



First-Principles Study on the Interaction of H₂O and Interface Defects in A-SiO₂/Si(100)

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The defects contained in amorphous SiO₂/Si (a-SiO₂/Si) interface have a considerable impact on the efficiency and stability of the device. Since the device is exposed to the atmospheric environmental conditions chronically, its performance will be limited by water diffusion and penetration. Here, we simulated the interaction of H₂O and interface defects in a-SiO₂/Si(100) by using the first-principles method. Our results suggest that H₂O penetrated into P_{b0} defect is more inclined to interact with the network in the form of silanol (Si-OH) group, while H₂O incorporated into P_{b1} defect is more likely to remain intact, which can be attributed to the location of P_{b1} defect closer to the interface than that of P_{b0} defect. Our research provides a powerful theoretical guidance for the interaction of H₂O and interface defects in a-SiO₂/Si(100).

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1 INTRODUCTION

Amorphous silicon dioxide (a-SiO₂) is an important component of Metal-Oxide-Semiconductor Field-Effect Transistors (MOSFET) (Sheikholeslam et al., 2016), optical fibers and solar cells (Stesmans and Afanas'ev, 1998; Watanabe et al., 1998; Matsudo et al., 2002; Kajihara et al., 2005; Pamungkas et al., 2011), which has a huge impact on the reliability of the device as a basic element of modern technology (Gerardi et al., 1986; Rochet et al., 1997; Lenahan and Conley, 1998; Kovačević and Pivac, 2014). In practical applications, it is usually plagued by defects and impurities such as hydrogen (H) (McLean, 1980; Stesmans, 1996; Chadi, 2001; Pantelides et al., 2007), water (H₂O) (Takahashi et al., 1993; Batyrev et al., 2008) and alkali metal (Perez-Beltran et al., 2015), and these factors will change the mechanical, electronic (Blöchl and Stathis, 1999) and optical (Griscom, 1991) properties of a-SiO₂. Among the electroactive defects in a-SiO₂/Si interface with different Si surface orientations, P_b center occupies a principal position. It is essentially a tripod-like structure consisting of an isolated Si dangling bond and three reverse-bonded Si atoms (·Si-(·Si)₃) (Helms and Poindexter, 1994). The most technologically crucial Si surface with (100) orientation contains two P_b variants, namely P_{b0} (Haneji et al., 1991; Yamasaki et al., 2003; Thoan et al., 2011) and P_{b1} (Stirling et al., 2000; Campbell and Lenahan, 2002; Kato et al., 2006). Although both of them have a common form of ·Si-(·Si)₃, they are quite different in terms of orientation and hyperfine parameters, and these interface defects are the main factors affecting device performance.

In reality, the interface dangling bond defects in device preparation can be passivated by H into a non-electrically active Si-H structure, which will not influence the quality of the device. However, if

exposed to the environment of ionizing radiation for a long time, the defects that have been passivated will be reactivated by the protons generated by the radiation, and the activated dangling bonds will accumulate charges by trapping the carriers, which may cause the degradation and even failure of the device performance (Michalske and Freiman, 1983; Lu et al., 1993; Kosowsky et al., 1997). Besides, H₂O plays a significant role in the properties of a-SiO₂/Si interface (Pfeffer and Ohring, 1981; Tomozawa, 1985; Bourg and Steefel, 2012). Due to the fascinating network pattern, it can form an amorphous structure with SiO₂ surface according to the strength of the interaction between them. Especially the interaction of H₂O and interface defects in a-SiO₂/Si(100), which is the main concern of scientific research. Since H₂O always exists in the atmosphere, its role in experiments is ineluctable under any achievable ultra-high vacuum pressure conditions. The interaction of H₂O and interface defects may cause hydrolysis in a-SiO₂/Si(100), which is possible to form a silanol (Si-OH) site attached to the silicon defect atom (Li et al., 2009; Yeon and Van Duin, 2016), and the remaining proton (H⁺) after hydrolysis can continue to interact with P_{b0}-H that has been passivated to generate P_{b0}⁺ and H₂. Even though little hydrogen is introduced in this process, as long as H₂O and interface defects are present, H₂O can be continuously converted into hydrogen molecules, thereby providing hydrogen that plays a key role in the ionization damage process (Schwank et al., 2008).

The present work focuses on the interaction of H₂O and P_{b0}/P_{b1} defects in a-SiO₂/Si(100) interface, and clearly reproduces the specific process of the reaction, as well as the variation of bond length and potential barrier during the reaction. The framework of the manuscript is as follows: The **Section 2** introduces the calculation method. The **Section 3** contains the simulation results and discussion, and the conclusion is drawn in the **Section 4**.

2 COMPUTATIONAL METHODS

All of the calculations are performed based on the density functional theory (DFT) using the projector augmented wave (PAW) method, which is implemented in Vienna ab-initio simulation package (VASP) (Parr, 1983; Blöchl, 1994; Hafner, 2008). Perdew-Burke-Ernzerhof (PBE) parametric generalized gradient approximation (GGA) is applied to deal with the exchange-correlation functional (Perdew et al., 1996). In addition, the models of P_{b0} and P_{b1} defects used in our study are both derived from Li et al. (Li et al., 2019a). Considering the large number of atoms in the system (more details in **Supplementary Material S1**), a step-by-step iterative method is adopted to optimize the structure to minimize the consumption of computing resources, where the cut-off kinetic energies are set to 300 and 500 eV, and the atomic positions are fully relaxed until the total energy and residual force ultimately converge to 10⁻⁵ eV and 0.01 eV/Å, respectively. The Brillouin zone (BZ) is only selected for sampling and integration at the Γ point (Monkhorst and Pack, 1976). Finally, the activation energy of H₂O embedded in a-SiO₂/Si(100) interface is measured by using the Climbing Image Nudged Elastic Band (CI-NEB) method, which makes a corrective improvement on the traditional NEB

method by combining the climbing mirror image and the definition of the tangent, so as to search for more accurate saddle points with fewer intermediate images (Henkelman et al., 2000; Sheppard et al., 2012).

3 RESULTS AND DISCUSSION

3.1 P_{b0} and P_{b1} Defects in A-SiO₂/Si(100) Interface

Firstly, the defects in a-SiO₂/Si(100) interface are briefly introduced to facilitate the subsequent simulation of its reaction with H₂O. There are two different types of defects in a-SiO₂/Si(100) interface, namely P_{b0} and P_{b1} defects. Among them, P_{b0} defect is located in the second unoxidized silicon atomic layer adjacent to the interface transition zone, and it is constructed by removing one of the Si atoms from the unoxidized top layer on the silicon side, which can be represented as ·Si(-Si)₃, as shown by the blue Si defect atom in **Figure 1A**. The dangling bond of it lies along the [111] direction (54.74°) (Li et al., 2019b), and the remaining dangling bonds of three defect atoms (silicon atoms 1, 2 and 3 in **Figure 1A**) can be passivated by hydrogen atoms into the form of Si-H, which has no effect on the defect model due to the lack of electrical activity.

In addition to P_{b0} defect, the a-SiO₂/Si(100) interface also involves another dangling bond defect, whose defect density is lower than that of P_{b0}, which is called P_{b1} defect. Different from P_{b0} defect, P_{b1} is created by removing one of the oxygen atoms from a-SiO₂/Si(100) interface, resulting in it being located on the top layer of bulk silicon and the dangling bond is roughly along the [112] direction (30–36°) (Li et al., 2019b), which is not consistent with any chemical bond orientation. A more intuitive schematic diagram of the location of P_{b0} and P_{b1} defects in a-SiO₂/Si(100) is shown in **Supplementary Figure S2**. Moreover, P_{b1} defect is extremely sensitive to the local structure deformation. Depending on the different positions of the defect, it can be divided into dimer, bridge and AOD (asymmetric oxidized dimer) models, and the latter has been verified to be superior to the former two in terms of Fermi contact value and dangling bond orientation (Poindexter et al., 1981) (more details in **Supplementary Figure S3**). Considering that the interface transition region of the amorphous model is more complicated than that of the crystal model, the atomic configuration around the defects of a-SiO₂/Si(100) interface will significantly influence the simulation results. In order to make the results more convincing, we simulated the reaction process between P_{b1} defect with AOD configuration at three random positions (as shown in **Figures 1B–D**) and H₂O, and compared the reaction results to obtain a more reliable conclusion.

3.2 The Interaction of H₂O and P_{b0} Defect in A-SiO₂/Si(100) Interface

Here, we will simulate the interaction of H₂O and P_{b0} defect concretely. First, place a H₂O at an appropriate position around the defect atom (Si89), and make sure that it does not bond with other atoms near the defect, so the initial state 00 for the reaction of P_{b0} defect and H₂O is constructed, as shown in **Figure 2A**. For the final state 05, one hydrogen atom (H39) in H₂O is attached to

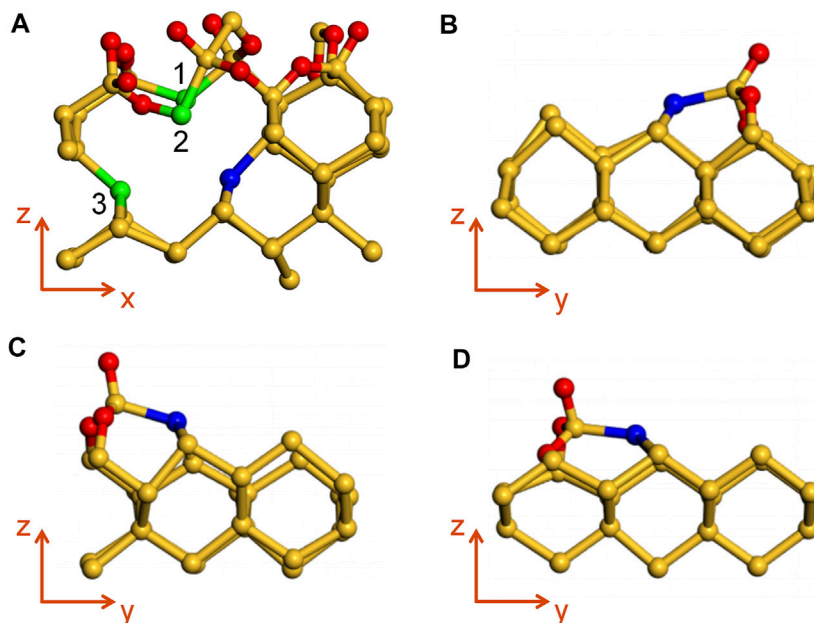


FIGURE 1 | (A) The schematic diagram of P_{bo} defect model. **(B–D)** P_{b1} defect models with AOD configuration at three randomly selected positions. Blue spheres indicate the location of P_{bo}/P_{b1} defects; green silicon atoms will be passivated by hydrogen atoms to saturate dangling bonds in subsequent reaction simulations; yellow and red spheres represent Si and O atoms, respectively.

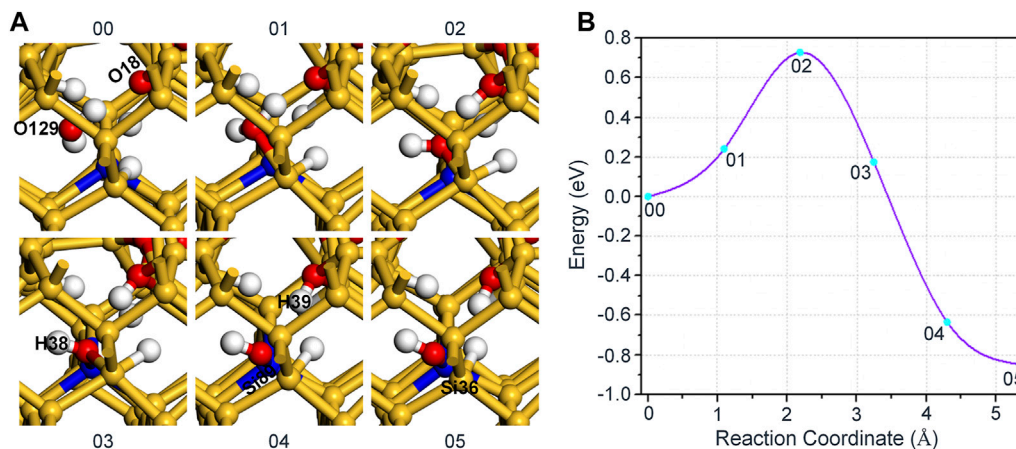
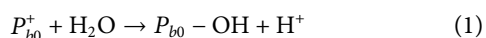


FIGURE 2 | (A) The schematic diagram of the reaction between P_{bo} defect and H₂O. H₂O in the initial state 00 exists freely around the defect atom (Si89), and three unsaturated silicon atoms marked by green spheres in **Figure 1A** are passivated by the adhesion of H atoms. **(B)** The variation of the energy with different reaction coordinates for the reaction of H₂O and P_{bo} defect.

the nearest neighboring oxygen atom (O18) with a form of proton, while the remaining hydroxyl group is bonded with Si89 to passivate the dangling bond defect, so that the system can reach a stable structure. Furthermore, we have subtracted an electron from the initial state to ensure the conservation of the electron numbers before and after the reaction, whose equation can be expressed as:



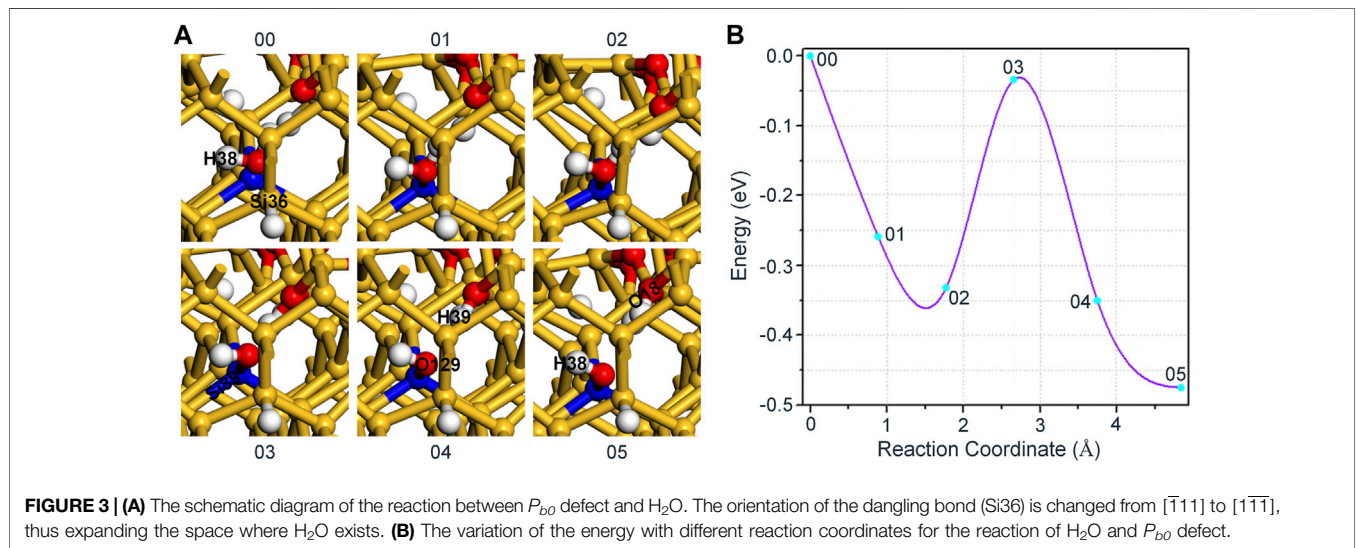
After determining the initial and final state of the reaction, the CI-NEB method based on first-principles calculation is employed to simulate the reaction process of H₂O and P_{bo} defect in a-SiO₂/Si(100) interface, as shown in **Figure 2A**. Compared with Si89, H₂O is closer to the silicon atom (Si36) passivated by H atom (2.500 Å), thereby the oxygen atom (O129) in H₂O prefers to connect with Si36 in the initial stage. This step can cause Si36 to change from a quadruple to a quintuple coordination, which will further increase the energy of the system, as shown in **Figure 2B**. Next, H39 in H₂O detaches from its

TABLE 1 | Changes in bond length for the reaction of H₂O and P_{b0} defect.

Bond length (Å)	00	01	02	03	04	05
Si89-O129	3.750	3.727	3.147	2.028	1.706	1.685
Si36-O129	2.500	2.024	1.770	1.975	2.614	2.873
H39-O18	1.769	1.597	1.045	0.994	1.000	1.006
H38-O129	0.979	0.979	0.986	0.996	0.983	0.981
H39-O129	0.995	1.017	1.594	2.376	2.908	3.203

TABLE 2 | Changes in bond length for the reaction of H₂O and P_{b0} defect.

Bond length (Å)	00	01	02	03	04	05
Si89-O129	2.901	2.079	1.927	1.829	1.726	1.715
H39-O18	1.775	2.002	1.895	1.185	0.991	0.997
H38-O129	0.982	0.990	0.991	0.983	0.979	0.979
H39-O129	0.997	1.000	1.004	1.309	2.270	2.871



parent and gradually approaches the position of O18 until after connection with it, thus reaching the saddle point 02 of the reaction. As the reaction continues, the distance between H₂O and Si89 gradually becomes smaller, so that they are linked to each other, and the energy of the system begins to show a downward trend. Moreover, the increase in the distance between H₂O and Si36 passivated by hydrogen atom leads to the cleavage of the Si36-O129 bond, then Si36 is also restored to a stable quadruple coordination from the preceding quintuple coordination. Finally, H39 in H₂O is bonded with O18 (1.006 Å) in a form of proton, while the remaining hydroxyl group is bonded with Si89 (1.685 Å) in order to passivate the dangling bond defect, so that the system reaches the most stable state. As shown in **Figure 2B**, the forward and reverse reaction barriers are 0.73 and 1.58 eV, respectively. In the whole reaction, the variation of interatomic bond length around the defect is shown in **Table 1**.

In the above reaction, we noticed that when the hydroxyl group in H₂O passivates the dangling bond, the position of O129 is closer to Si36 passivated by hydrogen atom than Si89. As a result, the hydroxyl group in H₂O is easier to interact with Si36, which makes Si36 vary from quadruple to quintuple coordination, resulting in an overall increase of the energy. In order to avoid this situation from adversely affecting the reaction, we adjusted the spatial position of Si36 accordingly. At the same time, the direction of the dangling bond passivated by hydrogen atom is also changed from the original $[\bar{1}11]$ to $[1\bar{1}\bar{1}]$, thereby expanding the space where H₂O exists.

As shown in **Figure 3A**, we reconstructed the initial and final state of the reaction according to **Eq. 1**. On this occasion, H₂O first moves towards the position of Si89 and bonds with it to achieve the expected purpose of passivating dangling bond, which results the energy of intermediate states 01 and 02 being lower than initial state 00 of the system, as shown in **Figure 3B**. Subsequently, the distance between O129 in H₂O and Si89 gradually decreases, and H39 in H₂O also combines with O18 in a form of proton, which makes the reaction reach the saddle point 03. Finally, with the continuous optimization of the positions about the hydroxyl group and proton, the energy of the system is also reduced to a minimum. In the entire reaction, the variation of interatomic bond length around the defect is shown in **Table 2**. Remarkably, the hydroxyl group used to passivate the dangling bond in the reaction is no longer interfered by Si36, which also proves that our adjustment to P_{b0} defect model is more conducive to simulate the real reaction of H₂O and P_{b0} defect in a-SiO₂/Si(100).

In addition, we can observe from **Figure 3B** that the energy of the intermediate state 02 is about 0.33 eV lower than that of the initial state 00. Taking into account that the structure of the intermediate state 02 can also meet the requirement of **Eq. 1** for the reactants. Therefore, the intermediate state 02 is subsequently selected as the initial state structure of the reaction between H₂O and P_{b0} defect, but the final state structure remains unchanged. In the following simulation, we directly choose the model with a

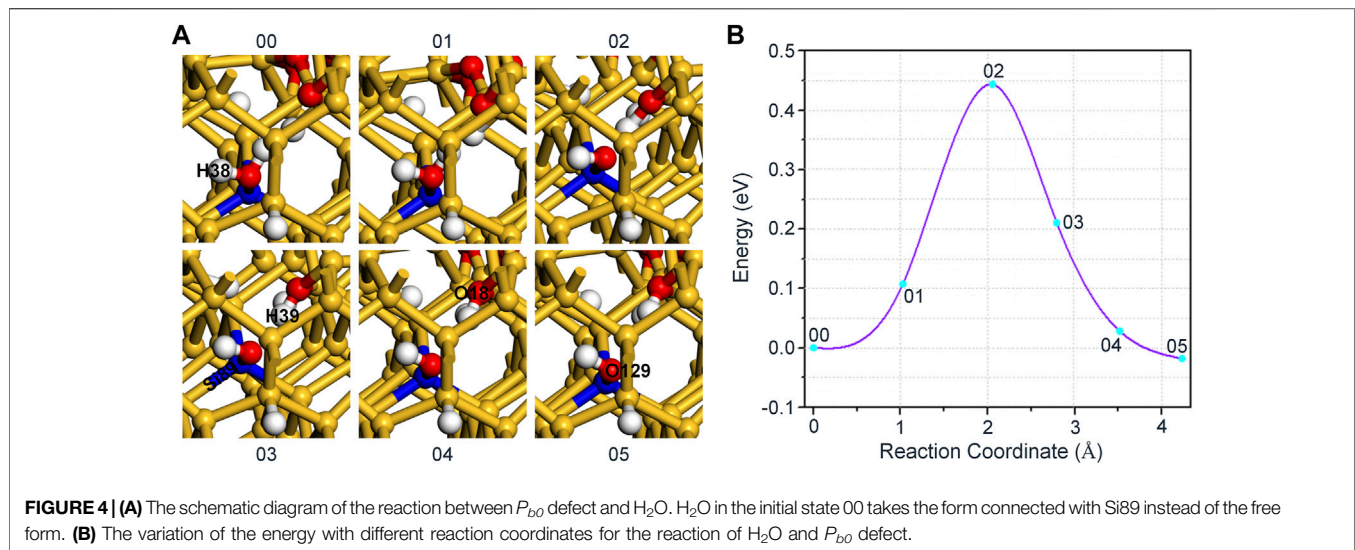


TABLE 3 | Changes in bond length for the reaction of H_2O and P_{bo} defect.

Bond length (Å)	00	01	02	03	04	05
Si89-O129	1.907	1.932	1.832	1.738	1.719	1.716
H39-O18	2.544	2.003	1.191	0.996	0.993	0.998
H38-O129	0.995	0.992	0.983	0.979	0.979	0.979
H39-O129	1.001	1.000	1.312	2.015	2.537	2.891

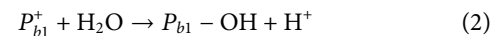
lower energy as the initial state of the reaction, which enables H_2O to passivate the dangling bond defect rather than existing in a free form.

In **Figure 4A**, the slight movement of H_2O towards Si89 leads to a decrease in the distance between H39 and O18, which in turn increases the energy of the system by 0.11 eV. When the absorbed energy reaches 0.44 eV, H39 in H_2O breaks with the oxygen atom connected to it, and combines with O18 in a form of proton to reach the highest energy state, which is the saddle point of the reaction. After that, with the continuous optimization of the positions about the hydroxyl group and proton, the system finally reaches an equilibrium state. In the whole reaction, the variation of interatomic bond length around the defect is shown in **Table 3**. Also, **Figure 4B** shows that, the forward and reverse reaction barriers are approximately the same, 0.44 and 0.46 eV, respectively. It is worth noting that the energy of the final state 05 is lower than that of the initial state 00 in all three cases, which indicates that the passivation of dangling bond defect by hydroxyl group is more stable than that by H_2O for the interaction of H_2O and P_{bo} defect in a-SiO₂/Si(100) interface.

3.3 The Interaction of H_2O and P_{bl} Defect in A-SiO₂/Si(100) Interface

Now, we perform computational simulations on the interaction of H_2O and P_{bl} defects at three random locations. According to the analysis in the last section, the initial state 00 of this part

adopts the form of bonding between H_2O and triple coordinated silicon defect atom (Si84). In addition, in order to ensure the conservation of the number of electrons in the system before and after the reaction, we have subtracted an electron from the structural model of the initial state 00. The reaction equation can be written as:



For the reaction of H_2O and P_{bl} defect with AOD configuration at random position 1, two oxygen atoms with different coordinates are selected to establish the final state, where the length of O-H bond is 1.502 Å and 1.907 Å, respectively. The relaxation results show that the hydrogen atom in H_2O cannot combine with the nearest oxygen atom in a form of proton, and it still appears that H_2O is intactly attached to the triple coordinated Si84. Our solution is to specify an oxygen atom (O89) that is relatively far away (3.153 Å) from the hydrogen atom (H37) in H_2O and bond with it as the final state 05 of the reaction. Hereby, we speculate that it is relatively stable to passivate the dangling bond defect with H_2O for the reaction of H_2O and P_{bl} defect with AOD configuration at random position 1.

The detailed reaction process is shown in **Figure 5A**. At first, H37 in H_2O gradually approaches the position of O89 and the reaction can reach the saddle point 02 when the distance between these two atoms is reduced to 2.083 Å. After that, the distance between H37 and O89 in H_2O keeps the trend of decreasing and eventually connects to each other (1.069 Å) after continuous optimization of the intermediate states. Intriguingly, the structural difference between any of the intermediate states (01, 02, 03 or 04) and the initial state 00 is only reflected in the bond length, as shown in **Table 4**. It also demonstrates that the passivation of dangling bond defect by H_2O is more stable than that by hydroxyl group for the reaction of H_2O and P_{bl} defect with AOD configuration at random position 1. **Figure 5B** shows the variation of the energy fluctuation during the whole reaction process, it can be seen that the forward reaction barrier of

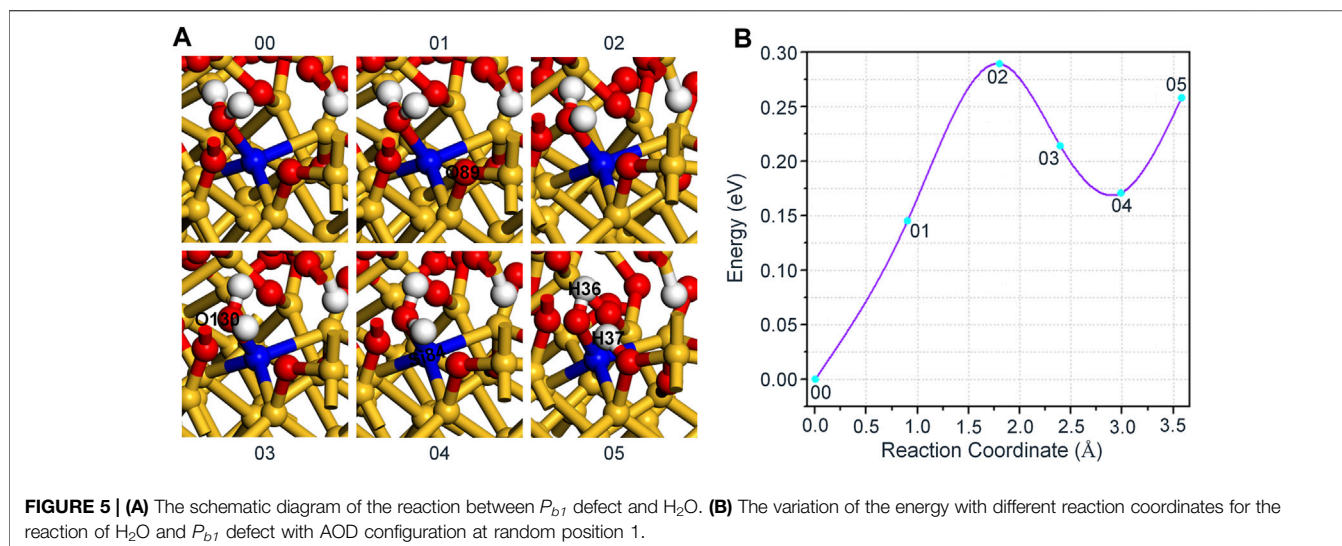


TABLE 4 | Changes in bond length for the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 1.

Bond length (Å)	00	01	02	03	04	05
Si84-O130	1.817	1.842	1.851	1.837	1.809	1.716
H36-O130	1.004	1.003	0.990	1.004	1.022	0.993
H37-O130	1.039	0.997	0.993	1.007	1.040	1.458
H37-O89	3.153	2.617	2.083	1.805	1.535	1.069

TABLE 5 | Changes in bond length for the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 2.

Bond length (Å)	00	01	02	03	04
Si84-O130	1.899	1.892	1.884	1.855	1.746
H36-O130	0.990	0.987	0.997	1.033	1.400
H37-O130	0.985	0.986	0.987	0.984	0.977
H36-O53	2.820	2.406	1.964	1.615	1.099

this reaction is 0.29 eV. In addition, the energy of the final state 00 is nearly 0.26 eV higher than that of the initial state 00, which again proves the stability of passivating P_{bd1} defect with H_2O .

Next, for the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 2, we conventionally choose an oxygen atom (O53) that is relatively far away (2.820 Å) from the hydrogen atom (H36) in H_2O and combine with it as the final state 04. In this reaction, the distance between O53 and H36 in H_2O gradually decreases, and the saddle point 01 can be attained when the distance between these two atoms is minished to 2.406 Å, as shown in **Figure 6A**. Since then, the position of H36 is getting closer and closer to that of O53, and finally bonds with it in a form of proton (1.099 Å), thus reaching the final state 04 of the reaction. As in the previous case, the structural

difference between any of the intermediate states (01, 02 or 03) and the initial state 00 is only embodied in the bond length, as shown in **Table 5**.

Figure 6B displays the change of the energy with different reaction coordinates in the interaction of H_2O and P_{bd1} defect with AOD configuration at random position 2. It can be seen that the forward reaction barrier of this reaction is only 0.02 eV. What's more, the energy of the final state 04 has a smaller raise (~0.08 eV) compared with that of the intermediate state 03, which verifies that the passivation of dangling bond defect by H_2O is more stable than that by hydroxyl group for the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 2.

Finally, we simulated and analyzed the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 3. Based on the above experience in constructing the final state, we still opt for an oxygen atom (O12) that is relatively far away (2.694 Å) from the hydrogen atom (H36) in H_2O and combine with it as the final state 04. As shown in **Figure 7A**, the movement of H36 in H_2O towards O12 leads to an increase in the bond length of O130 and H36. As the reaction proceeds, H36 in H_2O firstly breaks with O130 attached to it, and then combines with O12 in a form of proton (1.056 Å), thereby reaching the saddle point 03 of the reaction. Afterwards, with the continuous optimization of the positions about the hydroxyl group connected to Si84 and the proton bonded to O12, the system finally reaches the final state 04. The variation of interatomic bond length around the defect in the whole reaction is shown in **Table 6**.

In **Figure 7B**, we can see from the energy change curve that the forward reaction barrier of this reaction is 0.31 eV. Besides, the energy of the final state 04 is only inferior to that of the saddle point 03, which is 0.30 eV higher than that of the initial state 00. It further indicates that the passivation of dangling bond defect by H_2O is more stable than that by hydroxyl group for the reaction of H_2O and P_{bd1} defect with AOD configuration at random position 3.

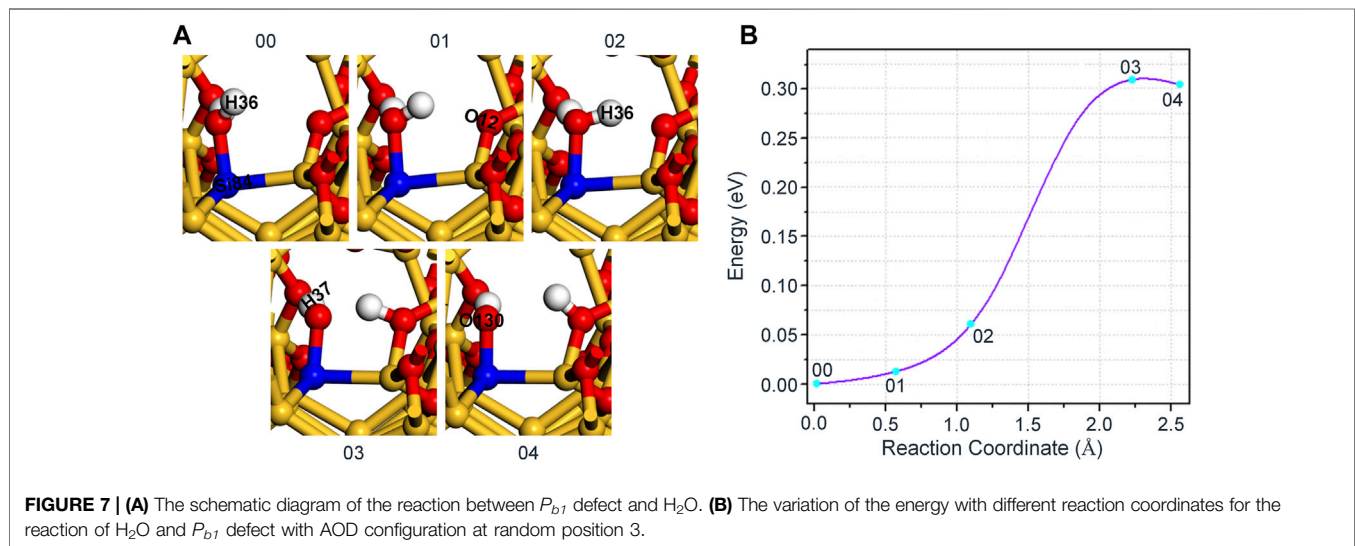
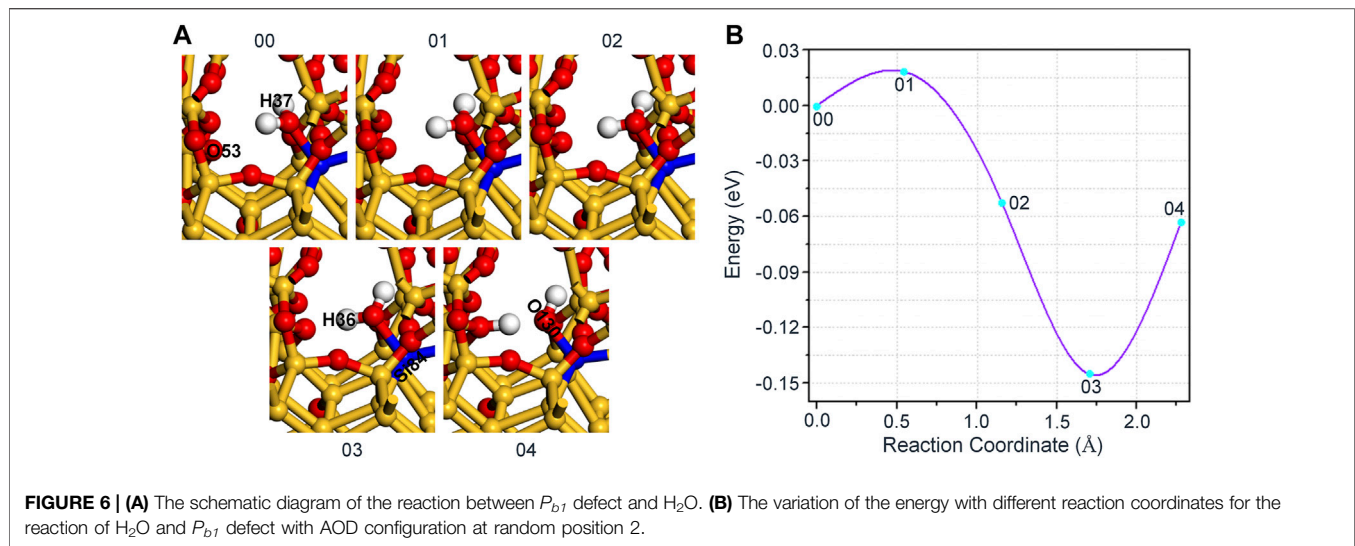


TABLE 6 | Changes in bond length for the reaction of H₂O and P_{b1} defect with AOD configuration at random position 3.

Bond length (Å)	00	01	02	03	04
Si84-O130	1.811	1.826	1.836	1.724	1.691
H36-O130	0.985	0.987	1.020	1.522	2.140
H37-O130	1.081	1.096	1.060	1.009	0.994
H36-O12	2.694	2.136	1.645	1.056	0.984

4 CONCLUSION

In summary, we have systematically simulated and analyzed the interaction of H₂O and interface defects in a-SiO₂/Si(100) by using the first-principles method. The reaction barrier is about 0.4 ~ 0.7 eV for the reaction of H₂O and P_{b0} defect, and the hydroxyl

group is more liable to passivate dangling bond defect than H₂O. Conversely, for the reaction of H₂O and P_{b1} defect, the passivation of dangling bond defect by H₂O is more stable than that by hydroxyl group, which can be attributed to the position of P_{b1} defect closer to the interface than that of P_{b0} defect. And there has been a precedent that high-pressure H₂O vapor heating can passivate the a-SiO₂/Si interface experimentally (Sakamoto and Sameshima, 2000). Our study offers a convective theoretical guidance for the interaction of H₂O and interface defects in a-SiO₂/Si(100).

DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding author.

AUTHOR CONTRIBUTIONS

All authors listed have made a substantial, direct, and intellectual contribution to the work and approved it for publication.

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SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fmats.2022.894097/full#supplementary-material>

REFERENCES

- Batyrev, I. G., Tuttle, B., Fleetwood, D. M., Schrimpf, R. D., Tsetseris, L., and Pantelides, S. T. (2008). Reactions of Water Molecules in Silica-Based Network Glasses. *Phys. Rev. Lett.* 100 (10), 105503. doi:10.1103/physrevlett.100.105503
- Blöchl, P. E. (1994). Projector Augmented-Wave Method. *Phys. Rev. B* 50 (24), 17953. doi:10.1103/PhysRevB.50.17953
- Blöchl, P. E., and Stathis, J. H. (1999). Hydrogen Electrochemistry and Stress-Induced Leakage Current in Silica. *Phys. Rev. Lett.* 83 (2), 372. doi:10.1103/PhysRevLett.83.372
- Bourg, I. C., and Steefel, C. I. (2012). Molecular Dynamics Simulations of Water Structure and Diffusion in Silica Nanopores. *J. Phys. Chem. C* 116 (21), 11556–11564. doi:10.1021/jp301299a
- Campbell, J. P., and Lenahan, P. M. (2002). Density of States of P_{b1} Si/SiO₂ Interface Trap Centers. *Appl. Phys. Lett.* 80 (11), 1945–1947. doi:10.1063/1.1461053
- Chadi, D. J. (2001). Intrinsic and H-Induced Defects at Si-SiO₂ Interfaces. *Phys. Rev. B* 64 (19), 195403. doi:10.1103/physrevb.64.195403
- Gerardi, G. J., Poindexter, E. H., Caplan, P. J., and Johnson, N. M. (1986). Interface Traps and P_b Centers in Oxidized (100) Silicon Wafers. *Appl. Phys. Lett.* 49 (6), 348–350. doi:10.1063/1.97611
- Griscom, D. L. (1991). Optical Properties and Structure of Defects in Silica Glass. *J. Ceram. Soc. Jpn.* 99 (1154), 923–942. doi:10.2109/jcersj.99.923
- Hafner, J. (2008). Ab-initio simulations of Materials Using VASP: Density-Functional Theory and beyond. *J. Comput. Chem.* 29 (13), 2044–2078. doi:10.1002/jcc.21057
- Haneji, N., Vishnubhotla, L., and Ma, T. P. (1991). Possible Observation of P_{b0} and P_{b1} Centers at Irradiated (100)Si/SiO₂ Interface from Electrical Measurements. *Appl. Phys. Lett.* 59 (26), 3416–3418. doi:10.1063/1.105693
- Helms, C. R., and Poindexter, E. H. (1994). The Silicon-Silicon Dioxide System: Its Microstructure and Imperfections. *Rep. Prog. Phys.* 57 (8), 791–852. doi:10.1088/0034-4885/57/8/002
- Henkelman, G., Uberuaga, B. P., and Jónsson, H. (2000). A Climbing Image Nudged Elastic Band Method for Finding Saddle Points and Minimum Energy Paths. *J. Chem. Phys.* 113 (22), 9901–9904. doi:10.1063/1.1329672
- Kajihara, K., Kamioka, H., Hirano, M., Miura, T., Skuja, L., and Hosono, H. (2005). Interstitial Oxygen Molecules in Amorphous SiO₂. III. Measurements of Dissolution Kinetics, Diffusion Coefficient, and Solubility by Infrared Photoluminescence. *J. Appl. Phys.* 98 (1), 013529. doi:10.1063/1.1943506
- Kato, K., Yamasaki, T., and Uda, T. (2006). Origin of P_{b1} Center at SiO₂/Si(100) Interface: First-Principles Calculations. *Phys. Rev. B* 73 (7), 073302. doi:10.1103/physrevb.73.073302
- Kosowsky, S. D., Pershan, P. S., Krisch, K. S., Bevk, J., Green, M. L., Brasen, D., et al. (1997). Evidence of Annealing Effects on a High-Density Si/SiO₂ Interfacial Layer. *Appl. Phys. Lett.* 70 (23), 3119–3121. doi:10.1063/1.119090
- Kovačević, G., and Pivac, B. (2014). Structure, Defects, and Strain in Silicon-Silicon Oxide Interfaces. *J. Appl. Phys.* 115 (4), 043531. doi:10.1063/1.4862809
- Lenahan, P. M., and Conley, J. F., Jr (1998). What Can Electron Paramagnetic Resonance Tell Us about the Si/SiO₂ System. *J. Vac. Sci. Technol. B* 16 (4), 2134–2153. doi:10.1116/1.590301
- Li, J., Wu, J., Zhou, C., Han, B., Karwacki, E. J., Xiao, M., et al. (2009). On the Dissociative Chemisorption of Tris(dimethylamino)silane on Hydroxylated SiO₂(001) Surface. *J. Phys. Chem. C* 113 (22), 9731–9736. doi:10.1021/jp900119b
- Li, P., Chen, Z., Yao, P., Zhang, F., Wang, J., Song, Y., et al. (2019). First-Principles Study of Defects in Amorphous-SiO₂/Si Interfaces. *Appl. Surf. Sci.* 483, 231–240. doi:10.1016/j.apsusc.2019.03.216
- Li, P., Song, Y., and Zuo, X. (2019). Computational Study on Interfaces and Interface Defects of Amorphous Silica and Silicon. *Phys. Status Solidi RRL* 13 (3), 1800547. doi:10.1002/pssr.201800547
- Lu, Z. H., Graham, M. J., Jiang, D. T., and Tan, K. H. (1993). SiO₂/Si(100) Interface Studied by Al K α x-ray and Synchrotron Radiation Photoelectron Spectroscopy. *Appl. Phys. Lett.* 63 (21), 2941–2943. doi:10.1063/1.110279
- Matsudo, T., Ohta, T., Yasuda, T., Nishizawa, M., Miyata, N., Yamasaki, S., et al. (2002). Observation of Oscillating Behavior in the Reflectance Difference Spectra of Oxidized Si(001) Surfaces. *J. Appl. Phys.* 91 (6), 3637–3643. doi:10.1063/1.1452764
- McLean, F. B. (1980). A Framework for Understanding Radiation-Induced Interface States in SiO₂ MOS Structures. *IEEE Trans. Nucl. Sci.* 27 (6), 1651–1657. doi:10.1109/tns.1980.4331084
- Michalske, T. A., and Freiman, S. W. (1983). A Molecular Mechanism for Stress Corrosion in Vitreous Silica. *J. Am. Ceram. Soc.* 66 (4), 284–288. doi:10.1111/j.1151-2916.1983.tb15715.x
- Monkhorst, H. J., and Pack, J. D. (1976). Special Points for Brillouin-Zone Integrations. *Phys. Rev. B* 13 (12), 5188–5192. doi:10.1103/physrevb.13.5188
- Pamungkas, M. A., Joe, M., Kim, B.-H., and Lee, K.-R. (2011). Reactive Molecular Dynamics Simulation of Early Stage of Dry Oxidation of Si(100) Surface. *J. Appl. Phys.* 110 (5), 053513. doi:10.1063/1.3632968
- Pantelides, S. T., Tsetseris, L., Rashkeev, S. N., Zhou, X. J., Fleetwood, D. M., and Schrimpf, R. D. (2007). Hydrogen in MOSFETs-A Primary Agent of Reliability Issues. *Microelectron. Reliab.* 47 (6), 903–911. doi:10.1016/j.microrel.2006.10.011
- Parr, R. G. (1983). Density Functional Theory. *Annu. Rev. Phys. Chem.* 34 (1), 631–656. doi:10.1146/annurev.pc.34.100183.003215
- Perdew, J. P., Burke, K., and Ernzerhof, M. (1996). Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* 77 (18), 3865–3868. doi:10.1103/physrevlett.77.3865
- Perez-Beltran, S., Ramírez-Caballero, G. E., and Balbuena, P. B. (2015). First-Principles Calculations of Lithiation of a Hydroxylated Surface of Amorphous Silicon Dioxide. *J. Phys. Chem. C* 119 (29), 16424–16431. doi:10.1021/acs.jpcc.5b02992
- Pfeffer, R., and Ohring, M. (1981). Network Oxygen Exchange during Water Diffusion in SiO₂. *J. Appl. Phys.* 52 (2), 777–784. doi:10.1063/1.328762
- Poindexter, E. H., Caplan, P. J., Deal, B. E., and Razouk, R. R. (1981). Interface States and Electron Spin Resonance Centers in Thermally Oxidized (111) and (100) Silicon Wafers. *J. Appl. Phys.* 52 (2), 879–884. doi:10.1063/1.328771
- Rochet, F., Poncey, C., Dufour, G., Roulet, H., Guillot, C., and Sirotti, F. (1997). Suboxides at the Si/SiO₂ Interface: A Si2p Core Level Study with Synchrotron Radiation. *J. Non-Cryst. Solids* 216, 148–155. doi:10.1016/s0022-3093(97)00181-6

- Sakamoto, K., and Sameshima, T. (2000). Passivation of SiO₂/Si Interfaces Using High-Pressure-H₂O-Vapor Heating. *Jpn. J. Appl. Phys.* 39, 2492–2496. doi:10.1143/jjap.39.2492
- Schwank, J. R., Shaneyfelt, M. R., Fleetwood, D. M., Felix, J. A., Dodd, P. E., Paillet, P., et al. (2008). Radiation Effects in MOS Oxides. *IEEE Trans. Nucl. Sci.* 55 (4), 1833–1853. doi:10.1109/tns.2008.2001040
- Sheikhholeslam, S. A., Manzano, H., Grecu, C., and Ivanov, A. (2016). Reduced Hydrogen Diffusion in Strained Amorphous SiO₂: Understanding Ageing in MOSFET Devices. *J. Mat. Chem. C* 4 (34), 8104–8110. doi:10.1039/c6tc02647h
- Sheppard, D., Xiao, P., Chemelewski, W., Johnson, D. D., and Henkelman, G. (2012). A Generalized Solid-State Nudged Elastic Band Method. *J. Chem. Phys.* 136 (7), 074103. doi:10.1063/1.3684549
- Stesmans, A., and Afanas'ev, V. V. (1998). Electron Spin Resonance Features of Interface Defects in Thermal (100)Si/SiO₂. *J. Appl. Phys.* 83 (5), 2449–2457. doi:10.1063/1.367005
- Stesmans, A. (1996). Passivation of P_{b0} and P_{b1} Interface Defects in Thermal (100) Si/SiO₂ with Molecular Hydrogen. *Appl. Phys. Lett.* 68 (15), 2076–2078. doi:10.1063/1.116308
- Stirling, A., Pasquarello, A., Charlier, J.-C., and Car, R. (2000). Dangling Bond Defects at Si-SiO₂ Interfaces: Atomic Structure of the P_{b1} Center. *Phys. Rev. Lett.* 85 (13), 2773–2776. doi:10.1103/physrevlett.85.2773
- Takahashi, J. I., Machida, K., Shimoyama, N., and Minegishi, K. (1993). Water Trapping of Point Defects in Interlayer SiO₂ Films and its Contribution to the Reduction of Hot-Carrier Degradation. *Appl. Phys. Lett.* 62 (19), 2365–2366. doi:10.1063/1.109391
- Thoan, N. H., Keunen, K., Afanas'ev, V. V., and Stesmans, A. (2011). Interface State Energy Distribution and P_b Defects at Si(110)/SiO₂ Interfaces: Comparison to (111) and (100) Silicon Orientations. *J. Appl. Phys.* 109 (1), 013710. doi:10.1063/1.3527909
- Tomozawa, M. (1985). Concentration Dependence of the Diffusion Coefficient of Water in SiO₂ Glass. *J. Am. Ceram. Soc.* 68 (9), C-251-C-252. doi:10.1111/j.1151-2916.1985.tb15804.x
- Watanabe, H., Kato, K., Uda, T., Fujita, K., Ichikawa, M., Kawamura, T., et al. (1998). Kinetics of Initial Layer-By-Layer Oxidation of Si(001) Surfaces. *Phys. Rev. Lett.* 80 (2), 345–348. doi:10.1103/physrevlett.80.345
- Yamasaki, T., Kato, K., and Uda, T. (2003). Oxidation of the Si(001) Surface: Lateral Growth and Formation of P_{b0} Centers. *Phys. Rev. Lett.* 91 (14), 146102. doi:10.1103/physrevlett.91.146102
- Yeon, J., and Van Duin, A. C. T. (2016). ReaxFF Molecular Dynamics Simulations of Hydroxylation Kinetics for Amorphous and Nano-Silica Structure, and its Relations with Atomic Strain Energy. *J. Phys. Chem. C* 120 (1), 305–317. doi:10.1021/acs.jpcc.5b09784

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