



# Hall Effect Study of the Metamagnetic Transition in the $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$ Nanosheet

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## OPEN ACCESS

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### Specialty section:

This article was submitted to  
Quantum Materials,  
a section of the journal  
Frontiers in Materials

Received: 16 January 2022

Accepted: 31 January 2022

Published: 11 March 2022

### Citation:

Wan J, Liu Y, Wan Y, Wu Q, Wang Y,  
Yang J, Mao Z and Wang J (2022) Hall  
Effect Study of the Metamagnetic  
Transition in the  
 $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  Nanosheet.  
Front. Mater. 9:856000.  
doi: 10.3389/fmats.2022.856000

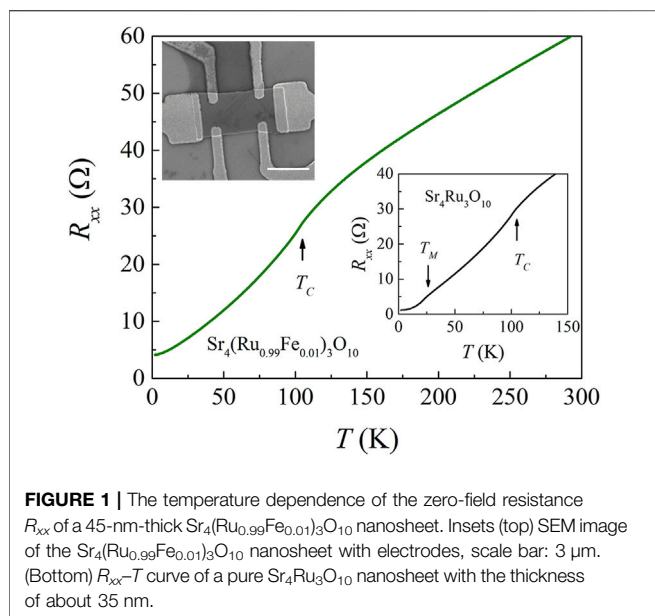
$\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  shows a ferromagnetic (FM) transition at  $T_C \sim 105$  K with the magnetic easy axis in the  $ab$  plane, followed by a metamagnetic transition (MMT) at low temperatures when the magnetic field  $H$  is applied along the  $c$  axis, which is in sharp contrast to that of the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$ , where the easy axis is along the  $c$  axis and the MMT is in the  $ab$  plane. Here, we studied the MMT in the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet by the Hall effect. It was found that the ordinary Hall coefficient of  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  is almost the same as that of the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$ , while a sudden increase in the Hall resistance  $R_{xy}$  is observed below  $\sim 50$  K, above which the  $R_{xy}$  presents the conventional anomalous Hall effect up to  $T_C$ . Analysis of the results indicates that the MMT has no direct correlation to the electronic structure but closely relates to the magnetic moment locking, where the magnetic-field-induced breakdown of the locked moments is responsible for the MMT.

**Keywords:** metamagnetic transition, Hall effect,  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$ , magnetic moment locking, ferromagnetic transition

## INTRODUCTION

The  $4d$ -electron-based perovskite ruthenium oxides  $\text{Sr}_{n+1}\text{Ru}_n\text{O}_{3n+1}$  ( $n = 1, 2, \infty$ ) with a strong inherent spin orbital coupling present rich and fascinating phenomena, such as unconventional superconductivity ( $\text{Sr}_2\text{RuO}_4$ ,  $n = 1$ ) (Mackenzie and Maeno, 2003), magnetic-field-tuned metamagnetic quantum criticality ( $\text{Sr}_3\text{Ru}_2\text{O}_7$ ,  $n = 2$ ) (Grigera et al., 2001), and itinerant ferromagnetism ( $\text{SrRuO}_3$ ,  $n = \infty$ ) (Koster et al., 2012). In contrast, the  $n = 3$  member  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  is an exotic ferromagnetic (FM) metal (Crawford et al., 2002). It possesses an orthorhombic unit cell composed of triple layers of corner-shared  $\text{RuO}_6$  octahedra separated by double rock-salt Sr-O layers. The  $\text{RuO}_6$  octahedra in the outer two layers of each triple layer are rotated by an average of  $5.25^\circ$  around the  $c$ -axis, while the octahedra of the inner layers are rotated in the opposite sense by an average of  $10.6^\circ$  (Crawford et al., 2002).  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  undergoes an FM transition at a Curie temperature  $T_C \sim 105$  K with the easy axis along the  $c$  axis, followed by a second transition at a critical temperature  $T_M \sim 50$  K (Crawford et al., 2002; Cao et al., 2003). Below  $T_M$ , a superlinear increase of magnetization, i.e., a metamagnetic transition (MMT), is observed when the magnetic field is applied in the  $ab$  plane (Cao et al., 2003).

The MMT in  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  has been widely investigated, but its nature remains elusive (Lin et al., 2004; Gupta et al., 2006; Mao et al., 2006; Jo et al., 2007; Schottenhamel et al., 2016). The magnetization and Raman spectrum studies suggest that the MMT is associated with a magnetic-field-induced breakdown of the antiferromagnetic (AFM) order in the  $ab$  plane below



$T_M$  through a spin-flip process (Cao et al., 2003; Gupta et al., 2006). However, no AFM order in the  $ab$  plane is detected by neutron diffraction measurements (Granata et al., 2013; Zhu et al., 2018). On the other hand, systematic transport studies on the  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  nanosheets show that the MMT is likely an in-plane magnetic-field-induced spin-flop process from the  $c$  axis to the  $ab$  plane (Liu et al., 2018).

Interestingly, when the Ru is substituted by a small amount of Fe [ $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$ ], the magnetic behavior is changed significantly (Liu et al., 2019). It is found that the magnetic easy direction is fully changed from the  $c$  axis to the  $ab$  plane below  $T_C$  and that the second transition at  $T_M$ , which usually occurs in pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  crystals, is not seen though the FM transition remains at 105 K. In addition, an MMT is observed when the magnetic field is applied along the  $c$  direction. The Fe doping opens a path to explore the nature of the MMT by using  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  as a model system. Here, we perform a systematic Hall effect measurement of the MMT in the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet, which contains both the ordinary Hall effect and the anomalous Hall effect. Results show that the MMT has less correlation to the electronic structure but depends on the magnetic moment locking.

## MATERIALS AND METHODS

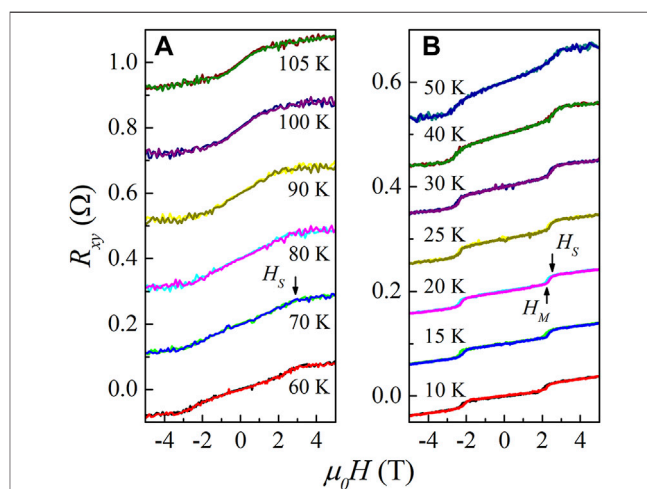
The  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  single crystals were grown by the floating zone method as described in Liu et al. (2019). The starting materials were  $\text{SrCO}_3$ ,  $\text{Fe}_2\text{O}_3$ , and  $\text{RuO}_2$ . Twenty-five percent excess  $\text{RuO}_2$  was added to compensate the evaporation of  $\text{RuO}_2$  from the melting zone. The mixed powder was ground for  $\sim 1$  h, and then pressed into pellets, after which, the pellets were sintered at a temperature of about  $900^\circ\text{C}$  for 12 h. The sintered pellets were ground for  $\sim 1$  h again, and the ground powder was pressed to a rod and sintered at a temperature of  $1,350^\circ\text{C}$  for 4 h.

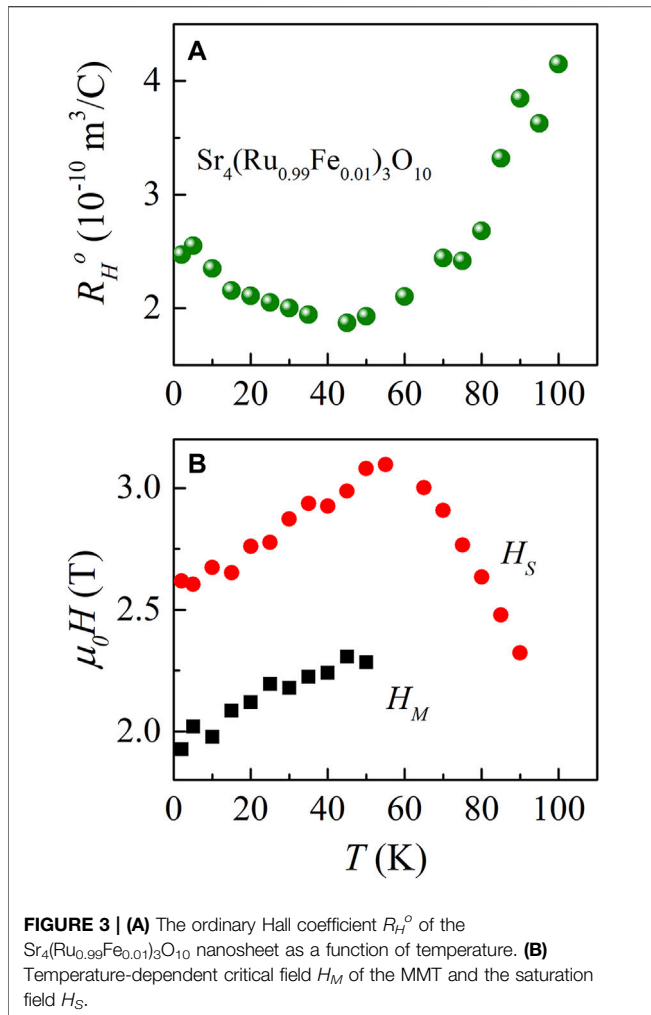
The pressure in the quartz tube was  $\sim 10$  bar (10%  $\text{O}_2$  + 90% Ar), and the growth speed was about  $\sim 15$  mm/h.

The concentration of Fe was analyzed roughly by energy-dispersive spectrum (EDS, Oxford Instruments).  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheets were obtained by the scotch tape-based micro-mechanical exfoliation method from the bulk single crystal and then transferred onto a  $\text{SiO}_2$  (300 nm)/Si substrate. Contacts were patterned using the electron-beam lithography technique followed by deposition of Ti/Au (5/100 nm). The scanning electron microscope (SEM) image of a device with patterned electrodes is shown in the top inset of **Figure 1**. The thickness of the nanosheet is about 45 nm, which is determined by the atomic force microscopy. The transport property of the nanosheet was measured by a physical property measurement system (PPMS, Quantum Design). The Hall resistance  $R_{xy}$  is determined from  $R_{xy} = [R_{xy}(H) - R_{xy}(-H)]/2$  in order to subtract the component of the longitudinal resistance  $R_{xx}$  arising from the small misalignment of transverse contacts.

## RESULTS AND DISCUSSION

**Figure 1** shows the temperature  $T$ -dependent longitudinal resistance  $R_{xx}$  of the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet, which is measured at zero magnetic field. It is found that the  $R_{xx}$  decreases monotonously as the temperature decreases, indicating a metallic behavior of the nanosheet. An anomaly of the resistance near  $T_C \sim 105$  K can be identified due to the FM transition, but no second anomaly at  $T_M$  is seen, which is in sharp contrast to that observed in the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  bulk or nanosheets, where two resistive anomalies at  $T_C$  and  $T_M$  can be found (Liu et al., 2018; Liu et al., 2017). For a comparison, the  $R_{xx}$ - $T$  property of a pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  nanosheet with thickness of about 35 nm is presented in the bottom inset of **Figure 1**. Note that the  $T_M$  of the pure nanosheet





is about 25 K, which is smaller than that of the pure bulk due to the size effect (Liu et al., 2016).

**Figure 2** shows the magnetic field  $H$ -dependent Hall resistance  $R_{xy}$  of the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet measured at various temperatures, where  $H$  is applied perpendicular to the  $ab$  plane of the nanosheet. It is found that all the  $R_{xy}$ - $H$  curves below  $T_C \sim 105$  K can be well described by Nagaosa et al. (2010)

$$R_{xy} = R_{xy}^o + R_{xy}^a \quad (1)$$

The  $R_{xy}^o$  is the ordinary Hall resistance determined by the electronic structure and can be written as  $R_{xy}^o = R_H^o H/t$ , with  $R_H^o$  and  $t$  being the ordinary Hall coefficient and the thickness of the sample, respectively;  $R_{xy}^a$  is the anomalous Hall resistance, which is proportional to the magnetization  $M_c$  along the  $c$  axis, i.e.,  $R_{xy}^a = R_H^a M_c/t$ , with  $R_H^a$  being the anomalous Hall coefficient. Therefore, the Hall effect measurement is a powerful tool to measure both the electronic structure and the magnetization  $M_c$  of the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet. It is seen that

(1) The  $R_{xy}$  increases almost linearly as  $H$  increases in the temperature range of  $T^* \sim 50$  K  $< T < T_C$  (**Figure 2A**) and then turns to saturation at a magnetic field  $H_S$ . This anomalous Hall

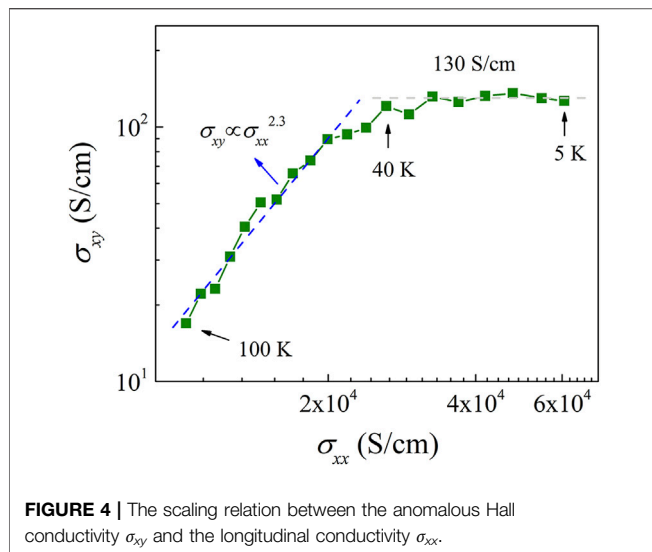
effect is very similar to that observed in many other FM materials (Nagaosa et al., 2010). Since the magnetic easy axis of  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  is in the  $ab$  plane, this behavior indicates that the magnetic moments at  $T^* < T < T_C$  are rotated collectively from the  $ab$  plane to the  $c$  axis by  $H$ .

(2) At  $T < T^*$ , the  $R_{xy}$  first increases linearly as the magnetic field increases and then followed by a sharp increase at a magnetic field  $H_M$ , which is slightly below the saturation field  $H_S$  as indicated in **Figure 2B**. Above  $H_S$ , the  $R_{xy}$  increases linearly. According to **Eq. 1**, this behavior demonstrates that there is a rapid increase of  $M_c$  at  $H_M < H < H_S$ , i.e., an MMT in the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet as the sweeping of  $H$  along the  $c$  direction, which is in contrast to that of the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$ , where the MMT is in the  $ab$  plane (Cao et al., 2003). The  $H_M$  can thus be defined as the critical field of the MMT.

To better understand the MMT, we firstly focus on the ordinary Hall effect of the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet. **Figure 3A** shows the ordinary Hall coefficient  $R_H^o$  as a function of temperature, which is extracted from the high field slope of the Hall isotherm at  $H = 5$  T ( $> H_S$ ). The  $R_H^o$  is positive in the whole temperature range, indicating that the dominant carriers in the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet are holes. Except for the  $R_H^o$  of the  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  nanosheet at low temperatures being slightly smaller, the trace of the  $R_H^o$ - $T$  curve here is very similar to that reported in the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  nanosheet (Liu et al., 2016). Considering that the FM moment in  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  is in the  $ab$  plane and the MMT is along the  $c$  direction, which is completely different to that in the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$ , where the FM moment is along the  $c$  direction and the MMT is in the  $ab$  plane, the almost identical  $R_H^o$  indicates that the MMT has no direct correlation to the electronic structure, but only the spin-flop from the  $ab$  plane to the  $c$  direction.

To give an insight of the MMT, we reinspect the Hall data shown in **Figure 2**. One can see that at 10 K, the slope of the  $R_{xy}$ - $H$  curve below  $H_M$  is almost the same as that at  $H > H_S$ . The change of  $R_{xy}$  above  $H_S$  is purely due to the orbital effect of  $H$  on carriers, i.e., the ordinary Hall resistance  $R_{xy}^o$ , while that below  $H_M$  is the sum of both the change of  $R_{xy}^o$  and  $R_{xy}^a$ . The almost identical slope of the  $R_{xy}$ - $H$  curve at  $H < H_M$  and  $H > H_S$  indicates that the  $R_{xy}^a$  is about zero (where  $M_c = 0$ ) at  $H < H_M$ . In other words, the magnetic moments are fully “locked” without any net moment along the  $c$  direction below  $H_M$ . As the  $T$  increases, the slope of the  $R_{xy}$ - $H$  curve at  $H < H_M$  becomes larger than that at  $H > H_S$ , indicating that some magnetic moments are unlocked due to the thermal activation. At  $T > T^* \sim 50$  K, all the magnetic moments become a collective rotation under the magnetic field as mentioned above; hence, the MMT vanishes. This result indicates that the spin-flop is associated with a magnetic-field-induced breakdown of the locking moments, which is responsible for the MMT.

An unexpected feature seen in **Figure 2** is that both  $H_M$  and  $H_S$  decrease with decreasing  $T$  below  $T^*$ , which means intuitively that it is easier to align the magnetic moments along the  $c$  direction with decreasing temperature by the magnetic field. To see clearly, the temperature dependence of  $H_M$  and  $H_S$  is shown in **Figure 3B**. This behavior is in contrast to that of conventional magnetic materials, in which the magnetic moments are usually difficult to



polarize to the magnetic hard axis at lower temperatures due to the reduction of thermal fluctuation.

We noticed that the  $T$ -dependent feature of the lattice parameter  $c$  of the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$  shows a minimum at  $T^*$  (Granata et al., 2013; Schottenhamel et al., 2016), indicating a slightly negative expansion below  $T^*$ . This negative expansion nicely echoes the anomaly in  $H_M$  and  $R_H^o$ . As known previously (Liu et al., 2017; Liu et al., 2018), the shrinkage of the  $c$  axis will reduce the magnetocrystalline anisotropy energy and favors the magnetic moment off the  $c$  axis, such as the Fe-doping effect (Liu et al., 2019). The tiny expansion of the  $c$  axis below  $T^*$  will favor the FM moment off the  $ab$  plane, resulting in the decrease of the pinning force for moment locking in the  $ab$  plane. The critical magnetic field  $H_M$  is actually the competition between the moment locking force below  $T^*$  and the  $c$  axis expansion-induced depinning effect. This is the reason that the  $H_M$  decreases with decreasing  $T$ .

Finally, as a metamagnetic metal of  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$ , we would like to present the scaling relation between the anomalous Hall conductivity  $\sigma_{xy}$  and the longitudinal conductivity  $\sigma_{xx}$  of  $\sigma_{xy} \propto \sigma_{xx}^\varphi$ , where  $\varphi$  is the scaling exponent, which is shown in **Figure 4**. The  $\sigma_{xy}$  is determined by  $\sigma_{xy} = \rho_{xy}^a / [(\rho_{xx})^2 + (\rho_{xy}^a)^2]$  with  $\rho_{xx}$  being the longitudinal resistivity and  $\rho_{xy}^a$  the anomalous Hall resistivity at zero field, which is obtained by extrapolating the high-field linear term of the Hall data to the zero field shown in **Figure 2**. It is found that at  $T > 60$  K,  $\varphi = 2.3$ , indicating that the side jump mechanism is dominant (Nagaosa et al., 2010). Below 40 K,  $\varphi = 0$  with  $\sigma_{xy} = 130$  S/cm, which can be attributed to the

intrinsic dissipationless topological Berry-phase contribution (Nagaosa et al., 2010). The scaling relation is almost the same as many conventional FM materials (Nagaosa et al., 2010), indicating a regular FM nature of  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  after being polarized.

## CONCLUSION

We have investigated the MMT of  $\text{Sr}_4(\text{Ru}_{0.99}\text{Fe}_{0.01})_3\text{O}_{10}$  along the  $c$  axis by Hall effect. Results show that the ordinary Hall coefficient is almost the same as that of the pure  $\text{Sr}_4\text{Ru}_3\text{O}_{10}$ . The magnetic moments are found to be fully locked without any net moments along the  $c$  direction at the ground state at 10 K and then gradually unlocked with increasing  $T$ . At about 50 K, the magnetic moments are fully depinned, and the MMT smears out. Our result indicates that the MMT has less correlation to the electronic structure but closely relates to the magnetic moment locking.

## DATA AVAILABILITY STATEMENT

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding author.

## AUTHOR CONTRIBUTIONS

JW, YL, YW, and QW carried out the experiments of EBL and transport measurement. YW and ZM synthesized the sample. JW, JY, and JW wrote and reviewed the manuscript with the help of all authors.

## FUNDING

This work was supported by the Fundamental Research Funds for the Central Universities (grant no. SWU019010), the Chongqing Municipal Program of Innovation and Entrepreneurship for Undergraduates (grant no. S202010635003), and the National Natural Science Foundation of China (grants nos. 11674323, 11804297, and U19A2093). The single-crystal growth effort was supported by the U.S. Department of Energy under EPSCoR Grant No. DE-SC0012432 with additional support from the Louisiana Board of Reagents.

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