It anchors the collagen fibrils from the deep zone to the subchondral bone providing optimal integration and as it is infrequently penetrated by blood vessels it prevents vascularization of the articular cartilage. The zone of calcified cartilage also acts as a transitional zone and is important for reducing stress concentrations at the cartilage-bone interface (Boushell et al., 2017). The subchondral bone

plate starts at the tidemark separating calcified and non-calcified cartilage. It is a supportive structure that consists of calcified cartilage and underlying subchondral bone that allows the build-up of hydrostatic pressure (Hwang et al., 2008). Damage to the integrity of the subchondral bone affects the generation of hydrostatic pressures and the repair of osteochondral defects often fails to recognise the importance of the subchondral bone plate. In its natural environment, cartilage is subject to a variety of different types of mechanical forces, including tension, compression, shear stress, and torsion. Physiological load on articular cartilage ranges from 5 to 8 MPa during walking and can reach up to 18 MPa when undergoing other activities such as rising from a chair (Clements et al., 2001). Due to the impermeable nature of the calcified cartilage and the low hydraulic permeability of the subchondral bone plate, the resistance to fluid flow within the cartilage results in the build-up of hydrostatic pressures (Hwang et al., 2008). Articular cartilage is resistant to these loads due, in part, to its viscoelastic behaviour resulting from the inter-relationship between the proteoglycan aggregates of the ECM (often referred to as the solid phase), and the interstitial fluid or liquid phase. The negatively charged carboxyl and sulphate groups of the proteoglycans attracts positive ions and creates an osmotic pressure, restrained by the tensile properties of the type II collagen network, which provides the cartilage with its compressive stiffness (Ateshian

et al., 2004). When a constant force is applied, the interstitial fluid pressure increases, forcing fluid out of the porous ECM in a time-dependent manner, creating frictional drag until equilibrium is reached. This frictional drag is inversely proportional to its permeability (Mak, 1986) and gives the cartilage its viscoelastic creep and stress-relaxation characteristics during compression (Mow et al., 1980; Halonen et al., 2014). When strain is kept constant, stress on the tissue increases until it reaches a peak which, due to redistribution of fluid within the cartilage, relaxes over time until equilibrium is reached. Similarly, when the load is removed, fluid flows back into the matrix allowing the cartilage to return to its original state, hence giving the tissue its mechanical properties and ability to withstand compressive loads (Figure 1B). Structural and biochemical variations relating to degenerative changes following injury or pathological conditions such as OA, alters the fluid flow dynamics throughout the tissue and can further affect load-bearing and compressive capability.

Cartilage was originally described as a biphasic material by Mow et al. (1980), composed of the liquid and solid phases as previously described. However, the model was adapted into a triphasic material by Lai et al. (1991) to include the mechano-electrochemical behaviour of monovalent ions and later the model accounted for the polyvalent ions in the interstitial fluid as forces acting as part of a separate liquid or ion phase (Gu et al., 1998). Although the triphasic model is a more recent theory that encompasses a structurally more accurate description of the composition of articular cartilage, the biphasic model highlights the importance of osmotic and hydrostatic pressure within the cartilage and how the tissue resists both compressive and tensile forces (Ateshian et al., 2004). It should be noted that any successful osteochondral implant has to accommodate these forces.

## The Effect of Physiological Loading, Overuse, and Disuse on Articular Cartilage

The high and complex range of physiological loads applied to cartilage are critical for maintaining healthy joint function. Mechanical loading, in the form of moderate exercise, is one of the most important factors for maintaining a homeostatic environment and balancing the anabolic and catabolic response of chondrocytes for ECM synthesis and degradation. Numerous studies have shown reduction in pro-inflammatory cytokines (IL-1 $\beta$ , IL-6, TNF- $\alpha$ ), inflammatory mediators (COX-2, PGE<sub>2</sub> and NO) (Chowdhury et al., 2001; Fu et al., 2019) and reduction in matrix-degrading enzymes (MMPs and ADAMTSs) in response to dynamic compression (Sun et al., 2012). *In vitro* studies also confirm anti-inflammatory effects of loading, with an increase in both gene expression, synthesis of type II collagen, aggrecan production (Buschmann et al., 1999; Waldman et al., 2006; Iseki et al., 2019) and stimulation of chondrocyte growth, differentiation, and proliferation. It is also important to note that chondrocytes from different regions of cartilage constitutively express mRNA for cartilage structural proteins in different baseline levels and respond differently to mechanical loading, suggesting that isolating chondrocytes from a non-load-bearing

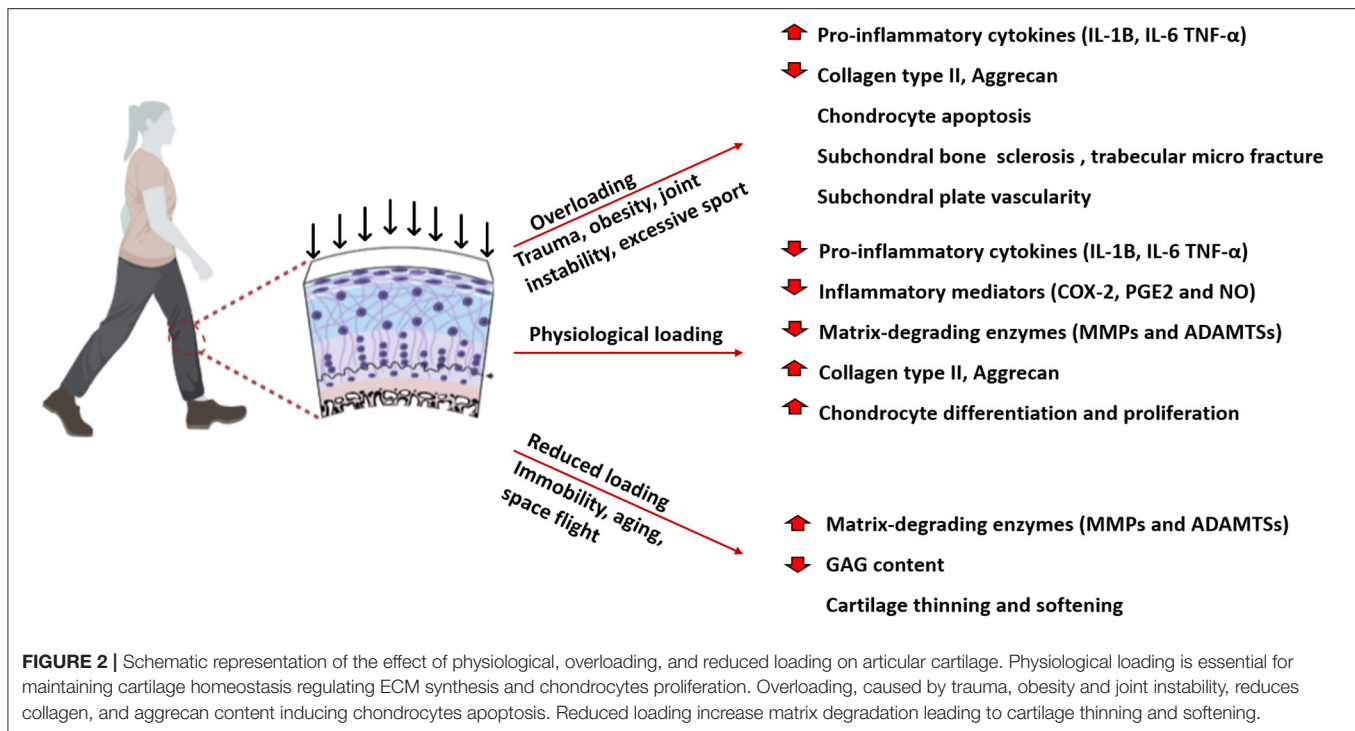
area might significantly affect the quality of the synthesised ECM (Bevill et al., 2009; Briant et al., 2015).

Although there is a genetic predisposition to the development of OA, loading plays a contributory role. Physiological loading is important for maintaining joint homeostasis (Figure 2), whilst abnormal loading caused by obesity, immobilisation, joint instability, overuse, or trauma can cause cartilage degradation and are the main risk factors linked to the development of OA (Arden and Nevitt, 2006). Overloading of the joint, either as a single impact load or cyclic loading causes increased catabolism, chondrocyte necrosis and apoptosis and damage to the collagen network in a dose-dependent manner (Chen et al., 2001; Clements et al., 2004; Hosseini et al., 2014). Most studies report a critical threshold with chondrocyte apoptosis, GAG loss and increased production of inflammatory cytokines above this threshold load (Clements et al., 2001; D'Lima et al., 2001). Kerin et al. (1998) indicated that loads above 10 MPa can result in apoptosis. In comparison, using bovine explants, Loening et al. (2000) showed that chondrocyte apoptosis can occur at 4.5 MPa as an earlier response to injury which is later followed by degradation of the collagen network at 7–12 MPa (Loening et al., 2000). On the other hand reduced mobility, which is associated with low loading conditions results in upregulation of MMPs, softening and a reduction in proteoglycan content and cartilage thinning (Jurvelin et al., 1986; Vanwanseele et al., 2002; Leong et al., 2011). Impaired joint loading significantly affects articular cartilage ECM composition and as consequence cartilage becomes thinner with reduced ability to absorb loads and shocks resulting in excessive load transmission to the underlying subchondral bone. Abnormal mechanical load can induce bone marrow oedema and subchondral sclerosis (Beckwée et al., 2015; Eriksen, 2015; Donell, 2019).

Articular cartilage is not only sensitive to the type of force applied and the magnitude of load but also to the duration, direction, and frequency of loading (Komeili et al., 2019). Párraga Quiroga et al. (2017) showed that higher strain rates cause more damage to the collagen network, while lower strain rates cause more damage to the non-fibrillar matrix components and that overall cartilage damage is both load and rate dependent. A study by Sadeghi et al. (2015) showed increases in crack length and surface damage with increasing loading frequency above a normal level of 1 Hz. There may also be variation in the material properties of the articular cartilage, in that weight-bearing areas may be more functionally prepared for loading compared to non-weight-bearing areas, and that non-weight-bearing areas may be more susceptible to damage and fibrillation when subject to the same tribological stresses (Moore and Burris, 2015). These factors highlight variability and therefore difficulty for a standard osteochondral graft material to be able to replicate native cartilage in different regions and locations within the same joint, let alone variability between different joints and that under different loading conditions.

## Variations in Strain and Stiffness

In native osteochondral tissue under normal loading conditions, cartilage can experience strains of 2–9% and can reach up to 20–30% during vigorous activity, whereas the underlying



bone experiences strains of  $<1\%$  (Sanchez-Adams et al., 2014; Steinmetz et al., 2015). In addition, calcified cartilage is  $\sim 100$  times stiffer than hyaline cartilage and 10 times less stiff than underlying subchondral bone and this transitional zone plays a crucial role in the transmission of loads between these regions (Mente and Lewis, 1994; Madi et al., 2020).

Strain distribution patterns vary depending on the type of loading, with more uniform strains during dynamic loading than static loading conditions and tension-compression non-linearity also causes variations in tensile stiffness (Huang et al., 2005; Komeili et al., 2019). Since cartilage is anisotropic material, the tensile moduli varies depending on the direction of testing and shows increased stiffness parallel to the local split-line patterns which also varies throughout the cartilage depth (Kempson et al., 1973). Variations have also been reported between anatomical locations within the joint, and differences in tensile modulus have been observed between high and low weight-bearing regions. For example, Wong and Sah (2010) showed regional variations in tibial and femoral cartilage, with more axial strain present in tibial cartilage during joint articulation. Several studies have also reported that in tibial and femoral knee condyle, higher strains are present on the medial side compared to the lateral compartment which provides an explanation for differences in mechanical stiffness and is related to contact biomechanics at these sites (Liu et al., 2010; Cotofana et al., 2011; Coleman et al., 2013; Halonen et al., 2014).

Asymmetric strain patterns of natural cartilage create numerous challenges for tissue engineers when analysing strain distribution throughout the cartilage and for replicating this mechanical environment *in vitro* in order to enhance

the maturation of the tissue engineered construct. However, mechanical stimulation has successfully proven to enhance the properties of tissue engineered osteochondral grafts which will be discussed in more detail in this review. It is important to note that *in vivo* the complex interplay of other supportive tissues such as the menisci, tendons and ligaments (Halonen et al., 2014) which may also be compromised by trauma and OA.

## CURRENT CLINICAL THERAPIES

Due to its avascular nature, the lack of abundant nutrients and low cell density, cartilage has limited regenerative capacity (Lo Monaco et al., 2018; Medvedeva et al., 2018). The treatment modality for repairing osteochondral injuries is dependent on the depth and area of the defect. Several clinical treatments are available to treat osteochondral defects such as microfracture (marrow stimulation), and the use of osteochondral autografts and allografts (Nukavarapu and Dorcenus, 2013; Freitag et al., 2017; Mathis et al., 2018). Microfracture is a surgical technique used to treat chondral defects, it involves perforating the subchondral bone with tiny holes allowing bone marrow mesenchymal stem cells and biomolecules to infiltrate the defect (Erggelet and Vavken, 2016). However, this often promotes the formation of mechanically inferior fibrocartilage with little evidence of type II collagen deposition (Redondo et al., 2018). For the treatment of larger osteochondral defects, where subchondral bone damage is seen tissue grafts of both cartilage and bone may be used. Osteochondral autograft transfer and mosaicplasty, have been used to treat full-thickness defects up to  $4 \text{ cm}^2$ .



During this procedure, chondral defects are replaced with plugs of the patient's own healthy articular cartilage and bone that are harvested from non-weight-bearing areas and transferred to pre-drilled holes at the defect site (Rowland et al., 2019). The outcome depends on age, sex and size of the lesion. In the case of large lesions, up to 8–9 cm<sup>2</sup>, multiple plugs can be used but with a risk of significant donor site morbidity (Richter et al., 2016; Kato et al., 2018). Unlike autograft, allografts use full-thickness cartilage that can be harvested from locations that correlate with the defects to be filled allowing more precise matching of the size and contour of the articular surface (Assenmacher et al., 2016; Haber et al., 2019). Even though allografts can be performed as a single staged procedure and have shown good survival rate in short to medium term (5–10 years), long term follow-up has shown considerable reoperation (30.2%) and high failure rates (18.2%) over time (Familiari et al., 2018). Moreover, allografts are limited by the lack of tissue supply, low cell viability due to graft storage and possible immunorejection (Yang et al., 2017; Mathis et al., 2018).

Among the current clinical therapies, multi-layered cell-free scaffolds have been considered and are currently under pre-clinical and clinical evaluation. TruFit CB (Smith and Nephew) is a synthetic plug designed to be used with microfracture in order to improve the mechanical stability of the defect. Initial studies showed positive results with the regeneration of cartilage in a goat model, however, clinical studies revealed that 70% of patients required reoperation and the plug failed specifically in restoring the subchondral bone (Williams and Gamradt, 2008; Joshi et al., 2012). The bone-layer of TruFit CB is made of PGA and calcium phosphate, two materials that degrade quickly post-implantation and mechanical failure has resulted in the plug being withdrawn from the market (Fraser et al., 2016; Tseng et al., 2020). D'Ambrosi et al. (2019), investigated the clinical and radiological efficacy of MaioRegen, a try-layered collagen-based scaffold, in restoring osteochondral knee defects. Despite the promising satisfactory and reliable results at mid-term follow-up, this systematic review revealed that, in terms of clinical improvement at follow-up, MaioRegen is not superior to conservative treatment or other cartilage techniques. Therefore, there is still an unmet need for an optimal biomaterial system that favours simultaneous bone and cartilage regeneration. Although current clinical approaches can reduce pain and improve the quality of a patient's life, none of them has routinely achieved complete healing of the osteochondral lesion. Non-biological man-made materials can be used to partially replace the joint (e.g., unicompartmental knee replacement) or when the whole joint is severely affected it is likely that a total joint replacement (TJR) will be required as an end-stage intervention. In the elderly TJR is a successful end stage treatment for OA however, younger patients have a significantly higher risk of undergoing revision due to implant limited lifespan (25 years), periprosthetic joint infection or aseptic mechanical failure (Meehan et al., 2014; Stambough et al., 2014; Bayliss et al., 2017; Evans et al., 2019). To overcome these limitations, in the last two decades, research has focused on tissue engineering (TE) as a possible solution for osteochondral regeneration and repair of cartilage.

## TISSUE ENGINEERING APPROACHES

The most common tissue engineering approach involves the use of a biocompatible scaffold, cells (e.g., stem cells) and/or a combination of bioactive molecules such as growth factors and cytokines. Autologous chondrocyte implantation (ACI) is a procedure for the regeneration of cartilage introduced by Brittberg et al. (1994), where autologous chondrocytes are isolated from a non-load-bearing site of the cartilage, expanded *in vitro* for 4–6 weeks and subsequently injected under a periosteal flap that is sutured onto the cartilage positioned over the defect (Könst et al., 2012). Although this technique has been used for two decades with successful surgical outcomes, the main issue is that two operations are required, one to obtain the cells, using arthroscopy and the other usually an open procedure to implant the cells (Minas et al., 2014; Mistry et al., 2017; Zikria et al., 2019). Matrix-induced autologous chondrocytes implantation (MACI) was originally developed to improve the biological performance of autologous chondrocytes cells and simplify surgical procedures (Andriolo et al., 2020). As with ACI, chondrocytes are isolated from a non-load-bearing area and cultured *in vitro*, however, this approach aims to deliver autologous chondrocytes in a biopolymer membrane. MACI<sup>®</sup> is also the name of a commercially available membrane of porcine collagen type I/III (Genzyme, United States). Several types of membranes and scaffolds have been developed for MACI procedures such as Novocart<sup>®</sup>3D (TETEC Tissue Engineering Technologies AG, Germany) a collagen-chondroitin-sulphate based membrane, CaRes<sup>®</sup>–Cartilage Regeneration System (Arthro-Kinetics, Germany) a collagen type I matrix and Cartipatch<sup>®</sup> (Tissue Bank of France, France) a monolayer agarose-alginate hydrogel (Vilela et al., 2018). However, MACI failed to prevent fibrocartilaginous healing and the integration of the scaffold into host hyaline cartilage is still unsatisfactory due to the intrinsic features of fully differentiated chondrocytes with their poor capability of tissue remodelling. Moreover, MACI still requires a two-step surgery, cartilage biopsy and cell cultivation, thus increasing the total cost (Behrens et al., 2006; Zikria et al., 2019). To further improve ACI outcomes and obtain a more reliable tissue repair, third generation of ACI has been developed, in which autologous chondrocytes are cultured in 3D to form spherical aggregates with a self-synthesised extracellular matrix. Spheroids of human autologous matrix-associated chondrocytes (Spherox) is an advanced tissue medicinal product with European Medicines Agency (EMA) market approval for the treatment of osteochondral defects up to 10 cm<sup>2</sup> (Niemeyer et al., 2020). However, due to differences in cartilage phenotype isolating chondrocytes from a non-load-bearing area might significantly affect the quality of the synthesised ECM (Bevill et al., 2009; Briant et al., 2015).

So far there appears to be little difference in outcomes of these cell therapies and tissue engineering approaches when compared with osteochondral autograft transfer system, mosaicplasty or microfracture surgery. Further, when harvested *in vitro*, chondrocytes undergo dedifferentiation exhibiting a flattened, fibroblast-like morphology. In these conditions they

produce a higher amount of collagen type I and collagen type X inducing the formation of fibrocartilage. An advantage of growing spheroids of chondrocytes isolated from biopsies is that the cartilage phenotype is better maintained than when cells are grown on flat tissue culture plastic. However, all of these approaches fail to fully repair the lesion in severe osteochondral defects, where both subchondral bone and articular cartilage are damaged (Davies and Kuiper, 2019). A significant proportion of research is focusing on the use of stem cells for cartilage repair since a large number of cells can be obtained from different sources such as bone marrow, peripheral blood, adipose tissue, dental pulp, placenta, and the umbilical cord (Tozzi et al., 2016). However, selectively promoting stem cell differentiation into appropriate cell lineages *in situ* is still challenging. An expanding field of research has demonstrated that mechanical cues from the environment could drive tissue formation and maturation, suggesting that combining scaffolds with mechanical properties that can drive stem cell differentiation could provide a solution for osteochondral defects where both bone and cartilage formation is required.

## OSTEOCHONDRAL GRAFTS MATERIALS THAT CAN BE USED TO REPLICATE THE NATURAL MECHANICAL ENVIRONMENT

The development of an osteochondral implant that replicates the structure of articular cartilage and subchondral bone remains challenging for tissue engineers. Material selection plays a pivotal role in the development of osteochondral grafts as it potentially contributes to the mechanical properties of the scaffold (Jahr, 2017; Bonani et al., 2018).

### Natural Materials

Natural materials such as collagen, chitosan, hyaluronic acid, silk, and alginate have been extensively used in TE for their biocompatibility, degradability and bioactivity (Jeuken et al., 2016; Li et al., 2018b). Natural materials are often used in the form of hydrogels with a highly hydrated viscoelastic matrix, tunable swelling behaviour and mechanical properties depending on the type and degree of crosslinking (Catoira et al., 2019; Mantha et al., 2019). Moreover, natural materials provide multiple binding sites for cell-ECM interaction. Multiple scaffolds for osteochondral TE in the clinical market are mainly composed of collagen type I (NOVOCART®3D, MACI®, CaReS®, NeoCart®, Maioregen®) (Kon et al., 2009; Crawford et al., 2012; Petri et al., 2013; Saris et al., 2014; Zak et al., 2014). Collagen can be extracted from various tissues and sources, for example, studies have reported that purified collagen can be isolated from vertebrate (generally rat, bovine, porcine and sheep) skin, tendon, cartilage and bone as well as from marine invertebrates (jellyfish, sponges, octopus, squid, cuttlefish, starfish) (Barzideh et al., 2014; Langasco et al., 2017). Even though collagen type I does not represent the main component of articular cartilage, several studies have demonstrated its pro-chondrogenic effects (Calabrese et al., 2017a,b; Xia et al., 2018). Preference of type I collagen in TE is largely attributed to its availability, its

general biocompatibility and safety approvals granted by various agencies; however, high production costs and poor mechanical properties of pure collagen scaffolds are still major limitations (Table 1) (Dong and Lv, 2016; Ghodbane and Dunn, 2016). In comparison, Gelatin is derived by thermal denaturation of collagen and can be manufactured at a much lower cost and in larger quantities (Grover et al., 2012). Gelatin shows low antigenicity, it possesses integrin-binding sites, and it is completely resorbable *in vivo*. However, at body temperature gelatin hydrogels are not stable, limiting their possible use as a biomaterial. Van Den Bulcke et al. (2000) first described gelatin methacrylate (GelMA), a chemically modified form of gelatin that can be stabilised through photo-crosslinking allowing the formation of a hydrogel that is stable at body temperature. The modulus of GelMA-based biomaterial can be controlled by varying the degree of substitution and macromer concentration (Sadeghi et al., 2017), for example, Gan et al. (2019) has modified GelMA hydrogels by intercalating oligomers of dopamine methacrylate obtaining flexible hydrogels with compressive modulus of 2.5 MPa and shape-recovery ability. GelMA has also been used in combination with hyaluronic acid (HA), which forms the backbone of aggrecan and therefore plays a critical role in maintaining the viscoelastic and mechanical properties of cartilage (Hemmati-Sadeghi et al., 2018). HA acts also as a biochemical cue enhancing chondrogenic differentiation of MSCs, promoting chondrocyte proliferation and preventing chondrocyte de-differentiation by activating CD 44 (Chen et al., 2018; Li et al., 2018a; Yamagata et al., 2018). The use of HA in TE affects matrix deposition by cells, thus enhancing the dynamic and equilibrium moduli during *in vitro* culture (Levett et al., 2014). Recently silk fibroin (SF) has also been investigated in the context of osteochondral TE due to its biocompatibility, low immunogenicity, slow degradation rate, and remarkable mechanical properties (Qi et al., 2017). Silk has a high tensile strength (around 300–740 MPa) and depending on the source and production method, it is possible to obtain elastic moduli ranging from 1 MPa to 17 GPa, making it a favourable biomaterial not only for cartilage repair but also for subchondral bone (Koh et al., 2015; Peng et al., 2019). Li J. J. et al. (2015), developed a bi-layered scaffolds for osteochondral regeneration using silk fibroin for the cartilage layer and a silk-coated strontium-hardystonite-gahnite ceramic scaffold for the bone layer. The silk layer exhibited highly elastic behaviour showing 91% strain at failure, indicating that the silk scaffold could stretch to approximately twice its original length before breakage, which is desirable for the cartilage phase. When tested under compression the biphasic scaffold approximated the biomechanical behaviour of osteochondral tissue, as it could maintain structural integrity under large compressive stresses while retaining the ability for shape recovery when hydrated, in addition the stiff bone phase could withstand large compressive stresses with minimal deformation.

Among natural polysaccharides, both alginate and chitosan have potential for cartilage repair (Xu et al., 2008; Yao et al., 2016; Ewa-Choy et al., 2017; Henrionnet et al., 2017; Merlin Rajesh Lal et al., 2017; Ruvinov et al., 2018; Huang et al., 2019). Alginate is a biodegradable and biocompatible material, derived

**TABLE 1** | Summary of advantages, disadvantages, and mechanical properties of naturally-derived materials.

Natural materials	Advantages	Disadvantages	Mechanical properties	References
Collagen type I	Low immunogenicity Degraded <i>in vivo</i> by MMPs	High production cost Low mechanical properties	Permeability 0.044–0.072 mm4/Ns Compressive modulus 3.5–3.7 kPa	Ghodbane and Dunn, 2016
Gelatin	Manufactured at a lower cost and in large quantities Low antigenicity Resorbable	Not stable at body temperature	Compressive modulus 0.75–6 kPa	Chen S. et al., 2016
GelMA	Stabilised form of gelatin Photocrosslinkable Varying the degree of substitution is possible to vary mechanical properties	UV crosslinking may have a negative effects on encapsulated cells	Compressive modulus 2–30 kPa	Sadeghi et al., 2017 Klotz et al., 2016
Hyaluronic acid (HA)	Enache MSCs chondrogenic differentiation Maintaining viscoelastic and mechanical properties in native cartilage Can be physically and chemically modified	Rapid degradation and poor mechanical properties	Elastic modulus of modified HA 1–70 kPa	Chen C. H. et al., 2016; Li et al., 2018a; Yamagata et al., 2018 Lee et al., 2018 Trombino et al., 2019
Silk fibroin	Biocompatibility Low immunogenicity Slow degradation rate Remarkable mechanical properties	Brittleness and swelling behaviour limits its applications in tissue engineering	High tensile strength 300–700 MPa and elastic modulus ranging from 1MPa to 17GPa	Koh et al., 2015; Chen et al., 2018; Peng et al., 2019
Alginate	Biodegradable Biocompatible Re-differentiate chondrocytes after monolayer culture Support chondrogenic phenotype Tunable mechanical properties	Lack of adhesion ligands	Elastic modulus 0.15–0.55MPa	Kaklamani et al., 2014
Chitosan	Biocompatibility Biodegradability Antibacterial properties	Display poor mechanical properties	0.13–0.199 MPa	Thomas et al., 2017

from seaweed that is composed of  $\alpha$ -D-mannuronic acid and  $\beta$ -l-glucuronic acid. Studies have shown that it can support chondrogenic phenotype promoting a rounded morphology of isolated chondrocytes and the synthesis of type II collagen and proteoglycans (Homicz et al., 2003; Caron et al., 2012; Angelozzi et al., 2017; Aurich et al., 2018). Chondrogenic differentiation of stem cells isolated from bone marrow, adipose tissues, Wharton's Jelly, and dental pulp has been promoted by growing cells within alginate gels (Huang et al., 2015; Reppel et al., 2015; Ewa-Choy et al., 2017; Mata et al., 2017; Baba et al., 2018). Although much lower than the compressive modulus of native cartilage the mechanical properties of alginate scaffolds can be modified to give values of 0.15–0.55 MPa using divalent ions ( $Mg^{2+}$ ,  $Ca^{2+}$ , and  $Sr^{2+}$ ) (Kaklamani et al., 2014). However, the main limitation of alginate-based materials is the lack of adhesion ligands that are essential for cell-attachment and to overcome this, bioactive components such as collagen may be incorporated (Bian et al., 2011; Lee and Mooney, 2012; Ganesh et al., 2013).

Another natural polymer employed is chitosan, derived from partial deacetylation of chitin, used in TE for its biocompatibility, *in vivo* degradation and antibacterial properties (Cheung et al., 2015; Varun et al., 2017; Huang et al., 2019). Chitosan hydrogels have been shown to support the proliferation of chondrocytes and MSCs *in vitro* and to improve the deposition of cartilaginous ECM both *in vitro* and *in vivo* (Griffon et al., 2006; Elder et al.,

2013; Faikruea et al., 2013; Sheehy et al., 2015; Huang et al., 2019; Scalzone et al., 2019). However, since chitosan display poor mechanical properties, crosslinking or combination with other materials is required to optimise the elastic modulus for osteochondral TE (Muzzarelli et al., 2015; De Mori et al., 2019; Kusmono and Abdurrahim, 2019; Scalzone et al., 2019). Thomas et al. (2017) tuned the stiffness of chitosan-hydrogels by blending increasing concentrations of hyaluronic acid dialdehyde and the degree of crosslinking to obtain hydrogels with a Young's modulus of 0.13 MPa and 0.199 MPa. However, a reinforced chitosan-based scaffold failed to regenerate bone and cartilage *in vivo* suggesting that the crosslinking treatment may have affected its overall degradation (Roffi et al., 2019). Therefore, a careful balance between the mechanical properties and degradation rate should be considered when designing osteochondral scaffolds using this material.

## Synthetic Materials

Synthetic materials are attractive substitutes for load-bearing tissues, since the mechanical properties can be tailored by altering the molecular weight and/or via the use of different processing methods (Grigore, 2017). Synthetic polymers, including poly(ethylene glycol) (PEG), polylactide (PLA) and its derivatives poly(L-lactide) (PLLA) and poly(lactic-co-glycolic acid) (PLGA), polyglycolic acid (PGA), poly( $\epsilon$ -caprolactone)

**TABLE 2 |** Summary of advantages, disadvantages, and mechanical properties of synthetic materials.

Synthetic materials	Advantages	Disadvantages	Mechanical properties	References
PEG	High solubility in water Hydrophilicity Biocompatibility Inertness Non-immunogenicity	Lack of specific binding motifs for cell attachment	Equilibrium modulus 0.01–2.46 MPa Hydraulic permeability 10–13–10–16 m <sup>2</sup> /Pa Tensile modulus 0.02–3.5 MPa	Nguyen et al., 2012 Zhu, 2010
PGA	Biocompatible Bioresorbable	Loses its mechanical integrity between 2–4 weeks <i>in vivo</i>	Tensile modulus 7 GPa	Woodard and Grunlan, 2018; Gorth and Webster, 2011
PLA	PLA is more hydrophobic compared to PGA, leading to a slower hydrolysis rate.	Lack of specific binding motifs for cell attachment	Tensile modulus 3 GPa Tensile strength 50–70 MPa	Narayanan et al., 2016 Samavedi et al., 2013
PLGA	Modulation of Young's modulus and degradation rate, Sustained mechanical integrity after implantation	Lack of specific binding motifs for cell attachment	Compression storage modulus 3.2–4.6 MPa	Baker et al., 2009 Gentile et al., 2014
PVA	Biodegradable Biocompatible Adjustable mechanical properties	Lack of specific binding motifs for cell attachment	Tensile strength 1–17 MPa Elastic modulus 0.0012–0.85 MPa Low friction coefficients ( $\mu$ ) 0.02–0.05	Lin et al., 2017 Teixeira et al., 2019 Sánchez-Téllez et al., 2017
PCL	Adjustable mechanical strength Possibility to produce hydrogel, porous scaffold, electrospun nanofibers	Lack of specific binding motifs for cell attachment	Compressive modulus 6.63–56.46 MPa Tensile Modulus 6.03–46.04 MPa	Olubamiji et al., 2016

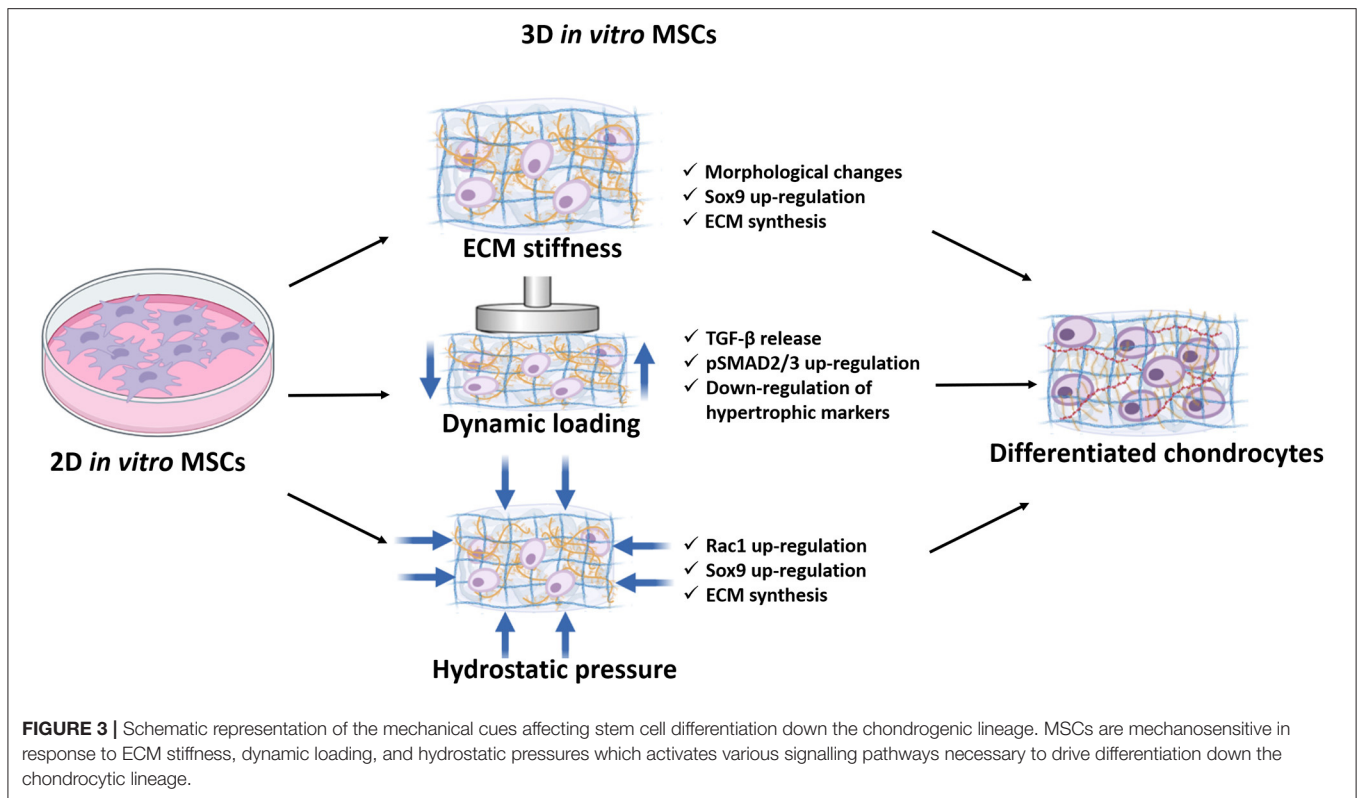
(PCL) and poly(vinyl alcohol) (PVA), are used to form hydrogels, porous scaffolds and nanofibrous scaffolds (Sánchez-Téllez et al., 2017; Yang et al., 2017; Castilho et al., 2018; Dai et al., 2018; Kudva et al., 2018; Critchley et al., 2020). The main disadvantage of these materials is the lack of specific binding motifs for cell attachment, but this can be improved through functionalization or by combining with more bioactive materials. PEG hydrogels have been used in TE due to their high solubility in water, hydrophilicity, biocompatibility, inertness, and non-immunogenicity (Table 2). They have also shown to maintain cell viability and promote chondrogenic ECM synthesis (Bryant and Anseth, 2002). By varying the molecular weight and the concentration of PEG precursors, Nguyen et al. (2012) obtained hydrogels with equilibrium modulus (0.01–2.46 MPa), hydraulic permeability [ranging from  $10^{-13}$  to  $10^{-16}$  (m<sup>2</sup>/Pa s)] and tensile modulus (0.02–3.5 MPa) similar to articular cartilage. Steinmetz et al. (2015) also developed a multi-layer PEG hydrogel resembling the zonal organisation of the osteochondral tissue. Although the compressive modulus did not match that of the native cartilage and bone when subject to mechanical loading, the strain distribution pattern was similar to osteochondral tissue with higher strain in the cartilage-like layer. When 7.5% apparent strains were applied to the hydrogel the local strains in the cartilage-like layer and in the bone-like layer were 15 and 2% respectively.

PGA exerts high tensile modulus (7 GPa) but due to its relatively hydrophilic nature and instability in aqueous solution loses its mechanical integrity between two and four weeks *in vivo* (Gunatillake and Adhikari, 2003; Gorth and Webster, 2011; Woodard and Grunlan, 2018). PLA exists in several isoforms and the presence of one extra methyl group makes it more hydrophobic compared to PGA, leading to a slower hydrolysis rate. PLA possesses a high tensile modulus (3 GPa) and strength (50–70 MPa) (Gorth and Webster, 2011; Samavedi et al., 2013).

PLGA can be synthesised using a different ratio of PGA and PLA that allows modulation of both Young's modulus and the degradation rate which can be, from a few weeks up to months, resulting in sustained mechanical integrity after implantation (Félix Lanao et al., 2013; Samavedi et al., 2013; Gentile et al., 2014). PVA is a biodegradable and biocompatible polymer, from which hydrogels can be prepared at different polymer concentrations to obtain tensile strengths in the cartilage range of 1–17 MPa as well as an elastic modulus up to 0.85 MPa (Karimi and Navidbakhsh, 2014; Lin et al., 2017; Teixeira et al., 2019). PVA hydrogels exhibit limited swelling when tested at osmotic pressures similar to that of articular cartilage, which is desirable for soft tissue engineering to preserve the initial size and shape and to prevent interfacial debonding (Holloway et al., 2011; Oliveira et al., 2019). A non-biodegradable PVA based hydrogel (Cartiva®) exerts biphasic behaviour similar to normal articular cartilage under compression and it is currently under clinical trial for first metatarsophalangeal joint hemiarthroplasty (Brandao et al., 2020).

PCL is an FDA approved biodegradable aliphatic linear polyester and it is one of the most investigated polymers for tissue engineering applications due to its adjustable mechanical strength. PCL can be used to produce porous scaffolds as well as electrospun nanofibers (Zhu et al., 2014; Panadero et al., 2016). Visser et al. (2015) incorporated PCL microfibers into GelMA obtaining reinforced hydrogels with mechanical properties similar to articular cartilage. Castilho et al. (2019) also used PCL to successfully develop a bi-layered construct that mimics the zonal structure as well as the functional properties of native cartilage. This construct incorporated a thin superficial tangential layer, mimicking the collagen organisation in the superficial layer of the cartilage, that improved the load-bearing properties of the micro-fibre reinforced hydrogel with a peak modulus of 473 kPa under unconfined compression as well as





exhibiting relaxation rates similar to those for native cartilage (Castilho et al., 2019). Controlling the mechanical properties of scaffolds for osteochondral TE is essential, not only to maintain structural integrity and withstand high mechanical loading *in vivo*, but also to provide environmental mechanical cues to selectively guide stem cell differentiation into the appropriate osteochondral phenotypes.

## MECHANICAL CUES AFFECTING STEM CELL DIFFERENTIATION

MSC commitment to the chondrocytic lineage is governed by TGF- $\beta$  and WNT/ $\beta$ -catenin signalling (Usami et al., 2016). In particular, the activation of TGF- $\beta$ /SMAD2/3 pathways is essential for the intracellular phosphorylation of Smad2 and Smad3, which then translocate to the nucleus to activate and stabilise the transcription factor Sex determining region Y (SRY) Box 9 (SOX9), that is the master regulator of chondrogenesis (Furumatsu et al., 2009; Coricor and Serra, 2016; Pfeifer et al., 2019). SOX9, along with SOX5 and SOX6 expression is required during embryonic development as well as in post-natal maintenance of articular cartilage regulating expression of ECM molecules, such as collagen (mainly types II, IX, XI) and proteoglycans (aggrecan, decorin).

To differentiate MSCs into chondrocytes the use of growth factors, such as TGF- $\beta$ , is usually required. However, its use in the clinic is limited as it leads to the expression of hypertrophic markers such as Col X, MMP13 and alkaline phosphatase

(ALP), which will eventually lead to cartilage mineralization. There is increasing evidence that environmental (such as low oxygen tension) and mechanical cues control stem cell fate. In particular (as described in **Figure 3**), MSCs are highly mechanosensitive and respond to both passive stimuli such as stiffness, and dynamic stimulation such as mechanical loading and hydrostatic pressure that signals through integrins and focal adhesion (FA) protein complex, transducing physical signals into biochemical signals.

## Regulation of MSCs Differentiation by ECM Stiffness

*In vivo* each tissue is characterised by a specific stiffness, which regulates tissue development and homeostasis by affecting cell migration, proliferation, morphology, cell phenotype and ECM protein production (Ehrbar et al., 2011; Handorf et al., 2015; Hwang et al., 2015; Du et al., 2016; Sun et al., 2017; Xia et al., 2017; Abbas et al., 2019; Chu et al., 2019; d'Angelo et al., 2019; Saidova and Vorobjev, 2020). Engler et al. (2006) originally explored the effect of stiffness on MSCs using polyacrylamide hydrogels mimicking the native elasticity of brain, muscle and osteoid. This work demonstrated that stiffness not only affects MSC morphology showing that the expression of the neurogenic marker  $\beta$  tubulin 3 was enhanced on soft substrates and Runx2 on stiff substrates. Interestingly, this work showed that growth factors tend to be less selective compared to matrix stiffness in driving lineage specification. MSCs pre-conditioned on a matrix with a specific stiffness for 3 weeks cannot be reprogrammed,

**TABLE 3** | Summary of the effect of mechanical cues on MSCs differentiation.

EFFECT OF MECHANICAL CUES ON MSCs				
Substrate stiffness	Soft substrates (1–30 kPa)	Intermediate substrate (0.1–5MPa)	Stiff substrates (5–300 MPa)	References
	MSCs on soft substrates exhibit a rounded chondrocyte-like morphology Higher expression levels of chondrogenic markers Unable to withstand mechanical loads	Mechanically competent Stiffness in the range of native cartilage MSCs found to express high level of SOX9 and Col II on substrates stiffness of 0.8MPa and 4.7MPa	Highly organised cytoskeleton Spindle-shape morphology	Engler et al., 2006; Park et al., 2011; Wang et al., 2016; Olivares-Navarrete et al., 2017
<b>Dynamic loading</b>	<b>0.15–1.5% compressive strain 1Hz</b>  Dynamic loading, delayed osteogenesis. Mineral deposits was diffuse in the unloaded condition while under dynamic loading was concentrated and spatially restricted to the central region	<b>10% compressive strain 1 Hz from day 0</b>  Compression from day 0 has negative effects on MSC chondrogenesis	<b>10% compressive strain 1 Hz after 1 week of pre-culture</b>  Dynamic culture increase synthesis of GAG aggrecan, Col II and increase expression SOX9 Upregulation of phosphorylated-SMAD2/3 MSCs under static culture MSCs exhibited higher of hypertrophic markers	Thorpe et al., 2010; Zhang et al., 2015; Sawatjui et al., 2018; Aziz et al., 2019; Cao et al., 2019
<b>Hydrostatic pressure</b>	<b>Low HP stimulation 100–300 kPa</b>  IHP upregulate osteogenic markers Increase expression of Runx2, ALP and osteopontin	<b>Physiological HP stimulation in the cartilage layer 1–10 MPa</b>  HP applied continuously it negatively affects SOX9, Coll II and aggrecan gene expression IHP positively affects SOX9 and Col II expression even without external growth factors and enhances cartilaginous matrix deposition	<b>High HP 25 MPa</b>  Inhibited aggrecan and Col II Pro-osteoarthritic effects	Correia et al., 2012; Li et al., 2016; Montagne et al., 2017; Stavenschi et al., 2018; Stavenschi and Hoey, 2019

suggesting that modulation of ECM modulus could be a powerful tool to drive stem cell differentiation. When cultured on stiff substrates, MSCs develop a highly organised cytoskeleton showing a spindle-shape morphology (Table 3). Conversely, MSCs on soft substrates exhibit a rounded chondrocyte-like morphology and express higher levels of chondrogenic markers. Park et al. (2011) compared collagen type II and GAG synthesis on a soft collagen hydrogel, on plastic coated with a thin layer of collagen and on polyacrylamide hydrogels with different stiffnesses (1 and 15 kPa). They showed an increase in expression of chondrogenic markers both on the soft collagen hydrogel and on the 1 kPa substrate. The effect of substrate stiffness together with biochemical cues was investigated by Wang et al. (2016). They showed that HA enhanced MSC chondrogenesis, evidenced by upregulated of aggrecan and Col II expression and this effect was more distinct when cells were grown in soft hydrogels (3 kPa), while this effect was reversed in the stiff hydrogel (90 kPa). It is important to note that cartilage stiffness varies between 0.1 and 6.2 MPa, and soft hydrogels will fail to maintain their structural integrity after implantation (Wang et al., 2016; Zheng et al., 2019; Guimarães et al., 2020). Olivares-Navarrete et al. (2017) compared both cytoskeletal organisation and gene expression of MSCs and auricular chondrocytes grown on methyl acrylate/methyl methacrylate (MA/MMA) polymer surfaces with elastic moduli ranging from 0.8 to 310 MPa mimicking the stiffness of articular cartilage and cortical bone. MSCs appeared to be elongated on the less stiff surfaces with a higher number of adhesion plaques on the 4.7 MPa substrate. After 7 days without

exogenous stimuli by cytokines or other factors associated with cartilage differentiation, the expression of SOX9, Col II, aggrecan and cartilage oligomeric matrix protein (COMP) in MSCs showed an increasing trend with decreasing stiffness. This work showed that mimicking the native elasticity of cartilage enhances chondrogenic phenotype without exogenous stimuli. Nevertheless, it is also important to consider that osteochondral tissue exhibits different stiffness among the different layers, and an implant displaying a layering or gradient approach with varying stiffness, might be more effective in reproducing the native architecture of the tissue as well as selectively promoting ECM synthesis.

## Role of Dynamic Loading in MSCs

During ambulation mechanical load plays an important role in maintenance and degeneration of articular cartilage affecting gene expression of Col II, aggrecan, and degenerative enzymes (MMPs). Interestingly dynamic stimulation also affects MSC differentiation and the quality of ECM synthesised. A study from Thorpe et al. (2008) revealed a negative effect of long term dynamic compression on MSCs cultured in agarose hydrogels. This study reported that unconfined compression at 10% strain and 0.5 Hz for 1 h/day significantly reduced GAG production and Col II synthesis compared to static culture. Interestingly, the application of dynamic compression from day 0 inhibits chondrogenesis even in the presence of TGF- $\beta$ 3 (Thorpe et al., 2010). In contrast, the inhibition of chondrogenesis in response to dynamic compression was not observed if the MSCs were

first allowed to undergo chondrogenesis. Consistent with these results, Sawatjui et al. (2018) studied the effect of dynamic compression of both MSCs and chondrocytes derived from osteoarthritic joints seeded on silk fibroin scaffold, pre-cultured for 1 week, and subsequently subjected to compression with 10% dynamic strain at 1 Hz, 1 h/day for 2 weeks. This study showed that dynamic compression significantly enhanced the synthesis of Col II and aggrecan along with an increase of compressive modulus. Cao et al. (2019) seeded rabbit derived MSCs into collagen scaffolds under 10% compressive sinusoidal strain at 1 Hz frequency, for 2 h/day for 21 days. Starting from the second week of culture, the morphology of MSCs in the dynamic culture group exhibited a rounded chondrocyte-like morphology, whereas cells remained spindle shaped in static culture. Dynamic culture also promoted GAG synthesis as well as aggrecan, Col II and SOX9 expression compared to the static culture. Zhang et al. (2015) demonstrated that delayed dynamic compression positively affected MSC chondrogenesis through phosphorylated-SMAD2/3 enhancing matrix deposition and suppressing hypertrophy. Further MSCs under free swelling condition exhibited higher expression of ERK (involved in chondrocyte hypertrophy) along with upregulation of MMP13, Runx2, and Col X. In addition, Gardner et al. (2017) demonstrated that multiaxial loads on MSC led to endogenous production and secretion of TGF- $\beta$ 1 as well as the activation of the secreted latent TGF- $\beta$ 1. Taken together these data suggest that dynamic load positively affects MSC chondrogenesis, however, MSCs should first be differentiated before applying loads. Consequently *in vitro* differentiation of stem cells prior to implantation could be critical for osteochondral tissue engineering.

## Hydrostatic Pressure and MSCs Differentiation

Cartilage ECM is characterised by a high water content and low permeability, and as a consequence when a load is applied the resistance of fluid flow generates hydrostatic pressure (HP). *In vivo* HP varies between 2 and 10 MPa with peaks of 18 MPa during intense activities such as jumping or running (Elder and Athanasiou, 2009; Correia et al., 2012). Several studies have demonstrated that the application of HP on MSCs might have a pro-chondrogenic effect. Angele et al. (2003) examined the effects of cyclic hydrostatic pressure on MSCs aggregates showing a significant increase in GAG and collagen content at days 14 and 28 compared to the unloaded control. Furthermore, Miyanishi et al. (2006a) studied MSCs in pellet culture exposed to intermittent hydrostatic pressure (IHP) and demonstrated an increase in expression of SOX9, Col II, and aggrecan with or without the addition of TGF- $\beta$ 3. In a second study the authors also demonstrated that the magnitude of loading modulated chondrogenic gene expression and cartilage matrix protein deposition in MSCs pellets in the presence of TGF- $\beta$ 3 suggesting that the magnitude of the load could enhance MSCs chondrogenesis (Miyanishi et al., 2006b). In fact, physiological levels of HP (5MPa) significantly enhance cartilaginous matrix deposition (Correia et al., 2012; Li et al., 2016). Conversely,

high HP (25 MPa up to 24 h) on the ATDC5 cell line markedly affecting the expression of matrix remodelling related genes, apoptosis-related genes and strongly inhibited aggrecan and Col II, suggesting that excessive loads induce pro-osteoarthritic effects (Montagne et al., 2017). Interestingly the use of low HP, in the range of 100–300 kPa, has been demonstrated to direct MSCs differentiation into the osteogenic lineage upregulating the expression of Runx2, ALP and osteopontin (Stavenschi et al., 2018; Stavenschi and Hoey, 2019). Not only the magnitude of load, but also the length of the stimulation affects matrix deposition. In fact, it has been shown that when the load is applied continuously, it negatively affects SOX9, Col II and aggrecan gene expression (Correia et al., 2012; Li et al., 2016).

One of the major limitations of cartilage tissue engineering is the formation of fibrocartilage, which has inferior mechanical properties compared to articular cartilage. HP appears to affect hypertrophic genes, increasing Col I, Col X and MMP13 (Ogawa et al., 2009, 2015; Li et al., 2016). Conversely, other studies revealed decreasing levels of Col I and Col X under IHP (Vinardell et al., 2012; Saha et al., 2017; Rieder et al., 2018). Freeman et al. (2017) demonstrated that HP without any external growth factors resulted in enhanced chondrogenesis along with reduction in hypertrophic markers. Additionally, when MSCs were stimulated with HP alone and subsequently induced to undergo osteogenic differentiation without any external mechanical stimulation, the production of hypertrophic markers was reduced compared to those exposed to chondrogenic growth factors alone. These studies suggested that the application of intermittent hydrostatic pressure could potentially lead to a stable differentiation of MSCs into the chondrogenic lineage without the use of growth factors. However, it is important to note that the intensity and the frequency of HP applied differ among studies, suggesting that standardisation is required to obtain consistent results.

## TESTING OSTEOCHONDRAL GRAFT MATERIALS

The success of osteochondral grafts depends on the restoration of surfaces representative of native articular cartilage to provide smooth joint movement during joint articulation. Implanted grafts also need to be structurally stable to withstand physiological loading conditions of up to 4–5 times body weight during walking (Morrison, 1970; Bellucci and Seedhom, 2001) with peak stresses in the knee ranging from 2 to 10 MPa and at a loading frequency of approximately 1 Hz (Brand, 2005; Sadeghi et al., 2015). Osteochondral defects cause high contact stresses at the rim, that vary depending on the size of the defect causing uneven strain distribution (Brown et al., 1991; Kock et al., 2008). These abnormal contact stresses and strains at the defect perimeter cause damage and chondrocyte death that could impair integration and healing of the graft, leading to reduced functionality of the joint, or cartilage damage (D'Lima et al., 2001; Wu et al., 2002). However, contact stresses can be restored to pre-operative levels, resembling intact cartilage depending on appropriate fitting, alignment, length and surface

of the graft (D'Lima et al., 2001; Koh et al., 2004; Kock et al., 2006, 2008). One of the major issues is that post-implantation osteochondral implants will be subject to continual cyclic loads, encompassing a range of shear and tensile forces which will affect the biological response of the graft and test the integration with the surrounding native cartilage. However, specific test methods to demonstrate the performance of these grafts have not yet been defined.

## Standardisation

Osteochondral grafts are classed by the International Standard Organisation (ISO) as implantable medical devices that as defined in ISO 13485, 2016, are implanted into the human body via surgical intervention and are intended to remain in place after the procedure. ISO 14630:2012 specifies the general requirements for non-active surgical implants, whereas ISO 21536:2007 is the level 3 standard referring more specifically to knee joint replacement implants. These standards include performance, design, materials, evaluation and sterilisation and the tests needed to demonstrate compliance with these regulations. More specific standards relating to tissue engineered cartilage constructs include the quantification of sulfated glycosaminoglycans (sGAG) (ISO 13019: 2018), and the evaluation of tissue morphology including collagen fibre orientation and anisotropy *in vivo* (ISO/TR 16379:2014) have also been defined. Despite these biological and clinical evaluation there are no specific requirements for mechanical testing, and there is uncertainty as to whether articular cartilage implants are classified as partial joint replacement implants and should therefore be subject to mechanical characterisation (Marchiori et al., 2019).

In contrast, the FDA provides more specific mechanical testing criteria for the use of tissue engineered cartilage constructs, which highlights inconsistencies with regard to global standardisation. The FDA guidance document for products intended to repair or replace knee cartilage includes specifications for *in vivo* animal studies (that will be discussed later in this review) and various *in vitro* mechanical tests. It states that “mechanical testing should address the following: the ability of the implant to withstand expected *in vivo* static and dynamic loading (i.e., compression, shear, and tension); analysis of fixation method (i.e., strength of integration between the product and surrounding native tissue); and propensity to generate wear debris.” It is also recommended that static mechanical behaviour such as the maximum recoverable compressive strain, the aggregate modulus (HA), the shear modulus (G), and permeability ( $\kappa$ ) as well as the dynamic complex shear modulus are included. Degradable scaffolds should also include assessment of failure properties over time and some examples of confined or unconfined compression and indentation are suggested for analysing the mechanical properties of implants.

## *In vitro* Compressive Testing (Confined, Unconfined and Indentation)

The most frequent *in vitro* test are usually biological assays to evaluate the biocompatibility (ISO 1099), cytotoxicity (ISO 10993-5), gene expression and matrix deposition (ISO 13019

quantification of sulfated glycosaminoglycans for evaluation of chondrogenesis) (Keong and Halim, 2009; Li W. et al., 2015). However, mechanical evaluation of osteochondral scaffolds are essential to ensure graft stability in the initial period following implantation (Bowland et al., 2015). As reviewed by Patel et al. (2019), compression testing is the most common test performed both on cartilage and tissue engineered construct. Compression test can be performed using unconfined and confined compression and indentation (**Figure 4**). For unconfined compression testing, the sample is placed between two impermeable steel plates allowing the Young's modulus to be measured directly from the linear portion of the stress-strain curve produced (Korhonen et al., 2002; Griffin et al., 2016). For confined compression the sample is either tested using a porous indenter or placed in a porous chamber with an impermeable indenter to ensure fluid flow. Confined compression allows the measurement of both the aggregate modulus (determined when fluid flow stops) of the specimen as well as the permeability (Boschetti et al., 2004). While unconfined and confined compression require the cartilage sample or the scaffold to be tested within a chamber, indentation allows the test to be performed on a whole osteochondral specimen (Griffin et al., 2016; Tozzi et al., 2020). Compression tests can be performed by applying a strain at a constant rate (ramp), by applying a strain to a target level and holding the strain constant (stress-relaxation) or applying a cyclic strain (dynamic) (Scholten et al., 2011; Vikingsson et al., 2015; Coluccino et al., 2016; Kundanati et al., 2016). Compression tests can be also load-controlled, applying a rapid load that is then kept constant, measuring sample strain over time (Oyen, 2014; Patel et al., 2019). Both the FDA and International Cartilage Repair Society (ICRS) recommend both static and dynamic compression tests to assess the mechanical behaviour of the osteochondral graft. However, specific guidelines on how to perform each test have not been agreed, which leads to inconsistent or non-physiologically relevant data. Cartilage and osteochondral grafts should be tested under the same conditions, as the strain rate influences the stress-strain curves, implying that the higher the strain rate the higher the modulus will be. As reviewed by Patel et al. (2019), 48% of the studies that analysed cartilage repair constructs under ramp mechanical testing, were tested to more than 20% strain, more than double the compressive strain that articular cartilage was tested to. Considering that the physiological average strain is 10% the data produced using higher strain might not be reliable (Chan et al., 2016).

In addition to the standard mechanical tests previously mentioned, implants need to be tested after periods *in vivo* (for dynamic and static loading) and under loading conditions of compression, tension, and shear. Analysis of fixation within the defect is also required (e.g., mechanical push-out tests to assess integration) and assessment of the bioreactivity of any device-generated wear debris.

## *In vivo* Animal Models

*In vivo* animal models are crucial preliminary studies to assess the safety and efficacy of newly developed cartilage TE implants. However, currently there are no exact guidelines



TEST	CONFIGURATION	RESULT
Confined compression	The sample is either tested using a porous indenter or placed in a porous chamber with an impermeable indenter to ensure fluid flow	Aggregate modulus and permeability
Unconfined compression	The sample is placed between two impermeable steel plates allowing to measure Young's modulus directly from the linear portion of the stress-strain curve	Elastic modulus (Young's modulus)
Indentation	The test can be performed in situ allowing to test the whole osteochondral plug	Elastic modulus (Young's modulus)

**FIGURE 4 |** Standard mechanical tests for assessing osteochondral grafts. Confined compression tests using either an impermeable chamber and porous indenter; or porous chamber with an impermeable indenter, are useful for defining the aggregate modulus. Unconfined compression tests and indentation tests can determine the elastic modulus (Young's modulus).

for the comparison of animal models, assessment of defect size/location and description of appropriate mechanical tests for the assessment of implantable devices such as osteochondral grafts to repair and regenerate articular cartilage. The FDA recommends using combined animal studies with appropriate mechanical testing to assess biological response, durability (using large animal models) and toxicology (U. S. Food Drug Administration, 2011). In addition, dose response (of bioactive scaffolds), lesion size and location, appropriate endpoints, and continual arthroscopic/MRI evaluation should also be taken into consideration (U. S. Food Drug Administration, 2011). Nevertheless, despite these recommendations there are no clearly defined protocols, test criteria, or test parameters for mechanical testing. It is also acknowledged that there is no optimal animal model for cartilage repair, which may also lead to variability between studies.

Small animal models (mouse, rat and rabbit) are mainly used for “proof of concept” studies as a translational step between *in vitro* tests and larger animal/human studies. Rabbit models have spontaneous intrinsic healing capabilities of cartilage defects that must be taken into consideration, therefore, they usually require additional validation in other animal models (Shapiro et al., 1993). Other variables to consider when choosing the most appropriate animal model are thickness of cartilage and joint suitability, skeletal/ cartilage maturity, defect type, size and location of the defect, availability and post-operative care (Hurtig et al., 2011). Canine models, like humans, often suffer

from diseases such as OA and OCD which makes them useful for assessing cartilage regeneration in pathologic conditions (Chu et al., 2010). Large animal models (goat, sheep, pig and horse) more closely reflect intended clinical use for assessing toxicity, integrity and inflammatory responses for both small and larger defects in load bearing environments. Since cartilage thickness in equine stifle joints (1.5–2.0 mm) is the most similar to human cartilage thickness (2.2–2.5 mm) the horse is the closest approximation for creating both partial and full thickness defects for preclinical cartilage repair studies (Frisbie et al., 2006; McIlwraith et al., 2011). Nevertheless, in most animal models the loading, thickness and geometry of the joint surface is very different to that of humans.

## Mechanical Push-Out Tests for Assessing Integration

Mechanical push-out tests are useful pre-clinical studies to evaluate the maximum forces needed for graft failure and for assessing integrative repair with host cartilage over time (Theodoropoulos et al., 2011). A biopsy punch is normally used to create a cylindrical defect filled with the TE scaffold or osteochondral graft to be tested. After a culture and/or treatment period to allow a certain amount of integration with the host tissue, the inner core is pushed out of the outer ring using a mechanical push-out rod. The calculated amount of force needed for displacement (or failure of the graft) allows an assessment of integrative strength at the interface to be assessed. A recent study

by Bowland et al. (2020) performed a series of push-in and push-out tests to assess the mechanical stability of bovine and porcine osteochondral grafts. Interestingly, the results showed that the harvesting method (using a trephine drill or chisel) showed no significant differences in graft stability (Bowland et al., 2020). However, preparation of the recipient site, the depth of insertion and dilation had more of an effect, showing that grafts with equal lengths to the site of insertion were more stable, and that dilation of grafts reduces the stability particularly in more skeletally immature tissue (Bowland et al., 2020). This research also highlights the importance of the underlying subchondral bone and the interrelationship between these tissues on regeneration and durability of focal defects consistent with other studies (Chan et al., 2012).

## Whole Joint Simulation Models to Mimic Joint Articulation

In contrast, *in vitro* whole joint simulations can be used to assess the tribological performance of osteochondral grafts, taking into consideration the interactions and biomechanical properties of the joint as a whole under physiological loading conditions. These types of test are relevant for comparing the efficacy of osteochondral grafts to other surgical interventions such as scaffolds and cell-based approaches (Bowland et al., 2018b). Bowland et al. (2018a) used an adapted method from Liu et al. (2015) using a whole joint simulator with six degrees of freedom and five controlled axes of motion to mechanically test grafts. The axial load was force controlled, tibial rotation ( $1.6\text{--}1.6^\circ$ ) and flexion/extension ( $0\text{--}21^\circ$ ) were displacement controlled at a frequency of 1 Hz. Anterior-posterior displacement was constrained using springs that generated rolling and sliding motions of the femur against the tibia, and mimicked ligament function. The medial-lateral axis was fully constrained and abduction/adduction was under passive motion. The main finding of this study was that allograft plugs fitted flush with the defect site to restore the articular surface caused the least wear and damage on the opposing joint surface after applying a complex range of motions. Similarly, Nebelung et al. (2017) combined a whole-knee joint loading device with MRI imaging to non-invasively assess the structural and functional responses of human articular osteochondral grafts in defect sites during *in situ* compressive loading. Whole joint simulation models highlight the importance of restoring the congruence of articular surfaces during an experimental setting that mimics more closely the physiological environment of joint articulation. However, the use of cadaveric tissue with diluted serum replicating the joint's synovial fluid is a useful approach but it fails to replicate large numbers of walking cycles due to limitations regarding the continual sterility and viability of the tissue.

## Shear Stress to Assess Tribology

Chondrocytes in the superficial layer produce lubricin that maintains low coefficients of friction of joints. Maintaining a low frictional interface is essential to prevent mechanical wear and erosion of the articular surface. The application of frictional shear stress has been shown to cause damage such as cracking and

peeling in cartilage TE constructs, which are not seen in native cartilage controls (Whitney et al., 2017). Therefore, assessing the tribology of osteochondral grafts is essential to ensure adequate integration and longevity. To measure the frictional coefficient, three different configurations of tribometer can be used: pin-on-disc, pin-on-plate, or rolling-ball-on-disc. In the first two settings a pin is glued to the sample and a disc or a plate are in motion, while for the rolling-ball-on-disc the disc and the ball can be moved independently. In each configuration a normal force is applied and a sensor measures the frictional force, the frictional coefficient can be derived by dividing the frictional force for the normal force applied. Different types of lubricant can be used (i.e., PBS or foetal bovine serum) which combined with the different testing configurations often lead to variable results, highlighting the need of standardisation procedure to test both cartilage samples and osteochondral TE constructs. Other mechanical tests such as frictional shear stress testing can assess the tribology, pressure distribution and the response to damage of osteochondral grafts and TE constructs in whole joint models under a complex range of sliding and torsional motions (Walter et al., 2013; Bobrowitsch et al., 2014).

## CONCLUSIONS

Despite tremendous advances in the field of tissue engineering, an optimal biomaterial system for osteochondral defects that is able to direct stem cell differentiation into chondrocytes for the cartilage and osteoblast for bone without the use of exogenous stimuli is elusive. Material selection is essential for creating a graft able to withstand the multiple forces that cartilage is subject to. Synthetic materials not only provide high tensile stress and compressive modulus, but they are easily modified, facilitating the creation of layered scaffolds which is a requirement for osteochondral grafts. However, the lack of cellular binding sites require them to be combined with natural materials, which are highly biocompatible and can provide biochemical cues for stem cell differentiation. The natural architecture of cartilage and the impermeable subchondral plate enhances the development of hydrostatic stress in the cartilage which promotes and maintains the chondrocytic phenotype, however few osteochondral implant designs replicate this subchondral barrier.

Although suitable mechanical properties are essential for ensuring graft stability *in vivo*, the optimal range of stiffnesses is yet to be determined. Conflicting results have been reported, as to whether high stiffness could enhance chondrogenic differentiation of MSCs or upregulate hypertrophic markers. The use of dynamic stimulation, such as hydrostatic pressure or dynamic loading, could promote a stable differentiation of MSCs into chondrocytes and enhance matrix deposition, thus preventing the use of TGF- $\beta$  which lead to the formation of hypertrophic cartilage.

Mechanical testing of TE constructs *in vitro* are essential to ensure graft stability *in vivo*, however, the lack of standardised procedures questions the reliability of the published data in providing an understanding of the long term endurance and

suitability of osteochondral grafts. In addition, only a small fraction of studies on cartilage constructs tests all of the mechanical properties requested from the FDA or the ICRS and this might, in part, explain why many scaffolds fail when tested *in vivo*.

## AUTHOR CONTRIBUTIONS

SD and TR completed a literature search, wrote the manuscript and critically revised the manuscript. MR, GB, and GT contributed ideas, content to the manuscript, and critically

revised the manuscript. All authors contributed to the article and approved the submitted version.

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# Metal-Organic Framework (MOF)-Based Biomaterials for Tissue Engineering and Regenerative Medicine

Moldir Shyngys<sup>1</sup>, Jia Ren<sup>1</sup>, Xiaoqi Liang<sup>1</sup>, Jiechen Miao<sup>1</sup>, Anna Blocki<sup>2,3,4</sup> and Sebastian Beyer<sup>1,2\*</sup>

<sup>1</sup> Department of Biomedical Engineering, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>2</sup> Institute for Tissue Engineering & Regenerative Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>3</sup> School of Biomedical Sciences, Faculty of Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong, <sup>4</sup> Department of Orthopaedics & Traumatology, Faculty of Medicine, The Chinese University of Hong Kong, Shatin, Hong Kong

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### \*Correspondence:

Sebastian Beyer  
sebastian.beyer@cuhk.edu.hk

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The synthesis of Metal-organic Frameworks (MOFs) and their evaluation for various applications is one of the largest research areas within materials sciences and chemistry. Here, the use of MOFs in biomaterials and implants is summarized as narrative review addressing primarily the Tissue Engineering and Regenerative Medicine (TERM) community. Focus is given on MOFs as bioactive component to aid tissue engineering and to augment clinically established or future therapies in regenerative medicine. A summary of synthesis methods suitable for TERM laboratories and key properties of MOFs relevant to biomaterials is provided. The use of MOFs is categorized according to their targeted organ (bone, cardio-vascular, skin and nervous tissue) and whether the MOFs are used as intrinsically bioactive material or as drug delivery vehicle. Further distinction between *in vitro* and *in vivo* studies provides a clear assessment of literature on the current progress of MOF based biomaterials. Although the present review is narrative in nature, systematic literature analysis has been performed, allowing a concise overview of this emerging research direction till the point of writing. While a number of excellent studies have been published, future studies will need to clearly highlight the safety and added value of MOFs compared to established materials for clinical TERM applications. The scope of the present review is clearly delimited from the general 'biomedical application' of MOFs that focuses mainly on drug delivery or diagnostic applications not involving aspects of tissue healing or better implant integration.

**Keywords:** metal-organic frameworks, biomaterial, tissue engineering, regenerative medicine, bone, cardio-vascular, nervous tissue

## INTRODUCTION

MOFs are new synthetic materials that emerged over the past three decades (Yaghi and Li, 1995; Yaghi et al., 1995). MOFs comprise of organic ligands that bridge metal ions. This leads to highly ordered, porous and 3-dimensional crystalline structures. The pore size and aperture is significantly smaller compared to polymeric structures (Pan et al., 2013) and can be accurately controlled (Eddaoudi et al., 2002; Banerjee et al., 2009). This accurate control makes MOFs predestined for drug loading and sustained or triggered release from nano-particulate

drug formulations or coatings (Wang et al., 2020). This has led to the classical understanding of “biomedical MOF applications” that primarily focused on diagnostic applications (Chapartegui-Arias et al., 2019) or drug delivery (Keskin and Kizilel, 2011; Banerjee et al., 2020). The present review is clearly delimited from these established fields of investigation by focusing on MOFs for TERM applications. In particular, the present review addresses MOFs that modulate the foreign body response to implants, MOFs that influence the inflammatory wound environment, or MOFs that improve the host tissue-implant interface and aid healing.

A systematic literature analysis was performed using Clarivate Analytics Web of Science database in combination with a structured key word search. In brief, a comprehensive list of works on MOFs was generated and those interfacing with biomaterials, tissue engineering, regenerative medicine or MOFs interfacing with neuronal repair were identified. Works solely focusing on drug-delivery or cancer treatment were excluded if not reporting aspect of tissue healing, integration or improvement of the foreign body response.

In addition to the drug loading capacity, the present review aims to highlight that MOFs may also have intrinsic bioactivity through delivering specific metal ions (Su et al., 2019), ligands (Rojas et al., 2017) or through catalytically generating messenger molecules such as nitric oxide in copper-based MOFs (Harding and Reynolds, 2012). Beyond individual components, structural aspects of MOFs are essential since they determine release kinetics as well as the accessibility of the metal centers providing catalytic effects. Most studies investigating MOFs for TERM application focused on orthopedic applications, followed by cardiovascular treatment, cutaneous wound healing, and applications in nervous tissue (Figure 1). Beyond these targeted organs, a specialized MOF was proposed for treatment of periodontitis (Li et al., 2019), while materials for other organs seem not yet to have been evaluated. In summary, the use of MOFs for TERM applications is an emerging field of research with only very few works published before the year 2018.

## Parallels of MOFs to Natural Minerals in TERM Applications

The concept of MOFs originates from natural minerals, which have a substantial track record in TERM applications. Calcium phosphate has intrinsic osteogenic effect (Albee, 1920) and its admixtures to biodegradable polymers led to clinical implants (de Groot, 1993). A calcium and strontium based MOF with bisphosphonate linker (Matlinska et al., 2019) appears to be the evolution of minerals with engineered bioactivity.

Another naturally occurring mineral is zinc oxide (ZnO), a non-porous solid devoid of any drug loading capacity, that has long been used to treat cutaneous wounds and pathological skin conditions (Lansdown et al., 2007). In contrast to ZnO, versatile drug loading, release and disintegration rate of porous Zeolitic Imidazolate Frameworks (ZIFs) can be designed leading to various envisioned biomedical applications (Maleki et al.,

2020). The most prominent member of the ZIF family is ZIF-8, which is a polymorph of zinc(II)-2-methylimidazolate and has a ligand-metal-ligand bonding angle similar to that of O-Si-O. In addition, ZIF-8 shares the crystal topology (SOD) with Sodalite, a naturally occurring zeolite of the sodium aluminum silicate framework family.

## PREPARATION METHODS OF MOFs SUITABLE FOR TERM LABORATORIES AND THEIR KEY PROPERTIES

### Solution Precipitation Synthesis of ZIF-8 and Loading of Bioactive Agents

ZIF-8 is easily synthesized through simple precipitation reactions facilitated by mixing the ligand (2-methylimidazole) and zinc salt solution at room temperature (Park et al., 2006; Beyer et al., 2016, 2018b; Kulow et al., 2019). ZIF-8 can be engineered to acquire bioactivity beyond that of zinc ions through encapsulation of drugs. Admixtures of Dexamethasone (Sarkar et al., 2019) or non-steroidal anti-inflammatory drugs (Ho et al., 2020) to these precursor mixtures during precipitation reactions led to their firm entrapment within ZIF-8. ZIF-8 shows a sustained release behavior for small molecules up to several weeks (Liédana et al., 2012). An interesting aspect of ZIF-8 is the observation that it crystallizes around biomacromolecules in aqueous solution by a process termed biomimetic mineralization (Liang et al., 2015a,b; Gao et al., 2019). This approach has been used for delivery of insulin (Duan et al., 2018) and may provide opportunities for e.g., growth factor delivery.

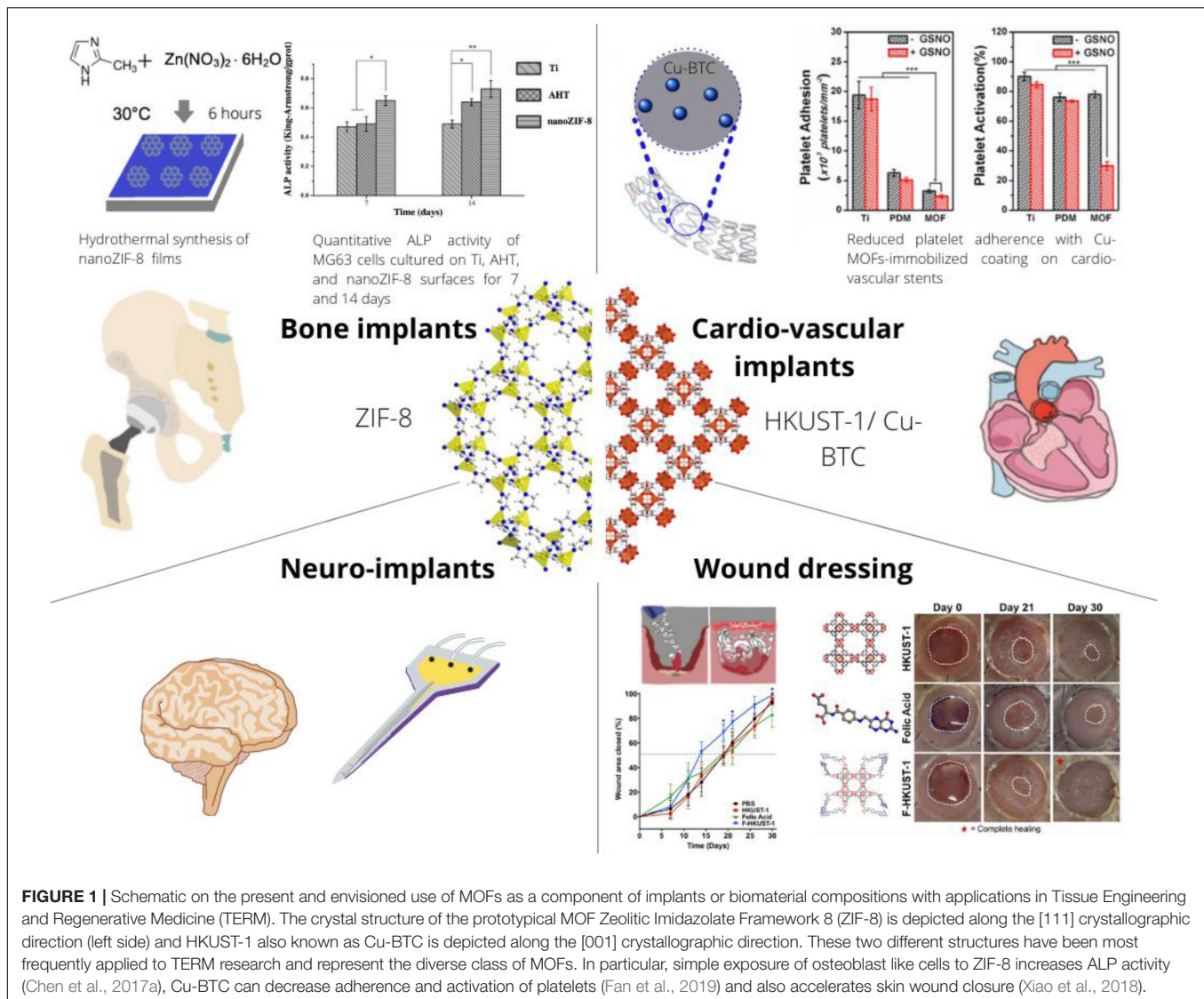
### Mechanochemical Synthesis of MOFs and Potential Use as Filler in Polymer Composites

MOFs may also be synthesized mechanochemically (Stolar and Užarevič, 2020) by mixing organic linkers and metal precursors in a small planetary ball-mill (Akhmetova et al., 2019), immediately yielding MOF powder in large quantities and short time. So far, the mechanochemical synthesis of calcium and strontium MOFs has been reported (Al-Terkawi et al., 2017), which is an excellent route to produce MOFs as bioactive fillers for e.g., polymeric biomaterials. However, mechanochemical syntheses routes for earth alkaline metal MOFs specifically for TERM applications have yet to be developed.

### Solvothermal Synthesis of CU-BTC

During solvo-thermal synthesis, two precursor solutions of organic ligands and metal ions are mixed and heated in a micro-autoclave beyond the boiling point of the solvent (Ni and Masel, 2006). Copper(II)-1,3,5-benzenetricarboxylate (Cu-BTC), also known as HKUST-1 (Chui, 1999) is typically synthesized under solvo-thermal conditions. The porous structure of Cu-BTC makes the copper metal center accessible to solutes. This renders Cu-BTC an interesting catalyst for the conversion of blood borne nitrosothiols to continuously release nitric oxide (NO) (Harding and Reynolds, 2012). Nitric oxide is a key





messenger molecule during inflammation (Sharma et al., 2007), platelet adhesion (Radomski et al., 1987), ECM deposition (Myers and Tanner, 1998), angiogenesis and other aspects of tissue healing and foreign body response (Schäffer et al., 1996). Beyond the intrinsic catalytic activity, constant release of copper ions is crucial for the biological activity of Cu-BTC. Copper ions are known to stimulate angiogenesis (Sen et al., 2002) and to increase extracellular matrix (ECM) deposition (Gérard et al., 2010). These properties together hold potential to benefit TERM applications of copper-based MOFs when correctly dosed for the intended application.

## Electrochemical MOF Coating With Potential Use in Sensors or Electrodes

Electro-deposition of MOFs (Campagnol et al., 2016) may be useful for coating of conductive, implantable electrodes and sensors especially with future applications in neuro-interface engineering.

## APPLICATIONS OF MOFs OR MOF BASED BIOMATERIALS IN TERM

### MOFs for Orthopedic Implants

Most reports on use of MOFs in TERM applications focus on the improvement of structural bone implants. Indeed, a small but significant fraction of clinical implants fail due to an unfavorable, non-healing wound environment that may be aggravated by bacterial infection (Inacio et al., 2013).

### Intrinsic Bioactivity of MOFs and Influence on Bone Tissue Biology

*In vitro* studies showed significantly improved corrosion resistance of magnesium alloys upon coating with bio-MOF-1, comprising of zinc ions and adenine as well as interconnections of 4,4-biphenyldicarboxylate linkers (Liu et al., 2019). Besides engineering corrosion resistance for better structural integrity, the ability to prevent bacterial infection

is crucial for successful integration of bone implants. Sr-HA (Sr-substituted hydroxyapatite) with MOF74 has demonstrated excellent abilities to inhibit *S. Aureus* and *E. Coli* bacteria and simultaneously induced apoptosis in Saos-2 cancer cells while promoting proliferation and maturation of primary osteoblasts (Zhang et al., 2019).

The effect of different metal ions on bone mineralization and osteogenesis is long known and mechanistic insights have been obtained e.g., for zinc (Kwun et al., 2010) or magnesium (Zhang et al., 2016). Consequently, MOFs were shown to promote osteogenic differentiation *in vitro*, an indicator for a better osteointegration of bone implants. ZIF-8 coatings on titanium implants enhanced ALP activity, better extracellular matrix mineralization and upregulation of osteogenic gene expression in MG63 cells, while suppressing *Streptococcus Mutans* proliferation (Chen et al., 2017a). Beyond zinc and magnesium, Ca-Sr MOFs could upregulate the expression of ALP, whereas simultaneous release of Ca and Sr provided osteo-inductive signals (Joseph et al., 2019). Further, simultaneous delivery of calcium and strontium ions as well as bisphosphonate linkers was noted to allow better bone mineralization (Matlinska et al., 2019). In the context of regenerative medicine, Mg/HCOOH-MOF was investigated for the treatment of osteoarthritis. This MOF was shown to up-regulate the expression of OCN, Axin2 and down-regulate iNOS and IL-1 $\beta$ , indicating a potentially favorable effect to ease osteoarthritis (Li et al., 2020).

*In vivo* investigations were less frequently reported, indicating that the field of MOFs for orthopedic applications is just at the beginning of being explored. Mg/Zn-MOF74, was demonstrated to have low toxicity, intrinsic anti-inflammatory and antibacterial features, while improving early osteogenic promotion and angiogenesis *in vivo* (Zhu et al., 2020).

### Acquired Bioactivity of MOFs Through Drug Loading for Enhanced Bone Regeneration

Acquired bioactivity refers to any additional biological effect that does not originate from the MOF itself, which can be acquired through encapsulation of drugs, ions or other bioactive agents. MOF-based drug carriers are interesting for orthopedic implant coatings due to their high drug encapsulation capacity with a negligible premature release (Tan et al., 2016).

*In vitro* studies showed that Poly-L-lactic acid (PLLA) scaffolds coated with copper loaded ZIF-8 promoted osteogenic differentiation as compared to pure PLLA scaffolds while preventing bacterial infection in the presence of Zn<sup>2+</sup> and Cu<sup>2+</sup> ions (Telgerd et al., 2019).

Beyond the release of ions, MOFs have unique sustained release abilities for small molecular weight drugs. A sustained release of dexamethasone for 4 weeks was achieved through integrating dexamethasone loaded ZIF-8 into cellulose-hydroxyapatite nanocomposites envisioned for load bearing orthopedic applications (Sarkar et al., 2019). A similar approach has led to silk fibroin-dexamethasone@zeolitic imidazolate framework-8-titanium (SF-DEX@ZIF-8-Ti) composite materials. MC3T3-E1 osteoblast precursor cells cultured on the SF-DEX@ZIF-8-Ti showed enhanced osteo-differentiation

(Ran et al., 2018). Hence, current research indicates a good potential of MOFs to facilitate osteointegration of bone implants, thereby improving their clinical outcome. Examples for treating bone diseases include the release of naringin, which improves osseointegration and prevents bacterial infection (Yu et al., 2017), delivery of antitubercular drug for treatment of osteoarticular tuberculosis (Pei et al., 2018) and release of vancomycin for treatment of osteomyelitis (Karakeçili et al., 2019). The versatility of MOF-based drug carriers principally allow the design of elaborate drug release mechanisms tailored to specific triggers. However, only one study describes pH and Ca<sup>2+</sup> ion dependent release of drugs into the bone environment using a Zr-MOFs capped with CP5-based pseudorotaxanes (Tan et al., 2016).

*In vivo* studies are less frequently reported. One study investigated implant materials modified with levofloxacin loaded ZIF-8 particles, that resulted in better attachment, proliferation, and differentiation of osteoblasts *in vitro*. Special focus was given on the pH-triggered release in slightly acidic environments resulting from inflammation and bacterial infection. The effectiveness of this implant material was confirmed in terms of osseointegration and antibacterial properties in a rat infected femur model (Tao et al., 2020).

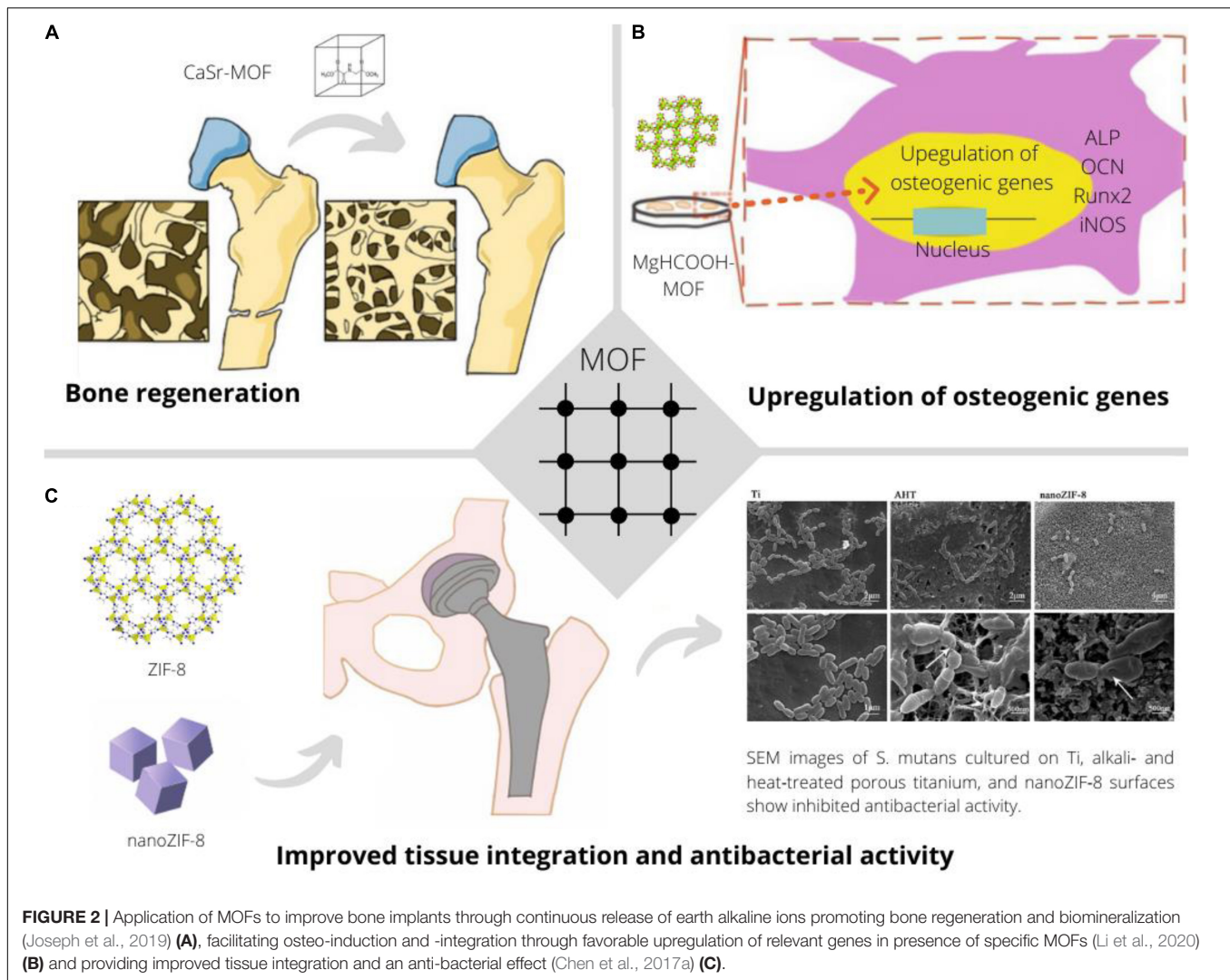
In conclusion, MOFs have been shown to be promising to enhance bone healing through intrinsic and/or acquired properties (Figure 2). While the here presented initial studies provide excellent perspectives for the use of MOFs for bone healing, more studies are needed toward their clinical applicability.

### MOFs for Cardio-Vascular Implants

Approximately 4% of the population in industrialized societies suffer from coronary artery disease (CAD) requiring intervention (Sanchis-Gomar et al., 2016). Traditionally, anti-thrombotic medication and small molecular weight nitric oxide donors acting as vasodilators are administered to prevent artery occlusion. In more severe cases, drug-eluting stents are implanted into the artery to prevent occlusion permanently. Copper-based MOFs recently emerged as component in anti-thrombotic coatings for cardio-vascular implants.

*In vitro*: Cu-BTC was established to be an efficient catalyst for the conversion of blood borne s-nitroso-cysteine to nitric oxide and cysteine (Harding and Reynolds, 2012). More complex Cu-MOFs have subsequently been investigated for catalytic conversion of s-nitrosoglutathione (Neufeld et al., 2017b). This has led to works on MOF/polymer composite materials for nitric oxide release as suitable leads for novel implant materials (Harding and Reynolds, 2014; Neufeld et al., 2015, 2016, 2017a; Zang et al., 2020). In addition to these initial studies, Cu-BTC was directly grown on stent surfaces as SURMOF and investigated *in vitro* showing favorable hemocompatibility (Zhao et al., 2019).

Evaluating non-copper based MOFs for cardiovascular implants, a MIL-101 (Fe) poly( $\epsilon$ -caprolactone) composite was investigated as material for metal free stents, using the MOF as mechanical re-enforcement, sustained drug release vehicle and Magnetic Resonance Imaging (MRI) contrast agent (Hamideh et al., 2020). Further, few studies exist that address the interaction of MOFs with cardiomyocytes (Al-Ansari et al., 2020).



*In vivo*, Cu-BTC was shown effective as cardiovascular stent surface modification through coating together with polydopamine (Fan et al., 2019). This significantly reduced protein absorption, thrombus formation, platelet adhesion and an increased vasodilating effect, especially upon systemic injection of nitric oxide donors requiring catalytic degradation. Using a 5-(1H-tetrazol-5-yl)isophthalic acid based MOF with  $\text{Ni}^{2+}$  and  $\text{K}^{+}$  ions in another study prevented arrhythmia through reducing the sympathetic excitability in a rat model (Ding et al., 2020). Future studies will need to compare the effectiveness and safety of MOFs with that of clinically used materials.

## MOF-Based Biomaterials for Cutaneous Wound Care

The clinical healing outcome of chronic wounds arising from pressure ulcers or injuries of diabetic patients is often unsatisfactory, which motivates research for better cutaneous wound care material. In addition, achieving fast hemostasis

and antibacterial properties is important for emergency care equipment and the potential role of MOFs for these two applications is elucidated in the following.

*In vitro* studies investigated polymer meshes with enclosed Cu-BTC (Zhang et al., 2020) and cellulose based biomaterials with the ability to release nitric oxide (Darder et al., 2020) as wound dressing. To ascribe increased mitochondrial respiration solely to nitric oxide, a titanium-based MOF devoid of intrinsic bioactivity and with controlled release of pre-loaded nitric oxide was proposed to be a copper free alternative (Pinto et al., 2020).

*In vivo* studies have shown the positive effect of Cu-BTC on skin healing using full-thickness splinted skin wounds in rodents. Dosage forms included folate stabilized Cu-BTC crystal suspensions (Xiao et al., 2018) and hydrogel compositions (Xiao et al., 2017).

Supplementation of zinc ions from ZIF-8 has shown to be similarly effective in a ZIF-8 hydrogel composite membrane (Yao et al., 2020) for wound healing. Combination of zinc and copper ions in MOFs that further encapsulated bioactive molecules promised to be most effective. Along this line of



thought, the encapsulation of niacin loaded copper-/zinc MOFs within alginate shells has shown antibacterial, antioxidant and angiogenic properties, which resulted in significant improvement of wound closure in an infected full-thickness skin defect model (Chen et al., 2019).

Future studies are necessary to ascribe these effects to the synergistic potential of MOFs and not to the presence of its individual components upon dissolution.

## MOFs for Neuronal Tissue Engineering and Implants

Traumatic injuries to the nervous system result in dysfunctional regeneration and glial scarring in an inflammatory process leading to loss of nervous tissue. The consequences include physical paralysis of muscles, neurosensory or mental impairment. This has motivated the development of biomaterials for nervous tissue regeneration (Liu and Hsu, 2020), and MOFs may be suitable to further augment positive materials characteristic toward this goal.

*In vitro* studies focused on assessing the response of sensory neurons to nano-particulate MOFs by employing a neurite outgrowth assay, indicating the potential of a material to promote axonal outgrowth, thus nervous regeneration (Wuttke et al., 2017). The MOFs investigated included iron and chromium salts of trimesic acid (MIL-100) and Zirconium fumarate MOFs. These MOFs did not show reduction in neurite outgrowth at a concentration of 100 µg/ml, indicating their suitability for drug delivery applications. In line with the thought that MOFs are compatible with neuronal biology, treatment of neuronal stem cells with a cobalt MOF having 3,3',5,5' - azobenzenetetracarboxylic acid as linker was envisioned to benefit therapies for spinal cord injury. This was due to molecular docking studies revealing potential binding of the MOF with the trkA receptor, which led to activation of PI3K/Akt signaling and enhanced nerve growth factor (NGF) production *in vitro*. NGF is responsible for neuron growth and survival (Cao et al., 2020).

MOF based drug delivery vehicles include cationic MOF-74-Fe(III) with low cytotoxicity on PC12 neuron cells and high drug loading due to electrostatic accumulation (Hu et al., 2014) of ibuprofen, which is known to reduce neuroinflammation (Wixey et al., 2019) and is passively released from the MOF. An example for actively triggered dopamine release by magnetophoretic means was demonstrated using MIL-88A (iron(III) fumarate) encapsulated iron oxide particles (Pinna et al., 2018). A drug delivery mechanism that is triggered upon synaptic transmission can be achieved using adenosine triphosphate (ATP) responsive release of drugs. Here, a MOF based on  $Zr^{4+}$  ions and amino-triphenyldicarboxylic acid was capped with aptamers specific for ATP, leading to an un-locking effect upon binding and drug release (Chen et al., 2017b). These three examples of passive-, triggered- and targeted drug release show the versatility of MOF based drug delivery for neurobiological applications. Systemic injection requires MOF particles to cross the blood-brain barrier. This ability was assessed for e.g., ZIF-8 particles through crossing an hCMEC/D3 brain endothelial cell membrane (Chiacchia

et al., 2015). Future studies in this direction would benefit from microfluidic vascular transport models for nanoparticles (Ho et al., 2017) involving pericytes to better emulate the blood-brain barrier and to study the biological response of further cell types in inflammation and healing (Beyer et al., 2018a; Blocki et al., 2018).

*In vivo* studies reported the use of a Cu(II) MOF in a mixed linker approach with 5-methylisophthalic acid and 1,3-bis(5,6-dimethylbenzimidazol-1-yl)propane (Tian and Tian, 2020). This MOF inhibited dopaminergic neurons, decreased expression of dopamine reporters, decreased dopamine release, reduced apoptosis and increased Wnt1-Nkx2.2 activation. The MOF was presumably injected on a systemic level into LgDel ± mice and biological effects were observed in a dose dependent manner up to 5 mg/kg body weight. No comparison of the biological effect upon injection of the MOF or a solution containing only the ligand was made.

A very inspiring study used MIL-100 (Fe) as a drug carrier for siSOX9 and retinoic acid in combination with co-encapsulated ceria based nanozymes to reduce oxidative stress (Yu et al., 2020). This complex drug formulation was incubated with neuronal stem cells (NSCs) isolated from newborn mice prior to NSC transplantation at sites of lesions. The co-release of soSOX9 and retinoic acid promoted neuro-differentiation and reduced oxidative stress prevented newborn neurons from apoptosis. The studies were carried out with 11-month old APP<sup>swe</sup>/PS1M146V/TauP301L triple transgenic mice, a model for Alzheimer's disease. The positive effect of the MOF based drug combination was shown through cognitive improvements in a Morris Water Maze test and histopathological improvements.

These inspiring *in vitro* and *in vivo* works have highlighted a clear potential of MOFs to treat neurological disorders. Very few macroscopic biomaterials have been developed along this thought and future studies should include evaluating the opportunity to improve implant-nervous tissue interfaces with these materials.

## CONCLUSION AND OUTLOOK

The presented *in vitro* and *in vivo* data are impressive in terms of the strong biological response that MOFs elicit toward a positive healing outcome in various tissues and applications. However, more efforts are required to understand the basic dose dependent response of individual cell types and tissues to different MOFs. In addition, concerns about the safety of some organic linkers of MOFs need to be eliminated or alternative linkers considered save for clinical applications must be shown effective in yielding MOFs with similar properties. This is essential before MOF based biomaterials for TERM applications will commence improving healing outcomes in the treatment of human patients.

## AUTHOR CONTRIBUTIONS

MS reviewed literature related to orthopedic implants and cutaneous wound care and designed the figures. JR reviewed



literature related to neuronal TERM applications. XL reviewed literature related to nitric oxide producing biomaterials. JM reviewed literature related to MOF & polymer composite materials. AB and SB wrote the manuscript. SB guided the review process. All authors contributed to the article and approved the submitted version.

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**Conflict of Interest:** The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Ultra-Thin Porous PDLLA Films Promote Generation, Maintenance, and Viability of Stem Cell Spheroids

Ya An Tsai<sup>1</sup>, Tianshu Li<sup>2</sup>, Lucia A. Torres-Fernández<sup>3</sup>, Stefan C. Weise<sup>3</sup>, Waldemar Kolanus<sup>3</sup> and Shinji Takeoka<sup>1,2\*</sup>

<sup>1</sup> Department of Life Science and Medical Bioscience, Graduate School of Advanced Science and Engineering, Waseda University (TWIns), Tokyo, Japan, <sup>2</sup> Institute for Advanced Research of Biosystem Dynamics, Research Institute for Science and Engineering, Waseda University, Tokyo, Japan, <sup>3</sup> Life and Medical Sciences Institute (LIMES), University of Bonn, Bonn, Germany

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### \*Correspondence:

Shinji Takeoka  
takeoka@waseda.jp

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Three-dimensional (3D) culture bridges and minimizes the gap between *in vitro* and *in vivo* states of cells and various 3D culture systems have been developed according to different approaches. However, most of these approaches are either complicated to operate, or costive to scale up. Therefore, a simple method for stem cell spheroid formation and preservation was proposed using poly(D,L-lactic acid) porous thin film (porous nanosheet), which were fabricated by a roll-to-roll gravure coating method combining a solvent etching process. The obtained porous nanosheet was less than 200 nm in thickness and had an average pore area of 6.6  $\mu\text{m}^2$  with a porosity of 0.887. It offered a semi-adhesive surface for stem cells to form spheroids and maintained the average spheroid diameter below 100  $\mu\text{m}$  for 5 days. In comparison to the spheroids formed in suspension culture, the porous nanosheets improved cell viability and cell division rate, suggesting the better feasibility to be applied as 3D culture scaffolds.

**Keywords:** ultra-thin film, porous nanosheet, PDLLA, 3D spheroid culture, cell viability, cell division

## INTRODUCTION

Two-dimensional (2D) cell culture on tissue plates is commonly performed in *in vitro* cell biology studies, allowing researchers to study how cells expand, behave under various stresses, proliferate and/or differentiate in response to stimuli. Monolayer incubation, however, lacks fundamental cell-cell and cell-matrix interactions as well as direct cell-to-cell exchange of chemical cues, thus limiting our understanding on how cells behave *in vivo*. Compared to 2D cell culture, three-dimensional (3D) culture bridges the gap between cell physiology and *in vitro* cell culture and improves *in vitro* to close-to-*in vivo* research. It provides more practical tool in basic and applied research to broaden the perspective of cell biology.

3D cell aggregates (spheroids) generate an inherent nutrient and oxygen gradient within their structure and constitute an *in vivo* microenvironment. Previous studies using 3D arranged cells uncovered mechanisms involved in tumorigenesis (Lv et al., 2017) and in maintaining expression of specific biomarkers due to enhanced extracellular matrix (ECM) (Bazou, 2010), and demonstrated the ability of spheroids to serve for drug evaluation and development. Stem cell spheroids are especially promising to improve the therapeutic effects of stem cell therapy through raised



cell viability, cell secretomes and differentiation ability (Park et al., 2017). Embryonic stem cell spheroids have shown multiple functional capabilities in development and differentiation (McKee and Chaudhry, 2017). In addition, mesenchymal stem cells cultured in 3D systems exhibit secretion of anti-inflammatory factors (Bartosh et al., 2010), enhanced cell survival (Emmert et al., 2013) and the osteogenic differentiation ability for bone regeneration application (Yamaguchi et al., 2014).

3D cultures made in different culture environments aid in our understanding of cell physiology changes by cell morphology observation, cytoskeleton examination, and cell-cell junction visualization within aggregates. Numerous techniques have been published for 3D cell cultures and classified according to the methodology of scaffold-free or scaffold-based systems. Scaffold-free techniques based on non- or limited support provide low-cost, simple and rapid cell aggregation via gravitation from the hanging drop and pellet culture methods (Santos et al., 2015; Xu et al., 2016) to form spheroids. These spheroids are easy to harvest but difficult to expand in a continuous passage (Lv et al., 2017). On the other hand, scaffold-based systems could construct a culture environment that is much closer to the *in vivo* conditions. For example, chitosan-based scaffolds or gel-supported embeddings chemically interact with cell spheroids, provide dense ECM signal connections and induce focal adhesion protein expressions (Huang et al., 2011; Zujur et al., 2017). However, these techniques require relatively complicated chemical reactions (Kim et al., 2013) and are costly in large-scale production (Lv et al., 2017). Therefore, economic biomaterials and easy-to-handle procedures are desired to achieve large-scale 3D cultures.

We have developed nanosheets with biocompatible and biodegradable organic polymers which are free-standing thin films, with tens or hundreds of nanometers in thinness (Fujie, 2016). They are suitable to be applied in a variety of biomedical applications such as an alternative to bio-membranes, for drug loading or as wound dressing materials (Fujie, 2016). They were also applied as wrapping materials to achieve high-quality live imaging of tissue and suspension cells (Zhang et al., 2017, 2018).

The tissue-biomaterial interface is pivotal in regulating the cellular interactions, for example cell adhesion, morphology, orientation, motility, proliferation, and differentiation as well as other intracellular events. Conventional lithographic techniques using photoresists and micro-patterned molds provide a feasible way to fabricate the nanomaterials, however, it may be not a suitable methodology to produce biomedical-use materials due to the necessary irradiate process. To address this problem, we previously proposed a convenient method to pattern murine fibroblasts by engineering the physiochemical properties of free-standing nanosheet (Fujie et al., 2011). Since then, various kinds of surface-modified polymeric nanosheets have been developed as novel scaffolds for cell organization, cell delivery, and cell hierarchical construction (Fujie et al., 2013, 2015; Shi et al., 2014; Otomo et al., 2020). Microporous thin films (referred to as porous nanosheets) were previously introduced to induce cell construction and cell alignment (Suzuki et al., 2016). ECM proteins such as fibronectin,

collagen and laminin penetrate through the micropores to stabilize cell hierarchies constructed between multi-layered porous nanosheets. Additionally, it has been reported that porous nanosheets facilitated long-term cell culture (up to 2 weeks) with more than 80% cell viability and oriented cell arrangement (Nishiwaki et al., 2019). Most recently, we reported for the first time that porous nanosheets prepared by a solvent etching method could support the adipose-tissue derived stem cell (ASC) spheroids for high-magnification imaging owing to its transparency (Suematsu et al., 2020). In this study, we further verified that free-standing poly(D,L-lactic acid) (PDLLA) porous nanosheet could improve the 3D spheroid cell viability and proliferation in comparison with the conventional suspension culture, demonstrating its potential to be applied as a novel 3D culture scaffold, which is cost-effective and easy-to-handle.

## MATERIALS AND METHODS

### Materials

The polymers, solvents and reagents used in this study were purchased from the following suppliers: Poly(D,L-lactic acid) (PDLLA, Mw = 300,000–600,000) from Polysciences, Inc. (Warrington, PA); Poly(vinyl alcohol) (PVA, Mw = 13,000–23,000), ethyl acetate and cyclohexane from Kanto Chemical, Co., Inc. (Tokyo, Japan); Polystyrene (PS, Mw = ~280,000) and bovine fibronectin from Sigma-Aldrich Co., LLC. (St. Louis, MO); poly (ethylene terephthalate) (PET) film (Lumirror 25T60) from Panac Co., Ltd. (Tokyo, Japan).

### Preparation of Porous Nanosheet

Using the approach proposed by Suzuki et al. (2016), the porous nanosheet were generated by Micro Gravure TM coater ML-120 (Yasui Seiki Co., Ltd., Kanagawa, Japan), a gravure coating method combining the roll-to-roll process. PVA solution (20 mg/mL) was coated first on the PET film to serve as a sacrificial layer, which could be dissolved and detach nanosheets from the PET substrate. From the polymer solution, the PET film runs through the coater, which rotates at 30 rpm, with a 1.3 m/min line speed over 7 meters. The solvent-coated film was high temperature air-dried (100°C) for about 5 min until the remaining solution has been evaporated. PDLLA and PS were then mixed in a 1:1 weight ratio and dissolved in ethyl acetate to make the polymer concentration of 20 mg/mL, which was optimized for cell culture purpose in terms of pore uniformity and diameters, porosity and thickness of nanosheets (Suzuki et al., 2016). This polymer solution was topped on PVA-coated film, and it went through the same coating process as the PVA layer except for a lower temperature (80°C) for drying. The obtained nanosheet was cut into 2 cm × 2 cm pieces, and then immersed into cyclohexane to selectively dissolve and remove PS regions by solvent etching. Without the PS regions, a random porous topography was produced on the remaining PDLLA nanosheet.

## Measurement of Thickness and Porous Diameter

Before the measurements, the substrate supported porous nanosheet was placed in deionized water to dissolve the PVA layer between the PET film and the porous nanosheet. Then the free-standing porous nanosheet was collected and reattached on a silicon wafer in deionized water in order to measure the thickness of porous nanosheet and porous diameter. The porous and non-porous nanosheets were examined with a profilometer (Dektak XT-S, Bruker BioSpin Co., Kanagawa, Japan), which uses two scarred scratches on the nanosheet attaching on a silicon wafer as the base to compare and obtain the relative height to represent nanosheet thickness. An atomic force microscope (AFM; VN-8000, Keyence Co., Ltd., Osaka, Japan) was applied to characterize the topographical surface. For the porous area, the ImageJ software package (U.S. National Institutes of Health, Bethesda, MD) was employed to measure the diameter using the phase-contrast images of porous nanosheets.

## Cell Culture

Murine adipose-derived stem cells (mASCs) (<10 passages; Cyagen Biosciences, Santa Clara, CA) were incubated at 37°C, 5% CO<sub>2</sub> atmosphere in a cell culture media [Dulbecco's Modified Eagle Medium: Nutrient Mixture F-12 (DMEM/F-12)] containing 10% (v/v) fetal bovine serum (FBS) and 1% (v/v) penicillin streptomycin. When the confluence reached 80%, cells were detached from the tissue-culture dish by using 0.25% (w/v) trypsin/0.1% (w/v) ethylenediaminetetraacetic acid (EDTA) after washing with Dulbecco's phosphate-buffered saline (DPBS; Thermo Fisher Scientific, Inc., Carlsbad, CA). Murine embryonic stem cells (mESCs) were cultured in DMEM-knockout medium containing 15% FBS, 1% PS, 1% Glutamax, 1% NEAA, 0.1% β-mercaptoethanol, 0.2% leukemia inhibitory factor (LIF) and two inhibitors (PD0325901 and CHIR99021) at 37°C, 5% CO<sub>2</sub> atmosphere.

## Spheroids Formation and Characterization

Collected stem cells ( $2 \times 10^5$  cells) were seeded on the porous nanosheet, which had been fixed on a glass-based dish (thickness: 0.15–0.18 mm, 35 ø glass base dish, AGC Inc., Tokyo, Japan) with silicone elastomer: poly(dimethyl-siloxane) (PDMS). The same cell density was applied to non-adherent 12-well plate as well. The culture media would be changed every 2 days to keep cell survival. Cell aggregations were observed and the images were captured using an inverted microscope (IX-71; Olympus) from day 1 to 5 after cell seeding. According to the phase-contrast images, the diameter of cell aggregations was analyzed on ImageJ.

## Live/Dead Staining

After 72 h of cell seeding, Live/Dead viability/cytotoxicity test kit (Thermo Fisher Scientific) was used to observe the living and dead cells distribution within the spheroids. The kit includes two staining solutions, calcein AM targeting living cells and ethidium homodimer labeling dead cells. Calcein AM

is hydrolyzed by esterase in living cells to become calcein, which retains in the cell cytoplasm and strongly emits green fluorescence; while ethidium homodimer only penetrates cells with disrupted plasma membrane and intercalates into DNA. Calcein AM and ethidium homodimer were mixed and diluted in PBS as instructed by the manual. After washing mASC spheroids collected from the nanosheet and the suspension culture media with PBS, 1 mL staining solution were added to each culture dish, followed by 15 min incubation at 37°C. Then, suspension cells were transferred into a glass-based dish prior to the observation under a confocal laser-scanning microscope (FV1000, Olympus Life Science). The green and red fluorescence intensity was separately analyzed by ImageJ and the ratio of living and dead cells in spheroids was compared between different groups. The corrected total cell fluorescence (CTCF) value was calculated using the formula (1) shown below:

$$\text{CTCF} = \text{Integrated density} - (\text{area of the selected cells} \times \text{mean fluorescence of background readings}) \quad (1)$$

## Cytoskeleton Staining

On day 3 after seeding, mASC spheroids were first fixed with 4% paraformaldehyde for 10 min and followed by treatment with 5% Triton X-100 (Wako Pure Chemical Industries, Osaka, Japan) in PBS for 15 min to disrupt the cell membrane after washing. In order to prevent background fluorescence, cells were treated with 10% bovine serum albumin (BSA, Sigma-Aldrich, United States) at room temperature for about 30 min for blocking of non-specific binding of antibodies. A dilution (1:500) of anti-Vinculin antibody (ab18058, Abcam, Cambridge, United Kingdom) was then used as the primary antibody to mark vinculin for overnight incubation at 4°C. After washing away the free anti-vinculin antibodies, a mixture of staining solution, containing Alexa Fluor 488-conjugated goat anti-mouse IgG (H + L) secondary antibody (A11001, Thermo Fisher Scientific), DAPI (4',6-diamidino-2-phenylindole, Invitrogen<sup>TM</sup>) and Alexa Fluor 594 Phalloidin (Invitrogen<sup>TM</sup>) were added and incubated with cells at room temperature for at least 45 min. DAPI and Phalloidin were used to label the nuclei and filamentous actin (F-actin), respectively. The suspension spheroids were transferred into glass-based dish prior to the observation using a confocal microscope.

## Proliferation Assay

Spheroid cell proliferation assays were performed by using eBioscience<sup>TM</sup> Cell Proliferation Dye eFluor<sup>TM</sup> 670 (ThermoFisher).  $1 \times 10^6$  stem cells were counted and stained before seeding. The culture medium containing serum were first washed away with DPBS, and then cells were resuspended in 1mL DPBS complemented with 5 μL cell proliferation eFluor<sup>TM</sup> 670 dye. After incubating and staining in dark at 37°C, the staining reaction was stopped by adding 4–5 mL cold FBS and incubating the cells on ice for 5 min. Cells were spun down (280 g, 5 min, 4°C) and washed once with culture medium. Then,  $2 \times 10^5$  cells were seeded in non-adherent 12-well plate as suspension culture or on a porous nanosheet to form spheroid. On day 3,

the spheroids were harvested and dispersed into single cells with 100  $\mu$ L trypsin/EDTA. After 5–8 min incubation at 37°C, the fluorescent intensity was monitored with a Cytomics FC500 flow cytometer (Beckman Coulter). The initial signal on day 0 was measured right after the staining process as the basis for the cell division calculation. The eFluor™ 670 dye specifically binds to primary amines in cellular proteins and it would be dispensed evenly when the cell divides. According to the fading fluorescence, number of cell divisions and cell cycle duration were calculated using the following formula (2) and (3):

$$\text{Number of cell divisions} = \log_2 [(F_0 - F_{\text{unstained}}) / (F_{\text{samples}} - F_{\text{unstained}})] \quad (2)$$

$$\text{Cell cycle duration (hours)} = \text{cell culture duration (hours)} / \text{number of cell divisions} \quad (3)$$

Where  $F_0$  and  $F_{\text{samples}}$  indicate the median fluorescence intensities of spheroid cells on day 0 and day 3, respectively;  $F_{\text{unstained}}$  indicates the background signal of unstained cells.

## Apoptosis Analysis

PE Annexin V Apoptosis Detection Kit I (BD Bioscience) was used to analyze the spheroid cell apoptosis cultured as suspension and on porous nanosheets. PE Annexin V detects the loss of membrane integrity accompanying the latest stage of cell death. It is often used in combination with 7-AAD, which labels dead or damaged cells without intact membranes, to identify viable cells (double negative), early apoptotic cells (PE annexin V positive, 7-AAD negative) and dead cells (double positive). The cells collected from both culture systems were washed with DPBS and treated with trypsin/EDTA in 37°C water-bath for 5–8 min to deconstruct the spheroids into single cells. Next, culture medium was added to stop the trypsinization, and the single cells were washed with DPBS. As the instruction describes, PE annexin V and 7-AAD were mixed and diluted in 1X Annexin V binding buffer, and samples were added to the staining solution. The cells were incubated and protected from light at room temperature for 15 min and then diluted with 1X Annexin V binding buffer before analyzing with the Cytomics FC500 Flow cytometer. For the positive control, cells were exposed under UV-light for 3–4 h.

## Statistical Analysis

All statistical analyses were performed using one-way ANOVA by StatsPlus software.  $P$ -value significance was determined as  $<0.005$  (\*\*\*),  $<0.01$  (\*\*), and  $<0.05$  (\*).

# RESULTS

## Preparation and Characterization of Porous Nanosheet

Porous nanosheets were integrated from two immiscible polymers, PS and PDLA (1:1 in weight) via a gravure-coating method (Figure 1A), followed by a solvent etching process with cyclohexane to remove the phase-separated PS regions (Figure 1B).

As indicated by the phase-contrast images in Figures 2A,B, the porous nanosheet obtained after cyclohexane treatment showed bubble-like, shiny and porous topography, whereas the PS regions remaining in the non-porous nanosheet block the light from optical microscopy and exhibited as little black circles. The thickness of nanosheet was calculated from the relative height of nanosheet surface to the substrate surface using a stylus profilometer (Figures 2C,D). The resulting porous nanosheets were 168 nm in thickness and had an average porous area of 6.6  $\mu\text{m}^2$  with a porosity of 0.887, whereas the non-porous nanosheet was 228 nm in thickness (Table 1).

## Spheroid Formation on Porous Nanosheet

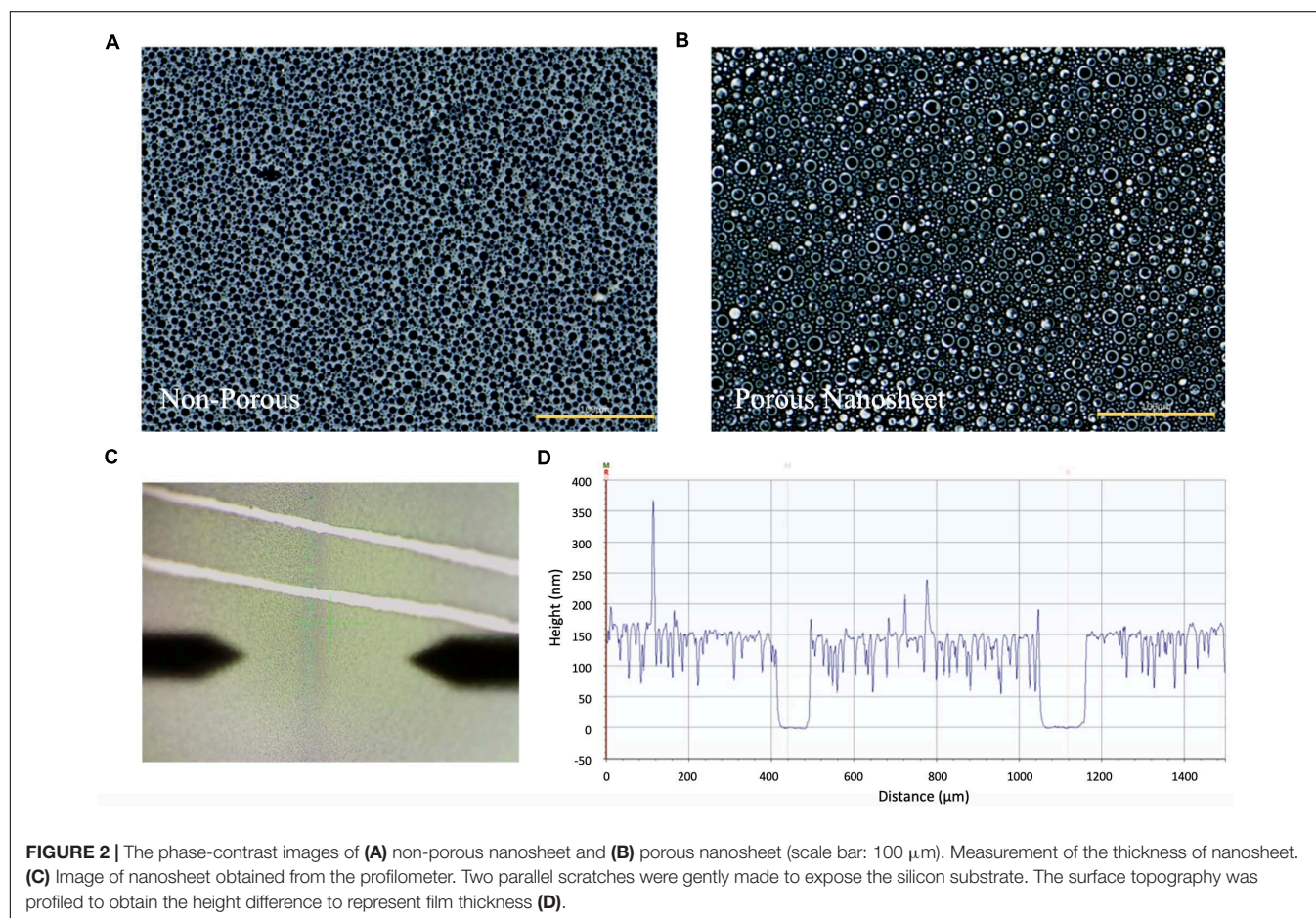
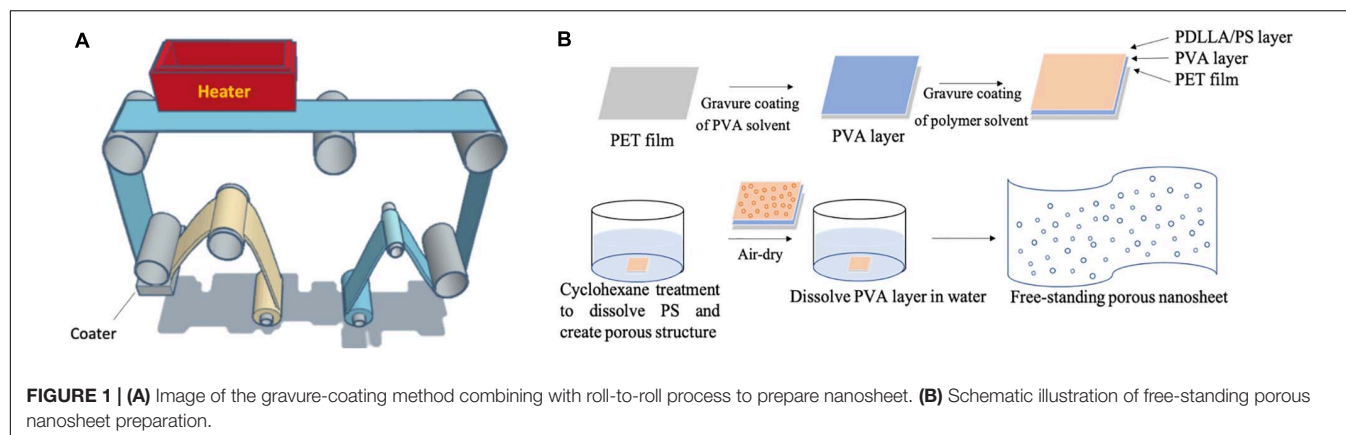
The edges of porous nanosheets were fixed on a glass-based dish with a silicone elastomer; poly-dimethyl-siloxane (PDMS), to proceed *in vitro* experiments. To observe how porous nanosheets affect the morphology of mASCs, phase-contrast images were captured every day after cell seeding. Representative images of day 1, 3, and 5 were chosen to demonstrate the differences in spheroid size and morphology among mASCs seeded on porous nanosheets and those in non-adherent dishes as suspension culture (Fang and Eglén, 2017; Lv et al., 2017).

Within the first 24 h, mASCs aggregated as spheroids both in suspension culture and on the porous nanosheet (Figure 3A). Spheroids in suspension culture tended to cluster together (represented by black arrows) as compared to spheroids on the porous nanosheet, which were evenly distributed. On day 1, spheroids had grown to an average diameter of 225  $\mu\text{m}$  in suspension culture and 90  $\mu\text{m}$  for spheroids on the porous nanosheet (Figure 3B). Furthermore, the spheroids cultured on the porous nanosheet started developing as a tethered state to the porous surface as indicated by the white arrows. Until day 5, the spheroids in suspension culture had grown to an average diameter of 250  $\mu\text{m}$  in irregular shapes, whereas spheroid size on the porous nanosheet remained below 100  $\mu\text{m}$  in average while keeping a spherical shape. Taken together, the porous nanosheets promoted a stable generation of mASC spheroids with a better homogeneity in size and shape.

## Live/Dead Staining of Spheroids

To evaluate cell viability within the spheroids, Live/Dead cell staining was conducted on day 3. Confocal images of spheroids on the porous nanosheet were captured through the porous nanosheet owing to its ultra-thin and transparent characteristics, which does not interfere with focusing the spheroids (Suematsu et al., 2020). It was found that dead cells mostly presented in the core region of spheroids regardless of culture environment (Figures 4A,B). Then, the fluorescence intensity was analyzed to confirm the Live/Dead cell ratio within the spheroids (Figure 4C). After eliminating the background intensity, the correlated total cell fluorescence (CTCF) indicates that the proportion of living and dead cells was approximately 9:1 in both culture systems. It is suggested that cell viability of spheroids was not affected by the external stress from the two





different culture environments and they supported cell viability to a similar extent. Note that spheroids of similar sizes were analyzed for comparison.

## Cytoskeleton Observation

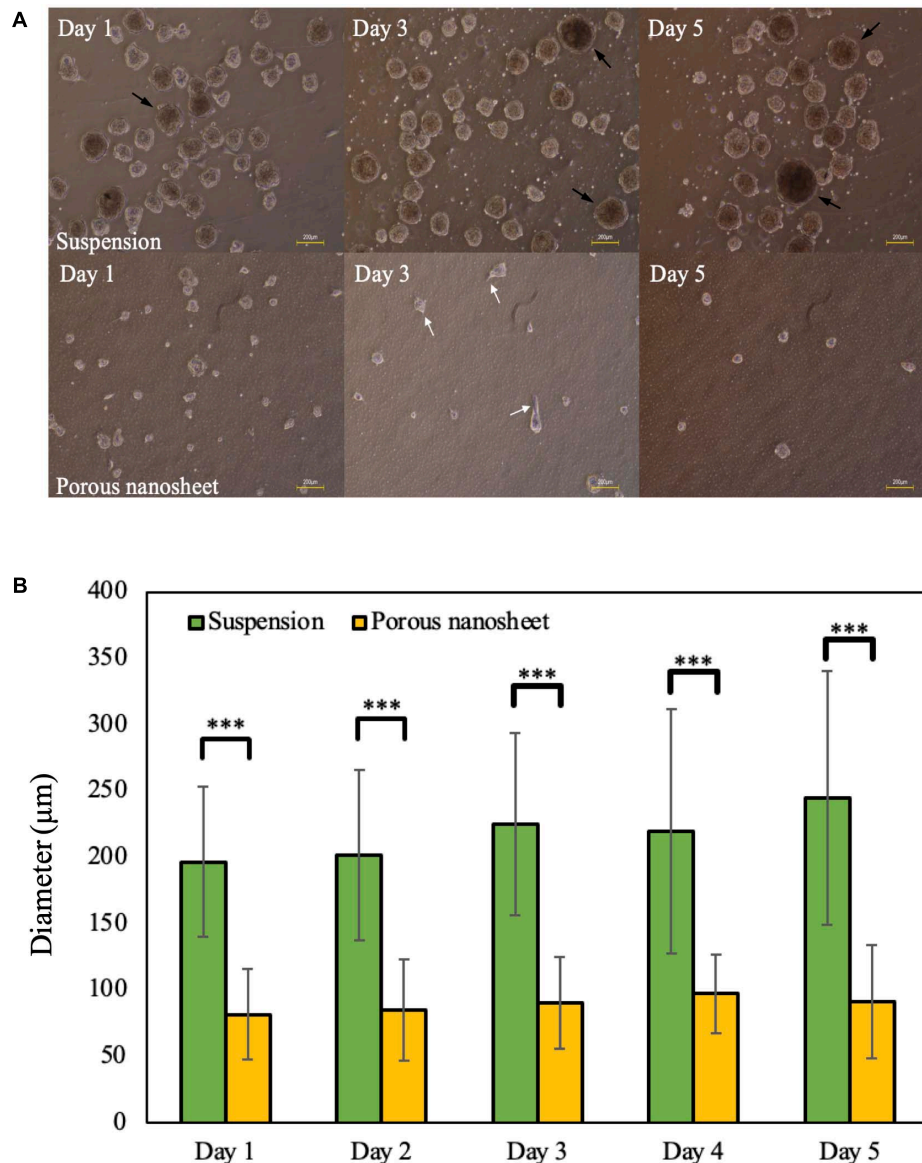
In order to confirm spheroid attachments on the porous nanosheets, vinculin and f-actin were visualized. Vinculin, a globular protein associated with cell adhesion and lamellipodia formation, interacts with f-actin and is identified as a

**TABLE 1 |** Characterization of non-porous nanosheet and porous nanosheet.

Sample	Thickness (nm)	Porous area ( $\mu\text{m}^2$ )	Porosity
Non-porous nanosheet	228	N/A	N/A
Porous nanosheet	168	6.6	0.887

mechano-transducer when developing focal adhesion (Tolbert et al., 2013). When cells anchor down on the substrate and spread like a monolayer culture, f-actin expands radially and

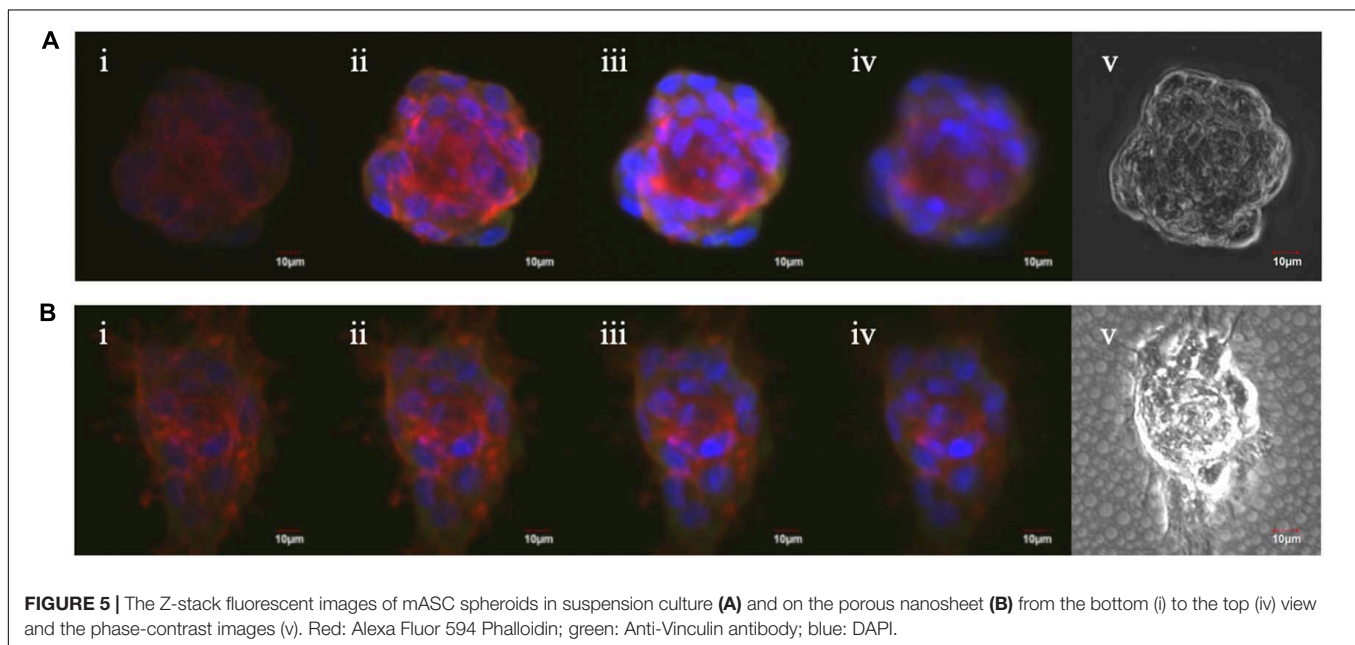
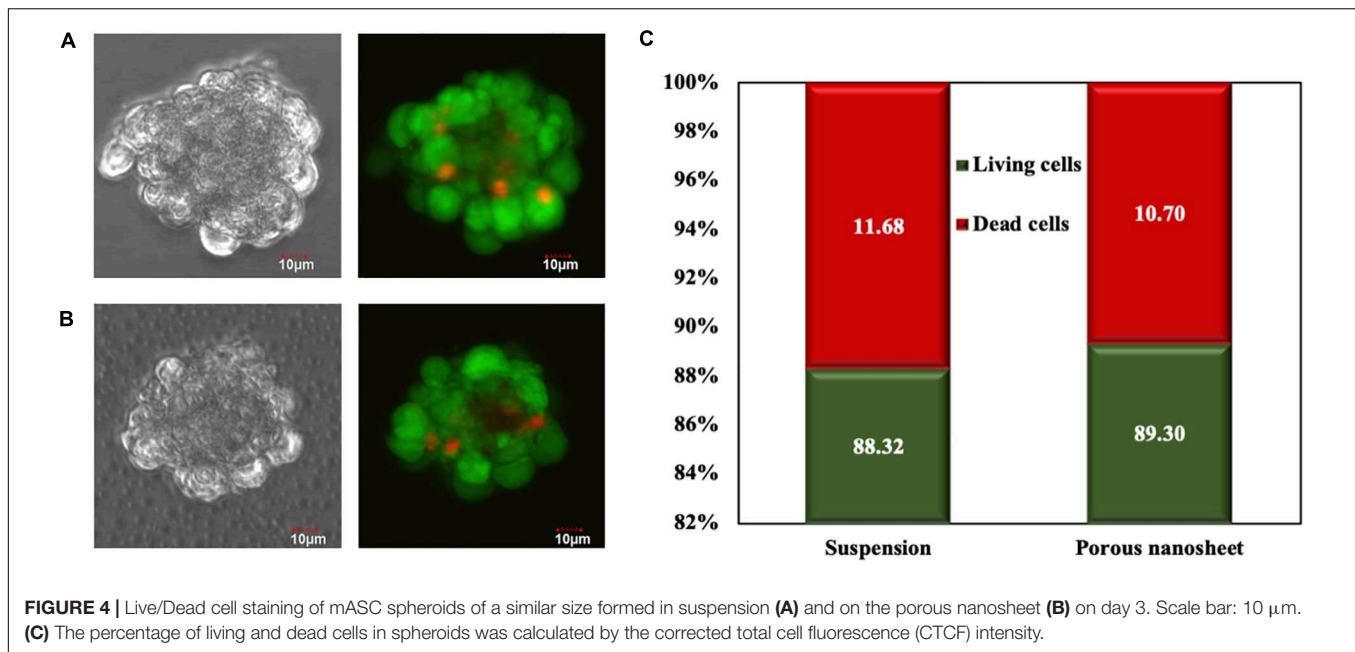




**FIGURE 3 | (A)** The morphology of mASC spheroids cultured in suspension (top) and on porous nanosheets (bottom) on days 1, 3, and 5. Spheroids showed the attachment on porous nanosheet on day 3 as indicated by white arrows and representative spheroid clusters formed in suspension culture were indicated by black arrows. Scale bar: 200 μm. **(B)** The average diameter of spheroids formed in suspension and on porous nanosheets. Data show the mean ± SD ( $n = 5$ , \*\*\* $p < 0.005$ ).

colocalizes with vinculin at the pointed end of actin filaments (Noriega and Subramanian, 2011). A confocal laser scanning microscope was used to capture z-stack images of the spheroids obtained from suspension culture or a porous nanosheet, starting on the bottom part of spheroids (i) and ending at the apex of spheroids (iv) (**Figure 5**). F-actin filaments in spheroids from the porous nanosheet noticeably expanded two-dimensionally and adhered on the nanosheet illustrated by the strongest intensity at the bottom half of the sphere. In contrast, focal adhesions were not observed for spheroids cultured in suspension and the f-actin fluorescence was mainly

apparent in the central part and on the edge of the spheroid. It is also interesting to find that the cell (nuclei) orientation is different. The spheroid cells in suspension culture appeared to be circumferentially oriented and tightly compressed, whereas those generated on the porous nanosheet did not show uniform orientation at the bottom part and only the cells at the apex part tended to be circumferentially oriented. Additionally, the colocalization of f-actin and vinculin was clearly observed in a circumferential manner surrounding the suspension spheroids. In comparison, when seeded on the porous nanosheet, f-actin seemed to be evenly distributed and

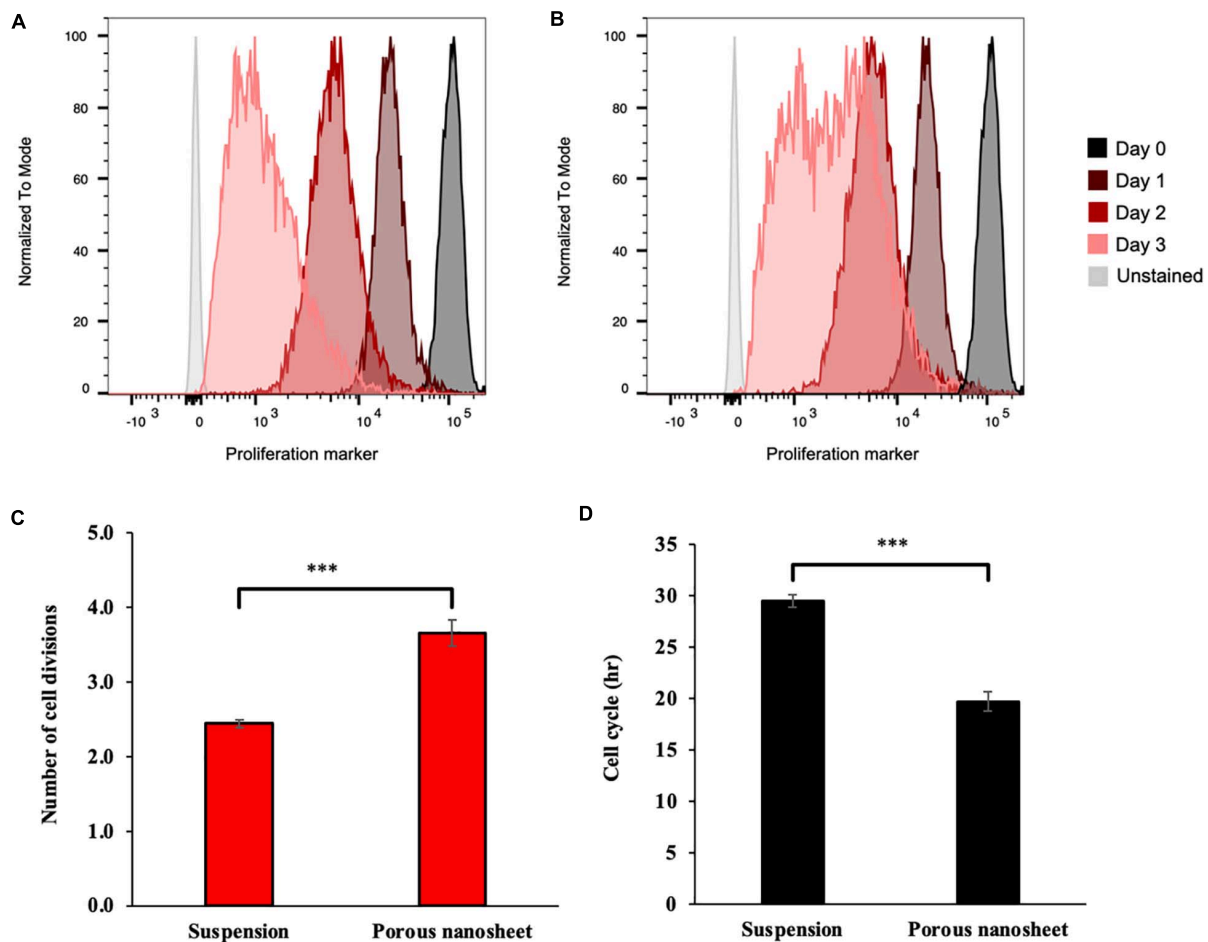


appeared with relatively weak signals only at the cell-junctions within the spheroids.

## Proliferation Assay

As mentioned above, the spheroids in both culture systems continued to develop during the culturing period, which might be due to the aggregation and fusion of spheroids or cell proliferation within the spheroid itself. Thus, a cell proliferation dye that binds to cellular proteins was used to analyze cell divisions overtime by flow cytometry, assuming that upon every cell division, the dye fluorescence intensity decreases to the half. As shown in **Figures 6A,B**, the cell division profiles of

two culture systems were remarkably different on day 3. In suspension culture, there was a portion of spheroid cells that have decelerated or terminated the division; whereas most of those seeded on the porous nanosheets continued to proliferate from day 2 to 3. It was estimated that the mASC of spheroids in suspension culture divided about 2.5 times while those adhered on the porous nanosheet divided approximately 4 times after 3 days of incubation (**Figure 6C**). From the number of cell divisions in this time frame, the cell cycle duration could be roughly gauged. It took nearly 30 h for the spheroid cells in suspension culture to divide once, whereas it took about 20 h for those seeded on the porous nanosheets (**Figure 6D**).



**FIGURE 6 |** Representative flow cytometry histograms of the proliferation assay, showing fluorescence reduction of cells from spheroids cultured on porous nanosheet (A) and in suspension (B) from day 1 to 3. Number of cell divisions (C) and average cell cycle duration (D) in hours (h) of mASC spheroids were calculated according to the loss of fluorescence on day 3 relative to initial fluorescence on day 0. Data show the mean  $\pm$  SD ( $n = 3$ , \*\*\* $p < 0.005$ ).

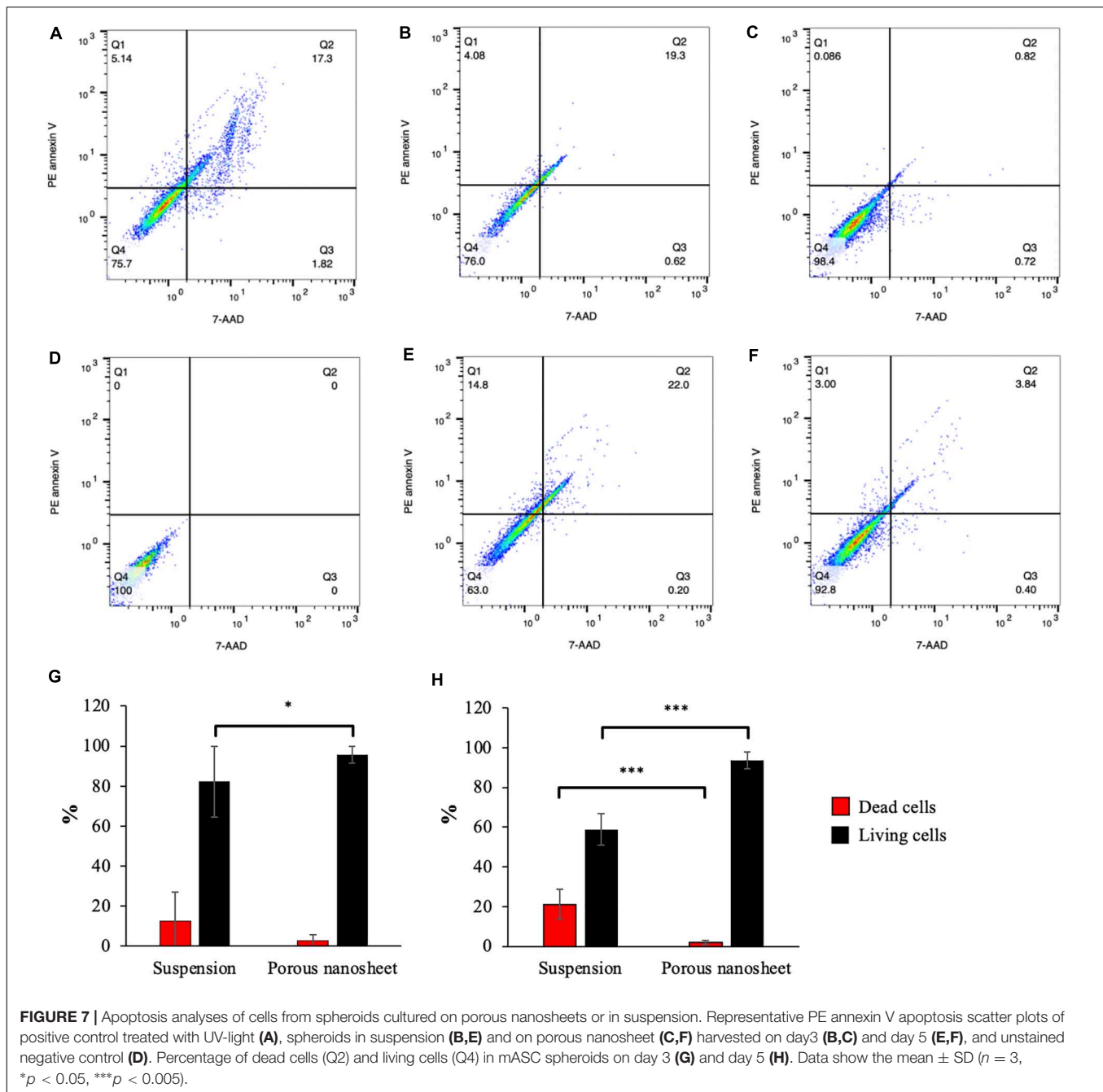
The mESC spheroids constructed on the porous nanosheets also showed a similar trend, with shorter, though not statistically significant, estimated cell cycle durations than the suspension culture (Supplementary Figure 1). Overall, these results showed that the porous nanosheet could sustain the spheroids in a prosperous state with higher proliferation rates than those in the suspension culture.

## Apoptosis Analysis

The Live/Dead staining allowed the visualization of cell viability in the spheroids of similar sizes; however, it is difficult to give an overall evaluation including all the spheroids. Therefore, cell apoptosis assay was performed by using PE annexin V and 7-AAD double staining in order to comprehensively analyze the cell death without size discrimination and investigate the progress of apoptosis.

The cells were harvested from mASC spheroids on day 3 and 5, dissociated by trypsinization of spheroids, labeled and analyzed by flow cytometry. PE annexin V and 7-AAD double negative cells were gated in Q4 on the scatter plots

(Figures 7A–F) to represent living cells. Early apoptotic cells were defined in Q1 gates (PE annexin V positive, 7-AAD negative) while double positive cells were identified in Q2 gates as dead cells. On day 3, the mASC spheroids in suspension contained almost 20% of dead cells whereas less than 3% of dead cells existed on the porous nanosheet (Figure 7G). On day 5, the population of living cells in the suspension spheroids shrank to 58%; in comparison, 93% of the spheroid cells on porous nanosheet remained alive (Figure 7H), the difference of which was remarkably significant. Notably, suspension culture also resulted in a small portion of early apoptotic cells on day 5, implying the progress of apoptosis. Similarly, it was confirmed with mESC spheroids that the porous nanosheet enhanced the cell viability with over 50% of living cells on day 5 in contrast to mESCs suspension spheroids, which contained about 40% of living cells (Supplementary Figures 2, 3). With the support of porous nanosheet, spheroids could be better preserved from apoptosis/necrosis than suspension culture and possibly used for long-term culture for comprehensive studies.



## DISCUSSION

PDLLA is a hydrophobic polymer with high biocompatibility and biodegradability. Due to its amorphous nature, it exhibits a lower mechanical strength and a faster degradation rate than the crystalline poly(L-lactic acid) (PLLA), making it more preferable to engineer soft tissue scaffold (Ulery et al., 2011). In contrast, PLLA is widely applied for bone tissue engineering after blending with hydrophilic polymers such as poly (glycolic acid) (PGA) to accelerate the degradation (Shuai et al., 2021). We previously reported that PDLLA porous nanosheets could induce the

generation of mASC spheroids owing to their porous topology and hydrophobic property (water contact angle:  $74^\circ$ ), which provides a weakly adhesive surface to allow the cell attachment as well as migration (Suematsu et al., 2020). In this study, we further demonstrated that this 3D culture scaffold had several advantages over the conventional suspension culture, as it better promoted cell viability and cell proliferation, as well as a better homogeneity of spheroid's size and shape.

3D culture provides a close-to-*in vivo* microenvironment to study cell-cell interaction and spatial effects. However, due to the lack of angiogenesis, radial gradients of nutrients, gasses



and growth factors are generated, reducing the viability of spheroid cells toward the core by decreased accessibility to nutrients and stimuli (Mehta et al., 2012; Cui et al., 2017). Therefore, the diameter of spheroids is thought to be a key factor in determining the cell viability. To verify whether the porous nanosheet imposes additional disturbance to affect the homeostasis, the spheroids of similar sizes were used for comparable quantifications, i.e., in the Live/Dead staining and cytoskeleton observation.

The mechanical properties within spheroids depend on the spatial position. In the core region, cells are progressively compressed in the radial direction, while the cells in the peripheral area further bear an increased circumferential stress, forming a contractile outer shell with more nuclei to drive the spheroid compaction (Lee et al., 2019). At an early stage of spheroid formation, cell-cell contact is relatively loose, and cells are either circumferentially, radially, or not oriented within the spheroids. Following the increase of spheroid size, structural anisotropy is generated, transducing compression forces toward the core which imposes mechanical anisotropy as a response of the cells. Such mechanical stress would lead to remodeling and changes in cell orientations, with the circumferentially oriented cells dominating the edge of spheroids, while the radially oriented cells occupying the central region (Dolega et al., 2017; Lee et al., 2019). Surprisingly, the cell orientation and the distribution of vinculin and f-actin were different in the two culture systems, indicating that the spheroids were bearing different kinds of mechanical stress and undergoing different fates. The porous nanosheet controlled the diameter of mASC spheroid by offering the porous scaffold for spheroids to cling on, which reduced cell aggregations. Therefore, the porous nanosheet might provide a culture system that favors the early stage of spheroid formation through proper size control.

In general, both the porous nanosheet and the non-adherent dish showed 3D spheroid construction capability. The suspended mASC spheroids had aggregated rapidly in the first 24 h, yet they developed into uneven sizes during the 5 days incubation. From Live/Dead cell staining, it was revealed that cell death started from the central region of the spheroids of both suspension and porous nanosheet culture systems, which may be caused by the shortage of oxygen and nutrition supplies; however, there was no distinguishable difference when analyzing by similar sizes. Living and dead cell ratios were almost 9:1 from both culture systems (Figure 4), in this regard, 20% of dead cells in suspended spheroids analyzed by flow cytometry (Figure 7) might have resulted from relatively large spheroids. Given the fact that the porous nanosheet provides limited area of attachment, stem cells could loosely and sparsely adhere to it to form spheroids instead of a monolayer (Suematsu et al., 2020). Meanwhile, the spheroid size would be controlled by lowering the opportunity of spheroid fusion or aggregation in terms of collision frequency and movement speed of spheroids. The crawling-like f-actin fluorescence depicts the spheroid attachments on the porous nanosheets, verifying the semi-adhesive property of porous nanosheet, which is crucial to control the spheroid size. As

a result, the spheroids anchoring on the porous nanosheet reduce interaction with each other so that preventing further development into larger spheroids as those floating in the suspension. The different culture environments contribute to the spheroid formation with different sizes, and the size difference affects spheroid biological performances including cell proliferation and apoptosis. Stem cells are characterized by a high potential of proliferation throughout the lifetime of organisms, and a high proliferation rate is thought to be crucial in maintaining human embryonic stem cell identity (Ruiz et al., 2011). It is no doubt that a sustained stem cell proliferation is a favorable feature for its application in tissue engineering and regeneration (Zhao et al., 2013; Zhou et al., 2016). Overall, mASC spheroids cultured on the porous nanosheet proliferated faster and survived longer when comparing to those forming in the suspension, implying a higher potential of PDLLA porous nanosheet in maintaining the stem cell functions and applicability for tissue engineering. However, further investigations on the pluripotency and differentiation potential of spheroid stem cells are necessary for a comprehensive evaluation. In addition to mASCs, porous nanosheets are also applicable to generate and sustain mESC spheroids although the improvement was not statistically significant. Note that advanced coating such as Matrigel is commonly used to support mESC culture *in vitro*, its lack on the porous nanosheet was likely to compromise the spheroid formation. Therefore, a combination of Matrigel and the porous nanosheet is expected to further improve the preservation of mESC spheroids.

In conclusion, porous nanosheets have shown its improvement in 3D cell constructions by providing efficient semi-attachment, which appears to limit the spheroid size. With preserved spheroids, porous nanosheets could be a promising scaffold for advanced long-term spheroid culture and close-to-*in vivo* study.

## DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/**Supplementary Material**, further inquiries can be directed to the corresponding author/s.

## AUTHOR CONTRIBUTIONS

YT, LT-F, and SW conducted the experiments and performed data analysis. YT wrote the first draft of the manuscript. TL guided the experiments, organized the manuscript, and performed data analysis. WK and ST conceived and supervised the study. All authors contributed to manuscript revision.

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The remaining authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Intervertebral Disk Degeneration: The Microenvironment and Tissue Engineering Strategies

Yiming Dou<sup>1†</sup>, Xun Sun<sup>1†</sup>, Xinlong Ma<sup>1</sup>, Xin Zhao<sup>2</sup> and Qiang Yang<sup>1\*</sup>

<sup>1</sup> Department of Spine Surgery, Tianjin Hospital, Tianjin University, Tianjin, China, <sup>2</sup> Department of Biomedical Engineering, The Hong Kong Polytechnic University, Kowloon, Hong Kong

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### \*Correspondence:

Qiang Yang  
yangqiang1980@126.com

<sup>†</sup> These authors have contributed  
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Intervertebral disk degeneration (IVDD) is a leading cause of disability. The degeneration is inevitable, and the mechanisms are complex. Current therapeutic strategies mainly focus on the relief of symptoms, not the intrinsic regeneration of the intervertebral disk (IVD). Tissue engineering is a promising strategy for IVDD due to its ability to restore a healthy microenvironment and promote IVD regeneration. This review briefly summarizes the IVD anatomy and composition and then sets out elements of the microenvironment and the interactions. We rationalized different scaffolds based on tissue engineering strategies used recently. To fulfill the complete restoration of a healthy IVD microenvironment, we propose that various tissue engineering strategies should be combined and customized to create personalized therapeutic strategies for each individual.

**Keywords:** intervertebral disk degeneration, microenvironment, tissue engineering, scaffold, regeneration, disk herniation, low back pain

## INTRODUCTION

Low back pain is prevalent in the society and is the number one cause of disability globally; 60–80% of adults experience varying degrees of low back pain (Murray et al., 2012; Hartvigsen et al., 2018). Although the etiology and pathology of low back pain are complex, evidence suggests that low back pain is strongly associated with intervertebral disk degeneration (IVDD) (Luoma et al., 1976). The intervertebral disk (IVD) is a fibrous cartilaginous tissue that connects two adjacent vertebral bodies consisting of three parts: the nucleus pulposus (NP), the annulus fibrosus (AF), and the cartilage endplate (CEP). The NP is a centrally located highly hydrated gel-like tissue surrounded by the AF, comprised of layer-by-layer collagen fiber lamellas. The CEPs are situated at the top and bottom of the vertebral bodies, and they interface the IVDs with the adjacent vertebrae (Lawson and Harfe, 2017). This complex structure plays a vital role in transmitting and absorbing mechanical loading onto the spine and maintaining motor function.

The IVD gradually degenerates due to aging and tissue damage caused by multiple stressors, resulting in vertebral instability, spinal canal stenosis, and spinal segment deformity, causing low back pain and mobility disability (Wang et al., 2016). There are three main events in IVDD: (1) inflammation and catabolic cascades, (2) continuous loss of cells, and (3) decline in cellular functions and anabolic activities (Frapin et al., 2019). NP gradually degenerates under the influence



of these events. The AF ruptures as the degree of degeneration increases, while the degenerated NP is extruded and oppresses nerve roots due to abnormal stress, resulting in nerve compression symptoms. The CEP becomes more calcified, which adversely affects the transport of nutrients and metabolites, making it difficult to maintain sagittal stress balance. Inflammatory factors stimulate nerve roots, causing or aggravating pain, and enhancing mobility disability. The deteriorating microenvironment further aggravates IVDD.

Traditional therapeutic strategies (surgical and non-surgical treatments) focus on solving the symptoms. Surgical treatments focus more on physically relieving symptoms at the organ level, such as decompressing nerve roots (discectomy) and removing the degenerated disk (fusion and disk replacement). Non-surgical treatments (non-pharmacological therapy, pharmacological therapy, and interventional therapies) perform limited intervention for the microenvironment through anti-inflammatory, analgesic, and spasmolytic effects. Traditional therapeutic strategies do not promote disk regeneration at the cell level and, therefore, do not reverse the progress of disk degeneration (Harmon et al., 2020). Some patients relapse after treatment, and the disk degeneration might even accelerate the degeneration of adjacent segments (Harper and Klineberg, 2019).

In the early stage of IVDD, the enclosed environment makes it difficult for the external stem cells to exert potential regenerative ability on the NP. In the later stage when the degeneration is severe, the AF will tear, and neovascularization and neoinnervation will occur in the NP. Inflammation caused by newly colonized immune cells further harms the microenvironment and accelerates degeneration. Thus, the deteriorating microenvironment and low self-regenerative ability of the IVD are primary obstacles for regeneration. How to customize suitable regenerative strategies for IVD has become a hot topic of recent research. Tissue engineering strategies combining material science, engineering, and life science will become new clinical therapeutic approaches in the future. Tissue engineering strategies contain three key elements: cells, scaffolds, and biomechanical or biochemical signals (Mhanna and Hasan, 2017). These elements have been applied and extended in IVD tissue engineering. Applications of stem cells in IVD and derivative methods (gene therapy and extracellular vesicle therapy) have shown promising therapeutic potential for IVDD. Illustrating the interactive mechanisms between the components [IVD cells, biological factors, extracellular matrix (ECM) components, and environmental factors] in the IVD microenvironment makes therapeutic strategies more rational. Scaffolds mimic the microstructure of the IVD ECM and provide proper structural support for IVD cells. Scaffolds can carry and release therapeutic biological factors. These findings contribute to IVD tissue engineering strategies. Tissue engineering strategies should be devised, considering the pathological changes of IVDD with the aim to reverse degeneration, namely: (1) cell proliferation should be promoted and cell apoptosis should be inhibited to ensure cell density; (2) the IVD microenvironment should be ameliorated to mitigate cellular

stress and inhibit inflammation; and (3) anabolic/catabolic balance should be mediated to ensure the quality and the quantity of the ECM.

## IVD AND THE MICROENVIRONMENT

### Anatomy and Composition of the IVD

The IVD is an articular cartilage structure located between the vertebral bodies and accounts for 25–30% of the overall length (height) of the spine. An IVD consists of the NP, the AF, and the CEP. The primary function of the IVD is to provide mechanical support for the vertebral body and allow movement of the spine (flexion, extension, and rotation) (Devereaux, 2007).

The NP is a centrally located highly hydrated gel-like structure consisting of NP cells and the ECM. The cell density in the NP is low (3,000/mm<sup>3</sup>), and the cell types are not completely clear, mainly containing small chondrocyte-like nucleopulpocytes (NPCs) and large vacuolated notochord cells (NCs) (Wang et al., 2014). The ECM of the NP is quite different comparing to that of hyaline cartilage and is mainly synthesized by the NPCs. It contains mostly aggrecan [a large proteoglycan (PG) aggregate], a high ratio of type II/type I collagen, hyaluronic acid (HA), and secondary components (type IX/VI/V collagen and small proteoglycans). PGs are widely found in cartilage, the brain, IVDs, tendons, and corneal tissues. They provide viscoelastic properties, retain water, maintain osmotic pressure, and arrange collagen tissue. Aggrecan in the IVD contains more keratan sulfated (a highly hydrated sulphated glycosaminoglycan). Thus, it provides more hydration ability (Iozzo and Schaefer, 2015; Frapin et al., 2019; Harmon et al., 2020). HA is a widely expressed glycosaminoglycan (GAG) located in the ECM, intracellular environment, and the cell surface that interacts with specific proteins, such as aggrecan, versican, lymphatic vessel endothelial HA receptor-1, tumor necrosis factor (TNF)-inducible gene-6 protein, and thrombospondin as well as membrane receptors, such as CD44, HA-mediated motility receptor, and Toll-like receptor 4/2. Thus, it plays a role in morphogenesis, cell migration, cell survival, apoptosis, inflammation, and tumorigenesis (Dicker et al., 2014). HA binds to PGs to form aggrecan, which highly hydrates NP and generates a hydrostatic pressure to effectively absorb stress, reduce vibration, and maintain the osmotic pressure and disk height of healthy IVDs (Raj, 2008; Brown et al., 2012; Wang et al., 2014). A small part of small leucine-rich proteoglycans (SLRPs) in PGs, which mediate tissue order, cell proliferation, matrix adhesion, and the responses between cell and biological factors, is an important signal transduction factor and receptor for the development, morphogenesis, and immunization activities of IVDs. Recent research has reported that SLRPs are related to IVDD, where the asporin gene in the SLRP family rapidly upregulates after the age of 22. SLRPs also exhibit an increased expression in degenerated IVDs (Song et al., 2008; Gruber et al., 2009). Type II collagen is the most critical collagen in the NP for forming irregular networks to support PGs and water. The distinctive arrangement equalizes the NP stress in different directions and, together with water, makes the NP elastic

(Colombier et al., 2014). As IVD degeneration progresses, type II collagen is gradually replaced by low-elasticity type I collagen, and the fibrotic NP gradually loses its biomechanical function.

The AF is comprised of 15–25 layers of angle-ply collagen fiber lamellas containing PGs, arranged in concentric circles outside the NP (Chu et al., 2018), and contains AF cells (AFCs) and ECM. The AF is divided into the outer AF and the inner AF. The outer AF mainly consists of dense and organized type I collagen. Thus, it has robust tensile strength, while the inner AF contains a lower ratio of type I collagen as the transition zone between the AF and NP. The density of AFCs is about 9,000/mm<sup>3</sup> and mainly consists of fibroblast cells (Raj, 2008). NP generates a hydrostatic pressure when IVDs are under axial stress and releases fluid shear stress to the AF. The multi-lamellated AF structure effectively converts the axial stress to interlamellar stress and produces annular stress to resist it (Liu et al., 2001; Iu et al., 2017). The AF can withstand compression and tensile stress during movement of IVDs.

The CEP is a hyaline cartilage structure that connects the vertebral bodies to the IVD. The CEP is comprised of hyaline cartilage cells and chondrocytes that produce PGs and type II collagen to transport nutrients and metabolites to the IVD, which is avascular tissue. The blood supply ends in the CEP, making the CEP a critical transportation junction. The CEP also bears the stress from the IVD to protect the vertebral bodies (McFadden and Taylor, 1989; Roberts et al., 1989; Fontana et al., 2015; Frapin et al., 2019).

As the core of this structure, the stable and enclosed microenvironment of the NP guarantees the expression of the ECM, which supports and separates the vertebrae, absorbs shock, permits transient compression, and allows for movement. NPCs also affect the expression of the AFC ECM. A healthy ECM (mainly type I collagen in the outer AF) is comprised of dense AF lamella, which provides protection for the NP microenvironment. Besides bearing sagittal stress, the CEP is the sole pathway of metabolite exchange for the avascular NP. These three parts together form a unique anatomical structure, which maintains homeostasis of the IVD microenvironment in unity and maintains healthy IVD function.

## Physiological Microenvironment

The IVD is in a unique microenvironment: avascular, hypoxic, hyperosmotic, acidic, and with low diffusion of metabolites and restricted by biomechanics (Roberts et al., 1996; Bartels et al., 1998; Wuertz et al., 2008). During early embryonic development, a rod-like notochord is located in the central area of the embryo and guides the ectoderm folds in on itself over the notochord to form neural tube mesenchymal cells (MSCs) to form vertebral bodies and the AF. NCs are trapped inside and participate in the formation of the NP. NCs are considered to be involved in the regeneration of the NP through cellular dialogue with other cells. NCs disappear in most human adults before the bone matures, but signs of IVDD occur not long after their disappearance (Hunter et al., 2003).

The cellular dialog between NCs and NP cells, which is triggered by multiple biological factors, maintains IVD homeostasis. These biological factors can also mediate the IVD microenvironment.

One of the critical factors mediating cell metabolism is hypoxia-inducible factor (HIF) (Semenza et al., 1994). HIF is a transcription factor that initiates a coordinated cellular cascade in response to a low-oxygen tension environment, including the regulation of numerous enzymes in response to hypoxia (Li et al., 2013). The HIF-1  $\alpha$ -subunits remain stable in hypoxic conditions but rapidly degrade under normoxic conditions. At the same time, HIF-1 $\alpha$  and HIF-2 $\alpha$  activities are stable in IVD, which illustrates that these two factors are positively related to IVD activity (Wang et al., 1995; Agrawal et al., 2008; Li et al., 2013).

Hypoxia-inducible factor is associated with most cell activities in the IVD. HIF-1 $\alpha$  regulates the glycolytic activity of NPCs to ensure an energy supply under hypoxia, which results in the acidic microenvironment in the IVD (Agrawal et al., 2007; Kalson et al., 2008). HIF promotes the synthesis of PGs, type II collagen, and GAGs by direct or indirect pathways (Agrawal et al., 2007; Li et al., 2013; Liu et al., 2017). It also inhibits Fas/FasL-mediated apoptosis of NPCs by inducing the expression of Galectin-3. The expression of HIF-1 $\alpha$  increases significantly in degenerated IVDs and is correlated with cell apoptosis (Ha et al., 2006; Li et al., 2013). HIF-2 $\alpha$  regulates the expression of aggrecan and type II collagen by regulating the expression of catabolic factors (MMPs-13 and ADAMTS-4). The expression of HIF-2 $\alpha$ , MMP-13, and ADAMTS-4 increases significantly, while the expression of aggrecan and type II collagen decreases significantly in degenerated IVDs (Huang et al., 2019), indicating that HIF plays an essential role in maintaining the IVD ECM. The hypoxic environment is destroyed after IVDD occurs; the absence of HIF can further accelerate degeneration (Meng et al., 2018).

Vascular endothelial growth factor (VEGF) is important during vasculogenesis and angiogenesis and mainly targets endothelial cells (Apte et al., 2019). Interestingly, the VEGF protein and its receptors are expressed in the avascular NP (Fujita et al., 2008). VEGF-A has a strong angiogenic activity and specific effects of mitosis and chemotaxis on endothelial cells (Risau, 1997). The expression of VEGF-A can be induced by hypoxia, leading to a local vascular invasion, but, typically, VEGF-A expressed by NPCs will not cause neovascularization. The reason may depend on the inhibition of endothelial cell adhesion and migration by the high aggrecan content in the IVD (Johnson et al., 1976). The complex of VEGF-A and its receptor VEGFR-1 also inhibit NP cell apoptosis (Fujita et al., 2008).

Nucleus pulposus cells stimulate NCs to secrete connective tissue growth factor (CTGF) and sonic hedgehog (Shh) by secreting transforming growth factor (TGF- $\beta$ ). CTGF and Shh together inhibit the expression of MMPs and stimulate the expression of tissue inhibitors of metalloproteinases (TIMPs), which inhibit MMPs, thereby suppressing the degradation of the ECM (Erwin et al., 2006; Erwin, 2008; Frapin et al., 2019). CTGF and Shh also stimulate the anabolism of the NP ECM and inhibit NP neovascularization and apoptosis by inhibiting VEGF, interleukin (IL)-6, and IL-8. Shh also promotes the proliferation of AF and CEP cells.

The bone morphogenetic protein (BMP) family promotes ECM synthesis by impacting the cellular dialog between NCs and NP cells and regulates the production of MMPs

(Leung et al., 2017; Frapin et al., 2019). BMP-2 and osteogenic protein-1 (BMP-7) upregulate the expression of aggrecan and type II collagen, promote the synthesis of GAGs, and concomitantly inhibit the expression of profibrotic genes. BMP-2 and BMP-7 are the most effective factors in the family stimulating the accumulation of PGs, while BMP-4 and growth differentiation factor-5 GDF-5 (BMP-14) stimulate the accumulation of collagen (Zhang et al., 2006; Leung et al., 2017; Li et al., 2017).

The characteristics of high osmotic pressure and high hydration ensure the biomechanical function of the IVD. As a weight-bearing organ, the metabolic and cellular activities of the IVD are closely related to the biomechanical microenvironment (Neidlinger-Wilke et al., 2014). External loads on the spine result in intense pressure on the disk. Intradiscal pressure varies from 0.1 to 2.3 MPa in different locations (Wilke et al., 1999; Neidlinger-Wilke et al., 2014). Severe degeneration can unbalance sagittal/coronal stress, resulting in abnormal stress of degenerated segments (Le Huec et al., 2011). NP cells in the IVD are mainly under extensive hydrostatic pressure, AF cells are under tensile strain, and the CEP is under compression due to its location at the interface (Le Maitre et al., 2004; Setton and Chen, 2006). The cytoskeleton of the IVD cells responds to the mechanical microenvironment. High osmotic pressure has positive effects on metabolic activity and matrix gene expression by IVD cells, and changes in hydrostatic pressure affect the synthesis of PGs by regulating the production of nitric oxide (Liu et al., 2001).

Interactions between different components in the IVD microenvironment maintain the homeostasis of the microenvironment (Figure 1). Due to aging, impaired disk function, loss of cells, and imbalance of ECM anabolism/catabolism, the homeostasis of the microenvironment breaks down, eventually resulting in pathological IVDD.

## Pathological Environment

The degenerative mechanisms are very complex and are related to various causes, such as age, genetics, the microenvironment, and biomechanics (Frapin et al., 2019). Early degeneration may be asymptomatic. Signs may be detectable by radiography, while the reduction in water content of NP can be visualized by magnetic resonance imaging due to the reduced synthesis of PGs and reduced disk height of the IVD on a computed tomography scan (Modic et al., 1988). Four main changes occur during degeneration: (1) cell senescence, (2) imbalance of ECM anabolism/catabolism, (3) inflammatory microenvironment, and (4) abnormal biomechanics (Figure 2).

## Cell Senescence

Cellular senescence, which is a fundamental mechanism that mediates age-related dysfunctions and chronic diseases, accumulates in human, bovine, and rat degenerative IVDs during aging (Roberts et al., 2006; Gruber et al., 2007; Kim et al., 2009; Muñoz-Espín and Serrano, 2014; van Deursen, 2014; Sharpless and Sherr, 2015). Unlike apoptosis, senescent cells are metabolically viable and arrest at the cell cycle transition, cease proliferation, and exhibit an altered expression of various catabolic cytokines and degrading enzymes (Campisi, 2005;

Davalos et al., 2010; Muñoz-Espín and Serrano, 2014; van Deursen, 2014; Sharpless and Sherr, 2015). Two intrinsic pathways are related to cellular senescence in IVDs: the p53-p21-RB pathway in a telomere-dependent manner and stress-induced premature senescence that activates the p16INK4a -RB pathway independently of telomere length (Ben-Porath and Weinberg, 2005). A stimulus from the microenvironment can cause damage to IVD cells, resulting in early senescence.

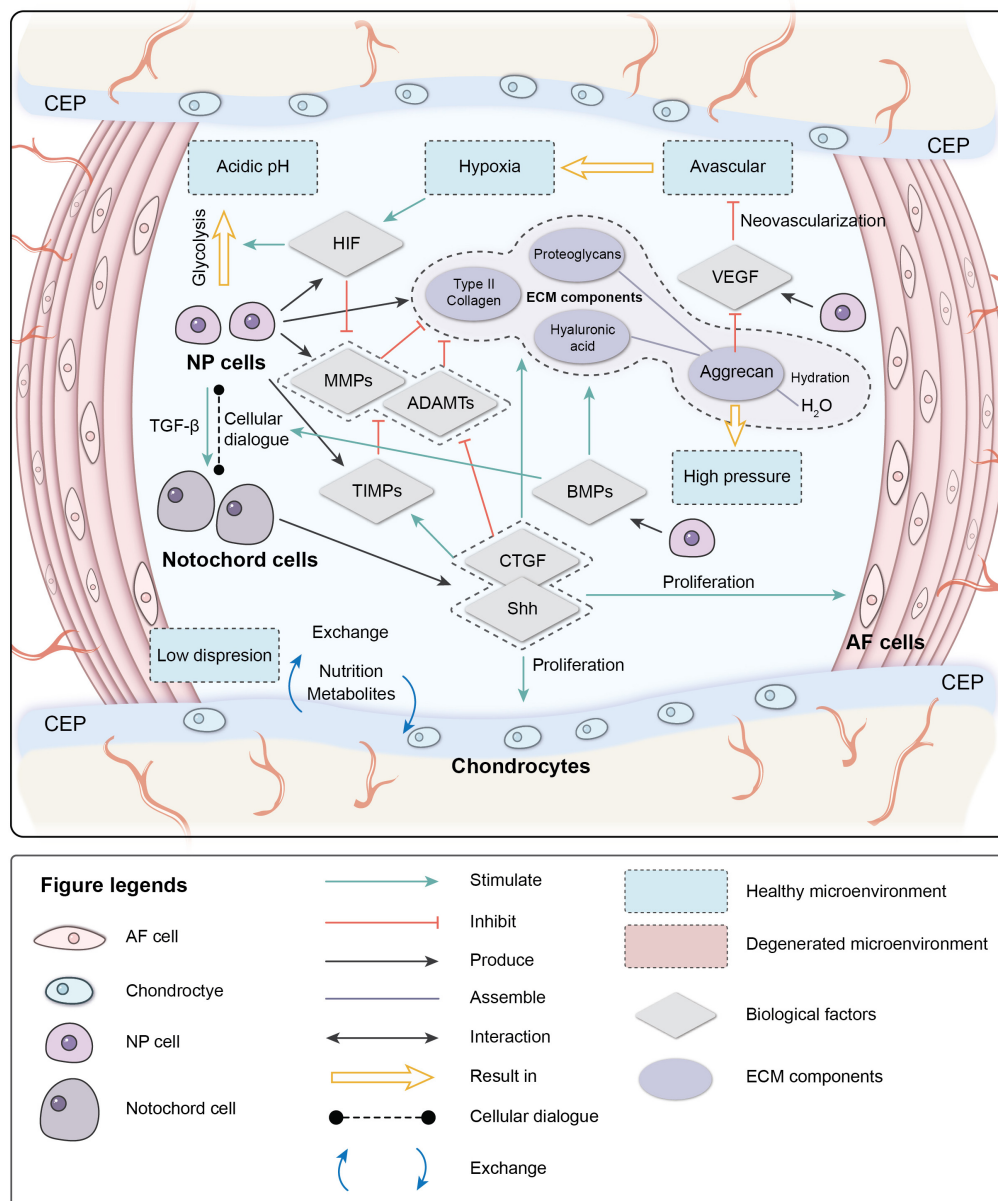
Senescent cells in degenerated disks can form senescent cell clusters, which can cause inflammatory stress by secreting pro-inflammatory cytokines and accelerate the senescence of neighboring cells (Wang et al., 2016). As a result, cellular senescence, impaired self-repair capacity, increased inflammation, and enhanced catabolism gradually lead to deterioration of the microenvironment and cause disorder in the cellular dialogue and an imbalance of anabolism/catabolism.

## Imbalance of ECM Anabolism/Catabolism

The ECM provides mechanical support for the IVD and is essential for maintaining the relatively avascular and aneural nature of a healthy disk. Proteoglycans, which are a primary component of the ECM, retain water and contribute to the osmotic pressure responsible for NP biomechanical properties. In the early stage of degeneration, proteoglycan content gradually decreases and is a sign of early degeneration. Type I collagen makes IVDs fibrotic, while type II collagen makes IVDs elastic. The proportions of collagen in the disk change with degeneration of the matrix. The absolute quantity of collagen changes little, but the type and distribution of the collagen can be altered. The ratio of collagen type II/type I decreases, and fibronectin content increases with increasing degeneration. As a result, the disk becomes more fibrotic and less elastic (Raj, 2008; Wang et al., 2016).

The balance between anabolism and catabolism is positively related to the IVD microenvironment, and the cellular dialogue, metabolic enzyme activity, and biomechanical changes affect this balance. As the population of NCs decreases with age, the cellular dialogue between NCs and NPCs gradually diminishes. Anabolic activity is mediated by biological factor disorders, while catabolic activity continuously progresses (Freemont, 2009).

MMPs, disintegrin, and metalloproteinase with thrombospondin motifs (ADAMTS) are important catabolic factors in disk degeneration. ADAMTS1, 4, 5, 9, and 15 are aggrecanases that degrade aggrecan, while MMP1, 8, and 13 are collagenases that cleave fibrillar collagen (Weiler et al., 2002; Le Maitre et al., 2004; Bachmeier et al., 2009; Vo et al., 2013). Upregulation of MMP and ADAMTS expression and enzymatic activity is implicated in the destruction of the disk ECM, leading to the development of IVDD (Bachmeier et al., 2009; Pockert et al., 2009; Vo et al., 2013). TIMP, which is a specific inhibitor of MMPs, binds to the active forms of MMPs to suppress catabolic enzyme activity. However, the activities of MMPs, ADAMTS, and TIMP are all regulated by inflammation-related factors, particularly pro-inflammatory factors. PGs in the IVD are proposed as a barrier to vascular and neural ingrowth into the NP (Khan et al., 2017). As PG contents decrease, the inhibiting effect for the neovascularization function



**FIGURE 1 |** Physiological microenvironment of intervertebral disk (IVD). AF, annulus fibrosus; NP, nucleus pulposus; CEP, cartilage endplate; ECM, extracellular matrix; HIF, hypoxia-inducible factor; MMPs, metalloproteinases; ADAMTs, metalloproteinase with thrombospondin motifs; TIMPs, tissue inhibitors of metalloproteinases; CTGF, connective tissue growth factor; Shh, sonic hedgehog; BMPs, bone morphogenetic protein; VEGF, vascular endothelial growth factor.

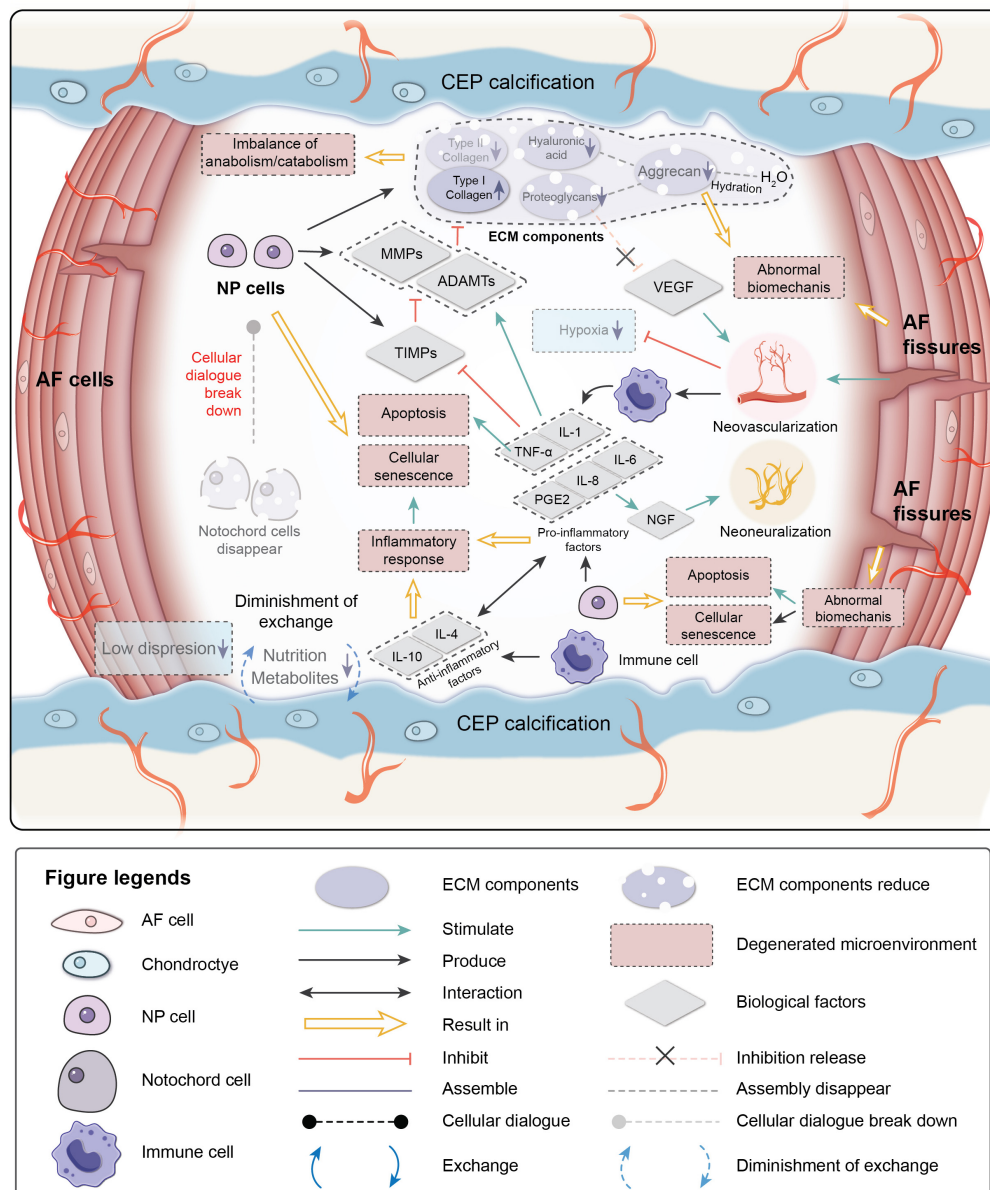
of VEGF weakens. Moreover, AF fissures caused by abnormal biomechanics allow neovascularization and neoinnervation from a lack of PG production. As a result, NP is invaded by immune cells transported by new blood vessels. Immune cells respond to the deteriorating microenvironment, eventually leading to inflammation.

### Inflammatory Microenvironment

Inflammation is an adaptive response triggered by noxious stimuli and conditions, such as infection and tissue injury. Regardless of the cause, inflammation presumably evolved

as an adaptive response to restore homeostasis. If the acute inflammatory response fails to eliminate the pathogen, the inflammatory process persists and acquires new characteristics (Majno and Joris, 2004; Medzhitov, 2008). The IVD is an avascular tissue until neovascularization occurs due to degeneration. Crystals and ECM breakdown products may be responsible for causing the IVD inflammatory response (Medzhitov, 2008). When anabolism/catabolism becomes unbalanced, degradation products trigger inflammation. In human IVD cells, HA fragments upregulate the mRNA expression levels of the inflammatory and catabolic genes IL-1 $\beta$ ,





**FIGURE 2 |** Pathological microenvironment of intervertebral disk (IVD). AF, annulus fibrosus; NP, nucleus pulposus; CEP, cartilage endplate. ECM, extracellular matrix; MMPs, metalloproteinases; ADAMTs, metalloproteinase with thrombospondin motifs; TIMPs, tissue inhibitors of metalloproteinases; VEGF, vascular endothelial growth factor; NGF, nerve growth factor; IL-1,4,6,8,10, interleukin-1,4,6,8,10; TNF- $\alpha$ , tumor necrosis factor; PGE2, prostaglandin E2.

IL-6, IL-8, cyclooxygenase-2, MMP-1, and MMP-13 (Quero et al., 2013). After neovascularization occurs, immune cells migrate to the NP to respond to the microenvironment. NP embryologically develops in an enclosed structure, which makes NP an avascular and immune-privileged tissue. Hence, immune cells invade when disk degeneration progresses to a certain stage, making IVD inflammation a chronic process. Inflammatory factors play essential roles in mediating IVD homeostasis and degeneration.

TNF- $\alpha$ , IL-1 $\beta$ , and IL-1 $\alpha$  are expressed by healthy IVD cells and immune cells (macrophages, monocytes, dendritic cells, B cells, and natural killer cells) (Le Maitre et al., 2005;

Richardson et al., 2009; Wang et al., 2013), where they have been observed at increased levels along with other inflammatory mediators in degenerated IVDs (Le Maitre et al., 2005, 2007). TNF- $\alpha$  is a cytokine that stimulates the inflammatory response, induces nerve swelling and neuropathic pain, and promotes cellular apoptosis *via* its cytotoxic effect (Pedersen et al., 2015). IL-1 $\beta$  induces a catabolic response by NP cells at the molecular level (Smith et al., 2011). Together they stimulate the production of MMPs and ADAMTs by NPCs and suppress the expression of TIMPs, resulting in decreased synthesis of aggrecan and type II collagen by NPCs (Frapin et al., 2019). Pro-inflammatory

factors are also related to pain. IL-6, IL-8, and prostaglandin E2 synthesis by NPCs stimulate nerve growth factor production, which induces abnormal nerve ingrowth and causes pain (Frapin et al., 2019). IL-6 and IL-8 are both higher in severe sciatica patients, and IL-6 is correlated to low back pain frequency (Khan et al., 2017). A herniated disk impinging on the nerve root is painless in 70% of patients, and it is likely that the secretion of products involved in the inflammation cascade in a torn AF sensitizes the nerve root or increases the number of innervations, thereby causing pain (Sharifi et al., 2015). Anti-inflammatory activity is also part of the IVD microenvironment. IL-4 and IL-10 are anti-inflammatory factors produced by activated macrophages and monocytes. They participate in pro-inflammatory/anti-inflammatory balance and inhibit the synthesis of pro-inflammatory cytokines (Khan et al., 2017).

### Abnormal Biomechanics

Intervertebral disk degeneration can also be affected by the biomechanical environment. A positive dose-response relationship is observed between cumulative lumbar load and early onset of symptomatic lumbar disk space narrowing: the disk disease onset time of workers with the highest exposure to heavy physical constraints is significantly advanced compared to others (Petit and Roquelaure, 2015). Xing et al. (1976) designed an amputated-leg rat model. Due to the forelimb amputation, the IVDs underwent abnormal mechanical loading. The results of senescence-associated  $\beta$ -galactosidase-positive staining showed that IVD cell senescence accelerates due to the abnormal loading of rat lumbar IVDs (Xing et al., 1976). Constant pressure on the disk can cause degeneration, which may be related to the upregulation of matrix degradation-related enzymes (Yurube et al., 2012; Neidlinger-Wilke et al., 2014). The mechanical load can also induce IVD cell apoptosis through the mitochondrial pathway (Rannou et al., 2004; Kuo et al., 2014).

Briefly, IVDs are continually adapting to changes in the microenvironment from embryogenesis to degeneration under the regulation of numerous factors (Colombier et al., 2014). The avascular and immune-privileged NPs regulate cell activity *via* cellular dialogue (Fontana et al., 2015; Henry et al., 2018). ECM synthesized by IVD cells ensures the biomechanical function of IVD. The CEP ensures transport of nutrients and metabolites (Zhang et al., 2018). When degeneration begins, interactions between components in the microenvironment break down, resulting in cascades of degeneration (Freemont, 2009). Therefore, therapeutic strategies should first consider how to relieve pain and regain IVD function in patients and, furthermore, correct the factors causing degeneration to eventually achieve ideal IVD regeneration and functional restoration.

## CURRENT IVDD TREATMENTS

### Non-surgical Treatments

Intervertebral disk degeneration is inevitable; 60–90% of patients can be treated non-surgically as long as no symptoms occur or mild symptoms that occur do not

affect daily life (el Barzouhi et al., 2013; Chiu et al., 2015). Non-surgical treatments include non-pharmacological treatments, pharmacological treatments, and interventional treatments. Non-pharmacological treatments, such as exercise, traction, acupuncture, massage, physical therapy, and spinal manipulation, are applied in daily life, but only for second-line or adjunctive treatment options with insufficient evidence for cure (Kreiner et al., 2014; Foster et al., 2018). More than 50% of patients with IVDD have used drugs. Non-steroidal anti-inflammatory drugs are most applied drugs for pain relief and improved function. Oral glucocorticoids alleviate the inflammation of nerve roots. At the same time, muscle relaxants can be useful to relieve muscle spasms, but these drugs are discouraged for lack of sufficient evidence (Kreiner et al., 2014; Ramaswami et al., 2017; Foster et al., 2018; Benzakour et al., 2019; Lee et al., 2019).

A meta-analysis reported that the therapeutic effect of an epidural injection intervention treatment is better than an intradiscal injection, percutaneous discectomy, traction, physical therapy/exercise, radiofrequency therapy, or chemonucleolysis (Lewis et al., 2015). However, interventional treatments are discouraged in the guidelines (Foster et al., 2018).

The goal of non-surgical treatments is to relieve symptoms and improve function, but they cannot halt the degeneration. Surgical treatments are required when non-surgical treatments are unable to relieve symptoms.

### Surgical Treatments

Mixter and Barr (1934) reported that a tumor in the spinal canal, which causes sciatica, was a herniated NP. Laminectomy, combined with excision of the NP, was performed, and satisfactory results were obtained (Mixter and Barr, 1934). Since then, surgical treatments to treat disk herniation have been developed. Four main types of surgical procedures are used: (1) decompression for neurological problems, (2) fusion to abolish motion at a functional spinal unit, (3) motion preservation/modifying surgery in the form of disk replacement/dynamic fixation devices, and (4) deformity surgery to realign the biomechanics between a large number of functional spinal units (Eisenstein et al., 2020). The purpose of decompression surgery is to relieve pain and numbness caused by nerve compression, and the most common procedure is discectomy.

A discectomy removes disk tissue that oppresses nerve roots in the intervertebral space. An emergency discectomy is required for patients who already have cauda equina syndrome and a new motor disorder. Selective discectomy is required for patients with persistent neurological symptoms that cannot be relieved by non-surgical treatments (Butler and Donnally, 2020).

Discectomy can be open or minimally invasive. Open discectomy is performed in a wide range of operations but with more tissue damage. Minimally invasive discectomy causes less tissue damage, but indications are limited. A meta-analysis showed that the incidence of postoperative complications and reoperations is similar, while less blood loss, shorter operating time, and shorter hospital stay are common after minimally invasive discectomy (Alvi et al., 2018;

Li et al., 2018a; Butler and Donnally, 2020). Discectomy relieves nerve root compression, relieves pain while retaining some of the structure, and restores some of the biomechanical functions of the disk. However, discectomy cannot change the deteriorating microenvironment. Therefore, further progression of degeneration cannot be prevented, and postoperative complications may occur (12.5, 13.3, and 10.8% for open microdiscectomy, microendoscopic discectomy, and percutaneous microdiscectomy, respectively) (Shriver et al., 2015). The AF must be broken to perform this surgery, and no new AF tissue is formed, so the opening remains open or is closed with the formation of scar tissue, which might cause reherniation (Sharifi et al., 2015).

Spinal fusion is a classic procedure. Hibbs and Albee treated Pott disease using fusion. Fusion abolishes pain by abolishing the motion of adjacent segments (Hibbs, 1964). Spinal fusion is widely used to treat many spinal diseases and is the gold standard for treating significant, chronic axial low back pain due to IVDD. About 60–65% of lumbar fusion procedures are performed for degenerative disk disease (Lee and Langrana, 2004; Auerbach et al., 2009). Spinal fusion completely removes the IVD and entirely relieves the oppression on the nerve roots, where the degenerated segments become integrated and lose motor function. Although the pain symptoms are relieved, the native IVD microenvironment is destroyed. The motor function is abolished, which may cause long-term complications. The mechanical environment of adjacent segments is affected and may develop into adjacent segmental disease (Lee and Langrana, 1984; Hilibrand and Robbins, 2004; Virk et al., 2014; Hashimoto et al., 2019).

Artificial disk prosthesis replacement surgery is performed to restore disk height, biomechanical structure, and motor function of the IVD and overcomes some deficiencies of spinal fusion. This procedure was first applied in the 1950s. As more and more prostheses have been developed, disk replacement surgery has developed quickly (Guyer and Ohnmeiss, 2003; Zhao et al., 2019). Disk replacement surgery effectively relieves pain and improves the quality of life, but it is not a superior substitute surgery for fusion. The average reoperation rate is 12.1% during follow-up after lumbar disk replacement or cervical disk replacement. Postoperative complications, such as prosthesis failure, infection, adjacent segmental disease, and pain, can occur, yet no evidence demonstrates that the surgical effect of disk replacement is better than fusion (Thavaneswaran and Vandepeer, 2014; Cui et al., 2018; MacDowall et al., 2019). The effect of disk replacement surgery greatly depends on the type of prosthesis and surgical technique of the surgeon. An artificial disk prosthesis replicates the anatomical structure of the IVD and attempts to mimic its biomechanical properties. The risk of adjacent segmental disease is reduced compared with fusion (Findlay et al., 2018). However, it is similar to fusion surgery in that the native IVD microenvironment is destroyed. Further follow-up studies are needed to investigate the long-term efficacy and safety of disk replacement surgery.

Several experimental surgeries have been developed. AF sutures strengthen the intensity of AF; thus, they restrict NP from herniating but cannot prevent the progression of IVDD

(Marcolongo et al., 2011). Annuloplasty is a minimally invasive method in which heat produced by electricity or radiofrequency radiation strengthens the collagen fibers and seals fissures in a process similar to tissue soldering (Helm Ii et al., 2017). Nucleoplasty releases the pressure on the outer AF, allowing the disk to return to normal size, thereby decompressing the nerve, with better therapeutic effects than non-surgical treatments. This procedure can be performed in the clinic, and the patient recovers quickly after the procedure, but the indications of this procedure are limited (Eichen et al., 2014; Liliang et al., 2016; de Rooij et al., 2017).

Patients who undergo surgery may have better pain relief, functional improvement, and satisfaction than those who receive non-surgical treatments. Nevertheless, long-term follow-up shows that disability outcomes are similar regardless of which treatment a patient receives (Atlas et al., 2001, 2005), suggesting that current strategies can only alleviate symptoms and that regenerative strategy is key to solving IVDD.

## TISSUE ENGINEERING STRATEGIES

Surgical treatments for IVDD have developed rapidly, but the limitations of surgical treatments are becoming apparent. Thus, researchers have turned their attention to tissue engineering strategies. Tissue engineering is “an interdisciplinary field that applies the principles of engineering and life sciences toward the development of biological substitutes that restore, maintain, or improve tissue function or a whole organ,” as defined by Langer and Vacanti (1993). Mizuno et al. (2004) fabricated the first documented IVD scaffold of polyglycolic acid and polylactic acid. The scaffold was seeded with AF cells. NP cells were injected into the center after 1 day. Then, the scaffolds were implanted in the subcutaneous space of the dorsum of athymic mice. The results showed that the engineered disks strongly resembled native IVDs and synthesized similar collagen components as native NP and AF (Mizuno et al., 2004; Hukins, 2005). This attempt was initially intended to identify an alternative strategy for IVD replacement due to its long-term deficiencies. The flaws in that study included the cell source, immunological rejection, and fixation of the scaffold, but that study inspired researchers to try tissue engineering strategies for IVDD. Various strategies have been formulated based on different treatment principles. It is important to note that there are no separate strategies, only those that are focused on in the study.

### Scaffold Strategy: From Natural ECM Mimetic to Multifunctional Platform

Scaffolds were initially designed as cell substrates to mimic the microenvironment where cells live. An ECM scaffold is similar to the native microenvironment of cells and has excellent biocompatibility and immunogenicity, which are beneficial for cell proliferation and metabolic activities. Yang et al. (2008) fabricated a decellularized ECM scaffold derived from human cartilage, and the scaffold successfully generated cartilaginous tissue in nude mice. As the AF is a fibrous cartilage structure and has a similar ECM



composition to cartilage, an ECM scaffold might be feasible to regenerate AF. McGuire et al. (2017) fabricated an ECM scaffold made of decellularized pericardial tissue. Collagen patches were obtained from treated pericardial tissue. These patches were assembled into a multi-laminate angle-ply scaffold. This scaffold mimicked the structure of AF, provided similar mechanical support, and supported cell viability, infiltration, and proliferation for bovine AF *in vitro* (McGuire et al., 2017).

Complete AF regeneration requires the recovery of biomechanical and structural properties of healthy AF and the restoration of the biological behavior of resident cells in AF (Driscoll et al., 2013). The alignment and organization of AF cells determine their biomechanical functions. As manufacturing technology develops, fabricating methods, such as electrospinning and 3-D printing, provide the possibility for microstructural scaffolds. Microstructural scaffolds guide the cells to form a specific order and are widely applied in AF tissue engineering due to their unique structure. Ma et al. (2018) developed a hybrid scaffold for AF tissue engineering. This hybrid scaffold consisted of traditional electrospun aligned nanofibrous scaffolds (AFS) as a baseline scaffold and electrospun aligned nanoyarn scaffolds (AYS). Morphological measurements showed that this hybrid scaffold replicates the tensile strength, axial compression, and anisotropic properties of AF tissues to some degree. Mechanical testing demonstrated that the tensile properties of AFS and AYS were qualitatively similar to those of native AF tissue. *In vitro* biocompatibility analyses demonstrated that the AYS and HS yield improved bone marrow mesenchymal stem cell (BMSC) proliferation (Ma et al., 2018). Gluais et al. (2019) fabricated electrospun aligned microfibrillar scaffolds that recruit neighboring healthy AF cells to migrate to the scaffold, which promotes cell colonization and proliferation. The results showed that, in addition to numerous dense collagen fibers in the aligned scaffolds, the fibers were arranged in the same direction as the scaffold (Gluais et al., 2019). Kang et al. (2018) combined electrospinning and 3D printing techniques to fabricate a biomimetic biodegradable scaffold that consisted of multi-lamella nano/microfibers. Each lamella contained one layer of aligned electrospun nanofibers and one layer of supporting 3D printed microfibers on each side, which were all aligned in the same direction, and the angle of the fibers between adjacent layers was 60°. The nano/microfibers aligned as native collagen tissue of AF. These results show that these scaffolds can form and integrate collagen fibers (Kang et al., 2018).

However, electrospun scaffolds often face several limitations, including low porosity that restricts uniform cell infiltration and a discrepancy of mechanical properties compared with native AF. To recapitulate the form and function of the complex anatomy of AF, Bhunia et al. (2018) adopted a directional freezing technique to fabricate silk-based multi-layered disk-like angle-ply ( $\pm 30^\circ$  of successive layers) bio-artificial scaffolds. The fabricated bio-discs supported the primary AF or human mesenchymal stem cell proliferation, differentiation, and deposition of a

sufficient amount of specific ECM *in vitro*. The subcutaneous implantation results showed a negligible immune response (Bhunia et al., 2018).

Three-dimensional printing precisely fabricates the scaffold structure. To generate a scaffold with angle-ply architecture similar to natural AF, Christiani et al. (2019) printed laminar constructs comprised of polycaprolactone (PCL) struts oriented at alternating angles of  $\pm 30^\circ$ . The mechanical characterization results showed that the mechanical properties of the scaffolds were similar to native AF tissue. They also cultured bovine AFCs with smooth-surface PCL and unidirectional channel-etched PCL to further study cell arrangement and ECM deposition, respectively. The SEM micrographs of the scaffolds showed that cells cultured on etched PCL had a tendency to align along the underlying surface, and the alignment of proteins with the underlying surface texture can be observed (Christiani et al., 2019).

However, PCL materials have poor hydrophilicity, have a long degradation time, and lack cell recognition sites. PCL materials bind poorly to surrounding host tissues after implantation *in vivo* (Cheng et al., 2017). Three-dimensional bioprinting achieves precise bionics according to the structure and size of native tissues and organs (Kang et al., 2016). As a novel technology, 3D bioprinting shows great potential and enormous advantages in the repair of IVDs. Sun et al. (2021) fabricated a dual growth-factor-releasing IVD scaffold using 3D bioprinting. They loaded CTGF and transforming growth factor- $\beta 3$  (TGF- $\beta 3$ ) on polydopamine nanospheres (PDA NPs). Then, CTGF@PDA and TGF- $\beta 3$ @PDA were mixed with rat BMSCs as the 3D bioprinting ink. A 3D model of the IVD scaffold was designed using AutoCAD software, and the support structures of CEP and AF were printed using a PCL polymer, while the parts of the model corresponding to AF and NP were printed using 3D bioprinting ink. *In vitro* experiments confirmed that the growth factors were released from the IVD scaffolds in a spatially controlled manner and induced the corresponding BMSCs to differentiate into NP-like cells and AF-like cells. After being implanted subcutaneously into the dorsum of nude mice, the reconstructed IVDs exhibited a zone-specific matrix: primarily type II collagen and glycosaminoglycan in the core zone and type I collagen in the surrounding zone (Sun et al., 2021).

In addition to providing mechanical support and topographic stimulus for cells, scaffolds can also be loaded with therapeutic drugs or cells. Cheng et al. (2011) fabricated an injectable thermosensitive chitosan/gelatin/glycerol phosphate hydrogel as a controlled release system for ferulic acid (FA) delivery. FA is an excellent antioxidant drug. The scaffolds were incorporated with FA, added to Transwells, and incubated with  $H_2O_2$ -induced NP cells *in vitro*. The results showed that these scaffolds achieved excellent antioxidant properties after loading with FA (Cheng et al., 2011).



Scaffolds are a crucial element of this strategy, as they provide a similar microenvironment for IVD cells by mimicking the structure of native IVDs. The scaffold strategy can also be combined with other strategies, such as delivery of therapeutic drugs or cells.

## Cell Therapy Strategy and Derivative Strategies

### Cell Therapy Strategy

Cell therapy is a classic strategy. An increasing number of studies have shown the efficacy of therapeutic cells in several IVDD animal models (Hiyama et al., 2008; Jeong et al., 2009; Henriksson et al., 2019). One study reported that stem cells and stem-like cells are found in almost all adult tissues (da Silva Meirelles et al., 2006).

Due to the potential of stem cells to differentiate, replacing damaged cells in target tissues, stem cells are ideal therapeutic cells for IVD tissue engineering (Krause, 2008; Meirelles Lda et al., 2009). Mesenchymal stem cells and induced pluripotent stem cells (iPSCs) are the most widely used in IVD cell therapy (Wei et al., 2014; Vickers et al., 2019). Several studies have demonstrated the ability of BMSCs and adipose-derived stem cells to differentiate into an NP-like phenotype, and *in vivo* studies have demonstrated the ability of implanted MSCs to enhance matrix production, particularly GAG synthesis, and increase disk height and hydration (Richardson et al., 2006; Gantenbein-Ritter et al., 2011; Stoyanov et al., 2011; Clarke et al., 2014). iPSCs differentiate into NCs or NP-like cells and reduce IVDD after transplantation *in vivo* (Sheyn et al., 2019; Xia K. et al., 2019).

Researchers have attempted different methods to transport therapeutic cells to diseased areas, such as direct injection of therapeutic cells or loading of cells onto scaffolds. Hiyama et al. (2008), Jeong et al. (2009), and Henriksson et al. (2019) injected human MSCs into rat, dog, and pig disk degeneration models, respectively. The results showed that transplantation of human MSCs has a positive repair effect on the xenogeneic animal disk degeneration model. Sheyn et al. (2019) induced human iPSCs to differentiate into notochordal cells *in vitro* and proved their regenerative capacity *in vivo* in an annular puncture pig model.

However, direct injection by needle puncture causes damage to the AF, and implanted cells could leak out through annular fissures. AF damage can lead to further degeneration and an increased risk of disk herniation (Oehme et al., 2015). Scaffolds could be pivotal to provide transplanted cells with a supportive environment. Bhunia et al. (2018) seeded MSCs on AF structure-like scaffolds and reported ideal results as mentioned above. GDF-5 inhibits IVDD and promotes NP-like differentiation of stem cells (Taylor et al., 2011). However, such a factor could require multiple injections due to its short life *in vivo* (Jin et al., 2016). Xia C. et al. (2019) generated polymeric gelatin microspheres, which can release growth

and differentiation factor-5 and act as a cell delivery vehicle for iPSC-derived NP-like cells. Then, they injected these cell-seeded polymeric microspheres into rat coccygeal IVDs, and the results indicated that disk height was recovered, water content was increased, and the NP ECM was partially restored (Xia K. et al., 2019). This study utilized growth factors to enhance the therapeutic efficacy of cells and prolong the life of growth factors *via* a polymeric gelatin microsphere as a sustained release platform. Multiple strategies complementing each other will become more important in IVD tissue engineering.

Recent studies have also revealed the existence of endogenous stem/progenitor cells in the IVD (Hu et al., 2018). Ying et al. (2019) found that stromal cell-derived factor-1 $\alpha$  expression is higher in the degenerative IVD microenvironment and showed a positive effect on enhancing the proliferation and recruitment of endogenous NP-derived stem cells into IVDs *in vitro* and *in vivo*. These cells might be a promising source for cell therapy.

Cell-based therapies have the advantages of modulating inflammation and concomitantly affecting the remodeling process, without presenting toxicity or immunosuppression (Lopes-Pacheco et al., 2016). These properties make cell therapy an exceptionally advantageous therapeutic approach for IVD tissue engineering. Nevertheless, various complications occur with stem cells, including tumorigenesis and immune reactions. Certain cases of tumorigenesis and immune reaction of iPSCs and embryonic stem cells (which are also a cell source for iPSCs) have been reported, and the application risks are always discussed (Pera et al., 2000; Taylor et al., 2011; Lee et al., 2013; Kim, 2014; Jin et al., 2016). Compared to iPSCs and embryonic stem cells, MSCs seem to be safer, as no case of tumorigenesis has been reported after MSC transplantation *in vivo* (Oryan et al., 2017).

Although the transplantation of stem cells may have risks, their efficacy cannot be denied. Risks may be hedged by enhancing the immune compatibility of stem cells. Immune rejection is caused by HLA mismatching. Xu et al. (2019) generated HLA pseudo-homozygous iPSCs through CRISPR-Cas9, with allele-specific editing of HLA heterozygous iPSCs and HLA-C-retained iPSCs, which evade T and NK cells *in vitro* and *in vivo*. HLA-C-retained iPSCs combined with HLA-class II knockout are immunologically compatible with >90% of the world's population, greatly facilitating iPSC-based regenerative medicine applications (Xu et al., 2019). Cell strategies have potential in tissue engineering, but improving safety and avoiding risks should be given more attention in future studies.

### Therapy Strategies Using Biological Factors

Biological factors are promising therapeutic drugs for IVDD as native mediators in the IVD microenvironment since they are key signaling factors in the cellular dialogue. Unlike conventional drugs, biological factors are secreted by IVD cells and have fewer side effects. Therapeutic biological factors

should be able to restore the healthy microenvironment of IVD. Therefore, biological factors should have at least one therapeutic function as follows: (1) pro-anabolism/anti-catabolism, (2) anti-inflammation, and (3) regulate cell activity. The application of biological factors is limited by their short life *in vivo*. Thus, biological factors are often used in conjunction with other strategies.

Researchers can customize biological factor scaffolds according to different conditions, with the key concept of restoring the balance of the microenvironment. Several widely used factors are introduced in **Table 1**.

## Gene Therapy Strategy

Biological factors intercellularly mediate cell activities, while microRNAs (miRNAs) mediate intracellularly. Gene therapy is a strategy to achieve the durable expression of a therapeutic gene or “transgene” at a level sufficient to ameliorate or cure disease symptoms with minimal adverse events. Two basic strategies have been proposed. An integrating vector is introduced into a precursor or stem cell so the gene is passed to every daughter cell or the gene is delivered in a non-integrating vector to long-lived post-mitotic or slowly dividing cells, ensuring the expression of that gene for the life of the cell (High and Roncarolo, 2019).

Although the application of gene therapy to the IVD has lagged behind other tissues, gene therapy shows excellent potential and safety for IVD tissue engineering. As an encapsulated and avascular tissue, the sealing property of IVD effectively prevents leakage of the disk contents to other sites in the body (Levicoff et al., 2005). Some researchers are using gene-editing techniques, such as CRISPR, to precisely alter DNA sequences or genetically modify immune cells to imbue them with the ability to fight cancer. TNF- $\alpha$  and IL-1 $\beta$  are inflammatory cytokines related to the inflammatory microenvironment in IVDD through TNFR1/IL1R1 signaling. To regulate this signaling, Farhang et al. (2019) produced CRISPR epigenome-edited lentiviral vectors based on TNFR1/IL1R1 targeted screens and delivered the genes into human NPCs by lentiviral transduction. The expression of the editing vectors was confirmed by qRT-PCR. Measurement of NF- $\kappa$ B activity (which is a downstream transcription factor that mediates TNFR1/IL1R1 signaling), apoptosis, and anabolic/catabolic changes in gene expression demonstrated that the lentiviral vectors significantly downregulated TNFR1 and IL1R1 and significantly attenuated the deleterious microenvironment (Farhang et al., 2019).

Gene therapy provides a potentially ideal tool for many diseases by delivering synthetic miRNAs to regulate gene expression (Deverman et al., 2018). However, there are obstacles to delivering miRNAs directly to target tissues due to their inactivation, low transfection efficiency, and short half-life. Chen W. et al. (2020) synthesized agomir, which is cholesterol-, methyl-, and phosphorothioate-modified miRNA, with good stability and enhanced transfection efficiency, in animals. Agomir penetrates the barriers of the cell membrane and tissues *in vivo* to enrich target cells. Agomir874 downregulates the expression of MMPs in NP by mimicking miRNA874. Chen et al. loaded agomir on an injectable polyethylene glycol hydrogel and injected it into a rat IVD. After 8 weeks, the rat IVD was gradually restored

to normal height, similar to the healthy group. These results show that agomir874 regulates the balance of synthesis/decomposition of the NP ECM and inhibits IVDD (Chen W. et al., 2020).

Six gene therapy products have been approved since 2016 (High and Roncarolo, 2019). Gene therapy is promising to be performed “one time,” with long-term and high-value therapeutic effects. However, there are still some problems to be solved, such as safety issues and high treatment expense. Gene therapy is aimed at specific targets, which have been studied thoroughly, to regulate cell activities. Thus, gene therapy cannot be used to regulate a series of therapeutic targets or pathways as in cell therapy. Due to the encapsulated microenvironment of the IVD, gene therapy is still promising and not fully explored for IVDD treatment. Studies on safety and therapeutic mechanisms should be conducted in the future.

## Extracellular Vesicle Therapeutic Strategy

Studies have confirmed that the mechanism of action of MSCs is predominantly paracrine (Lu et al., 2017; Zhu et al., 2020). Almost all types of cells secrete extracellular vesicles (EVs). As the main components of paracrine activity, EVs derived from MSCs achieve regenerative functions. The three main kinds of EVs are exosomes, microvesicles, and apoptotic bodies (El Andaloussi et al., 2013). Exosomes from MSCs have various effects on IVD regeneration, such as antioxidant, anti-inflammatory, anti-apoptosis, and proliferation-promoting effects (El Andaloussi et al., 2013; Lu et al., 2017; Xia C. et al., 2019). Due to the low immunogenicity and high efficiency of exosomes compared to MSCs, exosomes are promising substitutes for MSCs in cell therapy. The mechanism of its regenerative ability remains unclear, but it is very likely to be related to miRNAs. The level of miR-532-5p was observed to be decreased in TNF- $\alpha$  induced apoptotic NPCs but abundant in bone marrow mesenchymal stem cell (BMSC)-derived exosomes. Zhu et al. (2020) demonstrated that exosomes from BMSCs could deliver miR-532-5p and suppress the IVDD *via* targeting RASSF5. Cheng et al. (2018) also reported that exosomes from BMSCs could deliver miR-21, which could activate the PI3K/Akt pathway in apoptotic NPCs, and further inhibit IVDD.

Exosomes were also confirmed to be associated with pathological changes in IVDD. circRNA\_0000253 has the maximum upregulation in degenerative NPC exosomes. Song et al. (2020) found that exosomes in NPCs were differentially expressed in degenerative and normal NPCs. The circRNA\_0000253 levels notably increased and the miRNA-141-5p levels markedly reduced in degenerative NPCs compared with normal NPCs. Further research confirmed that the circRNA\_0000253 could increase IVDD by adsorbing miRNA-141-5p and downregulating SIRT1 *in vivo* and *in vitro* (Song et al., 2020). This study also demonstrated that, in a degenerative microenvironment of IVD, communication between NPCs by secreting exosomes may aggravate degeneration. Utilizing or blocking degeneration-related exosomes might become a promising therapeutic strategy for IVD degeneration.

**TABLE 1** | Several widely used therapeutic biological factors.

Factors	Targets	Scaffolds	Effects	Functions
GDF-5	Rat adipose-derived mesenchymal stem cells(ADSCs) <i>in vitro</i> and rat tail puncture model <i>in vivo</i>	Gelatin methacryloyl (GelMA) microspheres	(1) Exhibited good mechanical properties, biocompatibilities and enhanced the ADSCs differentiation into NP-like phenotypes <i>in vitro</i> . (2) Maintained NP tissue integrity and accelerated the synthesis of ECM <i>in vivo</i> .	Pro-anabolism and induce differentiation (Xu et al., 2020)
IGF-1	Human NPCs <i>in vitro</i>		IGF-1 protein treatment upregulated the IGF-1R and ER- $\alpha$ expression and promoted NP cell proliferation, collagen-II, and PCNA expression.	Pro-anabolism, promote proliferation and anti-inflammation (Chen R. S. et al., 2020)
SDF-1	Rat BMSCs <i>in vitro</i> and rat tail puncture model <i>in vivo</i>	Albumin/heparin nanoparticles (BHNPs)	(1) Induce migration of BMSCs <i>in vitro</i> . (2) Induce regeneration of annulus fibrosus and nucleus pulposus <i>in vivo</i> .	Pro-anabolism and induce migration (Zhang H. et al., 2018)
CCL5	Bovine AFCs and organ-cultured IVDs <i>in vitro</i> . Sheep AF rupture model <i>in vivo</i> .	Fibrin Gel	(1) Dose-dependent recruitment effect of CCL5 on AF cells were confirmed. (2) In the organ culture study CCL5 did not stimulate homing of AF cells toward the defect sites. (3) The pilot animal study did not show any repair effect of CCL5.	Induce migration (Zhou et al., 2020)
KGN	Human ADSCs <i>in vitro</i> and rat tail puncture model <i>in vivo</i>	Amphiphilic copolymer PEG-PAPO	(1) Enhanced the viability, autophagic activation (P62, LC3 II), ECM-related transcription factor (SOX9), and ECM (Collagen II, Aggrecan) maintenance in human ADSCs <i>in vitro</i> . (2) The injection of PAKM with human ADSCs yielded higher disk height and water content in rats.	Pro-anabolism and induce differentiation (Yu et al., 2021)
BMP-2	Human NPCs and AFCs <i>in vitro</i>		BMP-2 antagonized the IL-18 induced upregulation of aggrecan, type II collagen, and SOX6, resulting in reversal of IL-18 mediated disk degeneration.	Anti-catabolism (Ye et al., 2016)
BMP-7	Human NPCs <i>in vitro</i>	Self-assembled peptide RADA-KPSS	Attenuate the expression of MMP-3, MMP-9, ADAMTS-4, NF- $\kappa$ B-p65, IL-1, IL-6, and PGE2, promote the accumulation of ECM proteins and suppress apoptosis of NP cells treated with TNF- $\alpha$ <i>in vitro</i>	Pro-anabolism/anti-catabolism, anti-apoptosis, and anti-inflammation (Li et al., 2018b)
Anti-TNF- $\alpha$	Rat lumbar IVD puncture model <i>in vivo</i>		Prevents long-term upregulation of intradiscal TNF $\alpha$	Anti-inflammation (vashwick-Rogler et al., 2018)
Combination of CCL5, TGF- $\beta$ 1, and GDF-5	Human ADSCs and lumbar ovine IVD organ <i>in vitro</i>	Pullulan microbeads	(1) Induced Human ADSCs migration <i>in vitro</i> (2) The overall NP cellularity, the collagen type II and aggrecan staining intensity, and the Tie2-positive progenitor cell density in the NP were increased at day 28 compared to the control groups.	Pro-anabolism and induce migration (Frapiin et al., 2020)

GDF-5, growth and differentiation factor 5. IGF-1, insulin-like growth factor 1. TGF- $\beta$ 1, transforming growth factor beta1. SDF-1, stromal cell-derived factor-1. CCL5, chemokine receptor type 5. KGN, kartogenin. BMP-2, bone morphogenetic protein 2. BMP-7, bone morphogenetic protein 7. TNF- $\alpha$ , tumor necrosis factor-alpha.

The components of exosomes should be thoroughly studied. EVs can also be used to carry therapeutic drugs. Dou et al. (2020) fabricated engineered EVs to deliver drugs targeting inflammation. EVs have also been proven to have therapeutic effects on IVDD, but they have not been applied in IVD tissue engineering. The combination of EVs and tissue engineering is prospective. The study of EVs is still a hot research field. As research progresses, EVs will be a

promising therapeutic drug or targeted drug delivery system for regenerating IVDs.

## CONCLUSION

Intervertebral disk degeneration is a complex and common disease, and tissue engineering strategies provide many

innovative methods, but most are still in the experimental stage. Surgical and non-surgical treatments are the most effective and safest strategies to relieve pain and regain mobility for patients with IVDD. However, there are obvious limitations in intervertebral discs: toward engineering tissue the current therapeutic strategies, and relief of symptoms do not mean that degeneration has stopped, as IVDs remain in a hostile microenvironment. Regenerative strategies are key to the reversal of degeneration and need to be further studied.

The application of genes, biological factors, and EV are research hotspots. These applications are better substitutes for therapeutic cells, but their long-term viability and efficacy remain uncertain, and the mechanisms are not fully understood. The analysis of active components and illustration of mechanisms should be emphasized in the next stage to achieve precise therapies. Tissue-engineered scaffolds are more like multifunctional platforms, which may be able to simultaneously regulate cellular activities, load, and the controlled release of therapeutic drugs or cells while providing proper structural support, anti-inflammation, and antibiosis. Compared to traditional materials, such as decellularized ECM and classic synthetic materials, manufacturing technology utilizes superior materials to most traditional materials. Although many biomaterials have been proven to be safe and stable, researchers must develop novel materials and adjust their characteristics according to other strategies.

Every therapeutic strategy has a deficiency. Researchers should combine different strategies to restore the healthy IVD microenvironment and try to customize the best treatment

strategy for individual patients according to their personalized microenvironments in the future.

## AUTHOR CONTRIBUTIONS

YD read the manuscript and wrote the draft. XS reviewed and improved the manuscript. QY set up the framework of this review. XM and XZ provided guidance for YD and XS as consultants. YD and XS contributed equally to this work. All the authors contributed to the article and approved the submitted version.

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## SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fbioe.2021.592118/full#supplementary-material>

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