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Enhanced non-linear optical properties of porphyrin-based polymers covalently functionalized with graphite phase carbon nitride

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In our work, a flurry of original porphyrin-based polymers covalently functionalized g-C₃N₄ nanohybrids were constructed and nominated as PPorx-g-C₃N₄ (x = 1, 2 and 3) through click chemistry between porphyrin-based polymers with alkyne end-groups [(PPorx-C≡CH (x = 1, 2 and 3)] and azide-functionalized graphitic carbon nitride (g-C₃N₄-N₃). Due to the photoinduced electron transfer (PET) between porphyrin-based polymers [PPorx (x = 1, 2 and 3)] group and graphite phase carbon nitride (g-C₃N₄) group in PPorx-g-C₃N₄ nanohybrids, the PPorx-g-C₃N₄ nanohybrids exhibited better non-linear optical (NLO) performance than the corresponding PPorx-C≡CH and g-C₃N₄-N₃. It found that the imaginary third-order susceptibility (*Im* [χ⁽³⁾]) value of the nanohybrids with different molecular weight (MW) of the pPorx group in the nanohybrids ranged from 2.5 × 10³ to 7.0 × 10³ g mol⁻¹ was disparate. Quite interestingly, the *Im* [χ⁽³⁾] value of the nanohybrid with a pPorx group's MW of 4.2 × 10³ g mol⁻¹ (PPor2-g-C₃N₄) was 1.47 × 10⁻¹⁰ esu, which exhibited the best NLO performance in methyl methacrylate (MMA) of all nanohybrids. The PPorx-g-C₃N₄ was dispersed in polymethyl methacrylate (PMMA) to prepare the composites PPorx-g-C₃N₄/PMMA since PMMA was widely used as an alternative to glass. PPor2-g-C₃N₄/PMMA showed the excellent NLO performance of all nanohybrids with the *Im* [χ⁽³⁾] value of 2.36 × 10⁻¹⁰ esu, limiting threshold of 1.71 J/cm², minimum transmittance of 8% and dynamic range of 1.09 in PMMA, respectively. It suggested that PPorx-g-C₃N₄ nanohybrids were potential outstanding NLO materials.

KEYWORDS

porphyrin-based polymer, graphite phase carbon nitride, nanohybrids, non-linear optical (NLO) materials, optical limiting

1 Introduction

Non-linear optical (NLO) materials have received great attention owing to their tremendous applications in the fields of photonic devices (Andréasson et al., 2011), optical limiting (Wang et al., 2019), optical switches (Wang C. et al., 2021), light converters (Miriya and Mani, 2021), etc. Porphyrins with a unique 18 π electron conjugated structure, good thermal stability and narrow bandgap possess excellent NLO performance (Manjunatha et al., 2020; Ou et al., 2021). Meanwhile, due to the structural adjustability, unique electronic and photophysical properties of porphyrins, porphyrins have broad application in various areas and good prospects (Zhang et al., 2018; Chen et al., 2020; Asghar et al., 2021; Liu and Cheng, 2021). Particularly, the NLO performance of porphyrins could be improved by the flexible modification of peripheral substituents or the hybridization with other materials (Zawadzka et al., 2013; Woller et al., 2016; Hu et al., 2018; Biswal et al., 2019; Samal et al., 2019; Liu J. L. et al., 2020; Liu Z. et al., 2020; Liu and Cheng, 2021; Ramasamy et al., 2022). However, previous research indicated the self-aggregate behavior of porphyrins could lead to the formation of large macro-scale and fractal structures, causing a negative impact in the development of NLO devices in practice (Kalachyova et al., 2014). Therefore, the development of novel porphyrin-based NLO materials remained challenging, which have become one of the hot issues in the NLO field.

Growing research have focused on the porphyrin-based polymer to inhibit the aggregation behavior of porphyrins and enhance the NLO performance in solution (Wang et al., 2022). For example, Qiu et al. used porphyrin as an initiator to prepare Por-PMMA *via* Atom Transfer Radical Polymerization. Due to the steric hindrance of PMMA, the aggregation behavior of porphyrin was effectively inhibited, and its NLO performance was improved in the solvent (Qiu et al., 2013). Du et al. reported the introduction of porphyrin into the main chain of poly (arylene ether sulfone) and a large third order optical susceptibility was obtained (Du et al., 2016). Although the aggregation was effectively reduced, but the decrease of the content of porphyrin in the polymer was decreased with the increase of the molecular weight (MW) of porphyrin-based polymer correspondingly, which has a negative influence on NLO property. Other studies showed that porphyrins and nanomaterials were prepared into nanohybrids, and the NLO performance of the nanohybrids could be improved because of the PET between porphyrins and nanomaterials (Wang et al., 2014; Wang A. et al., 2020; Fu et al., 2022). Consequently, the combination of porphyrin-based polymers and nanomaterials might offer a better solution to further increase the NLO performance of porphyrin-based materials (Wan et al., 2012; Liu et al., 2013; Garg et al., 2017; Liang et al., 2022).

Graphite phase carbon nitride ($g\text{-C}_3\text{N}_4$) is a semiconducting nanomaterial with a stacked conjugated structure (Ji et al., 2017). Due to the medium bandgap, fast electron-hole separation efficiency and high carrier mobility (Alenizi et al., 2019; Zhang and Liang, 2019; Daraie et al., 2021; Vavilapalli et al., 2021), $g\text{-C}_3\text{N}_4$ has recently been utilized for constructing NLO materials. Park et al. prepared nanohybrids by combining metal oxide with $g\text{-C}_3\text{N}_4$, which showed excellent imaginary third-order susceptibility ($Im[\chi^{(3)}]$) in ethanol (Sridharan et al., 2015). By combining Ag quantum dots with $g\text{-C}_3\text{N}_4$, Sridharan et al. observed good NLO properties of the nanohybrid due to the energy transfer and electron transfer mechanisms (Sridharan et al., 2014). The electron transfer phenomenon also existed when porphyrin was combined with $g\text{-C}_3\text{N}_4$. Zhu et al. reported that the remarkable photoinduced electron transfer (PET) process was observed under visible light irradiation by combining porphyrin with $g\text{-C}_3\text{N}_4$ (Zhu et al., 2020). Thus, combining porphyrin with $g\text{-C}_3\text{N}_4$ might be a feasible method to improve NLO performance. However, most of the research on the NLO performance of porphyrins has been conducted in liquid systems, which was not conducive to the practical application (Wang A. et al., 2021; Rohal et al., 2022). Moreover, the direct doping of porphyrins and $g\text{-C}_3\text{N}_4$ in solid matrices would jeopardize NLO performance due to poor dispersion (Qiu et al., 2013). Therefore, it may be feasible to covalently bond $g\text{-C}_3\text{N}_4$ with porphyrin-based polymers to improve the dispersibility and NLO performance in the solid matrix. As far as we know, there was no literature on the porphyrin-based polymers covalently functionalized $g\text{-C}_3\text{N}_4$ nanohybrids for NLO research.

In our previous report, the comb-shaped porphyrin-based polymers [PPorx-C \equiv CH ($x = 1, 2$ and 3)] had been constructed by Reversible Addition-Fragmentation Chain Transfer Polymerization (Liang et al., 2022). In our work, a flurry of original porphyrin-based polymers covalently functionalized $g\text{-C}_3\text{N}_4$ (PPorx- $g\text{-C}_3\text{N}_4$ ($x = 1, 2$ and 3)) nanohybrids were synthesized *via* azide-alkyne click chemistry. The PPorx-C \equiv CH with high grafting density and highly flexible main chains could effectively inhibit the aggregation of porphyrin molecules (Chang et al., 2022). In addition, a significant PET process could occur between porphyrin-based polymer [PPorx ($x = 1, 2$ and 3)] group and $g\text{-C}_3\text{N}_4$ group in the nanohybrids. With the unique structural features of the nanohybrids, the obtained PPorx- $g\text{-C}_3\text{N}_4$ exhibited improved NLO performance through the synergistic effect of the minimal aggregation properties caused by PPorx-C \equiv CH and the PET process between p Porx group and $g\text{-C}_3\text{N}_4$ group. Considering practical use, a series of nanohybrid-doped polymethyl methacrylate (PMMA) composites were constructed through solution casting technology and further research the NLO performance.

This research presents a novel design strategy for preparing high-performance NLO materials.

2 Experimental

2.1 Materials and characterization

Details for materials, synthesis and characterization of $g\text{-C}_3\text{N}_4$, benzoic acid functionalized $g\text{-C}_3\text{N}_4$ ($g\text{-C}_3\text{N}_4\text{-BA}$) and PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites were listed in the Supporting Information. The porphyrin-based polymers PPorx-C \equiv CH were synthesized according to our previous work (PPor1-C \equiv CH: M_n , GPC, 2.7×10^3 g mol $^{-1}$; M_w , MALD, 3.5×10^3 g mol $^{-1}$; M_w/M_n (GPC) = 1.01; PPor2-C \equiv CH: M_n , GPC, 4.2×10^3 g mol $^{-1}$; M_w , MALD, 5.3×10^3 g mol $^{-1}$; M_w/M_n (GPC) = 1.06; PPor3-C \equiv CH: M_n , GPC, 7.0×10^3 g mol $^{-1}$; M_w , MALD, 7.0×10^3 g mol $^{-1}$; M_w/M_n (GPC) = 1.03) (Liang et al., 2022).

2.2 Preparation of azide groups functionalized $g\text{-C}_3\text{N}_4$ ($g\text{-C}_3\text{N}_4\text{-N}_3$)

In a typical procedure, $g\text{-C}_3\text{N}_4\text{-BA}$ (50 mg) and anhydrous DMF (1 ml) were dispersed in SOCl_2 (30 ml) and stirred for 24 h in ice-water bath. The mixture was depressurized to remove residual SOCl_2 to obtain a brown powder. After that, NaN_3 (185 mg, 30 mmol) in anhydrous DMF (30 ml) was added to the above mixture and stirred in ice-water bath for another 24 h. The obtained mixture was centrifuged, washed alternately with deionized water and absolute ethanol to eliminate the sodium salts and residual DMF and then dried under vacuum over night to obtain $g\text{-C}_3\text{N}_4\text{-N}_3$ as a yellow powder (52 mg).

2.3 Preparation of PPorx- $g\text{-C}_3\text{N}_4$ ($x = 1, 2$ and 3)

The PPor1- $g\text{-C}_3\text{N}_4$ was synthesized *via* copper-catalyzed azide-alkyne click chemistry. $g\text{-C}_3\text{N}_4\text{-N}_3$ (5 mg) was dispersed in anhydrous DMF (5 ml). PPor1-C \equiv CH (3 mg, 0.001 mmol), CuBr (0.14 mg, 0.001 mmol) and *N, N, N', N'*-pentamethyldiethylenetriamine (PMDETA, 2 μ l, 0.001 mmol) were added into the above mixture. The dispersion was bubbled with N_2 gas for 30 min and placed in an oil bath at 45°C. After stirring for 8 h, the reaction was precipitated in cold methanol to remove the organic residues. The final product was centrifuged, washed and dried to obtain PPor1- $g\text{-C}_3\text{N}_4$ as a brown powder (6 mg).

PPor2- $g\text{-C}_3\text{N}_4$ and PPor3- $g\text{-C}_3\text{N}_4$ were prepared in a similar way of PPor1- $g\text{-C}_3\text{N}_4$, with PPor2-C \equiv CH (3 mg, 0.7 μ mol) and PPor3-C \equiv CH (3 mg, 0.4 μ mol), respectively.

3 Results and discussion

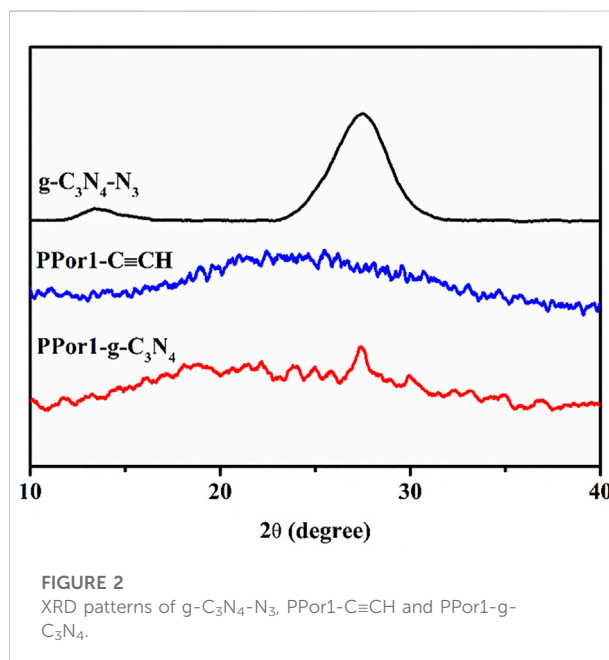
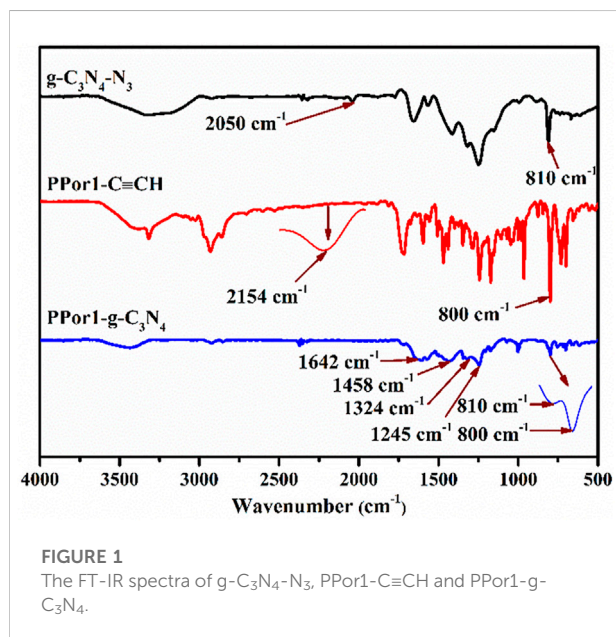
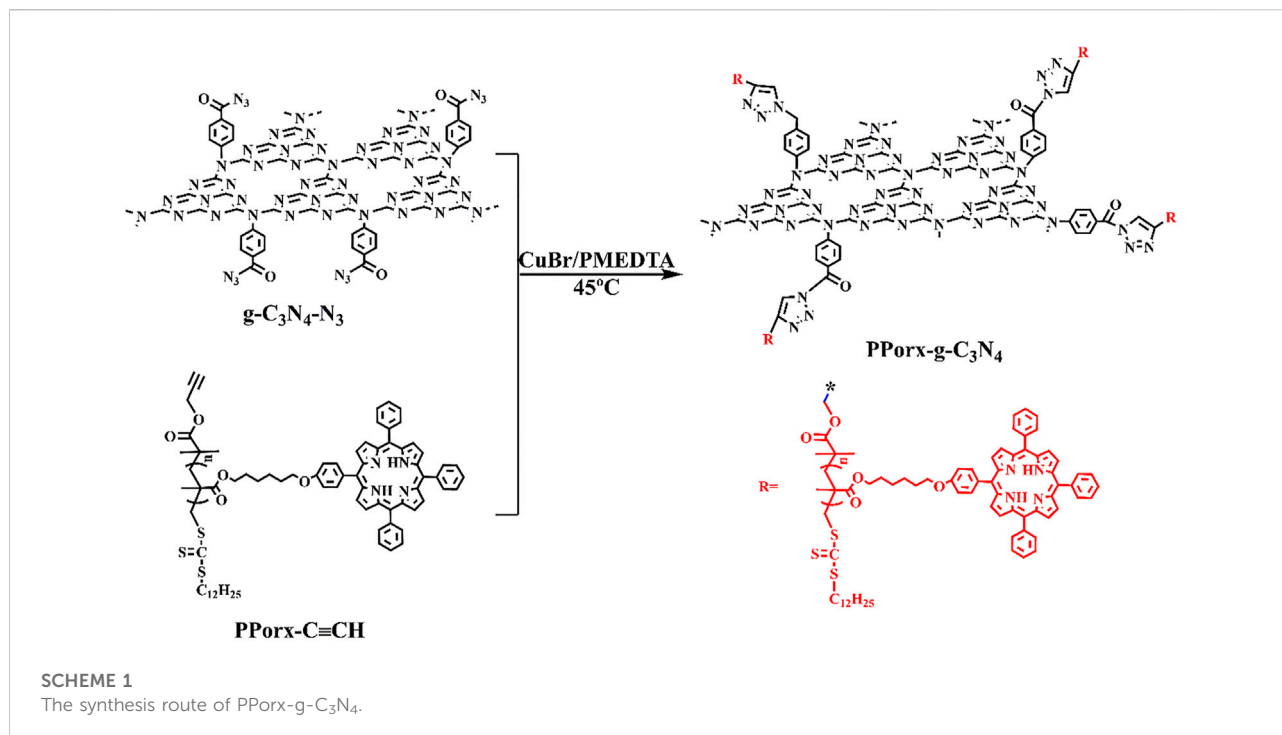
3.1 Synthesis and characterization

The synthetic process of PPorx- $g\text{-C}_3\text{N}_4$ nanohybrid was illustrated in Scheme 1. The PPorx- $g\text{-C}_3\text{N}_4$ nanohybrids were obtained through the click chemistry between PPorx-C \equiv CH and $g\text{-C}_3\text{N}_4\text{-N}_3$. Due to their similar structure, PPor1- $g\text{-C}_3\text{N}_4$ was analyzed to dissect the structure of PPorx- $g\text{-C}_3\text{N}_4$. Figure 1 shows the Fourier transform infrared (FT-IR) spectra of $g\text{-C}_3\text{N}_4\text{-N}_3$, PPor1-C \equiv CH and PPor1- $g\text{-C}_3\text{N}_4$. The stretching vibration of -C \equiv CH (2154 cm $^{-1}$) and -N $_3$ (2050 cm $^{-1}$) were observed for PPor1-C \equiv CH and $g\text{-C}_3\text{N}_4\text{-N}_3$, respectively. For PPor1- $g\text{-C}_3\text{N}_4$, the -N $_3$ and -C \equiv CH peaks disappeared, the characteristic peaks at 810, 1245, 1324, 1458 and 1642 cm $^{-1}$ from the $g\text{-C}_3\text{N}_4$ group, and 800 cm $^{-1}$ from porphyrin group were observed, indicating the successful combination of PPor1-C \equiv CH with $g\text{-C}_3\text{N}_4\text{-N}_3$.

The crystal structure of PPor1- $g\text{-C}_3\text{N}_4$ was further determined by X-ray diffractometer (XRD). As indicated in Figure 2, a broad peak in the range from 15° to 35° of PPor1-C \equiv CH was due to its indeterminate structure. The peaks of (100) and (002) at 13.4° and 27.4° of $g\text{-C}_3\text{N}_4\text{-N}_3$ indicated the interlayer stacking of aromatic rings and the in-plane repeat period in $g\text{-C}_3\text{N}_4$ (Xu et al., 2021). The amorphous structure of PPor1- $g\text{-C}_3\text{N}_4$ was due to the destruction of the ordered structure of $g\text{-C}_3\text{N}_4\text{-N}_3$ by the combination with PPor1-C \equiv CH.

X-ray photoelectron spectroscopy (XPS) was used to confirm the covalent attachment between PPor1 group and $g\text{-C}_3\text{N}_4$ group. In the survey spectra (Supplementary Figure S4), PPor1- $g\text{-C}_3\text{N}_4$ was constructed by C, N, O and S. As shown in Figure 3, the six characteristic peaks in PPor1- $g\text{-C}_3\text{N}_4$ could be divided into three groups: the characteristic peaks at 398.8 and 400.1 eV belonging to the N of NH and C=N in the pyrrole ring of PPor1 group; the characteristic peaks at 398.3, 399.5 and 400.6 eV belonging to the N of C-N=C, N-(C) $_3$ and C-NH $_2$ of $g\text{-C}_3\text{N}_4$, respectively (Xu et al., 2021); the characteristic peak located at 402.0 eV assigned to the N of the triazole ring in PPor1- $g\text{-C}_3\text{N}_4$ (Wiperman et al., 1991; Liu et al., 2016), indicating that PPor1 group was covalently attached to $g\text{-C}_3\text{N}_4$ group *via* the click reaction. In addition, compared with $g\text{-C}_3\text{N}_4\text{-N}_3$ and PPor1-C \equiv CH, the peaks of $g\text{-C}_3\text{N}_4$ group in PPor1- $g\text{-C}_3\text{N}_4$ shift towards lower binding energy, while the peaks of porphyrin shift towards higher binding energy, respectively, which could be attributed to the disappearance of the alkynyl group in PPorx-C \equiv CH and the azide group in $g\text{-C}_3\text{N}_4\text{-N}_3$ after the click reaction between PPorx-C \equiv CH and $g\text{-C}_3\text{N}_4\text{-N}_3$ and the formation of the triazole ring, leading to the change of the chemical environment of N in porphyrin and $g\text{-C}_3\text{N}_4$ (Li X. et al., 2022; Shi et al., 2022). The results of XPS proved the covalently linking between PPor1 group and $g\text{-C}_3\text{N}_4$ group.

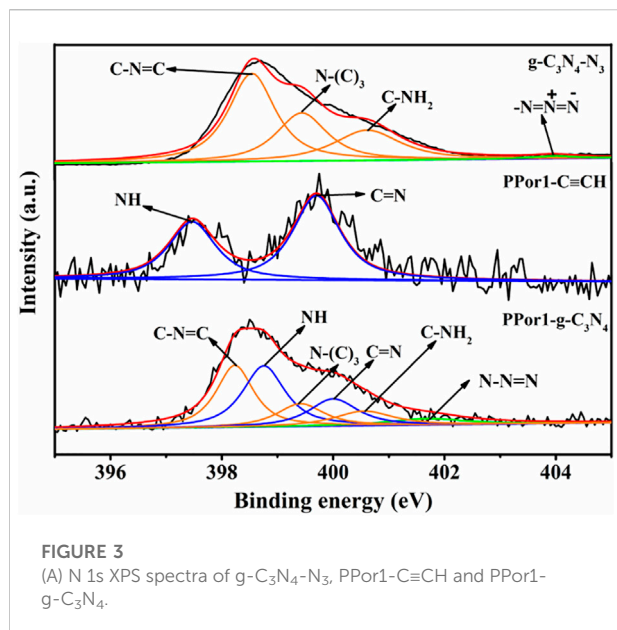
To further investigate the morphology of material, scanning electron microscope (SEM) was carried out. As shown in Figure 4A, $g\text{-C}_3\text{N}_4\text{-N}_3$ showed tubular morphology with a



smooth surface, and its tube diameters varies from 0.896 μm to 2.608 μm . In **Figure 4B**, PPor1-g-C₃N₄ also exhibited an obvious tubular structure, and there were irregularly shaped particles with a size of several hundred nm on the surface, which might relate to the introduction of pPorx group. The FT-IR, XRD, XPS, and SEM together confirmed the successful preparation of PPor1-g-C₃N₄.

3.2 Optical and physical properties

As shown in the UV-vis diffuse reflection spectra (DRS) (**Figure 5A**), the g-C₃N₄-N₃ exhibited an absorption edge at ca. 450 nm, which is consistent with the literature report (Zhou et al., 2018). A broad peak from 462 to 600 nm derived from the $n\text{-}\pi^*$ transition of heptazine ring unit in the g-C₃N₄-N₃ (Xu et al.,

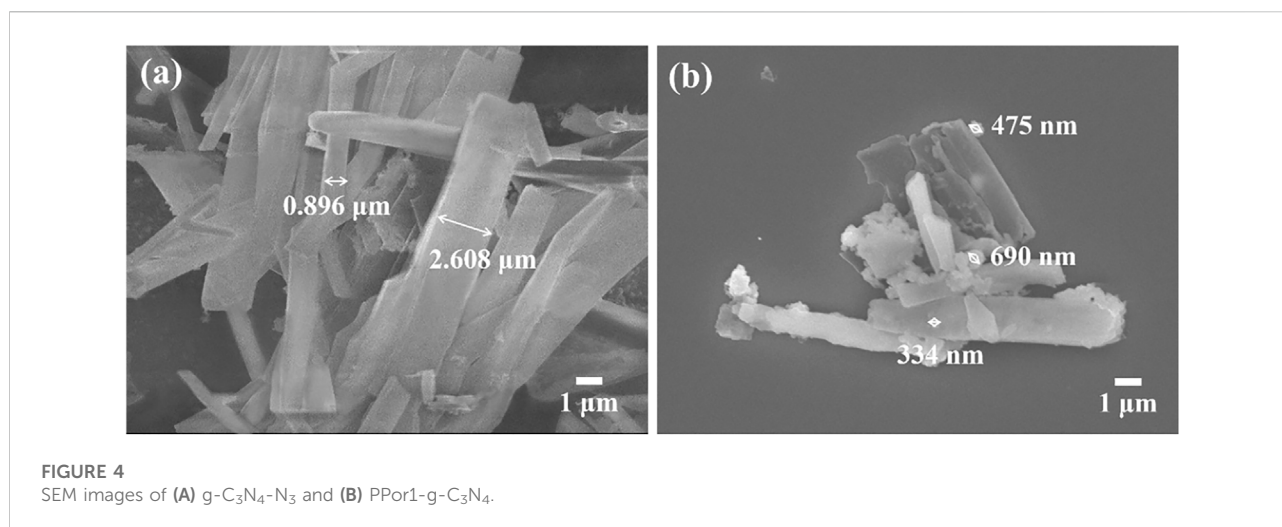


2019). PPor1-C≡CH possessed a Soret band at around 417 nm and four weak bands at 500–700 nm attributed to Q bands of porphyrin group. After the click chemistry between g-C₃N₄-N₃ and PPor1-C≡CH, the peaks of porphyrin group in PPor1-g-C₃N₄ were red-shifted compared with that in PPor1-C≡CH, which might be involved in the electron interactions between PPor1 group and g-C₃N₄ group. From the photoluminescence (PL) spectra (Figure 5B), the three porphyrin-based polymers PPorx-C≡CH exhibited two emission bands at 661 and 725 nm. Besides, the fluorescence emission intensity of PPorx-C≡CH decreases as the degree of polymerization of PPorx-C≡CH increases, which could be explained as follows: as the increase of the degree of polymerization of PPorx-C≡CH, the concentration of porphyrin increases,

resulting in a self-quenching phenomenon which is caused by the concentration quenching effect, and this phenomenon becomes more obvious with the increase of the degree of polymerization of PPorx-C≡CH (Grenoble et al., 2005; Loman-Cortes et al., 2021; Li C. et al., 2022). Notably, the PPorx-g-C₃N₄ exhibited obvious fluorescence quenching compared with PPorx-C≡CH, and the fluorescence characteristic peak (666 nm) had a redshift of 5 nm, which might be owing to the PET between pPorx group and g-C₃N₄ group in PPorx-g-C₃N₄.

The photocurrent response experiment of PPor1-C≡CH, g-C₃N₄-N₃ and PPorx-g-C₃N₄ (x = 1, 2 and 3) were used to further investigate the PET effect between pPorx group and g-C₃N₄ group in PPorx-g-C₃N₄. As shown in Figure 6, the photocurrent response of PPor1-g-C₃N₄ was enhanced compared to PPor1-C≡CH and g-C₃N₄-N₃, indicating that the PET existed between PPor1 group and g-C₃N₄ group in PPor1-g-C₃N₄ under visible light irradiation. In addition, with the increase of MW of pPorx group in PPorx-g-C₃N₄, the electron transfer effect of PPorx-g-C₃N₄ was observed to first increased and then decreased. Among them, PPor2-g-C₃N₄ showed the strongest current density, proving that PPor2-g-C₃N₄ has the strongest electron transfer effect.

The electrochemical experiments were further carried out to evaluate the effect of the MW of PPor-C≡CH on the electron transfer effect of PPorx-g-C₃N₄. As shown in Figure 7A, the Mott-Schottky (MS) plots of g-C₃N₄-N₃, PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH exhibited a positive slope, which indicated that all samples were n-type semiconductors (Yang et al., 2020). The flat band potentials (E_{fb}) of g-C₃N₄-N₃, PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH were -0.52 V, -0.94 V, -0.70 V and -0.57 V, respectively, which was measured from the intersection of $Cs^{-2}-0$ linear curve. And the CB edge potential (E_{CB}) of g-C₃N₄-N₃, PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH were calculated to be -0.50 V, -0.92 V, -0.68 V and -0.55 V,



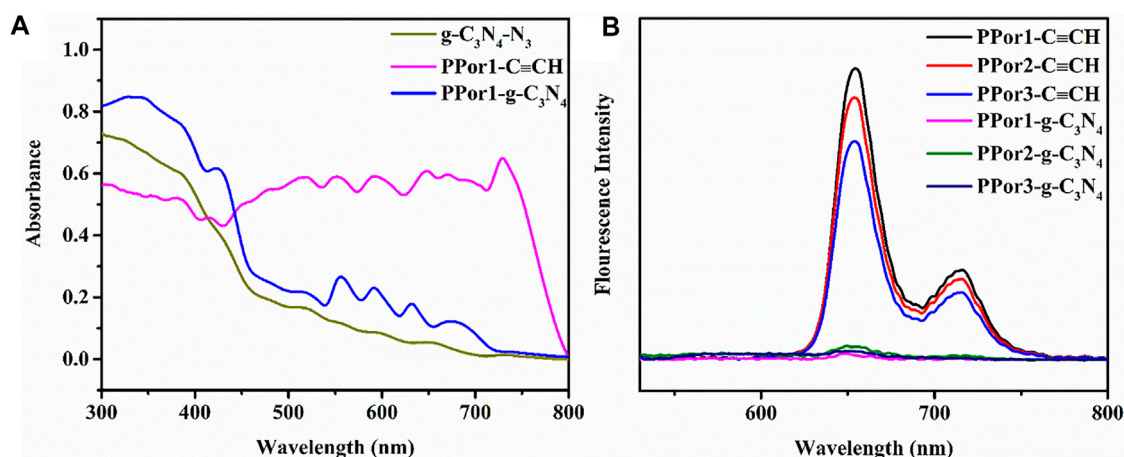


FIGURE 5

(A) DRS of PPor1-C≡CH, g-C₃N₄-N₃ and PPor1-g-C₃N₄. (B) PL emission spectra of PPor1-C≡CH (6 μmol/ml), PPor2-C≡CH (3.63 μmol/ml), PPor3-C≡CH (2.26 μmol/ml) and PPor1-g-C₃N₄~PPor3-g-C₃N₄ (0.05 mg/ml) under 370 nm excitation.

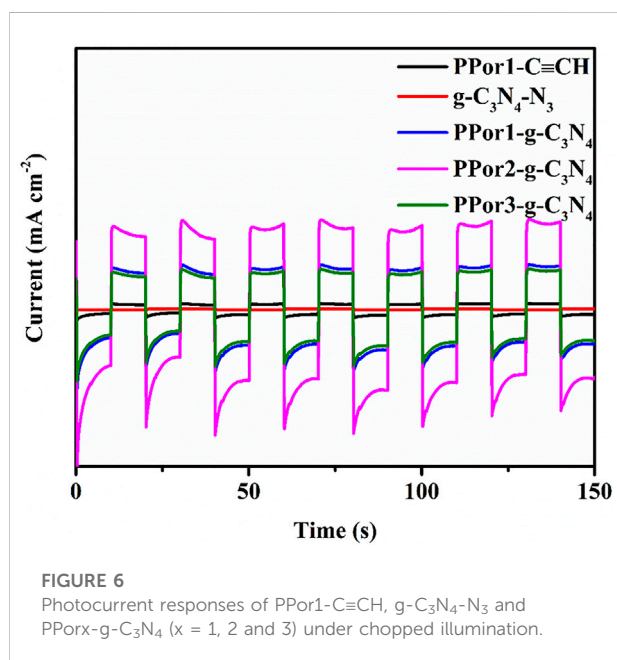


FIGURE 6

Photocurrent responses of PPor1-C≡CH, g-C₃N₄-N₃ and PPorx-g-C₃N₄ (x = 1, 2 and 3) under chopped illumination.

respectively, which according to the formula (E_{CB} (NHE, pH = 7) = E_{fb} (SCE, pH = 7) + 0.225 - 0.2) (Wang et al., 2017). Furthermore, the bandgap energy (E_g) values of g-C₃N₄-N₃ was calculated to be 2.50 eV, PPor1-C≡CH, PPor2-C≡CH, and PPor3-C≡CH were calculated to be 1.87 eV, 1.85 eV and 1.83 eV, respectively, based on the Tauc Plot transformed from UV-vis DRS spectra (Supplementary Figure S5 and Figure 7B) (Xu et al., 2019). The reduction of the bandgap of PPorx-C≡CH is associated with the increased conjugacy of porphyrin, which becomes more pronounced as the MW of PPorx-C≡CH increases

(Qiu et al., 2013). Based on the empirical formula $E_{VB} = E_{CB} + E_g$, the VB edge potential (E_{VB}) of g-C₃N₄-N₃, PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH were calculated to be 2.00 V, 0.95 V, 1.17 V and 1.28 V, respectively. Consequently, the interlaced band structures of g-C₃N₄-N₃ and PPorx-C≡CH could be obtained (Ismael, 2022).

The electron transfer process in the PPorx-g-C₃N₄ is shown in Figure 8. Under irradiation, pPorx group and g-C₃N₄ group could be excited and produce abundant e⁻ and h⁺ at the same time. Due to their staggered band structure, the electron transfer process was as follows: e⁻ could transfer from the CB of PPorx group to the CB of g-C₃N₄ group and h⁺ on the VB of g-C₃N₄ group could move to the VB of PPorx group. Thus, PPorx group could behave as the electron donor, and g-C₃N₄ group as the electron acceptor in PPorx-g-C₃N₄ nano hybrids. Moreover, as the MW of PPorx-C≡CH increased, the energy difference between the E_{CB} of PPorx-C≡CH and the E_{CB} of g-C₃N₄-N₃ decreased, which might result in lower electron transfer effect. However, the electron transfer efficiency of PPorx-g-C₃N₄ was observed to increase first and then decrease based on the photocurrent response experiment, which might be explained as follows. The PPorx group in the PPorx-g-C₃N₄ was increased after the click chemistry reaction with the increase of the MW of PPorx-C≡CH, causing the enhanced electron transfer effect. However, the MW of PPorx-C≡CH is gradually increased to a certain extent, and it contributes to steric hindrance increase, which hinders the click chemistry reaction between PPorx-C≡CH and g-C₃N₄-N₃, and then, the pPorx group in the PPorx-g-C₃N₄ was decreased, leading to a diminished electron transfer effect. Consequently, the electron transfer effect exerts a trend of first increasing and then decreasing.

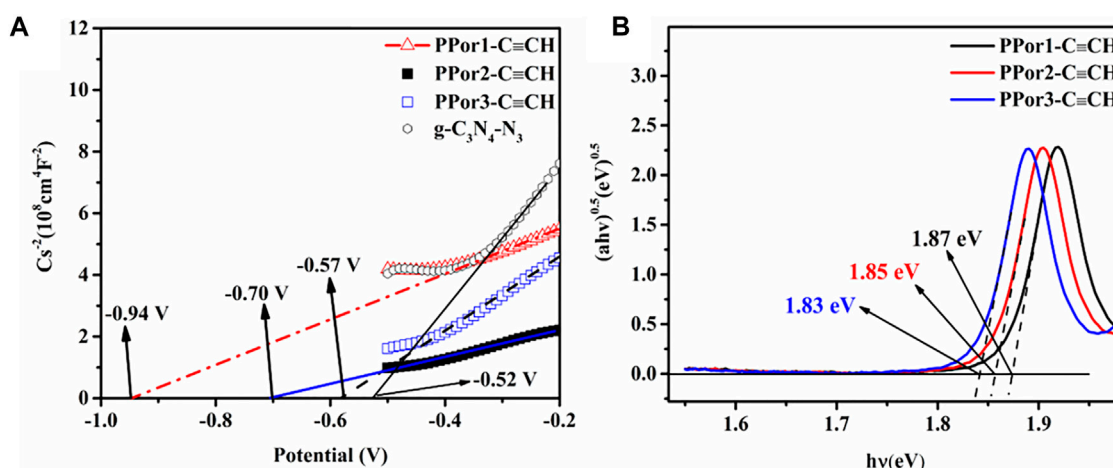


FIGURE 7
 (A) Mott-Schottky plots of g-C₃N₄-N₃, PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH. (B) Tauc plots of PPor1-C≡CH, PPor2-C≡CH and PPor3-C≡CH.

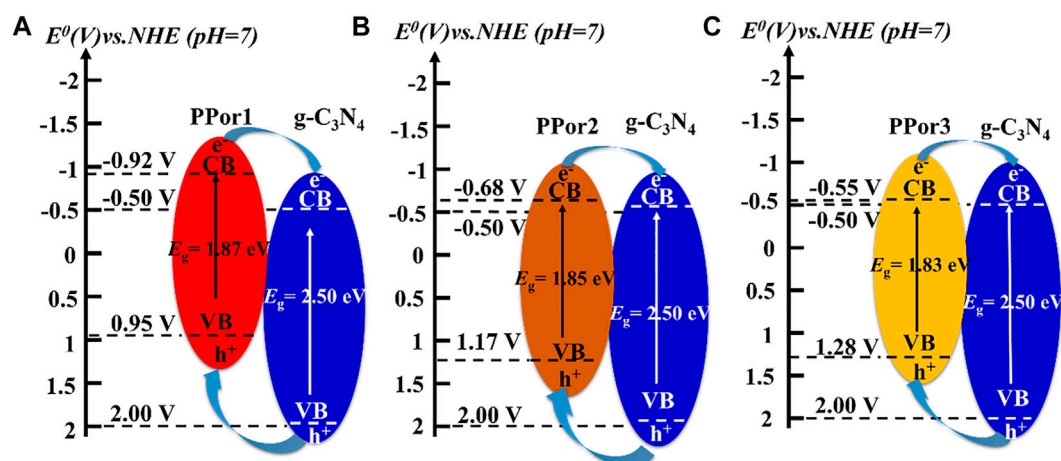


FIGURE 8
 The schematic diagrams of the proposed electron transfer under visible light irradiation in (A) PPor1-g-C₃N₄, (B) PPor2-g-C₃N₄ and (C) PPor3-g-C₃N₄.

3.3 Non-linear optical properties of PPorx-g-C₃N₄ nanohybrids

The NLO performances of PPorx-g-C₃N₄ nanohybrids were investigated in MMA by the Z-scan technique with 7 ns laser pulses of 532 nm. Generally, the value of the non-linear absorption coefficient (β_{eff}) was used to evaluate the reverse saturable absorption (RSA) performance. The excellent RSA performance would result in a large β_{eff} and a deep “V” shaped absorption curve. As shown in Figure 9, the RSA performance of PPor1-g-C₃N₄

nanohybrid was better than the corresponding PPor1-C≡CH and g-C₃N₄-N₃. The β_{eff} of PPor1-g-C₃N₄ was calculated to be 3.4×10^{-9} m/W, and $Im[\chi^{(3)}]$ value was calculated to be 1.11×10^{-10} esu, which was ca. 5.76 times higher if compared to PPor1-C≡CH ($Im[\chi^{(3)}]$ of 0.16×10^{-10} esu), attributing to the PET behavior between PPor1 group and g-C₃N₄ group in PPor1-g-C₃N₄. The NLO properties of PPorx-g-C₃N₄ in MMA are listed in Table 1 [for comparison, the results of covalently linked 5-(4-hydroxyphenyl)-10,15,20-triphenylporphyrin-g-C₃N₄ (Por-g-C₃N₄) is also provided (Supplementary Figure S6)]. From Table 1, PPor1-g-C₃N₄ and

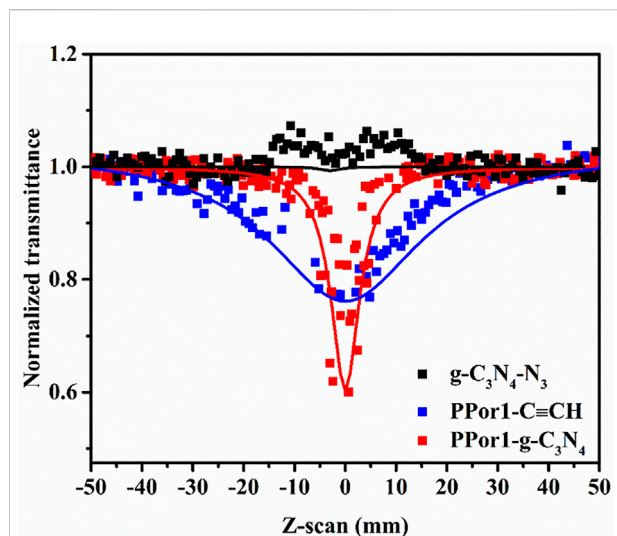


FIGURE 9
The open aperture Z-scan tests of $g\text{-C}_3\text{N}_4\text{-N}_3$, PPor1-C \equiv CH and PPor1- $g\text{-C}_3\text{N}_4$ in MMA (0.05 mg/ml).

PPor2- $g\text{-C}_3\text{N}_4$ show larger $Im[\chi^{(3)}]$ value than that of Por- $g\text{-C}_3\text{N}_4$, proving that the introduction of porphyrin-based polymer into $g\text{-C}_3\text{N}_4$ effectively improve the aggregation behavior of porphyrins and enhanced the NLO performance of the nanohybrids. PPor2- $g\text{-C}_3\text{N}_4$ exhibited the best NLO performance among PPorx- $g\text{-C}_3\text{N}_4$ with β_{eff} of 4.5×10^{-9} m/W and $Im[\chi^{(3)}]$ of 1.47×10^{-10} esu, respectively, due to the efficient PET from PPor2 group to $g\text{-C}_3\text{N}_4$ group in PPor2- $g\text{-C}_3\text{N}_4$, so that PPor2- $g\text{-C}_3\text{N}_4$ exhibits the best NLO performance in MMA.

To study the practicality of PPorx- $g\text{-C}_3\text{N}_4$, the PPorx- $g\text{-C}_3\text{N}_4$ was doped PMMA to form PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites *via* solution casting technology (described in supporting information). Figure 10A showed the RSA performance of PPorx- $g\text{-C}_3\text{N}_4$ /PMMA (0.05 mg/ml), and the results were listed in Table 2. Compared with PPorx- $g\text{-C}_3\text{N}_4$ in MMA (Table 1), the NLO performance of PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites was improved, which might be owing to the weaker aggregation

effect in the solid matrix (Wang J. et al., 2020). Among them, PPor2- $g\text{-C}_3\text{N}_4$ /PMMA composite exhibited the deepest trough, with the excellent β_{eff} of 7.2×10^{-9} and $Im[\chi^{(3)}]$ of 2.36×10^{-10} esu, respectively. Furthermore, the photographs of PPor1- $g\text{-C}_3\text{N}_4$ /PMMA, PPor2- $g\text{-C}_3\text{N}_4$ /PMMA and PPor3- $g\text{-C}_3\text{N}_4$ /PMMA composites were shown in Figure 10B, all of them showed excellent transparency, demonstrating the great potential in practical application.

The optical limiting (OL) properties of the PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites are shown in Figure 11A. Under the condition of low input fluence, the output fluence of PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites promoted with the increase of input fluence, displaying a linear optical property. However, as the input fluence was further increased, PPor1- $g\text{-C}_3\text{N}_4$ /PMMA, PPor2- $g\text{-C}_3\text{N}_4$ /PMMA and PPor3- $g\text{-C}_3\text{N}_4$ /PMMA composites showed obvious non-linear trends, and their initial thresholds were determined to be 0.401 J/cm², 0.058 J/cm² and 0.464 J/cm², respectively. Figure 11B shows the relationship between the input fluence and the normalized transmittance, where the black dotted line represents 50% of the initial transmittance. From Figure 11B, the decreasing normalized transmittance of all PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites was associated with the increase of the input fluence. Among them, the PPor2- $g\text{-C}_3\text{N}_4$ /PMMA composite exerted the best OL performance, with the limiting threshold of 1.71 J/cm², the minimum transmittance of 8% and the dynamic range of 1.09, respectively, which might be owing to the excellent PET from PPor2 group to $g\text{-C}_3\text{N}_4$ group in PPor2- $g\text{-C}_3\text{N}_4$. Some reported OL performances of the porphyrin-based materials are summarized in Table 3, and our OL data demonstrate that PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites are among the best performing materials for this purpose. In practical application, the damage threshold was an important criterion to measure the stability of the material, interpreting no optical damage occurrence under this input fluence condition. There was no obvious damage observed for PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites even if the input fluence reached 16 J/cm², which could be owing to the good thermal stability of each fraction in PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites. These results proved that the PPorx- $g\text{-C}_3\text{N}_4$ /PMMA composites had good application prospects in the OL field.

TABLE 1 NLO properties of as prepared samples in MMA.

Sample ^{ab}	Linear transmittance (%)	β_{eff} (m/W)	$Im[\chi^{(3)}]$ (esu)
PPor1-C \equiv CH	65	0.5×10^{-9}	0.16×10^{-10}
$g\text{-C}_3\text{N}_4\text{-N}_3$	70	0.1×10^{-10}	0.03×10^{-10}
Por- $g\text{-C}_3\text{N}_4$	70	0.8×10^{-9}	0.27×10^{-10}
PPor1- $g\text{-C}_3\text{N}_4$	68	3.4×10^{-9}	1.11×10^{-10}
PPor2- $g\text{-C}_3\text{N}_4$	65	4.5×10^{-9}	1.47×10^{-10}
PPor3- $g\text{-C}_3\text{N}_4$	69	0.7×10^{-9}	0.22×10^{-10}

^aPeak intensity for each independent Z-scan measurement was -13 μJ . The excitation source was 7 ns laser pulses of 532 nm wavelength.

^bc = 0.05 mg/ml.

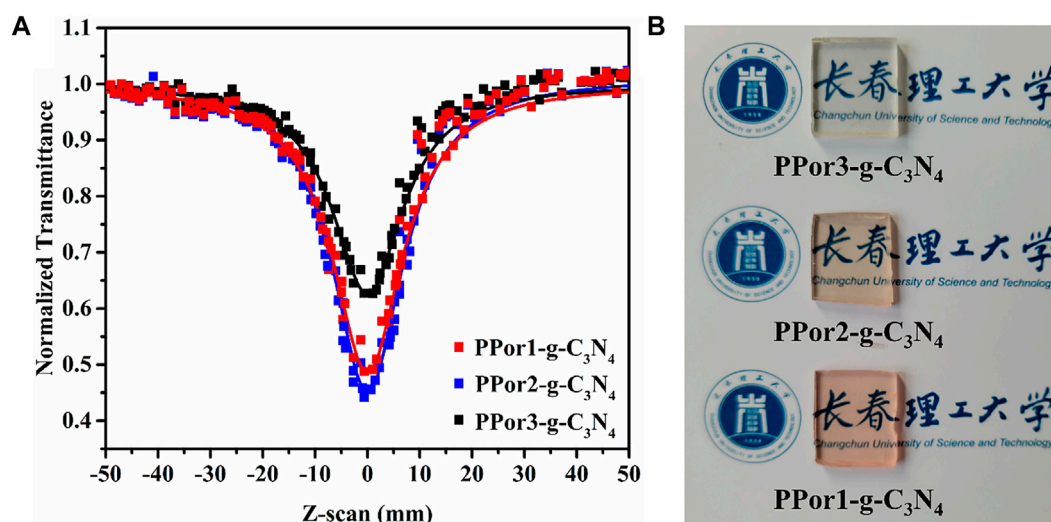


FIGURE 10

(A) The open aperture Z-scan tests of PPorx-g-C₃N₄/PMMA composites with the doping concentration of 0.05 mg/ml. (B) Photographs of PPorx-g-C₃N₄/PMMA composites with 0.05 mg/ml doping concentrations.

TABLE 2 NLO properties of as prepared samples in PMMA.

Sample ^{ab}	Linear transmittance (%)	β_{eff} (m/W)	$Im[\chi^{(3)}]$ (esu)
PPor1-g-C ₃ N ₄ /PMMA	65	4.8×10^{-9}	1.58×10^{-10}
PPor2-g-C ₃ N ₄ /PMMA	67	7.2×10^{-9}	2.36×10^{-10}
PPor3-g-C ₃ N ₄ /PMMA	63	2.8×10^{-9}	0.92×10^{-10}

^aPeak intensity for each independent Z-scan measurement was 13 μ J. The excitation source was 7 ns laser pulses of 532 nm wavelength.

^bc = 0.05 mg/ml.

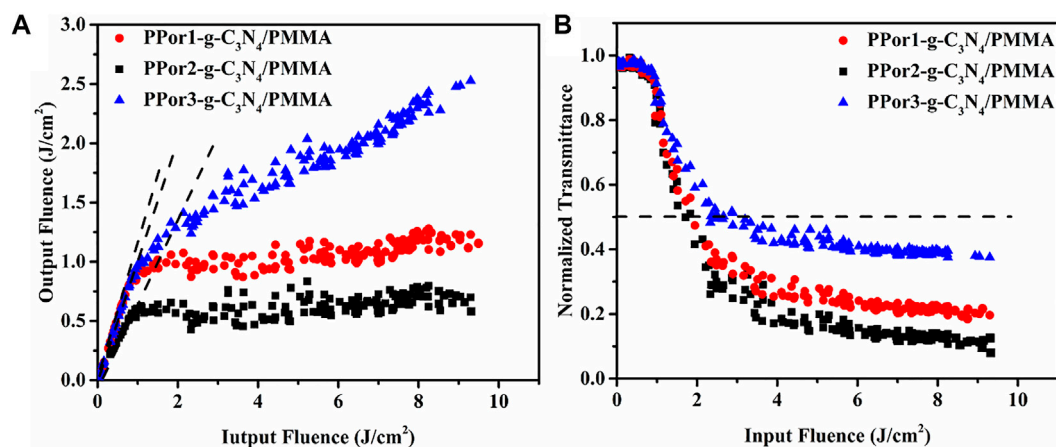


FIGURE 11

(A) The OL performance and (B) the non-linear transmittance of PPorx-g-C₃N₄/PMMA composites with the doping concentration of 0.05 mg/ml.

TABLE 3 Summary of the OL parameters of PPorx-g-C₃N₄/PMMA and reported works.

Sample	Initial threshold (J/cm ²)	The dynamic range	References
PPor1-g-C ₃ N ₄ /PMMA	0.401	0.43	this work
PPor2-g-C ₃ N ₄ /PMMA	0.058	1.09	this work
PPor3-g-C ₃ N ₄ /PMMA	0.464	0.70	this work
Pm@HPA in DMF	0.531	0.60	Hassan et al. (2018)
1-GO in DMF	–	0.45	Garg et al. (2017)
Penta (ZnP)C ₆₀ in DMF	–	0.66	Kulyk et al. (2020)
SWCNT-TPP1 in DMF	–	0.92	Wang et al. (2019)
LaPc in DMF	–	0.27	Ou et al. (2021)

4 Conclusion

In summary, a flurry of novel porphyrin-based polymers functionalized g-C₃N₄ nano hybrids PPorx-g-C₃N₄ had been prepared. The PPorx-g-C₃N₄ nano hybrids exhibited improved NLO performance compared to single g-C₃N₄-N₃ and PPorx-C≡CH using the Z-scan technique under 532 nm in ns regimes. Among them, due to the suitable molecular weight and steric hindrance, the efficient PET from PPor2 group to g-C₃N₄ group in PPor2-g-C₃N₄ gave PPor2-g-C₃N₄ the best NLO performance among PPorx-g-C₃N₄ with β_{eff} of 1.47×10^{-10} esu. For practical application, the PPorx-g-C₃N₄ doped PMMA composites were prepared by the solution casting method. PPor2-g-C₃N₄/PMMA composite exhibited the best $Im[\chi^{(3)}]$ of 2.36×10^{-10} esu, initial threshold of 0.058 J/cm² and dynamic range of 1.09, indicating its great potential for practice. This research provided a new strategy for the design of porphyrin-based nano hybrid for NLO application.

Data availability statement

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding authors.

Author contributions

CL: Investigation, Data curation, Writing-Original draft. XC: Formal analysis, Conceptualization, Writing—Reviewing and Editing. WD: Writing—Reviewing and Editing, JQ:

Writing—Reviewing and Editing, QD: Funding acquisition, Project administration, Writing—Reviewing and Editing.

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Conflict of interest

The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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Supplementary material

The Supplementary Material for this article can be found online at: <https://www.frontiersin.org/articles/10.3389/fchem.2022.1102666/full#supplementary-material>

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