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Process method of Si₃N₄ ceramic brazing sealed cavity for high-temperature application

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The process method of a Si₃N₄ ceramic sealed cavity is realized by vacuum brazing and chemical reaction at 1,100°C and 0.5 MPa pressure. Through the combination of Si₃N₄ ceramic polishing and thinning, inductively coupled plasma etching, and high-temperature metal filler (Ti-Zr-Cu-Ni) brazing process, a vacuum-sealed cavity suitable for high-temperature environments was prepared. The cross section of the bonding interface was characterized by scanning electron microscope (SEM) and energy dispersive spectrometer (EDS), which indicated that the two Si₃N₄ ceramic were well bonded, the cavity structure remained intact, and the bonding interface strength exceeded 5.13 MPa. Furthermore, it retained its strong bonding strength after in high-temperature environments of 1,000, 1,050, and 1,100°C for 1 h. This indicates that a brazed vacuum-sealed cavity can be used in high-temperature environments. Through the proposed method, pressure sensor that can withstand high temperatures can be developed.

KEYWORDS

Si₃N₄ ceramic, vacuum brazing, brazing interface, sealed cavity, high temperature

Introduction

In high-temperature environments, *in situ* and accurate pressure measurements are required for industrial applications, such as energy extraction, environmental monitoring, turbine engines, high-speed aircraft, and other aerospace applications (Lü et al., 2018; Chen et al., 2022; Li et al., 2022; Zhang et al., 2022), to improve component conditions and ensure engine reliability (Bonham et al., 2017; Miura et al., 2017; Han et al., 2020; He et al., 2021). Materials are crucial as the cornerstone of pressure sensors, as sensors made of ordinary materials cannot operate optimally in high-temperature environments. Therefore, high-temperature-resistant materials are preferred, and by using these, developing pressure sensors that can be used in ultra-high temperature environment. It is crucial and significant to develop an original test of pressure parameters in high-temperature environments (Vysotina et al., 2021).

The operating temperatures of conventional silicon pressure sensors and some silicon-on-insulator (SOI)-based pressure sensors are generally in the range of

200-300°C, which cannot be adopted in high-temperature environments (Cheng et al., 2021; Tang et al., 2021). Hence, ceramic pressure sensors have been developed (Li J. et al., 2019; Meng et al., 2021). The technology of these sensors is mature and they are small-sized. However, the operating temperatures of these sensors generally do not exceed 500°C, imposing significant limitations when operating in higher temperature environments (Naderi and Moghaddam, 2020; Salvatori et al., 2020; Abdel-Motaleb and Dittakavi, 2021). Most LC-type high temperature pressure sensors are fabricated using either a low-temperature co-fired ceramic (LTCC) or high-temperature co-fired ceramic (HTCC) technology (Lin et al., 2018; Zhu et al., 2018; Raynaud et al., 2021; Zhang et al., 2021; Seek and Makarovi, 2022). In these methods, sensors are using combined processes of multilayer ceramics and laminations, high-temperature sintering, and screen-printing processes, however, the manufacturing process is complex. In this process, the embedded sealed cavity is easily deformed or even collapses, thereby resulting production difficulties. Therefore, in recent years, several researchers have developed pressure sensors from some high-temperature resistant materials (such as sapphire/sapphire and MgO/MgO, etc.) via direct process bonding, which can be applied to hightemperature environments. However, its preparation process is complicated and expensive (Li et al., 2017; Li W. et al., 2019; Liu et al., 2019; Liu et al., 2020). Si₃N₄ ceramic has excellent properties such as high hardness and strength, low thermal expansion coefficient, optimal oxidation resistance, and high corrosion resistance (Gu, 2004; Usherenko et al., 2019; Xiao and Han, 2019; Ye et al., 2021), in particular, hot-pressed Si₃N₄ ceramic, which is extremely resistant to high temperatures, can be utilized as a high-temperature sensor core substrate while preparing high temperature pressure sensor cavity.

In this study, we propose a high-temperature pressure sensor sealed cavity fabrication method based on Si₃N₄ ceramic, fabricated by etching Si₃N₄ ceramic via metal masks and brazing Si₃N₄ ceramic with a high-temperature active brazing filler (Ti-Zr-Cu-Ni). For the brazing filler, joints brazed with mature and widely used Ag-Cu-Ti (Deng et al., 2021) brazing fillers exhibit excellent mechanical properties. However, owing to the limitation of the working temperature, their application in high-temperature environments is limited. It is necessary to select appropriate refractory metals to maximize excellent high-temperature performance of Si₃N₄ ceramic. Accordingly, Cu-Ni is a more suitable choice, than noble metals. Nevertheless, to ensure that the brazing filler can react with the ceramic, the active metal Ti-Zr was selected to achieve optimal wettability with the ceramic. Furthermore, tensile tests and microstructural observations of the bonded interface were performed on the brazed samples. The connection method proposed in this study can realize the high-strength connection of Si₃N₄ ceramic/Si₃N₄ ceramic and form a well-sealed cavity, thus providing a basis for the next production of pressure sensors. Compared with conventional SOI-based pressure sensors, this type of ceramic can be used in higher-temperature environments. Furthermore, compared with HTCC technology and directly bonded technology, this type of method has a simpler manufacturing process, is less expensive, and provides broad application prospects in high temperature pressure sensors.

Experiment

Polishing and thinning of samples

The raw materials used in this experiment are Si₃N₄ ceramic with a purity of 99.9%. In the manufacturing process of the sealed cavity, three key processes are mainly included: the Si₃N₄ ceramic polishing reduction process, Si₃N₄ ceramic etching process, and Si₃N₄ ceramic bonding process. According to the designed preparation process, first of all, the thickness and roughness of the sample ceramics should be reduced. Because the close contact between the sample ceramics cannot be guaranteed if the roughness is too large, and so that the brazing quality cannot be guaranteed. At the same time, it will affect the high-temperature brazing filler metal to fully expand on the ceramics, resulting in loose connection. The process of thinning can ensure the flatness of the sample ceramics. The Si₃N₄ ceramic material is harder. The abrasive used in the grinding process is boron carbide material. In the grinding process of chemical machinery, while the ceramic sheet is flattened, residual stress will be introduced on the surface of the chip due to mechanical grinding. The thinned Si₃N₄ ceramic was subjected to silicon oxide suspension polishing to satisfy the surface appearance requirements of the subsequent brazing process. The root mean square roughness was approximately 0.2 µm measured by atomic force microscopy (AFM). Its process flow is illustrated in Figure 1.

Chemical etching process of Si_3N_4 ceramic

Subsequently, a metal mask was utilized for etching. The specific process is as follows:

1) Sputtering aluminum film: A 3- μ m thick aluminum film was sputtered onto a Si₃N₄ ceramic substrate *via* physical vapor deposition (PVD), the power and Ar flow adopted were 3 kW and 20 standard cubic centimeter per minute (sccm), respectively. The sputtering rate is 6 nm/s;

2) Spin coating photoresist: The positive photoresist AZ6130 was evenly applied on the Si_3N_4 ceramic by rotating the gelatinizer, with an average glue speed of 500 r/min for 10 s, and 3,000 r/min for 60 s, to obtain a layer of photoresist;

3) Patterned exposure: The Si_3N_4 ceramic coated with photoresist were placed upward, and the mask plate with lithography pattern placed downward into the lithography machine to ensure that the mask plate and the silicon wafer



were fixed. The pattern on the mask printed by the was transferred to the ${\rm Si}_3{\rm N}_4$ ceramic at a ratio of 1: 1;

4) Corroding aluminum: The exposed aluminum was subsequently corroed using phosphoric acid (H_3PO_4) at 85°C;

5) ICP etching Si_3N_4 ceramic: After etching starts, sulfur hexafluoride (SF₆) and trifluoromethane (CHF₃) were used as etching gases, oxygen (O₂) was used to clean the chamber and took away the exhaust gas produced by etching, helium (He) was used as protective gas, and the pressure of the process chamber was set to 12 mTorr with the etching power 2500 W;

6) Etched Si_3N_4 ceramic: After etching, the remaining aluminum film was etched away;

7) Coating brazing filler: Apply high-temperature brazing filler on the etched the samples;

8) Prepare brazing samples: Complete the preparation of samples.

After the experimental samples were prepared, they were brazed using a filler metal coating. As ordinary components do not wet the surface of Si_3N_4 ceramic, it is necessary to add a small amount of active elements to the conventional metal brazing technology, which promotes the wetting of the brazing filler on the ceramic surface. In this study, Ti and Zr were adopted as active elements to achieve wetting *via* the following reactions.

$$Si3N4 + 4Zr = 4ZrN + 3Si$$
 (1)

$$Si3N4 + 4Ti = 4TiN + 3Si$$
 (2)

We selected a Ti-Zr-Cu-Ni brazing filler with optimal stability and oxidation resistance to maximize the excellent high-temperature performance of Si_3N_4 ceramic in the active brazing of Si_3N_4 ceramic. Connected Si_3N_4 ceramic remain significantly high-temperature resistant and can resist oxidation resistance and corrosion resistance at high temperatures. First, acetone and alcohol were sequentially used in an ultrasonic cleaner for 20 min in sequence, and finally cleaned with deionized water (DI) and dried with high-purity nitrogen. The prepared brazing material was then applied

to the Si_3N_4 ceramic, thus completing the preparation of the prebrazing samples. Figure 2 illustrates the Si_3N_4 ceramic etching process and sample preparation before high-temperature active brazing.

High-temperature active brazing process

The prepared samples were then placed in a graphite mold. The mold comprises an upper indenter, a sleeve, and a lower indenter. Owing to the pressure provided by the upper indenter, ceramic tightness can be guaranteed during the brazing process. Contact with lower indenter was combined to apply pressure evenly on the sample surface. The assembled mold was then introduced into a brazing furnace for high-temperature brazing. The brazing furnace device rapidly heated the Si₃N₄ ceramic to be welded using electrodes, with the heating speed controlled by controlling the power of the workpiece with a pressure pump and a header displacement control system. Simultaneously the hotpressing furnace table was provided with a column connection on the movable beam to ensure the verticality of the compression. A schematic diagram of the brazing furnace and the sample processing process is presented in the Figure 3. To avoid contaminating the cavity, during the coating process of the brazing filler, a membrane with the same size as the cavity was utilized to cover the cavity, and after the coating was completed, the membrane was gently removed with tweezer.

Finally, the assembled mold was introduced into a brazing furnace for high-temperature brazing. When the vacuum degree in the vacuum heating furnace reached approximately 2.8×10^{-3} Pa, the sample to be welded was heated to 200° C at a rate of 6° C/min, and maintained for 30 min. The sample was then heated to the brazing temperature at a rate of 10° C/min, and maintained for a certain time. During the heat preservation process, a pressure of 0.5 MPa was applied on the sample simultaneously. Figures 4A,B illustrate the brazing process and its principle.



FIGURE 2

Etching process and brazing process of Si_3N_4 ceramic: (A) Sputtering aluminum film. (B) Spin coating photoresist. (C) Patterned exposure. (D) Corroding aluminum. (E) ICP etching Si_3N_4 ceramic. (F) Etched Si_3N_4 ceramic. (G) Coating brazing filler. (H) Prepare brazing samples.



After brazing, the temperature was reduced to 200°C at a rate slightly lower than the heating rate, and then cooled in the furnace. Figure 4C shows the complete heating process curve. After cooling the vacuum furnace to room temperature, the Si_3N_4 ceramic samples were removed.

Results and discussion

Tensile test

Tensile tests were conducted on the brazed ceramic samples to evaluate the brazing strength of the $\rm Si_3N_4$





ceramic brazing interface. Form Figure 5A, the sealed cavity of the brazed sample remains intact, and the sample is adhesively fixed to two vertical aluminum blocks and then mounted on the vertical fixture of the stretching machine. The final bonded interface fractured under a load of 513 N, and the bonded interface remained intact, thus indicating that the tensile strength of the bonded interface exceeded 5.13 MPa. To test the connection strength after the high temperature experiment, we placed the Si₃N₄ ceramic

samples in a high temperature furnace at 1,000, 1,050, and 1,100°C for 1 h, then they were fixed on the tensile machine with adhesive for tensile test, and finally they were pulled apart under the tensile force of 362, 346, and 258 N, respectively. The curves of test time and test force are illustrated in Figure 5B. The tensile strength tests after different temperature are 5.13, 3.62, 3.46, and 2.58 MPa, as presented in Figure 5C. These values exceed the joint strength required to fabricate some pressure sensors.



FIGURE 6

SEM images of the cross section of the brazed interface at different magnifications.



Scanning electron microscope characterization

To investigate the quality of the internal connections, the samples were cut along the cross sections, which were then gold-sprayed and imaged under a scanning electron microscope (SEM). Figure 6 presents the SEM images of the brazed sample's cross section at different magnifications. It can be observed that the brazing interface is neat, smooth and tightly connected.

Energy dispersive spectrometer characterization

The qualitative analysis of the bonding interface elements by the energy dispersive spectrometer (EDS) attached to the electron microscope was carried out. The result presents in Figure 7 indicates that there was no formation of new elements in the bonding interface.

Conclusion

In this study, a process method of Si_3N_4 ceramic sealed cavity was proposed. The cavity was fabricated by combining polishing and thinning techniques with chemical etching. The vacuum-sealed cavity structure was completed using the chemical reaction between the materials through a hightemperature active brazing process. The tensile strength of the brazed samples exceeds 5.13 MPa, and can maintain high brazed strength after high-temperature environments of 1,000, 1,050, and 1,100°C for 1 h. In addition, the SEM and EDS characterizations of the brazing interface verify that the brazing quality is optimal, no impurities are introduced, and the cavity structure is complete. Accordingly, the brazing interface was uniform and seamless. This method obtains a tightly sealed cavity, which can be adopted for fabricating high-temperature pressure sensors.

Data availability statement

The original contributions presented in the study are included in the article/Supplementary Material, further inquiries can be directed to the corresponding author.

Author contributions

LC will complete the polishing and etching process of silicon nitride ceramics and finally complete the writing of the article. ZF designed the process flow during the brazing process and completed the brazing of the samples. SB completes the application of the solder and completes the preparation of the samples to be treated. YH directed and completed the pull test at high temperature. JX summarizes and processes the collected data. AX did the drawing work in the article. XG processed the data and made adjustments to the article format.

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Conflict of interest

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