

Multicomponent Precipitation and Strengthening in Intermetallic-Strengthened Alloys

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Advanced structural materials with superior mechanical properties are of technological importance for industrial applications. Multicomponent precipitation provides a potential approach for designing high-performance alloys and has been receiving increasing attention from both academia and industry. In this Perspective, we highlight the recent advances and future perspectives in multicomponent-precipitation-strengthened alloys, including multicomponent steels, superalloys, and high-entropy alloys. The emphasis is placed on the unique microstructural features and challenges of multicomponent precipitation as well as their impacts on the strengthening behavior. Finally, future research areas for this class of materials are critically discussed.

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INTRODUCTION

Metallic materials with excellent mechanical properties are highly desirable for a wide range of engineering applications, such as aerospace, transportation, construction, marine, and energy industries (Reed, 2008; Fine et al., 2010; Yen et al., 2011; Kapoor et al., 2014; Miracle and Senkov, 2017; George et al., 2019). It is commonly understood that the mechanical properties of crystalline materials are determined by the microstructure and dislocation interactions. Basically, strengthening crystalline materials involves the controlled creation of internal defects to impede the motion of dislocations. Intermetallic precipitation strengthening has been extensively used in many metallic systems, such as steels and superalloys (Raabe et al., 2009; Mulholland and Seidman, 2011; He et al., 2016; Bleck et al., 2017; Sun et al., 2017; Liang et al., 2018; Yang et al., 2018). The degree of precipitation strengthening is highly dependent on the microstructure and micro-mechanical properties of precipitates as well as the nature of the dislocation-precipitate interaction. Recent studies reveal that intermetallic precipitates in multicomponent metallic systems are often of multicomponent type, the formation of which involves complicated elemental partitioning and sublattice occupancies. The complex compositions and lattice structures can significantly influence their nucleation, growth, and coarsening behaviors, thereby affecting the precipitate microstructure and strengthening behavior (Jiang et al., 2015; Trotter et al., 2016; Song et al., 2017; Zhao et al., 2018). For example, B2-ordered Ni(Al,Mn)-type precipitates can form in Fe-Ni-Al-Mn multicomponent alloys, in which Mn partitions to the precipitates and occupies the Al sublattice (Zhang et al., 2013). The Mn partitioning significantly reduces the critical energy for nucleation and increases the number density of precipitates by more than one order of magnitude, leading to a doubled strengthening

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effect. In fact, the beneficial effects of multicomponent precipitates on the thermal, physical, and mechanical properties of materials are frequently observed in many precipitation-strengthened systems. As a result, the development of multicomponent-precipitation-strengthened alloys has attracted extensive and increasing interest from both academia and industry. In this Perspective, we highlight the development recent of multicomponent-precipitationstrengthened alloys and critically discuss their unique features, challenges, and future visions, as schematically illustrated in Figure 1. In particular, we focus on the complicated elemental partitioning and sublattice occupancies in multicomponent steels, superalloys, and high-entropy alloys. The unique microstructural features bring new challenges for understanding and control of the precipitation and strengthening behaviors. To further accelerate the advancement of the multicomponentprecipitation-strengthened alloys, fundamental and technological areas should be further explored in future studies, such as computation-aided alloy design, atomic structure, intrinsic characteristics, and overall properties of these materials.

MULTICOMPONENT PRECIPITATES IN INTERMETALLIC-STRENGTHENED ALLOYS

Multicomponent Precipitates in Steels

Fe-based steels undergo a phase transformation from a facecentered cubic (fcc) austenite phase at high temperatures to a body-centered cubic (bcc) structure at low temperatures. Some fcc elements, such as Ni, Al, and Cu, can be dissolved to a considerable extent in the fcc austenite phase but have a low solubility in the bcc structure, which results in the precipitation of fcc elements at low temperatures. For example, B2-ordered NiAl particles can precipitate coherently from the ferritic/martensitic matrix, and their compositions often deviate from the stoichiometric composition of binary NiAl, especially at an early stage of precipitation (Teng et al., 2012; Kim et al., 2015; Jiang et al., 2017; Song et al., 2017). Depending on the alloy composition and processing conditions, Fe can partition to the B2-ordered precipitates by occupying the Ni or Al sublattice, leading to the formation of (Ni,Fe)Al or Ni(Al,Fe) precipitates. In addition, Mn can also partition to NiAl-type precipitates by replacing parts of Al, resulting in the formation of Ni(Al,Mn) precipitates (Jiao et al., 2015). In terms of mechanical properties, the Ni(Al,Mn) precipitates increase the yield strength of a Fe-5Ni-1Al-3Mn (wt%) steel from 685 MPa in the asquenched condition to 1225 MPa in the peak-aged condition, which is almost a double increase; meanwhile, the ductility decreases slightly from 16% in the as-quenched condition to 14% in the peak-aged condition (Jiao et al., 2015). Another example is the Ni₃Ti-type precipitation in maraging steels. In the initial stages of precipitation, Mo partitions to the Ni₃Ti-type precipitates by occupying the Ti sublattice, which leads to the formation of high densities of Ni₃(Ti,Mo) precipitates (Niu et al., 2021). These multicomponent precipitates provide effective strengthening responses, leading to the development of advanced ultrahigh-strength steels with 2 GPa strength and >10% ductility.

Multicomponent Precipitates in Superalloys

Ni-based superalloys are known to have a complex composition, with more than ten alloying elements in most alloys. They are key structural materials for high-temperature applications in aerospace, particularly the turbine blades of jet engines (Pope and Ezz, 1984; Pollock and Tin, 2006; Reed, 2008). Precipitation-strengthened superalloys have a unique γ/γ' microstructure, in which ordered γ' precipitates are distributed in the γ matrix. The γ phase is an fcc solid solution with randomly distributed different species of atoms, whereas the γ' phase is a L1₂-ordered Ni₃Al-type intermetallic compound. Generally, the Ni

sublattice is comprised of mostly electronegative elements, such as Ni and Co., whereas the Al sublattice constitutes more electropositive elements, such as Al, Ti, Nb, and Ta (Reed, 2008). Generally, dislocations move into the L1₂-ordered γ' phase by forming antiphase boundaries, whereas dislocations travel across the y matrix by forming stacking faults. The antiphase boundary energy of the γ' phase is higher than the intrinsic stacking fault energy of the y phase, which thus hinders the motion of dislocations in the y phase. The high volume fractions of coherent γ' precipitates with high strength and thermal stability enable the γ/γ' superalloys to achieve superior mechanical properties at high temperatures (up to 0.9 Tm) (Reed, 2008). Similar precipitation behaviors are also observed in Co-based γ/γ' superalloys, in which the γ' phase is the L1₂-ordered Co₃Al-type intermetallic compound. The binary Co₃Al phase is known to be metastable at high temperatures, but which can be stabilized by alloving elements, such as Cr, V, Nb, Ta, and W. In the recent decades, a series of novel Co-based superalloys strengthened by stable γ' precipitates, such as Co₃(Al,W), Co₃(Al,Cr,Ta), and Co₃(Al,Ti,V,Nb), have been developed (Sato et al., 2006; Oohashi et al., 2008; Xue et al., 2013; Suzuki et al., 2015). These materials show promising mechanical properties as potential candidates for hightemperature applications.

Multicomponent Precipitates in High-Entropy Alloys

High-entropy alloys, also known as multiple-principal-element alloys, have emerged as a new class of metallic materials because of their unique microstructure and excellent mechanical properties at cryogenic temperatures (Gludovatz et al., 2014). Among these alloys, the face-centered cubic (FCC) alloys, such as CoCrFeMnNi and CoCrNi, have received considerable attention because of their exceptional ductility and toughness, even down to the liquid nitrogen temperature. The yield strength of these FCC high-entropy alloys are generally in the range of 200-500 MPa (Wu et al., 2014). Recent studies reveal that intermetallic precipitation is effective in improving the mechanical strength of high-entropy alloys without causing a significant reduction in ductility (Zhang et al., 2014; Wang et al., 2018). Relating to the multicomponent nature of high-entropy alloys, the intermetallic precipitates are also complex in composition and sublattice occupancy. For instance, L12ordered (Ni,Co)₃(Al,Ti) precipitates can coherently form in the Fe-Co-Ni-Cr-Al-Ti system, which leads the to development of precipitation-strengthened high-entropy alloys with gigapascal strengths (Daoud et al., 2015; He et al., 2016; Liu et al., 2018; Zhao et al., 2018). Recently, even more complicated L1₂-ordered precipitates with a composition of 43.3Ni-23.7Co-10.0Fe-14.4Ti-8.6Al (at%) are formed in Fe-Co-Ni-Al-Ti alloys (Yang et al., 2018). The sublattice occupancy in the multicomponent precipitates was investigated by atomistic modelling, and the results indicate that these L12-type multicomponent precipitates can be regarded as the (Ni_{43.3}Co_{23.7}Fe₈)₃(Ti_{14.4}Al_{8.6}Fe₂) phase (Yang et al., 2018). In addition, multicomponent B2-(Ni,Co)(Al,Cr,Fe) precipitates

strengthened high-entropy alloys are developed, which exhibit outstanding fatigue resistance (Feng et al., 2021). This research area is still immature and may lead to new exciting developments.

DISCUSSION ON THE MULTICOMPONENT PRECIPITATION AND STRENGTHENING BEHAVIORS

Precipitation Behavior

As the formation of multicomponent precipitates involves the atomic interaction and cooperative diffusion of various species of elements, their precipitation behavior can be significantly different from that of compositionally simple precipitates, especially in the nucleation stage. According to the classical nucleation theory, the critical energy for nucleation depends on the interplay among the chemical driving force, interfacial energy, and strain energy. Any factors affecting the chemical driving force, interfacial energy, and/or strain energy can have an impact on the nucleation behavior. An accelerated or decelerated nucleation behavior is often observed in multicomponentprecipitation-strengthened alloys. For example, Fe partitions to B2 precipitates in maraging steels, leading to the formation of Ni(Al,Fe)-type precipitates. The lattice misfit between Ni(Al,Fe) and bcc-Fe is much smaller than that between NiAl and bcc-Fe, which results in an extremely small strain energy between the precipitates and matrix, thereby substantially accelerating the NiAl nucleation (Jiang et al., 2017). When Cu is added to NiAlstrengthened steels, Cu partitions to NiAl precipitates in the early state of precipitation, which leads to the formation of B2-Ni(Al,Cu) precipitates. The incorporation of Cu into NiAl precipitates enhances the attractive interaction between Ni and Al, which substantially increases the chemical driving force for nucleation, thereby promoting the NiAl precipitation (Shen et al., 2017; Zhou et al., 2021). In addition, the coarsening behavior of multicomponent precipitates is also influenced by the precipitate composition and lattice structures. In high-entropy alloys, for example, the long-range diffusion is found to be sluggish because of the necessity for cooperative diffusion of solute atoms to maintain a proper elemental partitioning and sublattice occupancy. Resulting from sluggish diffusion, the precipitate coarsening rate in L12-(Ni,Co)3(Al,Ti)-strengthened highentropy alloys is one to two orders of magnitude lower than that in traditional Ni-based superalloys (Zhao et al., 2018). Therefore, the compositional complexity of multicomponent precipitates can result in a considerable variation in the thermodynamic, structural, and physical properties, which provides a useful approach to tailoring the precipitate microstructure. Meanwhile, it brings new challenges for understanding and control of the precipitation behavior.

The processing method also has an important impact on the precipitation behavior of multicomponent alloys. Traditionally, metallic parts are manufactured by casting and forging processes, and the precipitate microstructure can be controlled through heat-treatments. There is now a growing interest in processing multicomponent alloys using new technologies for better material utilization, higher degree of design freedom, and/or better physical and mechanical properties. For example, the electron beam melting of CoCrFeNiMn-xTi results in the formation of multicomponent σ , γ , and η precipitates, the growth of which is limited at the boundaries of FCC columnar grains due to the rapid solidification nature of the process (Cagirici et al., 2021). The selective laser melting of a (FeCoNi)86Ti7Al7 alloy produces a heterogeneous precipitate microstructure consisting of L1₂ precipitates inside cellular structures and B2 precipitates along the cellular boundaries, which is quite different from the traditionally heat-treated microstructure with a uniform distribution of L1₂ nanoparticles (Huang et al., 2022). The sputtered Inconel 718 films generally show the precipitation of multicomponent δ -Ni₃Nb precipitates, whereas in bulk 718 alloys metastable γ'' -Ni₃Nb precipitates form first and transform to the stable δ -Ni₃Nb phase after prolonged aging times (Burns et al., 2014). These observations suggest that different processing methods can result in different microstructures in terms of precipitate types, sizes, volume fractions, morphologies, and distributions. The new processing technologies provide new opportunities for tailoring the precipitate microstructure for improved properties. However, a clear understanding of the underlying precipitation mechanism and processingmicrostructure correlation of these alloys remains elusive and needs further investigation.

Strengthening Behavior

The precipitation strengthening is based mainly on the interaction of dislocations with precipitates (Gladman, 1999). There are two types of dislocation-precipitate interaction depending on the precipitate size (inter-precipitate spacing). When the precipitate size is larger than a critical size, the Orowan bypassing mechanism is operative; when the precipitate size is smaller than the critical one, the dislocations cut through or shear the precipitates. In the precipitate shearing mechanism, three factors contribute to the strength increase, i.e., coherency strengthening, modulus strengthening, and order strengthening. From the strengthening models, the precipitation strengthening effect depends on not only the precipitate microstructure (precipitate size, volume fraction, and morphology) but also the intrinsic properties of the precipitates, such as lattice constant, shear modulus, Young's modulus, and antiphaseboundary energy (Embury et al., 1971; Ardell, 1985). When modelling the strengthening of multicomponent precipitates, the values of shear modulus, Young's modulus, and antiphase boundary energy are often borrowed from compositionally simple intermetallic compounds, such as binary Ni₃Al and NiAl. However, it is known that a small change in composition and lattice occupancy can have a great impact on the intrinsic properties of intermetallic compounds. For example, Vittori et al. found that the antiphase-boundary energy of Ni₃(Al,Ti) can be two times higher than that of binary Ni₃Al (Vittori and Mignone, 1985). Kim et al., (2010) reported that the partial substitution of Ni by Cr can significantly reduce the shear modulus of Ni₃Al. Thus, the intrinsic mechanical properties of multicomponent precipitates can be significantly different from those of stoichiometric precipitates, and the use of parameters of stoichiometric precipitates for strengthening modelling may produce large inaccuracies and even wrong conclusions. Therefore, for a better

understanding of the strengthening mechanism of multicomponent precipitates, a quantitative study of the intrinsic mechanical properties of multicomponent precipitates is essential.

OUTLOOK

Although multicomponent-precipitation-strengthened alloys are promising in terms of physical and mechanical properties, several fundamental and technological areas should be further explored in future studies.

Computation-Aided Alloy Design

Because of the complex interaction among various alloying elements, the classical "trial-and-error" method no longer suffices for the design of multicomponent-precipitation-strengthened alloys. Alternatively, computational calculations and genetic algorithms offer a potentially effective way for rapid alloy screening and design. For example, ab initio calculations can compute thermodynamic and structural properties for both stable and metastable phases, whereas thermodynamic calculations can provide useful information on phase equilibrium and transformations (Kresse and Hafner, 1993). For example, Kannan et al. developed a thermo-kinetic model to predict the precipitation and austenite reversion kinetics during various post-fabrication heat treatments in multicomponent steels (Kannan and Nandwana, 2021). In addition, machine learning can capture the highlycomplex non-linear input/output relationships, to help filter or select the potential good materials rapidly (Zhao et al., 2021). These computational methods can serve to guide experimental efforts by effectively narrowing down the search space. Future work should place an emphasis on the integrated computationalprediction and experimental-validation approach for designing novel multicomponent-precipitation-strengthened alloys with desired properties.

Atomic Structures and Intrinsic Micro-Mechanical Properties

For a full understanding and control of the multicomponent precipitation behavior, the knowledge of the structure, chemistry, and energetics of multicomponent precipitates is necessary. The multicomponent precipitates are complex in terms of lattice occupancy, elemental partitioning, and interfacial segregation, and in some cases, the precipitates undergo phase transformations during the growth and coarsening processes. Future attention needs to be paid to the atomic-level characterization of multicomponent precipitates through advanced nano-analytical tools (such as aberration-corrected transmission electron microscopy and 3D atom probe tomography) (Seidman, 2007; Miller, 2012) and computational calculations (such as ab initio calculations) (Kresse and Hafner, 1993). In addition, as aforementioned, the strengthening effect of multicomponent precipitates is related to their micromechanical properties, such as Young's modulus, shear modulus, and anti-phase boundary energy. The state-of-the-art peak-force tapping atomic force microscopy allows simultaneous capture of topography and maps of nano-mechanical properties, thus

providing new opportunities to understand the micro-mechanical properties of multicomponent precipitates. Furthermore, atomistic modellings can play an important role in the understanding of intrinsic mechanical properties of multicomponent precipitates, such as elastic properties and energetics. Therefore, it is of fundamental importance to obtain detailed insights into the atomic structure and intrinsic mechanical properties of multicomponent precipitates through a combination of experimental and computational approaches.

Overall Properties

While most studies on multicomponent-precipitation-strengthened alloys focus on the general mechanical properties, such as strength, ductility, and strain hardening, the understanding of their overall properties, including processability, toughness, fatigue, creep, corrosion, and welding properties, remains limited. Particularly, alloys for industrial applications are generally used in various forms, such as sheets, plates, bars, tubes, and other structures. Thus, the processability of these materials is critical for manufacturing and industrial applications and needs to be carefully evaluated. In addition, creep resistance is critically important for applications that require long-term dimensional stability under load at elevated temperatures, whereas impact toughness is a key parameter in influencing the performance of engineering applications at low temperatures. Fatigue resistance is an essential property in preventing catastrophic failure of engineering components in service conditions. In order to develop advanced multicomponent-precipitation-strengthened alloys for industrial applications, future work needs to evaluate the overall properties of these materials under different conditions. In addition, it is also important to understand the fundamental mechanisms and key factors governing the overall properties of multicomponentprecipitation-strengthened alloys and to establish the relationship between the precipitate microstructure/properties and bulk alloys properties.

CONCLUDING REMARKS

This Perspective highlights the recent advances in the multicomponent precipitation and strengthening of

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intermetallic-strengthened alloys. The formation of multicomponent precipitates involves complicated elemental partitioning and sublattice occupancies, which results in a considerable variation in their structural, physical, and mechanical properties, thereby influencing the precipitation and strengthening behaviors. Understanding the composition-microstructure-property relationship is essential for both fundamental understanding and technological applications of multicomponent-precipitation-strengthened alloys. Future work should place on the computation-aided alloy design of advanced alloys with superior mechanical properties and the fundamental understanding of the atomic structure and micro/macro-mechanical properties of multicomponent-precipitation-strengthened alloys.

DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding author.

AUTHOR CONTRIBUTIONS

ZJ designed the research. JF, WL, TY, YW, and ZJ discussed and wrote the manuscript. All authors read and approved the final manuscript.

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