



Boosting Oxygen Reduction Performance of Manganese Oxide in Alkaline Media by Three-Dimensional Highly Ordered Conductive Porous Framework

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Xiao X, Fang Z and Yu D (2019) Boosting Oxygen Reduction Performance of Manganese Oxide in Alkaline Media by Three-Dimensional Highly Ordered Conductive Porous Framework. Front. Mater. 6:219. doi: 10.3389/fmats.2019.00219 MnO₂ has been widely used as an alternative candidate for oxygen reduction reaction owing to its abundance, low cost, and environmental compatibility. However, bulk MnO₂ as electrocatalysts is still suffering from serious active sites agglomeration and poor conductivity, resulting in low utilization of catalysts and sluggish reaction kinetics. In view of this, we fabricated a 3D highly ordered porous MnO₂@Ni-pc nanocomposite and significantly enhance ORR performance of MnO₂ with high onset potential of 1.04 V and half-wave potential of 0.89 V as well as a large limiting current density of 4.07 mA cm⁻², being even comparable to the commercial Pt/C. This nano-engineering technique is suitable for various catalysts that can be attached onto the 3D highly ordered porous framework, which provides new opportunities in designing electrode structures to boost the performance of bulk materials.

Keywords: manganese oxide, electrocatalyst, oxygen reduction reaction, nanocomposite, 3D ordered porous scaffold

INTRODUCTION

Electrocatalysts for the oxygen reduction reaction (ORR) are of vital importance for many renewable energy applications such as metal-air batteries and fuel cells (Bashyam and Zelenay, 2006; Lefèvre et al., 2009; Wu et al., 2011). Noble Pt-based materials have shown the most efficient ORR performance but restricted severely by high cost, poor stability, and scarcity (Jaouen et al., 2011; Morozan et al., 2011b). Hence, a variety of catalysts based on non-precious metal oxides (Co_3O_4 , FeO_x , and MnO_x etc.) have been widely investigated as alternative candidates for ORR owing to their abundance, low cost and environmental compatibility (Zhou et al., 2011). Nonetheless, bulk metal oxide catalysts as electrocatalysts are always suffering from some problems such as poor conductivity and serious agglomeration of active sites. Recently, many researches have demonstrated that there are various methods to effectively boost the performance of bulk metal oxide electrocatalysts through regulating their chemical components, conductivity, and microstructures. In addition, macrostructure, mechanical properties, and electrocatalyst/substrate contact of practical catalyst electrodes are also very important to achieve the maximum utilization

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of electrocatalysts. Some materials such as graphene (Gong et al., 2013), Ni network (Chang et al., 2011), or carbonaceous nanofoam (Morozan et al., 2011a) have been widely used as conductive substrate to enhance the catalytic performance of electrocatalysts. However, most of reported nanocomposites can only satisfy one or several requirements of ideal electrocatalyst (**Figure 1a**). There are still big challenges to concurrently achieve highly ordered porous nanostructure and efficient electrons/ions pathways for electrocatalysts (Liu et al., 2011; Morozan et al., 2011a).

Due to the long-range periodicity of nanostructure, high volume fraction, and unique optical response behaviors, photonic crystal structures offer a way out for solving difficulties abovementioned. The unique 3D ordered porous frameworks have been adopted for a myriad of applications from thin-film electronic and optoelectronic devices (Wang et al., 2011b) to energy conversion and storage (Li et al., 2011), such as Li-ion batteries (Liu et al., 2011; Wang et al., 2011a), supercapacitors (Wang Y. J. et al., 2011), and photocatalysis etc. The bicontinuous conductive scaffold used as electrode substrate can provide (1) an interconnected electrolyte-filled pore network that enables rapid mass transport; (2) a large electrode surface area for full exposure of active sites; and (3) high electron conductivity (Liu et al., 2011; Wang Y. J. et al., 2011). These merits give us the chance to address the aforementioned electrocatalyst design limitations in one system. Herein, through a welldesigned electrode architecture, we fabricated MnO₂@Ni-pc by using the unique 3D highly ordered conductive scaffold as substrate, which significantly boosted the ORR performance of MnO2 with high onset potential of 1.04 V and half-wave potential of 0.89 V as well as a large limiting current density of 4.07 mA cm⁻², being even comparable to the commercial Pt/C electrocatalyst.

The preparation procedure of the 3D highly ordered bicontinuous porous Ni scaffold (Ni-pc) is schematically illustrated in Scheme 1 (Please see supporting information for details). Typically, the opal template was prepared by vertical growth of polystyrene spheres (PS, 450 nm, Figure 1b) on the surface of conducting glass. Then heating treatment at different temperatures (101, 102, 103, and 104°C. Here 104°C was chosen as the optimum temperature for the following research) for 1 h was carried out to enhance the connection strengths and void spaces between the PS spheres, facilitating Ni electrodeposition (electrodeposition for 1, 2, and 3 h. Here 1 h was chosen as the optimum temperature for following investigation) into the opal template. After removal of PS template, a 3D highly ordered bicontinuous porous Ni scaffold (Figure 2c and Figures S1, S2) with narrow interconnections between two adjacent spherical voids was formed on the conducting glass. Finally, catalytically active material (MnO₂) for ORR was deposited on the Ni-pc by anodic pulse electrodeposition (electrodeposition for 150, 300, 450 segments were carried out, and 300 segments was choosed for the ORR performance study, denoted as MnO2@Ni-pc, Figure 1d and Figure S3).

The thickness of this uniform and 3D bicontinuous Ni-pc was about 10 layers (ca. $2.5\,\mu$ m) as described in Figure 2c. To keep its unique porous structure and obtain both excellent

mechanical and electronic contact between the active material and the conductive framework, deposition time of MnO2 should be carefully controlled (300 segments for 1 h). Relatively short deposition time would give rise to insufficient mass loading of catalyst while prolonged time could cause excessive mass loading, thereby blocking the pores and resulting in poor catalytic performance. As seen from Figure 2d, MnO₂ flakes with a moderate thickness evenly grown on the Ni-pc, which gained a high pore volume fraction. This porous structure of MnO2 @Ni-pc is conducive to sufficient exposure of the active sites of catalysts. Moreover, the nanochannels of Ni-pc remain unblocked during MnO2 deposition process according to the backside SEM image of MnO₂@Ni-pc (Figure 1e and Figure S4), which benefits both the O₂ diffusion and electrolyte transmission. Besides, energy dispersive X-ray (EDX) element analysis (Figure 1f) for MnO2@Ni-pc verified the existence of Mn, Ni, and O elements.

The powder X-ray diffraction (PXRD) (**Figure 2a**) pattern was performed to confirm the crystal structure of MnO₂@Ni-pc. The diffraction pattern for MnO₂@Ni-pc has apparent peaks at 37.1, 39.0, and 65.7° , corresponding to (311), (222), and (440) lattice plane of MnO₂ with *Fd*-3*m*(277) space group (JCPDS 44-0992), respectively. The preferred orientation of the electrodeposited manganese oxide on the highly ordered Ni-pc has led to the disappearance of other crystal peaks. In addition, the broadening of peaks indicates the small particle size/thickness of the asdeposited MnO₂. The sharp peak at 44.5° can be indexed to Ni (011) (JCPDS 45-1027), which is attributed to the Ni-pc scaffold.

Surface analysis of MnO₂@Ni-pc was further carried out using X-ray photoelectron spectroscopy (XPS) (**Figures 2b-d**). The XPS survey (**Figure 2b**) indicated the existence of Mn, Ni, O elements. The magnitude of peak splitting is indicative of the Mn oxidation sate (Wu et al., 2012). The high-resolution spectra of Mn 2p and Mn 2s are shown in **Figures 2c,d**, it can be seen that the relative position of Mn $2p_{3/2}-2p_{1/2}$ (ΔE 2p) doublet and the multiplet splitting width of Mn 3s (ΔE 3s) are 11.8 and 4.52 eV, respectively, which correspond to Mn(IV) in MnO₂ (Zhang et al., 2011).

To evaluate the enhancement effect of unique 3D highly ordered porous Ni-pc toward catalytic activity of MnO₂, ORR performance of MnO2@Ni-pc was measured in N2 or O2saturated 0.1 M KOH (Figure 3). As seen from Figure 3a, CV curves of MnO₂@Ni-pc show no peaks in N₂-saturated electrolyte. However, there exists a well-defined peak potential at about 0.62 V in O₂-saturated electrolyte, which arises from the O₂ reduction reaction (Duan et al., 2013). The linear sweeping voltammograms (LSV) were tested in O2-saturated 0.1 M KOH with a rotating disk electrode (RDE) at 1600 rpm (Figure 3b). For comparison, the bare 3D Ni scaffold (Ni-pc), MnO₂@Ni foil, and 20 wt % Pt/C catalyst were also measured. The LSV curve of $MnO_2@Ni-pc$ shows a half-wave potential (E_{1/2}) of 0.89 V, which is even better than Pt/C (0.88 V). The onset potential and limiting current density at 0.2 V of MnO2@Nipc (1.04 V and 4.92 mA cm⁻²) compared favorably with that of Pt/C (1.05 V and 4.56 mA cm⁻²). Moreover, MnO₂@Ni-pc exhibits much better catalytic performance than Ni-pc (0.64 V



SCHEME 1 | Schematic illustration of the preparation process for MnO₂@Ni-pc.



and 0.89 mA cm $^{-2})$ and MnO2@Ni foil (Figure 2a 0.71 V and 1.65 mA cm $^{-2}).$

The number of electrons transferred is an important parameter to evaluating the promotional role of the unique

structure of MnO₂@Ni-pc for ORR kinetics. Thus, LSV curves are recorded at different rotating rates (ω) from 400 to 1,600 rpm and corresponding Koutecky-Levich plots (K-L plots) are shown in **Figure 3c** (Zhang and Braun, 2012). The electron



FIGURE 2 | (a) PXRD pattern of MnO₂@Ni-pc; (b) XPS survey (0–1,300 eV) of MnO₂@Ni-pc; (c,d) High-resolution XPS spectra of (c) Mn 2p and (d) Mn 3s in MnO₂@Ni-pc.



FIGURE 3 | (a) CV curves of MnO₂@Ni-pc in N₂ and O₂-saturated 0.1 M KOH; (b) LSV curves of various materials at 1,600 rpm with a sweep rate of 10 mV s⁻¹ in O₂-saturated 0.1 M koh; (c) LSV curves of MnO₂@Ni-pc at different rotating speeds; (d) I-t curve of MnO₂@Ni-pc tested at 0.57 V vs. RHE in O₂-saturated 0.1 M KOH.

transfer number (n) per O₂ of MnO₂@Ni-pc according to the K-L equation (**Figure S5**, see Supporting Information for more details) is \sim 3.4 within the potential ranging from 0.2 to 0.5 V,

suggesting a great enhancement of ORR kinetics compared to MnO_2 (3.1, 2.7, and 2.3 for pure α -, β -, and γ -MnO₂, respectively) reported before (Wu et al., 2012; Han et al., 2015;

Li et al., 2015, 2019; Lang et al., 2016; Sun et al., 2016; Deng et al., 2017; Wang et al., 2017). Stability of $MnO_2@Ni$ -pc was further tested under a static potential of 0.57 V (**Figure 3d**). The reaction current density for ORR remains stable over 13 h, which exemplify the excellent structural and electrocatalytic stability of $MnO_2@Ni$ -pc. All these experimental results highlight the critical role of the Ni-pc structure for boosting the ORR activity of MnO_2 . Ni-pc not only provides bicontinuous conductive scaffold for efficient electron and ion transmission path, but also is in favor of sufficient exposure of active sites and O_2 diffusion (**Figures 2c,d**; Wang et al., 2011a).

In summary, a new $MnO_2@Ni$ -pc nanocomposite with 3D bicontinuous and highly ordered nanoporous structure was fabricated and showed substantially enhanced ORR activity compared with bulk $MnO_2@Ni$ foil (**Figure 1a**), which was even comparable to benchmark Pt/C catalyst. The unique structure of $MnO_2@Ni$ -pc is advantageous for exposing more active sites, accelerating oxygen diffusion, while providing efficient ion and electron pathways during the catalytic process. Rather than being confined to MnO_2 and Ni-pc systems, the presented methodology of attaching active component onto 3D highly ordered conductive skeleton is envisaged to be viable toward improving the performance of other materials such as transition metal dichalcogenides for novel electrochemical applications in supercapacitors, photovoltaic and lithium ion batteries.

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DATA AVAILABILITY

All datasets generated or analyzed for this study are included in the manuscript and/or the **Supplementary Files**.

AUTHOR CONTRIBUTIONS

DY conceived the project and designed the experiments. XX and ZF performed the sample preparation, characterization, and electrochemical studies. DY and XX wrote the paper. All authors discussed the results and commented on the manuscript.

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SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: https://www.frontiersin.org/articles/10.3389/fmats. 2019.00219/full#supplementary-material

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Conflict of Interest Statement: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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