



# Fe-Vacancy-Ordered Fe<sub>4</sub>Se<sub>5</sub>: The Insulating Parent Phase of FeSe Superconductor

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We carried out a detailed study to investigate the existence of an insulating parent phase for FeSe superconductors. The insulating Fe<sub>4</sub>Se<sub>5</sub> with  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order shows a 3D-Mott variable-range-hopping behavior with a Verwey-like electronic correlation at around 45 K. The application of the RTA process at 450°C results in the destruction of Fe-vacancy order and induces more electron carriers by increasing the Fe<sup>3+</sup> valence state. Superconductivity emerges with  $T_c \sim 8$  K without changing the chemical stoichiometry of the sample after the RTA process, resulting in the addition of extra carriers in favor of superconductivity.

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## INTRODUCTION

The FeAs-based [1] and FeSe-based [2] superconductors are among the most investigated materials in condensed matter physics since their discovery in 2008. The observation of a wide range of superconducting transition temperatures, with the highest confirmed Cooper pair formation temperature up to 75 K in monolayer FeSe films [3], provides a unique opportunity to gain more insight into the origin of high-temperature superconductivity. The multiple-orbital nature of the Fe-based materials, combined with spin and charge degrees of freedom, results in the observation of many intriguing phenomena such as structural distortion, magnetic or orbital ordering [4], and electronic nematicity [5, 6]. There are suggestions that the orbital fluctuation may provide a new channel for realizing superconductivity [7, 8].

The parent compounds of FeAs-based materials exhibit structural transitions from a high-temperature tetragonal phase to a low-temperature orthorhombic phase, which is accompanied by an antiferromagnetic (AF) order [9, 10]. Upon doping, both the orthorhombic structure and the AF phase are suppressed and superconductivity is induced. On the other hand, FeSe undergoes a tetragonal-to-orthorhombic transition at  $\sim 90$  K [2, 11, 12]. However, no magnetic order is formed at ambient pressure [12, 13] and superconductivity below  $\sim 8$  K [2] is crucially related to this orthorhombic distortion. The nematic order coexists with superconductivity but not with long-range magnetic order which has led to arguments that the origin of the nematicity in FeSe is not magnetically but likely orbital-driven [14, 15].

However, recent studies show that the nematic states in the FeSe systems are far more complex [16–20]. There exist strong high-energy spin fluctuations [20] which suggest that the nematicity and magnetism may be still intimately linked. It was also found that there are many interesting features in the band structures of the nematic state. More surprises came as one applied pressure to FeSe. The application of pressure leads to the suppression of structural transition, the appearance of a

magnetically ordered phase at  $\sim 1$  GPa [13, 21], and  $T_c$  increases to a maximum  $\sim 37$  K [22–27] at  $\sim 6$  GPa. An even more dramatic enhancement of  $T_c$  was achieved on monolayer FeSe grown on SrTiO<sub>3</sub> substrate [28–31].

The above observations lead to questions that exist since the discovery of FeSe superconductors: what is the exact chemical stoichiometry of the compound and what is the exact phase diagram for the FeSe system? Earlier studies showed that the superconducting property of FeSe is very sensitive to its stoichiometry [2, 12, 32]. The fact that higher superconducting transition temperature exists in monolayer FeSe on SrTiO<sub>3</sub> substrate suggests that the commonly accepted phase diagram, derived from assuming that FeTe is the nonsuperconducting parent compound of FeSe [33], is questionable. Studies have observed the trace of the superconducting feature with  $T_c$  close to 40 K in samples of nanodimensional form [34].

It has been a debate whether there exists an antiferromagnetic Mott insulating parent phase, similar to the cuprate superconductors, for FeSe superconductors [35–37]. Chen et al. first reported the existence of tetragonal  $\beta$ -Fe<sub>1-x</sub>Se with Fe-vacancy orders, characterized by analytical transmission electron microscopy [38]. The authors further argued the Fe<sub>4</sub>Se<sub>5</sub> phase with  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order to be the parent phase of FeSe superconductors [38]. The Fe-vacancy order observed in the Fe<sub>4</sub>Se<sub>5</sub> phase is identical to the Fe-vacancy order observed in the A<sub>2</sub>Fe<sub>4</sub>Se<sub>5</sub> (A = K, Tl, Rb), which has been shown to be an antiferromagnetic [35, 39–42] and is the parent phase of the superconductor A<sub>2-x</sub>Fe<sub>4+x</sub>Se<sub>5</sub> [43, 44]. The detailed studies of the Fe vacancy in K<sub>2</sub>Fe<sub>4+x</sub>Se<sub>5</sub> reveal that its order/disorder is directly associated with superconductivity. A recent study shows that the Fe-vacancy-ordered Fe<sub>4</sub>Se<sub>5</sub> nanowire is the nonoxide material with the Verwey-like electronic correlation [45]. It suggests that a charge-ordered state emerges below  $T = 17$  K. The question remains unanswered is whether this Fe-vacancy-ordered phase is the parent compound of superconducting FeSe?

In this paper, we present the results of structure, electrical transport, and magnetic measurements on the polycrystalline sample of Fe<sub>4</sub>Se<sub>5</sub> treated by rapid-thermal-annealing (RTA) process at a proper temperature and time. After RTA treatment, the sample shows superconductivity with  $T_c \sim 8$  K without changing its chemical stoichiometry. Our findings confirm that the Fe<sub>4</sub>Se<sub>5</sub> with Fe-vacancy order is the parent compound of FeSe superconductors.

## EXPERIMENTAL TECHNIQUES

### Sample Preparation

Fe<sub>4</sub>Se<sub>5</sub> nanosheets were prepared by a chemical coprecipitation method. First, 200 ml of ethylene glycol was mixed with NaOH and SeO<sub>2</sub> powder and slowly heated up to 160°C for mixing well. The volume of 2.4 ml hydrazine hydrate was then added as the reducing agent. Then, at 160°C, the Fe precursor solution was added and reacted for 12 h in order to form Fe<sub>4</sub>Se<sub>5</sub> nanosheets. The Fe precursor solution is made by dissolving the amount FeCl<sub>2</sub> in ethylene glycol. The reaction was done under N<sub>2</sub> gas purging to

avoid the formation of oxide impurity. To clean the Fe<sub>4</sub>Se<sub>5</sub> nanosheets, the reacted product was dispersed in acetone with absolute ethanol, and high-speed centrifugation is applied to precipitate the nanosheets and remove the capping ligand dissolved in the above organic solvent. The nanosheets were finally dried in a vacuum for 24 h and collected. The process for the rapid thermal annealing (RTA) is as follows: the as-grown Fe<sub>4</sub>Se<sub>5</sub> nanosheets were heated at 450°C for 10 min in a tube furnace with 1 atm Ar gas inside to maintain a nonoxidation environment as a rapid thermal treatment process. After the rapid thermal treatment, an air-quenching process was taken by flowing room-temperature Ar gas through the tube. All the samples were stored in the oxygen-free glove box.

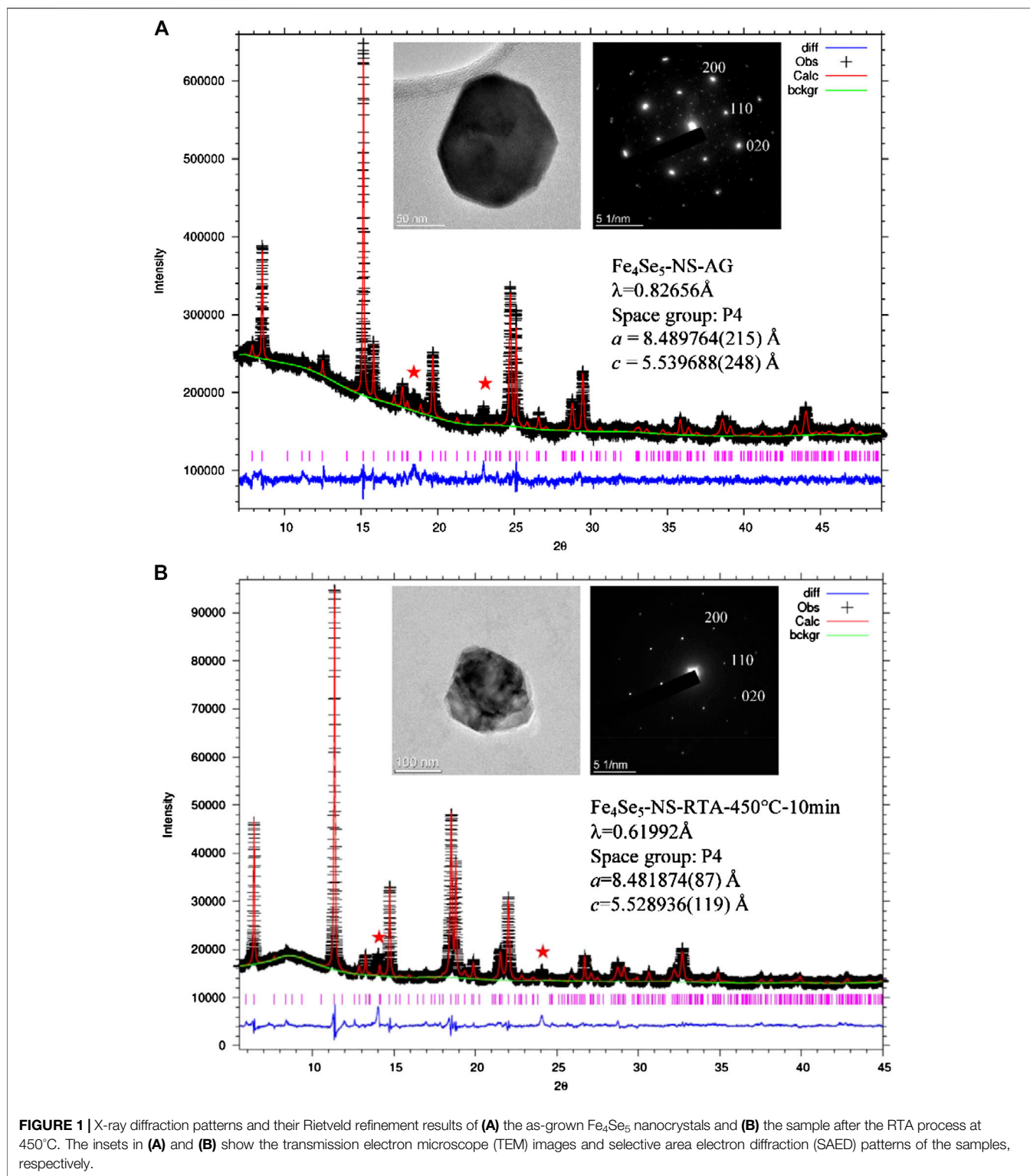
### Analysis

The crystal structure observation of the Fe<sub>4</sub>Se<sub>5</sub> samples was carried out by the high-resolution transmission electron microscope (HRTEM, JEOL JEM-2100F) and 4-circle x-ray diffractometer with the incident beam (12.4 keV) of wavelength 0.82656 Å at beam-line BL13A and wavelength 0.61992 Å at beam-line TPS09A in NSRRC. The temperature-dependent structural information of Fe<sub>4</sub>Se<sub>5</sub> samples was analyzed by the high-resolution neutron powder diffraction (high-resolution powder diffractometer Echidna with the wavelength of 2.4395 Å at ANSTO). The Fe<sub>4</sub>Se<sub>5</sub> nanosheet powder was pressed into the pallet under 200 kg/cm<sup>2</sup> at 100°C for 1 h for the following measurements: To identify the stoichiometry, the energy-disperse X-ray spectrometer (EDS) setup with the SEM (JEOL JSM-7001F Field Emission Scanning Electron Microscope) was applied. To affirm the valence states of the Fe ions, the X-ray photoemission spectroscopy (XPS, VG Scientific ESCALAB 250) measurements for the samples were performed. For polycrystalline bulk samples, the resistance was measured by using the standard four-probe method with silver paste for electrical contact and the Hall measurement by a Hall-bar configuration was done by the Quantum Design Physical Properties Measurement System (PPMS, Model 6000). The magnetic property was measured by the Quantum Design Superconducting quantum interference device (SQUID, VSM).

## EXPERIMENTAL RESULTS AND DISCUSSIONS

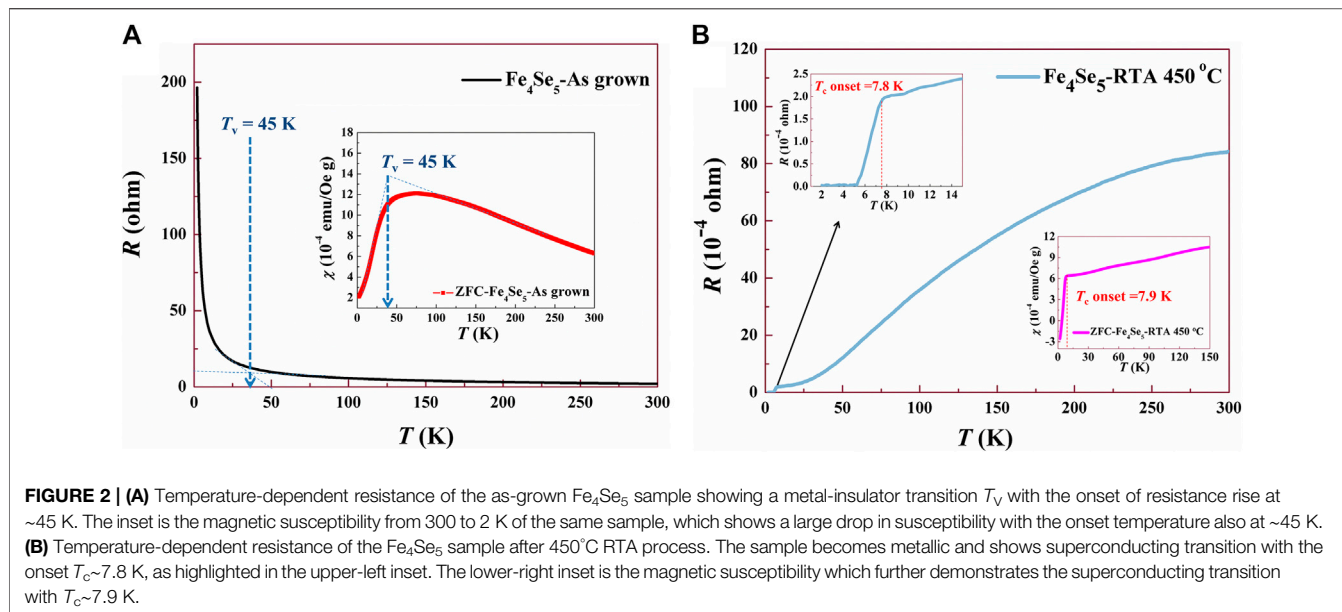
### Structural Analysis of Fe<sub>4</sub>Se<sub>5</sub>

Figure 1A shows the X-ray diffraction (XRD) patterns of the as-grown Fe<sub>4</sub>Se<sub>5</sub> sample at room temperature. The diffraction pattern, which exhibits superstructure peaks, is refined with a tetragonal  $P4$  symmetry with  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order instead of the tetragonal  $P4/mmm$  symmetry as observed in FeSe [2]. The insets are the TEM image of as-grown Fe<sub>4</sub>Se<sub>5</sub> nanocrystal and its TEM-SAED (selective area electron diffraction) patterns along the  $c$ -axis. The observation of extra diffraction points among the main diffraction points in the SAED pattern confirms the  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order in the as-grown Fe<sub>4</sub>Se<sub>5</sub> nanocrystal [35]. After RTA treatment at 450°C, the superstructure peaks observed



in XRD and TEM-SAED due to the  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order disappear, as shown in **Figure 1B** and its inset. The refinement results, using the same *P4* symmetry, reveal that the occupations of Fe at vacancy 4d sites and the originally occupied 16i sites are almost the same, indicating that the Fe vacancies become

disordered after RTA treatment. It is noted that the XRD patterns of the RTA treatment sample are nearly ten times smaller in the intensity response. This is due to the different synchrotron beamlines we used for our measurements. However, the difference does not affect the refined results.



It is noted that the  $\text{FeSe}_4$  tetrahedron in as-grown  $\text{Fe}_4\text{Se}_5$  is highly distorted due to the existence of the Fe-vacancy order. As the Fe vacancies disordered by the RTA treatment, the  $\text{FeSe}_4$  tetrahedron becomes more symmetric. The refined structure parameters are tabulated in **Supplementary Tables S1 and S2**. The EDS analysis confirms that the stoichiometries of samples keeps  $\text{Fe}_4\text{Se}_5$  with Fe/Se ratio of 44.5 : 55.5 and 45.3 : 54.7 before and after the RTA treatment, respectively, as shown in **Supplementary Figure S2** in the supplementary information.

## Temperature Dependence of the Resistance and the Magnetic Susceptibility Measurement of $\text{Fe}_4\text{Se}_5$

**Figure 2A** shows the temperature dependence of the resistance for the as-grown polycrystalline pellet sample made of  $\text{Fe}_4\text{Se}_5$  nanosheets. The inset is the magnetic susceptibility of the as-grown sample from 300 to 2 K. The R-T results of  $\text{Fe}_4\text{Se}_5$  exhibit a metal-insulator transition with the sharp rise in resistance at  $\sim 45$  K. The data can be well-fitted to the three-dimensional Mott variable-range-hopping model (3D-MVH):  $\rho(T) = \rho_0 \exp(T_0/T)^\nu$ , where  $T_0$  is the variable-range-hopping characteristic temperature, and the exponent  $\nu$  is  $1/(d+1)$  with  $d = 3$  (the fitting results are shown in **Supplementary Figure S2**. A transition temperature  $T_V$  is marked as the onset temperature of 3D-MVH behavior). The variable-range-hopping characteristic temperature  $T_0$  calculated is  $\sim 1,400$  K for the as-grown sample. The magnetic susceptibility of the same sample shows paramagnetic behavior as the sample cools down from 300 K, and a sudden drop in susceptibility appears at about the same temperature as the resistance transition temperature ( $T_V$ )  $\sim 45$  K. The sharp resistive rise and the diamagnetic drop are the two signatures for the Verwey transition observed in  $\text{Fe}_3\text{O}_4$ , which occurs at 125 K. These results are also in line with those reported results in the  $\text{Fe}_4\text{Se}_5$  nanowire, which was

recently demonstrated to exhibit the Verwey-like electronic correlation [45].

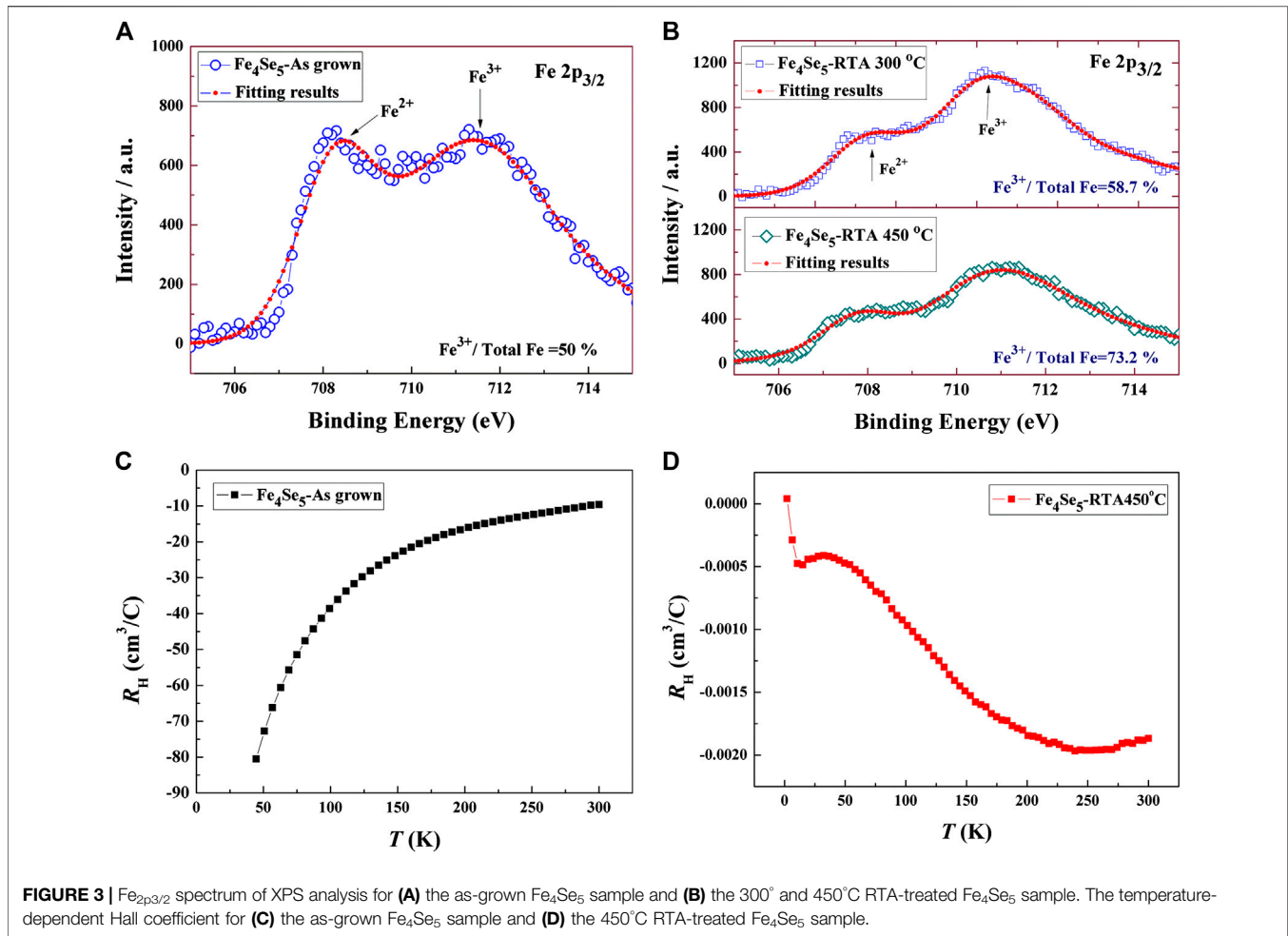
**Figure 2B** shows the temperature-dependence resistance for the  $\text{Fe}_4\text{Se}_5$  samples after  $300^\circ$  and  $450^\circ\text{C}$  RTA process. The sample after  $300^\circ\text{C}$  remains to behave like semiconductor. The sample treated at  $450^\circ$  and  $300^\circ\text{C}$  changes to metallic and becomes superconducting below  $\sim 5$  K with the onset superconductive critical temperature ( $T_c$ )  $\sim 7.8$  K, as evidenced in the upper-left inset of **Figure 2B**. The lower-right inset is the magnetic susceptibility, which further demonstrates the superconducting transition with onset  $T_c \sim 7.9$  K. The evolution to superconductivity in this RTA-treated sample is similar to that reported in the  $\text{K}_{2-x}\text{Fe}_{4+y}\text{Se}_5$  system, where superconductivity appears after Fe vacancies becoming disordered through high-temperature annealing and rapid quenching processes [43, 44].

## XPS and Hall Measurement of $\text{Fe}_4\text{Se}_5$

In order to gain more insight into the observed Verwey-like electronic correlation, XPS at room temperature and temperature-dependent Hall measurements on the samples were performed. **Figure 3A** is the observed XPS results for the as-grown  $\text{Fe}_4\text{Se}_5$  sample. **Figure 3A** is the observed XPS results for the as-grown  $\text{Fe}_4\text{Se}_5$  sample. The XPS spectrum clearly reveals two peaks showing the existence of mixed-valence states of Fe. The observed two peaks, at 708.5 and 711.5 eV, can be associated with the  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$  states, respectively. The best data fitting gives the ratio between  $\text{Fe}^{2+}$  and  $\text{Fe}^{3+}$  close to 1 : 1. This result is similar to that observed in the magnetite  $\text{Fe}_3\text{O}_4$ .

**Figure 3B** displays the XPS results for samples with RTA treated at  $300^\circ$  and  $450^\circ\text{C}$ . After the RTA treatment, the  $\text{Fe}^{3+}$  state becomes dominant. The extracted  $\text{Fe}^{3+}$  ion to total Fe atoms ratio is 58.7% for  $300^\circ\text{C}$  RTA-treated and 73.2% for  $450^\circ\text{C}$  RTA-treated samples, respectively, indicating a substantial increase in electron carriers in these samples. It should be noted that the sample after  $300^\circ\text{C}$  still exhibits temperature-dependent behavior like





**FIGURE 3** |  $Fe_{2p_{3/2}}$  spectrum of XPS analysis for (A) the as-grown  $Fe_4Se_5$  sample and (B) the 300° and 450°C RTA-treated  $Fe_4Se_5$  sample. The temperature-dependent Hall coefficient for (C) the as-grown  $Fe_4Se_5$  sample and (D) the 450°C RTA-treated  $Fe_4Se_5$  sample.

semiconductor. There is no specific difference of Se 3d peak at 54.7 eV before and after the RTA process of the  $Fe_4Se_5$  sample according to the XPS result, as shown in **Supplementary Figure S3**.

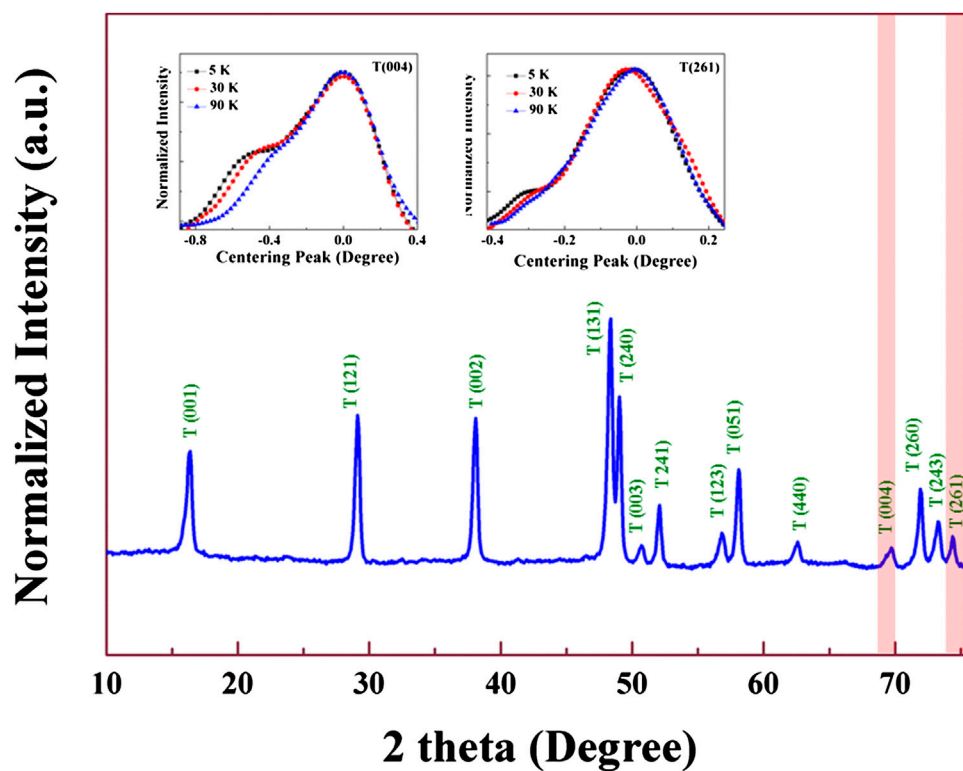
It is known that tetragonal FeSe is a metal with two-band based on the first-principles electronic structure calculation, for example, by T. Xiang et al., [46]. T. Xiang et al. also reported that the electronic structure of  $Fe_4Se_5$  with  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy order is a pair checkboard antiferromagnetic insulator. The calculation shows that the Fe-vacancy-ordered  $Fe_4Se_5$  has a single-band structure with n-type carrier dominated and a bandgap  $\sim 290$  meV. Berlijn et al. [47] investigated the effect of disordered Fe vacancies on the normal-state electronic structure of the alkali-intercalated FeSe system, where the  $KFe_4Se_5$  exhibits exactly the same Fe-vacancy order as that in  $Fe_4Se_5$ . They found that the disorder of Fe-vacancy can effectively raise the chemical potential giving enlarged electron pockets without adding carriers to the system.

It is noted that, as reported by Chen et al. [38], there exists a series of  $Fe_xSe_y$  compounds with  $x/y = 1/2, 2/3, 3/4, 4/5$ , etc. We have carried out a systematic study using the coprecipitation method to successfully prepare tetragonal  $Fe_{(1-x)}Se$  with stoichiometry of  $Fe_3Se_4$  and  $Fe_4Se_5$ . Based on the XPS results, the observed  $Fe^{3+}/Fe^{2+}$  ratio is

2 and 1 for tetragonal  $Fe_3Se_4$  and  $Fe_4Se_5$ , respectively, as shown in **Supplementary Figure S4A** and **Figure 3A**. These data imply that  $Fe_3Se_4$  would be hole-doped and  $Fe_4Se_5$  be electron-doped if there are additional carriers based on the simple charge balance picture by considering  $Fe_3Se_4$  to be the combination of  $Fe^{2+}Se$  and  $Fe_2^{3+}Se_3$ , whereas  $Fe_4Se_5$  is from  $2(Fe^{2+}Se)$  and  $Fe_2^{3+}Se_3$ . Indeed, our Hall measurement results show at 300 K a hole concentration of  $1.20 \times 10^{19}/cm^3$  for  $Fe_3Se_4$  (**Supplementary Figure S4B**) and electron concentration of  $-6.52 \times 10^{17}/cm^3$  for  $Fe_4Se_5$ .

Both of the as-grown and RTA-treated  $Fe_4Se_5$  show a single-band behavior with n-type carrier from the Hall resistivity measurements, as shown in **Supplementary Figure S5**. The Hall coefficient of the as-grown sample at room temperature is  $-9.59$   $cm^3/C$ , corresponding to the electron carrier concentration of  $6.52 \times 10^{17} cm^{-3}$ , and the carrier concentration decreases by about a factor of 8 at the transition temperature  $T_V$ , as shown in **Figure 3C**.

After the  $Fe_4Se_5$  sample is RTA-treated at 450 °C, the  $Fe^{3+}/Fe^{2+}$  ratio becomes close to 3 : 1, which means that a large number of electrons are introduced and subsequently induced superconductivity. Indeed, the Hall measurement results, as shown in **Figure 3D**, show that the carrier concentration at 300 K increases to  $-3.34 \times 10^{21}/cm^3$  (Hall coefficient  $-1.87 \times 10^{-3} cm^3/C$ ) for 450 C RTA-treated sample. The electron carrier



**FIGURE 4** | The neutron diffraction patterns of as-grown  $\text{Fe}_4\text{Se}_5$  nanosheet at 300 K. The tetragonal  $P4$  symmetry is identified. The insets are the diffraction peaks of (004) and (261) at low temperatures showing the growth of additional peaks, indicating a structural change at low temperature.

concentration is about four orders of magnitude increase comparing with the as-grown  $\text{Fe}_4\text{Se}_5$ . Obviously, the RTA treatment disrupts the Fe-vacancy long-range order and leads to the increase of electron carriers.

### Neutron Diffraction of $\text{Fe}_4\text{Se}_5$

It is well known that the Verwey transition in magnetite exhibits a structural transition accompanied by the sharp resistive and magnetic susceptibility changes. To examine whether such a structural change exists for the as-grown  $\text{Fe}_4\text{Se}_5$ , we have carried out the neutron diffraction at low temperatures.

The detailed structural information of the as-grown  $\text{Fe}_4\text{Se}_5$  sample measured by neutron diffraction at different temperatures is shown in **Figure 4**. At room temperature, the neutron data, consistent with XRD results, fit well with the  $P4$ -tetragonal symmetry. At low temperatures, a distortion appears at temperatures below 30 K. The data at 5 K, with evident peak emergence shown in the inset of **Figure 4**, indicates a possible structural change. This result further supports that the as-grown  $\text{Fe}_4\text{Se}_5$  nanosheets, similar to the results observed in  $\text{Fe}_4\text{Se}_5$  nanowire, show the Verwey-like correlation. The Verwey-like transition temperature of  $\sim 45$  K in nanosheets is higher than that observed in the nanowire, which was found to be  $\sim 30$  K. This shows the size dependence of  $T_V$ , which is also noticed in Verwey transition [48–50]. Currently, we are waiting for the results of the detailed high-resolution XRD at low temperatures using a synchrotron source to determine exactly the low-temperature phase and the transition temperature.

## CONCLUSION

We carried out a detailed study to investigate whether there exists an insulating parent phase for FeSe superconductors. Our studies unambiguously show that 1) the  $\sqrt{5} \times \sqrt{5}$  Fe-vacancy-ordered  $\text{Fe}_4\text{Se}_5$  is a Mott insulator with Verwey-like transition at low temperature; 2)  $\text{Fe}_4\text{Se}_5$  is the parent compound of the FeSe superconductors. The application of the RTA process at  $450^\circ\text{C}$  disrupts Fe-vacancy order and induces more electron carriers by increasing the  $\text{Fe}^{3+}$  valence state. Superconductivity emerges with  $T_c \sim 8$  K without changing the chemical stoichiometry of the sample after the RTA process. Consistent with the observations in  $\text{K}_2\text{Fe}_{4+x}\text{Se}_5$ , superconductivity is directly related to the disappearance of Fe-vacancy long-range order. In the  $\text{Fe}_4\text{Se}_5$  case, no extra Fe doping is required as the random occupation of Fe atom in the vacancy sites, resulting in the addition of extra carriers in favor of superconductivity. More detailed evolution of superconductivity by varying the RTA temperature and time is currently underway in order to gain more insight into the exact phase diagram of the FeSe superconductors.

## DATA AVAILABILITY STATEMENT

The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation.

## AUTHOR CONTRIBUTIONS

MJW and MKW designed research. KYY, TSL, and YRC performed research. MJW and KSCL contributed new reagents/analytic tools. KYY, TSL, PMW, YRC, KSCL, MJW, and MKW analyzed data and took part in physics discussions. KYY, MJW, TSL, PMW, and MKW wrote the paper.

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## SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fphy.2020.567054/full#supplementary-material>

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**Conflict of Interest:** Author PMW is employed by the company BitSmart LLC.

The remaining authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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