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# Effect of solutions acidity on Haacke's Figure of Merit of ZnO and ZnO:F thin films deposited by ultrasonic spray pyrolysis

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Fluorine-doped zinc oxide (ZnO:F) thin films are valued for their potential as transparent conductive materials, particularly in optoelectronic applications. In this work, the deposition of highly conductive and transparent fluorine-doped ZnO thin films, deposited by chemical spray on glass substrates is reported. The effect of acetic (AcAc) in the initial fresh solution on Haacke's Figure of Merit ( $\Phi$ ) of both zinc oxide (ZnO) and fluorine-doped zinc oxide (ZnO:F) thin films was studied. The substrate temperature was fixed to 450°C, and two deposition times (8 and 14 min) were tested. Accordingly, the optical and transport properties, as well as the structural and morphological characteristics of the films were measured. The results indicate that the samples are polycrystalline in all cases and exhibit a wurtzite-type structure of ZnO. The variation of AcAc in the starting solution causes a switch in preferential growth from (002) to (001). As the AcAc content increases, the surface morphology of the films reveals the formation of well-defined hexagonal grains, and the grain size increases. Conversely, the transmittance decreases as the acetic acid increases, which can attributed to carbon incorporation into the film. In addition, the band gap values of the films varied from 3.2 to 3.4 eV. Also, as the acetic acid concentration increased, a drop in the electrical resistivity of ZnO thin films on the order of 0.016  $\Omega$  cm was found for the ZnO:F films deposited with fresh solution for 14 min. This can be ascribed to a dense acetic cloud formed during the synthesis process, trapping volatile F species that incorporate into the ZnO lattice. This enhances Haacke's Figure of Merit of ZnO:F films deposited with a fresh solution, demonstrating their potential use for application as transparent conductive films. This contrasts with other synthesis methods that require longer aging time in the precursor solution, making these findings useful for large-scale and more efficient production of transparent conductive films.

#### KEYWORDS

zinc oxide films, fluorine-doped zinc oxide films, ultrasonic spray pyrolysis, transparent conductive films, Haacke's Figure of Merit

# **1** Introduction

Transparent thin films with low resistivity manufactured from semiconductor oxides have attracted the interest of the scientific community due to their promising applications for electronic and optoelectronic devices. Among the relevant characteristics of transparent conductive films (TCOs) are low electrical resistance and high optical transmittance, which can be evaluated by Haacke's Figure of Merit ( $\Phi_{TC}$ ) that correlates transparency and sheet resistance of the thin films (Haacke, 1976).

Haacke's Figure of Merit is an indispensable indicator to evaluate transparent conductors (Veron et al., 2022) in typical applications such as transparent electrodes (Arora et al., 2022), transparent sensors (Gu et al., 2023), touch screens (Wu, 2020), photovoltaics (García-Mejía et al., 2024), and smart windows (DiGregorio et al., 2022). The traditional figure of merit (FoM) allows the evaluation of the quality and the efficiency of different transparent conductive materials to be compared, considering both electrical conductivity and optical transparency. On the other hand, Haacke's (FoM) can be applied to optimize materials in the manufacturing of transparent conductors, ensuring maximum light transmission while maintaining high conductivity. (Dhar and Alford, 2012; Hubarevich et al., 2019).

In the literature, indium tin oxide (ITO) films have been extensively researched and commercialized due to their low electrical resistivity of  $10^{-4} \Omega$  cm and high optical transmittance of 90% in the visible light range. However, the supply of ITO is constrained by the limited availability and high cost of indium, presenting challenges for the expanding market for TCO-dependent electronic devices. Therefore, alternatives to ITO are under study, ZnO-based TCOs promise to replace common TCOs demanded by the industry, for instance (Mallick and Basak, 2018). Zinc oxide (ZnO) is a costeffective, widely available, and environmentally benign material. It typically exhibits a hexagonal wurtzite structure with a direct bandgap around of 3.4 eV. This intrinsic bandgap allows ZnO to effectively absorb UV light while remaining transparent to visible light, which is crucial for its use in various optoelectronic applications. Doping ZnO with elements from group III of the periodic table, such as aluminum or gallium, significantly improves its electrical conductivity and other transport properties, making it suitable for use as a transparent conductive oxide (TCO) in numerous electronic and photovoltaic applications (Xin, 2021; Vallejo et al., 2020).

Anionic doping with fluorine or chlorine has demonstrated considerable advantages. When ZnO is doped with fluorine, both the electrical and optical properties of the material are enhanced. Fluorine doping introduces extra electrons into the ZnO lattice, which increases the electron concentration, thereby improving its conductivity (Jiamprasertboon et al., 2019; de la L Olvera et al., 2002). According to Hu and Gordon, (1991) the fact that fluorine and oxygen have close ionic radii (for  $O^{2-}$  is 0.138 nm and for  $F^{1-}$  is 0.131 nm) facilitates that F ions incorporate easily in the ZnO lattice by replacing O atoms which result in a reduction in the electrical resistivity of ZnO films by increasing the level of free electrons in the ZnO structure.

When it comes to cost-effective chemical deposition methods, careful attention is essential to produce high-quality

ZnO:F films. Recent reports have highlighted improvements in ZnO films achieved through various manufacturing techniques (Hurma and Caglar, 2020). The ultrasonic spray pyrolysis (USP) technique presents different advantages over others that are more sophisticated, whether physical or chemical; among the most evident can be its simple setup and consequently low cost, ease of operation, and mainly provides the possibility of deposition of high-quality films in large areas. However, as with all deposition techniques, it also presents certain disadvantages, among which is the lack of thickness control, which depends largely on the configuration of the system and the external environmental conditions. Fluorine, which is highly volatile during synthesis, can be effectively incorporated by using high doping levels in the solution, up to 60 atomic percentage (at%). The composition of this solution is crucial for determining the properties of the resulting films. Specifically, the pH level, influenced by the presence of acetic acid (AcAc), has been shown to improve both the conductivity and optical properties of ZnO:F films (Kang and Park, 2018; Edinger et al., 2015). Research by Edinger et al. (2015) highlighted the positive impact of AcAc concentration in the precursor solution on the electrical characteristics of ZnO thin films produced via USP. Additionally, Jiao et al. (2011) reported that incorporating acetic acid during the USP synthesis of ZnO and ZnO:In films also enhanced their transport properties.

The production of high-quality ZnO:F thin films using chemical spray techniques faces a challenge due to the necessity of aging the starting solution for several weeks. Despite this limitation, the advantages of the spray method justify efforts to address the aging issue (Workie et al., 2023; Habas et al., 2010). The solution to this problem depends on the composition of the starting solution. Acetic acid facilitates the formation of complex species, which help to integrate fluorine ions into the ZnO lattice during the synthesis, as was previously reported for ZnO:F films made from aged solutions (Rodríguez-Báez et al., 2007). However, there has been no reported research on the impact of acetic acid in films produced with fresh solutions.

This work reports the beneficial effect of acetic acid concentration in the starting solution on Haacke's Figure of Merit of ZnO and ZnO:F thin films deposited with a fresh solution. Additionally, the structural, electrical, and optical properties, as well as Haacke's Figure of Merit of fluorine-doped ZnO and ZnO:F thin films, were studied.

# 2 Experimental

# 2.1 Synthesis of ZnO and ZnO:F films

ZnO and fluorine-doped ZnO (ZnO:F) films were deposited on glass substrates by the USP technique. A 0.2 M zinc acetate dihydrate [Zn (CH<sub>3</sub>COO)<sub>2</sub>.2H<sub>2</sub>O] (Merck, purity of 99%) solution was prepared with a mixture of deionized water, methanol, and acetic acid at different volume proportions. The water content was fixed to 100 mL/L according to Chávez-Vargas et al. (2018). Table 1 shows the volume proportions of acetic acid, deionized water, and methanol. A 0.6 M aqueous solution of Ammonium fluoride  $[NH_4F]$  (Merck,

ID Sample	Desposition time (min)	AcAc (mL)	Water (mL)	Methanol (mL)
Zn0_20_8	8	20	100	880
Zn0_100_8		100	100	800
Zn0_200_8		200	100	700
Zn0_20_14	14	20	100	880
Zn0_100_14		100	100	800
Zn0_200_14		200	100	700
Zn0:F_20_8	8	20	100	880
Zn0:F_100_8		100	100	800
Zn0:F_200_8		200	100	700
Zn0:F_20_14	14	20	100	880
Zn0:F_100_14		100	100	800
Zn0:F_200_14		200	100	700

TABLE 1 Samples used and the proportions by volume of acetic acid:water:methanol for the preparation of the different starting solutions.



XRD spectra of the undoped ZnO thin films with different acetic acid quantity added in the precursor solution at different deposit times (8 and 14-min).

purity of 99%) was prepared (a source of fluorine). Then, this solution was added to the starting zinc acetate solution for a [F]/[F + Zn] ratio of 30 at%. The substrate temperature was also

fixed to 450°C (Sanchez-Juarez et al., 1998). The deposition temperature is selected because it allows for optimal material synthesis, reducing resistivity by ensuring proper crystallization



TABLE 2 Crystallite sizes of plane (002) of ZnO and ZnO:F films according to Equation 1.

Sample	2θ (°)	FWHM	Crystallite size (nm)
ZnO_20_8	34.58	0.0041	35
ZnO_100_8	34.68	0.0041	35
ZnO_200_8	34.42	0.0034	42
ZnO_20_14	34.54	0.0037	39
ZnO_100_14	34.402	0.0034	42
ZnO_200_14	34.718	0.0054	26
ZnO:F_20_8	35.26	0.0109	13
ZnO:F_100_8	35.11	0.0030	47
ZnO:F_200_8	33.79	0.0109	13
ZnO:F_20_14	37.71	0.0020	71
ZnO:F_100_14	35.25	0.0044	33
ZnO:F_200_14	35.160	0.0027	53

and homogeneous incorporation of the fluorine dopant into the ZnO lattice. Lower temperatures lead to poor conductivity, while higher temperatures risk dopant loss and morphological changes, according to (Maldonado et al., 2005), for deposition times of 8 and 14 min are chosen to optimize film thickness (Perednis and Gauckler, 2005), uniformity and dopant incorporation. An 8 min deposition yields a thin, uniform layer with high optical transparency, while 14 min increases thickness and enhances electrical conductivity. The solution flow was fixed to 1 mL/min for all depositions. In addition, the experimental setup of the USP technique used in this work can be seen in Supplementary Figure S1.

# 2.2 Characterization of morphology, optical, and electrical properties

X-ray diffraction (XRD) analysis of ZnO films was performed on PANa-lytical Xpert Pro equipment, Cu-K $\alpha$  ( $\lambda$  = 0.154056 nm), the technique  $\theta$  –  $2\theta$  was used to analyze the crystallographic properties of the films in conventional Bragg–Brentano mode (Xian et al., 2024). In addition, the crystallite sizes were estimated from the (002) plane using the Debye–Scherrer equation:

$$D = \frac{0.9\lambda}{\beta \cos\theta} \tag{1}$$

where  $\lambda$  is the wavelength of the X-ray used,  $\beta$  is the broadening of the preferred diffraction peak at half of its maximum intensity (Full Width at Half Maximum or FWHM), and  $\theta$  is the Bragg's angle.





The optical transmittance of these films was evaluated by using a UV-visible spectrometer Jasco, Model V-670. Spectra were recorded from 300 to 900 nm and using a scanning rate of 60 nm per minute. The optical band gap,  $E_g$ , was determined using Tauc's plot equation defined by:

Finally, the optical and electrical properties measured were used to calculate Haacke's Figure of Merit ( $\Phi_{TC}$ ) as follows (Haacke, 1976):

$$\Phi_{TC} = \frac{T^{10}}{R_s} \tag{3}$$

$$\alpha h \nu = \left( h \nu - E_g \right)^2 \tag{2}$$

where  $\alpha$  is the absorption coefficient and  $h\nu$  is the photon energy.

On the other hand, the surface morphology of thin films was observed using a scanning electron microscope FE-HRSEM Auriga 3916. ImageJ software was used to measure the grain size of each sample.

Furthermore, for the electrical properties of the films, the resistivity,  $\sigma$ , was acquired by Hall effect measurements using the van der Pauw method (Magnetic field of 5000 Gauss, Keithley brand multimeters, and current source).

where *T* is the transmittance at 550 nm and  $R_s$  is the sheet resistance measured by Hall effect.

# **3** Results

### 3.1 Structural properties

Figures 1, 2 present the X-ray diffraction (XRD) patterns for ZnO and ZnO:F films, respectively. All films exhibited a polycrystalline nature with the hexagonal wurtzite structure characteristic of ZnO, as identified

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TABLE 3 Optical properties of ZnO and ZnO:F films with AcAc variation according to Equation 2.

Sample	Transmittance at 550 nm (%)	E <sub>g</sub> (eV)
ZnO_20_8	93	3.25
ZnO_100_8	65	3.31
ZnO_200_8	75	3.34
ZnO_20_14	94	3.24
ZnO_100_14	56	3.26
ZnO_200_14	73	3.29
ZnO:F_20_8	90	3.21
ZnO:F_100_8	87	3.32
ZnO:F_200_8	84	3.40
ZnO:F_20_14	79	3.27
ZnO:F_100_14	80	3.33
ZnO:F_200_14	77	3.20

by JCPDS card number 01-075-0576. The (002) plane showed preferential growth in the undoped ZnO samples, consistent with findings from other studies (Lee and Lin, 1996). In contrast, ZnO:F films produced from a fresh solution (Figure 2) showed contributions from the (002), (100), and (101) planes. This phenomenon occurs because incorporating fluorine (F) into ZnO alters its crystal structure. When fluorine atoms replace oxygen atoms within the lattice, it affects the formation energy and stability of the crystallographic planes. As a result, there is less growth preference along the (002) plane, which instead promotes increased growth in other directions like (100) and (101). This finding aligns with what was reported by (Anandhi et al., 2012; Kim et al., 2006).

On the other hand, Edinger et al. (2015) demonstrated that the concentration of acetic acid in the initial solution influences the structural, optical, and, notably, transport properties of chemically sprayed undoped ZnO thin films. Jiao et al. (2011) also reported that the concentration of acetic acid affects sprayed ZnO films, causing a shift in the preferred crystalline planes, changes in morphology from flake to triangle, and enhanced electrical transport. Smith, (2000) suggests that these changes in sprayed ZnO films occur because the precursors do not easily integrate into the growing film, resulting in a deposit primarily composed of flat equilibrium faces. In contrast, small and neutral complex molecules are more likely to form non-equilibrium faces. Finally, these non-equilibrium faces, which are polar, can be stabilized during film growth if the surrounding atmosphere contains polar gases that can adsorb onto these faces and slow down their growth (Smith, 2000; Smith and Rodriguez-Clemente, 1999). In our study, the observed change in the preferred crystal growth planes in ZnO films is likely due to the varying concentrations of acetic acid during the synthesis process.

Table 2 shows the crystallite sizes calculated by using Equation 1, as well as the FWHM and  $2\theta$  values obtained from all samples. The effect of increasing acetic acid quantity in the starting solution increases the crystallite size, *t*, for undoped ZnO films, as is shown in Table 2. In the case of ZnO:F films, the variation in crystallite size with AcAc concentration was almost the same. It can be observed that increasing the deposition time also increases the *t*-value.

### 3.2 Optical properties

Figures 3, 4 show the optical transmittance spectra of synthesized thin films of ZnO and ZnO:F, respectively. In the case of ZnO films (Figure 3) it can be observed that the transmittance of the films decreases as the acetic acid content increases. This is attributed to the carbon content in the film for the AcAc excess due to the selective adsorption of acetic acid on specific surface facets of ZnO, such as (001) planes reported previously (Smith and Rodriguez-Clemente, 1999; Hagendorfer et al., 2014). In addition, as the acetic acid increases in the starting solution, the free carrier absorption into the ZnO also increases, as is shown in the electrical properties below, which in turn means that the corresponding transmittance decreases. Namely, it negatively affects the optical properties. The average transmittance at 550 nm for ZnO films ranged between 60% and 90%, as was expected. On the other hand, in the thin films of ZnO:F (Figure 4), the transmittance at 550 nm varied between 77% and 98%, which is an appropriate value for transparent conductive electrodes (Edwards et al., 2004; Elsokary et al., 2024; Hwang and Lee, 2024). It is worth mentioning that at 310-320 nm, the absorption values significantly increased, indicating the fundamental absorption region of ZnO (Elmas et al., 2019).

Table 3 shows the transmittance at 550 nm and the bandgap calculated by Equation 2 of each sample. In general, the bandgap of ZnO and ZnO:F films increases with the rising content of acetic acid. For ZnO films, the highest value is 3.34 eV, while for ZnO:F films, the bandgap decreases with higher acetic acid content, reaching the lowest value of 3.20 eV.

## 3.3 Surface morphology

Figures 5, 6 show SEM micrographs of thin films of ZnO and ZnO:F, respectively. The films exhibited the following characteristics: uniformity, good adhesion, absence of cracks, and absent pores. Furthermore, Table 4 shows the average grain size measured by ImageJ software for doped and undoped films. The acidity and fluoride content significantly influence the morphology of the films in terms of shape and size distribution. In fact, for undoped ZnO films, only hexagonal-shaped grains with a narrow size distribution were observed (Figure 5). In addition, in Figure 6 it can be observed the ZnO:F films morphology, it was found that as the acetic acid content increases, the grain size also increases. This effect has been reported by Jiao et al. (2011). According to (Chen et al., 2015), the incorporation of acetic acid caused the grains to grow and eventually develop into pyramid shapes, enhancing both surface roughness and the ability to scatter light. Larger quantities of acetic acid resulted in the formation of well-defined pyramidal grains with six uniform side facets, leading to highly rough surfaces with significant light-trapping capabilities. Edinger et al. (2015) demonstrate that adjusting the acetic acid concentration in the precursor solution can effectively control the morphological characteristics of ZnO thin films produced by spray pyrolysis, which is vital for enhancing their performance in applications such as photovoltaics and transparent conductive oxides.

The presence of AcAc in the precursor solution is necessary to avoid zinc hydroxide compounds and favor zinc acetate complexes,



FIGURE 5

SEM images of undoped ZnO with different acetic acid added in the precursor solution. (A) ZnO\_20\_8, (B) ZnO\_20\_14, (C) ZnO\_100\_8, (D) ZnO\_ 100\_14, (E) ZnO\_200\_8, (F) ZnO\_200\_14.

as has been shown by Caillaud et al. (1993), who also reported significant growth rates around 2  $\mu$ m per hour.

## 3.4 Electrical properties

The resistivity and effective mobility of ZnO and ZnO:F thin films as a function of acetic acid content in the starting solution are presented in Table 5. Figures 7, 8 show the resistivity vs. mobility graphs of both films. It can be observed that the electrical properties of ZnO:F films improved as the acetic acid content increased. It is worth mentioning that the resistivity, carrier concentration, and mobility for the lowest acetic acid content are not shown in Table 5 because they were out of range during the measurements.

On the other hand, both ZnO and ZnO:F films deposited from solutions with a low acetic acid content (20 mL/L) showed a high sheet resistance with the lowest transmittance in the UV-vis region. In addition, it was shown that the resistivity of ZnO and ZnO:F thin

films decreased as the acetic acid content in the starting solution increased. The lowest value found was 0.016  $\Omega$ · cm for the film deposited at 450°C and 14 min. In contrast to usual reports that add a few drops of AcAc to the starting solution to avoid early precipitation (Yoon and Cho, 2000), in this study, an excess of acetic acid showed a beneficial effect on the transport properties of ZnO and ZnO:F thin films.

# 3.5 Figure of merit

The Haacke's Figure of Merit for each sample was determined using Equation 3. Figures 9, 10 illustrate the variation of  $\Phi_{TC}$  for ZnO and ZnO:F thin films as a function of the AcAc content in the initial solution. It can be observed that the  $\Phi_{TC}$  for ZnO:F films with a 14-min deposition is  $3.2 \times 10^{-5} \Omega^{-1}$ , reaching a maximum effective carrier concentration of  $n = 3.131 \times 10^{18}$  cm<sup>-3</sup> at a doping concentration [F]/[F + Zn] of 30 at%. Compared to what exists



FIGURE 6

SEM images of ZnO:F with different acetic acid added in the precursor solution. (A) ZnO:F\_20\_8, (B) ZnO:F\_20\_14, (C) ZnO:F\_100\_8, (D) ZnO:F\_ 100\_14, (E) ZnO:F\_200\_8, (F) ZnO:F\_200\_14.

in the literature, Zhang et al. (2023) have deposited F and Al-doped ZnO thin films by sol-gel, requiring an aging time for the deposition of the films. The best results reported for carrier transport were on the order of  $1x10^{-3}\Omega cm$ , after annealing treatment at 500°C in  $N_2$  atmosphere. Altamirano-Juárez et al. (2004) reported F-doped ZnO films deposited also by sol-gel, and the best results were  $10^{-3}\Omega cm$  for two at% and after being annealed in 96:4  $N_2/H_2$ . It is worth mentioning that the spray pyrolysis technique, when removing the aging time of the starting solution, is an alternative for the deposition of F doped with low resistivity with fresh solution and no extra annealing step required.

# 4 Discussion

The primary objective of this work was the manufacture of high-quality TCOs based on semiconductor oxides. The mechanisms developed to achieve high conductivity and transmittance in the visible spectrum begin with two types of deposition techniques: the physical methods involvs high vacuum processing (Chen et al., 2021) and economical chemical methods suitable for large-scale production (Chen et al., 2015). Regarding chemical techniques, which often struggle to attain high Figure of Merit values, the strategy focuses on a comprehensive study of the key variables that influence transport properties. Each chemical method has specific variables that impact the characteristics of the films (Maldonado et al., 2005). For instance, in chemical spray processes, reports highlight the importance of the starting solution composition in ZnO thin films. Given the lack of studies on F-doped ZnO and the role of acetic acid, this work proposes a thorough investigation to explore the effects of both low and excess AcAc concentrations in the starting solution. Hence, this research has demonstrated the positive effect of acetic acid in enhancing transport properties, which subsequently improves the Figure of Merit.

TABLE 4 Average grain size of the ZnO and ZnO:F films deposited with different concentrations of acetic acid and with deposition times of 8 and 14-min.

Sample	Grain size (nm)
ZnO_20_8	173.22 ± 33.39
ZnO_20_14	181.71 ± 32.19
ZnO_100_8	258.0 ± 70.5
ZnO_100_14	379.5 ± 85.9
ZnO_200_8	214.5 ± 57.0
ZnO_200_14	-
ZnO:F_20_8	$104.4 \pm 19.0$
ZnO:F_20_14	$149.5 \pm 32.8$
ZnO:F_100_8	192.2 ± 38.8
ZnO:F_100_14	244.0 ± 52.6
ZnO:F_200_8	165.6 ± 38.0
ZnO:F_200_14	428.4 ± 121.4

TABLE 5 Electrical properties of ZnO and ZnO:F films.

Sample	Resistivity (Ω· cm)	Carrier concentration $(cm^{-3})$	$\mu_{\mathrm{H}} \ (rac{\mathrm{cm}^2}{\mathrm{Vs}})$
ZnO_20_8	—	—	_
ZnO_100_8	0.23	$-2.87 \mathrm{x10}^{19}$	0.94
ZnO_200_8	0.34	$-3.47 \mathrm{x10}^{19}$	0.52
ZnO_20_14	_	_	_
ZnO_100_14	0.33	$-5.05 \mathrm{x10}^{19}$	0.36
ZnO_200_14	0.27	-2.43x10 <sup>19</sup>	0.97
ZnO:F_20_8	_	_	_
ZnO:F_100_8	0.14	-3.43x10 <sup>19</sup>	1.26
ZnO:F_200_8	0.26	$-3.55 \mathrm{x10}^{19}$	0.61
ZnO:F_20_14	_	—	_
ZnO:F_100_14	0.05	$-8,74 \times 10^{19}$	1.29
ZnO:F_200_14	0.016	$-8.84 \mathrm{x10^{19}}$	4.40

The AcAc addition in the starting solution enhanced zinc acetate precursor dissolution, and it avoided a milky finish that decreased the transmittance of ZnO films. A complete study of chemical species in solution and the effect of inorganic acids and AcAc content in solution on growth rates has been reported in the case of the deposition of ZnO by pyrosol technique (Caillaud et al., 1993). It has been shown that a pH solution in the range of four to five leads to an optimum growth rate. According to Caillaud et al. (1993) from zinc acetate dissolved in water, the acid variation gives rise to the following chemical species:  $Zn^{2+}$ ,  $Zn(Ac)^+$  and  $Zn(Ac)_2$ .

Under normal conditions, it has been established that for pH in the range of 3.5–4.3, where the acetate complexes are the major species, the ZnO growth rate reaches maximum values. The ZnO growth rate

decreases for pH values less than 3.6. Therefore, the zinc acetate complex  $Zn(Ac)_2$  is the required precursor for film formation.

Simultaneously, the decomposition of ammonium fluoride occurs according to the Equation 4 (Agashe and Major, 1996):

$$NH_4F \to NH_3\uparrow + HF\uparrow$$
 (4)

The decomposition temperature of ammonium fluoride is below 100°C. Therefore, it occurs almost instantaneously and forms an HF cloud in the vicinity of the growing ZnO:F film.

The boiling point of AcAc, methanol, and water are 117.9°C, 65°C, and 100°C, respectively. During the synthesis process, a rise in the temperature of the traveling drop occurs abruptly, from room temperature to 450°C (Sears and Gee, 1988). A decrease in the volume of the drops occurs due to the quick boiling of solvents, which can cause instantaneous precipitation of zinc hydroxides, diffusion on the substrates, and rapid ZnO formation with characteristics determined from solution composition.

The excess AcAc in the starting solution generates an acid cloud during the pyrolytic decomposition that drags down on the hot substrate surface due to the nitrogen flow. The high reactivity of F can form a  $F^-$  (organic chains) complex that increases the F ions trapped in the ZnO lattice during the synthesis process. As the AcAc increases, the transmittance of ZnO films decreases as a result of carbon incorporation. The latter is attributed to the fact that free carrier absorption has more effect at high carrier concentrations or due to the thickness of the film (Peelaers et al., 2012; Chen et al., 2021).

In the case of low AcAc content, film formation proceeds at a low rate due to significant pH variation, which leads to a decrease of  $Ac (Zn)_2$  species.

It should be noted that the results obtained in this work are compared with previous reports where the effect of varying the acetic acid concentration on the synthesis of ZnO and ZnO:F films is also studied. For instance, Harun et al. (2017) discussed the influence of AcAc on the preparation of ZnO thin films by the sol-gel method, with a focus on optoelectronic applications; they observed that the acetic acid concentration affects both the morphology and the optical and electrical properties of the films and they determined the AcAc concentration at which crystallization is enhanced and resistivity of the films is reduced. On the other hand, Shikha et al. (2015) made a similar analysis where variations in grain size and surface texture are highlighted, which impact visible light transmittance and conductivity. However, in this work we address how the AcAc concentration is impacted by the USP method which is considered more suitable for large-scale applications since it is faster and cheaper than the sol-gel method, since the latter involves the use of high-purity precursors such as metal alkoxides which tend to be more expensive and also involves several processing steps, which increases manufacturing time and costs. In contrast, USP is a continuous and more direct process, allowing for greater efficiency (Kaneva et al., 2012).

Finally, the obtained results suggest that this method for synthesis of ZnO and ZnO:F thin films can be useful in the production of Transparent Conductive Oxides. Moreover, this methodology presents an advantage over other reported methods by using a fresh starting solution making the process more efficient and cost-effective, besides it avoids the aging of the solution, which can delay the production of these films.









# 5 Conclusion

In conclusion, this work has demonstrated that the ultrasonic chemical spray technique can be used to produce highly transparent and conductive thin films of ZnO and ZnO:F. The variation in acetic acid

content in the starting solution has been shown to play a key role in the modulation of electrical, morphological, and optical characteristics of the films. It was found that ZnO:F films with the lowest resistivity value (0.016  $\Omega$ · cm) were obtained through a 14-min deposition at 450°C from an initial solution containing 200 mL/L of acetic acid. Additionally, higher

resistivities were observed with lower acetic acid contents. All the films, ZnO and ZnO:F, exhibited a wurtzite-type structure and preferential growth in the (002) direction, indicating their polycrystalline nature. Although an acetic acid excess does not shift the pH of starting solutions to the lowest value beyond 4, under normal conditions, the synthesis reactions are modified in such a way that enhances the transport properties of the resulting films. These findings suggest that the produced films have the potential to be used as thin transparent conductors (TCOs) for several technological applications. Additionally, reducing the aging time lowers the overall manufacturing costs of ZnO-based TCOs, utilizing a cost-effective technique suitable for mass production.

# Data availability statement

The raw data supporting the conclusions of this article will be made available by the authors, without undue reservation.

# Author contributions

GB-B: Methodology, Writing-review and editing. AM: Conceptualization, Supervision, Writing-review and editing. MO: Conceptualization, Supervision, Writing-review and editing. JZ-J: Writing-review and editing.

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# Conflict of interest

The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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# Supplementary material

The Supplementary Material for this article can be found online at: https://www.frontiersin.org/articles/10.3389/fnano.2024.1445269/ full#supplementary-material

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