



Defective Titanium Dioxide-supported Ultrasmall Au Clusters for Photocatalytic Hydrogen Production

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Ultrasmall precious metal clusters have attracted extensive attention for providing a very specific surface and promoting electron transfer. In this work, ultrasmall Au clusters based on defective TiO₂ nanosheets (Au/D-TiO₂) were prepared and introduced into photocatalytic hydrogen evolution. Different defects of TiO₂ nanosheets (D-TiO₂) were constructed using a heating process and then loaded with Au clusters. Compared with bare TiO₂, Au clusters established on defective TiO₂ nanosheets with a narrower band gap showed higher light absorption performances, resulting in obviously enhanced photocatalytic hydrogen evolution activity of 3,142.33 μ mol h⁻¹ g⁻¹, which was over 45 times than the pure TiO₂. The results showed that the catalysts had good prospects in the field of photocatalytic hydrogen production.

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1. INTRODUCTION

Photocatalytic H_2 evolution, which utilizes water decomposition under solar energy, is a promising route to overcome the growing energy crisis and environment issues. However, in this process, the photocatalyst plays a key role. At present, photocatalysts such as nitrides [1, 2], metal sulfides [3, 4], and metal oxides [5, 6] have been demonstrated in hydrogen evolution systems. Among the traditional photocatalysts, anatase is one of the most extensively studied photocatalysts due to its reasonable price, nontoxicity, and remarkable photochemical stability [7, 8]. However, TiO₂ with a comparatively large band gap (3.2 eV) can absorb only the ultraviolet portion of the solar spectrum, resulting in insufficient photocatalytic activity of TiO₂ for H₂ production.

Photocatalytic efficiency of TiO_2 for water decomposition is limited due to the high recombination rate of photogenerated carriers. Hence, many measures have been taken to solve the problem of low photocatalytic performance of TiO_2 , involving the mingling of transition metals [9, 10], adding of the nonmetal ions [11, 12], loading of the precious metals [13, 14], the surface for dye-sensitized [15], and the generation of hybrid semiconductor [16, 17]. The manufacture of metal-semiconductor oxide composite materials is another very dynamic research area that can improve the photocatalytic activity of semiconductor oxide photocatalysts [18]. It has been reported that the transfer of electrons from semiconductor oxides to metals may reduce electron hole recombination events and improve the photocatalytic performances of semiconductor oxide-based catalysts. Thus, the electron hole pairs in the recombined semiconductor oxide produced by photons can be reduced by a large part [19, 20].

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Therefore, in this work, the electron hole recombination rate was reduced, we have demonstrated a method of constructing defects on metal oxides to support Au clusters. Both oxygen vacancies and a Ti-O-Ti structure were formed in this process. Oxygen atoms were lost through the formed oxygen vacancies, and the original oxygen vacancies were occupied by metal clusters, thus effectively reducing the recombination rate of photogenerated carriers. Au clusters built on defective TiO₂ nanosheets (Au/D-TiO₂) were applied to the production of photocatalytic hydrolysis hydrogen and showed significantly enhanced performances. Compared with the traditional TiO₂ catalyst, the catalysts of Au/D-TiO2 could produce up to 45 times more hydrogen than TiO₂. The results demonstrated that the structural defects on the surface of metal oxide could improve the catalytic performance of Au/D-TiO2 catalysts. This work laid a foundation for the preparation of catalysts in the future.

2. EXPERIMENTAL SECTION

2.1. Materials

The chemical reagents Tetra-n-butyl Titanate $(Ti(OC_4H_9)_4, 99.0\%, AR grade)$, hydrofluoric acid solution (HF, 40.0 wt%, AR grade), ethanol (99.7%, AR grade), and ammonium carbonate (40%, AR grade) were bought from Tianjin Damao Chemical Reagent Factory, China. HAuCl₄·3H₂O (99.9%, AR grade) was bought from Aladdin, China. All experimental materials were used directly in the experiment after purchase.

2.2. Sample Preparation 2.2.1. Preparation of D-TiO₂ Nanosheets

 TiO_2 nanosheets were obtained by hydrothermal method. In a common synthesis, $Ti(OC_4H_9)_4$ (50 ml) and HF (6 ml) were

added to the Teflon-lined autoclave. Then, the hydrothermal reaction occurred at 180°C for 24 h. After the reaction, the white sediment of the Teflon-lined autoclave was centrifuged and rinsed three times with water and ethanol, and dried in an oven at a temperature of 80°C for 12 h. Different defects of D-TiO₂ sample were obtained after being calcined at reducing atmosphere (10 vol.% H₂ and 90 vol.% Ar, 2 h) with unequal temperatures (The D-TiO₂ were treated at 100°, 150°, and 200°C, respectively).

2.2.2. Synthesis of Au/D-TiO₂

Au/D-TiO₂ was composite according to the depositionprecipitation method. First, 1 g of pure or defective TiO_2 nanosheets powders were suspended in 50 ml distilled water and stirred for 20 min. Then, the aqueous solution (25 ml) of 2.4 g (NH₄)₂CO₃ and 0.01 g HAuCl₄·3H₂O were mixed into the above solution drop by drop and stirred for 1 h at room temperature. And then, the samples were collected by centrifugation, and washed with distilled water and ethanol for three times, respectively. Au/D-TiO₂ samples were gained after being dried at 70°C for 6 h and calcined (200°C, 10 vol.% H₂ and 90 vol.% Ar) for 2 h. For comparison, Au/TiO₂ was synthesized with the same method on TiO₂ nanosheets instead of D-TiO₂.

2.2.3. Characterization

The crystalline structures of the acquired samples were analyzed by a powder X-ray diffractometer (XRD, BRUKER D8 ADVANCE) with a scan range of 10°–90° and a step size of 0.02°. The measurement of electron paramagnetic resonance (EPR) was performed at room temperature using a BRUKE A300 EPR spectrometer. The transmission electron microscopy (TEM) and high angle annular dark-field (HAADF) were performed on a JEM-2100HR operating at





200 kV. The surface analysis of every sample was examined by X-ray photoelectron spectroscopy (XPS) using a Thermo Fisher Scientific K-Alpha spectrometer. The Metal contents of samples were analyzed by analysis of inductively coupled plasma spectroscopy (ICP, Agilent 700). The UV-vis diffuse reflectance spectra (DRS) recorded on a Lambda 750 spectrophotometer. The photoluminescence (PL) spectra gained by a F-4600 FL Spectrophotometer.

2.2.4. Photocatalytic Hydrogen Evolution

The photocatalytic hydrolysis reaction was carried out in a heatresistant glass reactor, which was connected to a sealed single channel glass system, a circulating water system, a controller, and a vacuum pump. A 300 w xenon lamp was applied as a light source and fixed with the distance of 1 cm from the glass reactor. The glass reactor was charged with 80 ml of distilled water, 20 ml of methanol solution and 50 mg of photocatalyst.

Before the start of the reaction, in order to keep the entire reaction device free of air, the vacuum pump was turned on for 1 h, and the valves of the glass system were rotated while evacuating, and the air of the system was extracted as much as possible. The circulating water and the controller were opened while the vacuum pump was being punched, and the magnetic stirrer placed under the glass reactor was turned at 500 rpm to avoid overheating of the reaction during the experiment and to guarantee uniform dispersion of the photocatalyst in the solution. In the photocatalytic hydrogen production experiment, a gas chromatograph was used to analyze the produced hydrogen gas, and the gas chromatograph was equipped with a packed bed column having a temperature of 70°C and a detector. In the hydrogen production experiment, the xenon lamp irradiation time was 1 h for each time, and the intake air was detected every hour with a total of 4 h [21, 22].



3. RESULTS AND DISCUSSION

The compositions and phase structures of the samples were studied by using XRD (**Figure 1**). It could be seen from the XRD patterns that both pure and defective TiO_2 were with a palpable structure of anatase phase (JCPDS no. 21-1272). In the case of Au/D-TiO₂, the crystallographic peaks of Au could be detected (**Figure 1A**). Defects in D-TiO₂ were tested by electron paramagnetic resonance (EPR) measurements (**Figure 1B**), which indicated an aerobic vacancy defect with a g value of 2.015 [23–25].

The preparation of defective TiO_2 was the first and an important step in the synthesis process of the catalyst Au/ D-TiO₂. The transmission electron microscopy (TEM) image (**Figure 2A**) demonstrated the nanosheet structure of D-TiO₂ with a length of about 50 nm. And the high-resolution transmission electron microscopy (HRTEM) showed a lattice spacing of approximate 0.358 nm (**Figure 2B**) corresponding to (101) planes of TiO₂. In addition, **Figure 2C** had many Au clusters distributed and marked by the red arrow. Further, the high angle annular dark-field (HAADF) and the element mapping image (**Figure 2D**) further indicated that the Au clusters were well dispersed on the defective TiO_2 carrier.

In order to further explore the surface defects of the pure TiO₂ and the defective TiO₂, the X-ray photoelectron spectroscopy (XPS) spectra were conducted. The O 1 score-level XPS spectrum manifested two peaks in Figure 3A, one of which at 530.0 eV was considered as the oxygen band of Ti-O-Ti, and another one at 531.5 eV could be attributed to oxygen vacancy. The peak area of 531.5 eV of 150°C treated sample was larger than that of the other samples, indicating the highest oxygen vacancy concentration (Figure 3A). Figure 3B showed the O 1 s XPS spectrum of Au/ TiO₂ and Au/D-TiO₂ (treated at 150°C). It was found that the binding energy of Au/D-TiO2 was significantly shifted to higher binding energies compared with Au/TiO2. And the strength of Ti-O-Ti of Au/D-TiO₂ (treated at 150°C) was greater than that of Au/ TiO₂. Usually, such binding energy transfer is explained by a strong interaction between the two components. Figure 3C displayed the Ti 2p XPS spectrum of TiO2 and Au/D-TiO2 (The D-TiO2 were treated at 100°, 150°, and 200°C, respectively) and demonstrated that they



width of synthesized of TiO₂, Au/TiO₂ and Au/D-TiO₂ (The D-TiO₂ were treated at 100°, 150°, and 200°C, respectively) (**D**) if width of synthesized of TiO₂, Au/TiO₂, and Au/D-TiO₂ (The D-TiO₂ were treated at 100°, 150°, and 200°C, respectively).

were not obvious differences. The Au 4f of Au/D-TiO₂ (treated at 150°C) core-level XPS spectrum displayed one special peak at 83.9 eV in **Figure 3D**, and this peak could be attributed to Au clusters [26, 27].

The measurements of UV-visible diffuse-reflectance spectrum (UV-vis DRS) were carried out to investigate the light absorption intensity. As can be seen from **Figure 4A**, the intensity of light absorption for Au/D-TiO₂ was significantly enhanced in the wavelength ranging from 250 to 800 nm compared with the untreated TiO₂ nanosheets. And Au/D-TiO₂ (treated at 150°C) showed the strongest absorption intensity. According to the UV-vis DRS, the optical band gap value E_g of all samples was estimated from these absorption profiles using the Tauc's relation (**Figure 4B**) [28]. As shown in **Figure 4B**, the band gaps of these samples were analogous. However, after combining with Au clusters, the band gap of the samples mildly narrowed, whereby the results of pure TiO₂, Au/TiO₂, Au/D-TiO₂ (treated at 100°C), Au/D-TiO₂ (treated at 150°C) and Au/D-TiO₂ (treated at 200°C) were 2.93, 3.05, 3.03,2.86 and 2.98 ev, respectively [29, 30].

Figure 5 presented a comparison of the PL spectra of synthesized TiO₂ and Au/D-TiO₂ (The D-TiO₂ were treated at 100°, 150°, and 200°C, respectively). PL emission in semiconductors was due to the recombination of free carriers. The PL peak at about 396 nm was attributed to the emission of the bandgap transition. At an excitation wavelength of 230 nm, the light energy was approximately equal to the bandgap energy of anatase (387 nm). As expected, the PL intensity of the prepared Au/D-TiO₂ was significantly reduced as compared with pure TiO2. This indicated that the recombination rate of electrons and holes of Au/D-TiO₂ sample was low. This might be owing to the fact that electrons were excited from the valence band to the conduction band, and then moved to Au, thereby preventing the direct recombination of electrons and holes. In general, low recombination rate of electrons and holes are often associated with high photocatalytic activity [31, 32].



FIGURE 5 | PL spectra of TiO₂, Au/TiO₂ and Au/D-TiO₂ (The D-TiO₂ were treated at 100°, 150°, and 200°C, respectively).

In order to understand the band structure change of TiO_2 nanosheets after Au loading, the Mott-Schottky experiment (MS) was conducted (**Figure 6**).

Based on the MS equation, the capacitance (C) depend on applied potential and could be fitted as follows [33]:

$$\frac{1}{C^2} = + \frac{2}{e\varepsilon\varepsilon_0 N_d} \left(E - E_{FB} - \frac{k_b}{e} \right) \tag{1}$$

Where the slope expresses the type of semiconductor (negative to p-type and positive to n-type). The N_d denotes the carrier density. The *C*, *E* (corrected by the AgCl vs 0.197 eV) denote the space charge capacitance and applied potential. $\boldsymbol{\varepsilon}$ denotes the relative permittivity. $\boldsymbol{\varepsilon}_0$ denotes the vacuum permittivity. Where *e*

denotes the electron charge. Besides, the intercept on the x axis denotes the E_{fb} (band potential).

As revealed in **Figure 6**, the positive slope in the Mott–Schottky plot for the samples indicated that TiO₂ nanosheets was typical n-type semiconductors according to the MS equation. Furthermore, the conduction band (CB) of the samples were obtained corrected by the AgCl by calculating, and the flat band potential (E_{fb} (Vs RHE)) of pure TiO₂, Au/TiO₂, Au/D-TiO₂ (treated at 100°C), Au/D-TiO₂ (treated at 150°C) and Au/D-TiO₂ (treated at 200°C) were gained to be -0.16, -0.13, -0.07, -0.04, and -0.01 V, respectively. Moreover, the valence band (VB) was calculated by adding the band gap value to the CB level. whereby the results of pure TiO₂, Au/TiO₂, Au/D-TiO₂



(treated at 100°C), Au/D-TiO₂ (treated at 150°C) and Au/D-TiO₂ (treated at 200°C) were 2.77, 2.92, 2.96, 2.82, and 2.97 V, respectively compared with NHE.

The photocatalytic H₂ production activities of the bare TiO₂, Au/TiO₂ and Au/D-TiO₂ were further examined in Figure 7. Under the same experimental conditions, the test was performed every 1 h, and a total of 4 h of experiments were performed. Bare TiO₂ and Au/TiO₂ only exhibited a very low photocatalytic H₂ production rate of 69.18 and 138.7 μ mol h⁻¹ g⁻¹, respectively. However, after coupling with Au clusters, the production rate of photocatalytic H₂ was significantly enhanced. Au/D-TiO₂ (treated at 150°C) exhibited the highest photocatalytic H₂ production rate of $3,142.33 \mu mol h^{-1} g^{-1}$, suggesting that the construction of Au/D-TiO2 could effectively boost the production activity of photocatalytic H₂. Besides, the production rate of photocatalytic H₂ first increased and then declined when the treated temperature of D-TiO₂ increased from 100° to 200°C for Au/D-TiO₂ samples, indicating that the treated temperature could efficiently adjust the defects of TiO₂ nanosheets and effect the H₂ production activity [34-36].

The stability test with Au/D-TiO₂ was carried out by a total of 16 h of cycling experiments, which were divided into four groups of 4 h, respectively, and an injection was performed every hour for data recording. The result suggested a good stability of Au/D-TiO₂ at fully photocatalytic hydrogen production experiment process without obvious decrease, which might be due to the Au clusters could be effectively stabilized on TiO₂.

A reaction mechanism for the evolution of hydrogen is shown in **Figure 8**. First, the valence band electrons of TiO_2 are excited to the conduction band, and then immediately transferred to Au through the intimate interface contacts. Since it is well known that H₂O has an ionization balance in the natural state, the water contains trace amounts of H⁺ and OH⁻. Therefore, the releasing H⁺ binding in the e⁻ of H₂O to produce H₂. H⁺ adsorption on the surface of the Au and accepting excited electrons is a key step in the







hydrogen evolution reaction. On the other hand, H_2O/OH^- reacts with h^+ in Au/D-TiO₂ to produce H_2 and OH. When H_2 is released, O_2 is also formed. Then O_2 reacts with the e^- to form O_2^- . Finally, the O_2^- reacts with CH₃OH to form H_2O . When TiO₂ is coupled with Au clusters, it can provide more active sites for the hydrogen release reaction due to the characteristics of the Au/ D-TiO₂ catalyst, thereby increasing photocatalytic activity [37, 38].

CONCLUSION

In conclusion, Au based on defective TiO₂ nanosheets (Au/D-TiO₂) has been successfully prepared and applied to the production of photocatalytic hydrolysis hydrogen. Compared with bare TiO₂ and Au/TiO₂ only exhibit a very low production rate of photocatalytic H₂ of 69.18 and 138.7 μ mol h⁻¹ g⁻¹, respectively, the production rate of photocatalytic H₂ was significantly enhanced after coupling with Au clusters. The Au/D-TiO₂ treated at 150°C exhibited the highest production rate of photocatalytic H₂ of 3,142.33 μ mol h⁻¹ g⁻¹. The stability test suggested a good stability of Au/D-TiO₂ at fully photocatalytic hydrogen production experiment process due to the Au clusters could be effectively stabilized on TiO₂. Overall, this could be an effective approach for enhancing photocatalytic hydrogen production efficiency and stability by catalyst.

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DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/supplementary material, further inquiries can be directed to the corresponding authors.

AUTHOR CONTRIBUTIONS

XZ: Conducted experiments and wrote manuscript; WZ: Conducted experiments, evaluated data; YX: Designed experiments, evaluated data, and revised manuscript; MJ: Acquired research funding, conceived research and revised manuscript.

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Conflict of Interest: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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