



(1-C5H14N2Br)₂MnBr₄: A Lead-Free Zero-Dimensional Organic-Metal Halide With Intense Green Photoluminescence

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Jiang X, Chen Z and Tao X (2020) (1-C5H14N2Br)₂MnBr₄: A Lead-Free Zero-Dimensional Organic-Metal Halide With Intense Green Photoluminescence. Front. Chem. 8:352. doi: 10.3389/fchem.2020.00352 Low-dimensional organic-inorganic hybrid materials have attracted tremendous attentions due to their fascinating properties as emerging star materials for light-emitting applications. Taking advantage of their rich chemical composition and structural diversity, here, a novel lead-free organic-manganese halide compound, (1-mPQBr)₂MnBr₄ (1-mPQ = 1-methylpiperazine, 1-C5H14N2) with zero-dimensional structure has been rationally designed and successfully synthesized through solvent-evaporation method. Systematical characterizations were carried out to investigate the structure, thermal and photophysical properties. The (1-mPQBr)₂MnBr₄ was found to crystallized into an orthorhombic crystal (P2₁2₁2₁) with lattice parameters of a = 8.272(6) Å, b = 15.982(10)Å and c = 17.489(11) Å. The structure consists of isolated [MnBr₄]²⁻ clusters and free Br⁻ ions as well as [C5H14N2]²⁺ molecules. Thermal analysis indicates that it is stable up to 300°C. Upon ultraviolet photoexcitation, the (1-mPQBr)₂MnBr₄ exhibits intense green emission centered at 520 nm with a narrow full width at half-maximum of 43 nm at room temperature, which should be assigned to the spin-forbidden internal transition (⁴T₁(G) to ⁶A₁) of tetrahedrally coordinated Mn²⁺ ions. The superior photoluminescence properties coupled with facile and efficient synthesis method of this material suggest its considerable promise to be utilized as light-emitting materials.

Keywords: organic-metal halides, photoluminescence, single crystal, lead-free materials, manganese

INTRODUCTION

In recent decades, organic metal halide materials have flourished as star materials in solution-processed optoelectronics fields (Kojima et al., 2009; Chen et al., 2017; Yang et al., 2019a), arising from their superior properties including high absorption coefficient, long electron-hole diffusion length, ultralow trap density and high photoluminescence quantum yield (PLQY) as well as facile synthesis including low cost, high efficiency and flexibility (Dang et al., 2015; Liu et al., 2015, 2018; Huang et al., 2017). Benefitting from their remarkable advantages, they have shown great potential for photovoltaic solar cells (Kojima et al., 2009; Cheng et al., 2019; Yang et al., 2019b), light-emitting diodes (Ling et al., 2016; Thirumal et al., 2017; Lin et al., 2018), photodetectors (Adinolfi et al., 2016; Ahmadi et al., 2017; Shrestha et al., 2017), field-effect transistors (Yu et al., 2018; Zhu et al., 2019), and lasers (Yakunin et al., 2015; Zhu et al., 2015; Gu et al., 2016). Lately, the certified power

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conversion efficiency of organic lead halide solar cells have achieved $25.2\%^1$, which outperforms that of CdTe and CuInGaSe₂ solar cells. In spite of their rapid development, the presence of toxic lead in these materials is deemed to be a seriously concern, becoming a huge hindrance in their way to wide-scale exploitation. Based on this circumstance, it is therefore of urgent need to look for the alternative lead-free hybrids for future commercial optoelectronic applications.

Hence, immense efforts in reducing lead contents or exploring lead-free substitutes offer a viable solution for high-performance eco-friendly optoelectronic devices. Noteworthy, this class of materials endows with rich chemical and structural diversities. In addition to modifying the length of the organic components, the diverse structural dimensionality, referring to three-dimensional (3D), two-dimensional (2D), one-dimensional (1D) and zerodimensional (0D) structures, can also be achieved by tuning the inorganic frameworks, which results in fascinating properties. In the first place, the most obvious alternative substitution to Pb²⁺, should be the elements in the same group in the periodic table, namely Sn²⁺ and Ge²⁺ (Zhumekenov et al., 2017; Fu et al., 2018; Ju et al., 2018a; Nazarenko et al., 2019). In addition, heterovalent elements in Group 15 (Bi³⁺ and Sb³⁺) (Abulikemu et al., 2016; Sun et al., 2016; Ji et al., 2017, 2018; Ju et al., 2018b; Zhang et al., 2018; Tao et al., 2019) and in Group 13 (In³⁺) (Zhou et al., 2019) have also been demonstrated as the alternative metals. Moreover, a range of divalent transition metal ions [Cu²⁺ (Cortecchia et al., 2016; Jun et al., 2018; Li et al., 2018; Park et al., 2018), Fe²⁺ (Han et al., 2014, 2015; Nakayama et al., 2017), Mn²⁺ (Bai et al., 2018; Park et al., 2018)] can also serve as substitutes for Pb²⁺. Among which, large number of researches have reported that organic manganese (Mn²⁺) halides possess brilliant photoluminescence ranging from green to red due to the variable metal-ion coordination geometry, with the photoluminescence lifetimes varying from microseconds to milliseconds. Many groups have made great efforts in exploring octahedral- coordinated Mn (Han et al., 2015; Lv et al., 2016; Nakayama et al., 2017; Bai et al., 2018) single crystals, such as 3D-structured (CH₃)₃NCH₂ClMnCl₃ (You et al., 2017) and (3-Pyrrolinium) MnX_3 (X = Cl, Br) (Ye et al., 2015), 2D-structured NH₃(CH₂)₅NH₃MnCl₄ (You et al., 2017) and (C₆H₅CH₂CH₂NH₃)₂MnCl₄ (Lv et al., 2016), 1D-structured (N-Methylpyrrolidinium)MnCl₃ (Ye et al., 2015), and (pyrrolidinium)MnBr₃ (Sun et al., 2017), as well as 0Dstructured (C4NOH10)₅Mn₂Cl₉·C2H5OH (Zhang et al., 2015), etc. In comparison, less attention has been paid on the search and investigation of tetracoordinated Mn²⁺ counterparts (Xu et al., 2017; Gong et al., 2019; Jana et al., 2019; Sun et al., 2019).

It is well-known that single crystals can exhibit better intrinsic properties of materials compared with the polycrystalline counterparts. Hence, in this work, we first rationally designed and synthesized a novel organic manganese halide, (1-mPQBr)₂MnBr₄ (1-mPQ=1-methylpiperazine, 1-C5H14N2) single crystal with a 0D structure. Systematical characterizations were applied to investigate the structures, photophysical and

thermal properties. The facile solvent-evaporation method, the intense green emission with high PLQY of 60.70% as well as good stability makes the $(1-mPQBr)_2MnBr_4$ suitable as green phosphors.

MATERIALS AND METHODS

Materials

Analytical-grade manganese (II) monoxide (MnO, 99.0%, Sinopharm Co. Ltd.), 1-methylpiperazine (1-C5H12N2, 99.0%, Sinopharm Co. Ltd.), hydrobromic acid (HBr, 40 wt% in H_2O , Sinopharm Co. Ltd.), and hypophosphorous acid aqueous solution (H_3PO_2 , 50% in H_2O , Sinopharm Co. Ltd.) were used as received without any further processing or refining.

Preparation of (1-mPQBr)₂MnBr₄ Single Crystal

As a typical process, the yellow crystals of $(1-mPQBr)_2MnBr_4$ were obtained by slowly evaporating HBr/H₃PO₂ (3:1) mixed solutions containing 1-methylpiperazine and MnO with molar amounts of 2:1.

Characterizations

Powder X-ray diffraction (PXRD) patterns were performed on a Bruker-AXS D8 Advance X-ray diffractometer with CuKα1 radiation ($\lambda = 1.54186$ Å) in the range of 10–90° (2θ).

Single crystal's structure was determined by Bruker SMART APEX-II diffractometer equipped with a CCD detector (graphitemonochromatized Mo-K α radiation, $\lambda = 0.71073$ Å) at 300 K. Data integration and cell refinement were performed using the APEX₂ software. The structure was analyzed by direct methods and refined using the SHELXTL 97 software package. All nonhydrogen atoms of the structure were refined with anisotropic thermal parameters, and the refinements converged for Fo² > 2σ IJFo². All the calculations were performed using SHELXTL crystallographic software package. Symmetry analysis on the model using PLATON revealed that no obvious space group change was needed. The crystallographic data was deposited in Cambridge Crystallographic Data Center (CCDC #1979443).

Fourier transform infrared (FTIR) spectrum in the region 700–4,000 cm⁻¹ was examined on a spectrometer (Nicolet 330) with KBr pellets.

X-ray photoelectron spectroscopy (XPS) measurements of the newly synthesized $(1-mPQBr)_2MnBr_4$ samples about 2 × 1 × 1 mm³ in size were performed on an ESCALAB 250 (Thermo Fisher Scientific) instrument under vacuum $(1.7 \times 10^{-10} \text{ mbar})$.

UV-vis diffuse reflectance spectroscopy was recorded using a Shimadzu UV 2550 spectrophotometer equipped with an integrating sphere over the spectral range 200–800 nm. The (1-mPQBr)₂MnBr₄ single crystals were ground into powders for tests. A BaSO₄ plate was used as the standard (100% reflectance). The absorption spectrum was calculated from the reflectance spectrum by using the Kubelka-Munk function: $\alpha/S = (1-R)^2/(2R)$, where α is the absorption coefficient, S is the scattering coefficient, and R is the reflectance.

¹National Renewable Energy Labs (NREL) Efficiency Chart. Available online at: https://www.nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.20200203.pdf.

Photoluminescence Measurements

The excitation wavelength dependent-photoluminescence (PL) spectra and PL excitation spectra (PLE) were carried out by a laser of 365 nm with a photomultiplier (PMTH-S1-CR131) and DSP lock-in amplifier (SRS 830). The time-resolved photoluminescence measurements (TRPL) were carried by FLS920 all functional fluorescence spectrometer (Edinburgh). The output laser wavelength was set to be 520 nm. The photoluminescence quantum yields (PLQY) were tested by an absolute PLQY measurement system (FLSP920) in an integrating sphere.

Thermalgravimetric Analysis (TGA) and Differential Scanning Calorimetry (DSC) Measurements

TGA and DSC curves were collected using a TGA/DSC1/1600HT analyzer (Metter Toledo Instruments). The polycrystalline sample was placed in an aluminum crucible and heated at a rate of 10 K/min from room temperature to 800°C under flowing nitrogen gas.

RESULTS AND DISCUSSION

Crystal structure of the (1-mPQBr)₂MnBr₄ single crystal was obtained through SCXRD test, which belongs to the orthorhombic crystal system (a non-polar D₂ and chiral space group $P2_12_12_1$) at room temperature, with lattice parameters of a = 8.272(6) Å, b = 15.982(10) Å and c = 17.489(11) Å. In this structure, four formula units of (1-mPQBr)₂MnBr₄ are present in the unit cell. Further details for crystallographic parameters are provided in Tables S1-S3 (in the Supporting Information) and the crystal structure and local structure descriptions are displayed in Figure 1. It is clearly seen that each Mn atom is coordinated by four Br atoms to form an isolated [MnBr₄]²⁻ tetrahedral cluster as inorganic part of the title compound. Such isolated [MnBr₄]²⁻ tetrahedral are surrounded by free Br⁻ and [C5H14N2]²⁺ molecules (organic part), featuring a 0D structure. In Figure 2 shows that the experimental PXRD pattern corresponds well to that calculated from SCXRD result with slightly varying intensities.

The FTIR spectrum in **Figure S1** further verifies the existence of the organic component. The broad peak around 3,353 cm⁻¹ belongs to the N-H stretching peak. The peaks in the range of 2,950–2,820 cm⁻¹ are assigned to the CH2 and CH3 symmetric and asymmetric stretching vibrations and the peak around 1,410 cm⁻¹ is ascribed to CH bending vibrations. Also, the strong signal at nearly 1,630 cm⁻¹ indicates asymmetric NH₃⁺ deformation. **Figure S2** shows the scanning electron microscope photograph of $(1-mPQBr)_2MnBr_4$ crystal. **Table S4** provides the detailed results of energy dispersive X-ray spectroscopy to further confirm the composition of inorganic and organic parts, respectively.

Additionally, the XPS spectrum in **Figure 3A** evidences the signatures of carbon (C 1s), oxygen (O 1s), nitrogen (N 1s), manganese(Mn 2p), and bromide (Br 3d), and the appearance of adventitious oxygen in the spectrum is generally ascribed to



FIGURE 1 | (A) Single crystal structural packing of $(1-mPQBr)_2MnBr_4$ viewed along a-axis direction (H atoms are omitted for clarity). Ball-and-stick scheme of **(B)** a single [C5H14N2]²⁺ organic cation and **(C)** a single [MnBr₄]²⁻ tetrahedron.



a contamination of the sample by physical adsorption during ambient exposure. We further recorded high-resolution spectra of the constituents in designated energy ranges (**Figures 3B–E**). The Mn doublet shows a spin-orbit splitting of 11.7 eV with the peaks corresponding to the binding energies of Mn 2p1/2 and 2p3/2 orbitals located at 653.3 and 641.6 eV, respectively. Similarly, the core-level spectrum of Br 3d contains a couple of split peaks at 69.2 and 68.1 eV corresponding to Br 3d3/2 and Br 3d5/2 orbitals with a separation of 1.1 eV, which are in good agreement with Br. By calculating the integrated peak areas of the XPS spectra, we can roughly estimate that Mn to Br possessed a molar ratio of 1:5.88. According to the analytical results of SCXRD, PXRD, EDS, and XPS, we can verify that



pure $(1-mPQBr)_2MnBr_4$ were successfully synthesized through the solvent-evaporation method.

Figure 4 displays the optical image of a rod-shaped $(1-mPQBr)_2MnBr_4$ single crystal and powders under ultraviolet light irradiation, and it is clearly seen that the title compound emits strong green light. To characterize the photophysical properties of $(1-mPQBr)_2MnBr_4$, the UV-vis absorption spectroscopy and PL spectroscopy were carried out at room temperature. In **Figure 5**, typical absorption spectra for organic manganese halides materials can be observed. The peaks below 340 nm originates from the transitions within the $[C5H14N2]^{2+}$ cation and peaks from 345 to 600 nm should be ascribed to the electronic transitions between the ground and the first excited triplet states of the Mn^{2+} ion in the crystal field, which is consistent with previous reports as listed in **Table S5**.

To further investigate the origin of this green emission, **Figures 6A,B** display the wavelength dependent PL spectra and PLE spectrum of $(1-mPQBr)_2MnBr_4$ single crystals. Upon excitation, the $(1-mPQBr)_2MnBr_4$ exhibits an intense green emission located at 520 nm with a narrow full width at halfmaximum (FWHM) of 43 nm, corresponding to a characteristic transition from the ground state of the d-electron configuration $(e_g)^2 (t_{2g})^3$ to the upper state of the configuration $(e_g)^3 (t_{2g})^2$ (Wrighton and Ginley, 1974; Jiang et al., 2019). It is clearly noted that the PLE spectra are consistent with the absorption spectra. In the region between 300 and 500 nm, the discernable peaks correspond to radiative transitions from the ground state 6A_1



of tetrahedral Mn (II) to the excited states of ${}^{4}T_{1}(G)$, ${}^{4}T_{2}(4G)$, ${}^{4}A_{1}(G)$, ${}^{4}E(G)$, 4E(D), ${}^{4}T_{1}(P)$, ${}^{4}T_{1}(F)$, and ${}^{4}A_{2}(F)$, respectively, according to the excited states of Mn²⁺ system (d5) in the Tanabe-Sugano diagram (Rodríguez-Lazcano et al., 2009).

The PLQY at room temperature is calculated to be about 60.70% and the Commission Internationale de l'Eclairage (CIE)

chromaticity coordinate for this green emission is determined to be (0.175, 0.589) (**Figure 6C**). Upon the excitation of 453 nm, the room-temperature TRPL decay curve is monitored for the



emission peak at 520 nm as shown in **Figure 6D**. The decay curve is modeled with the biexponential decay function:

$$I(t) = A_1 e^{-\frac{t}{\tau_1}} + A_2 e^{-\frac{t}{\tau_2}}$$

where I(*t*) is the time-resolved PL intensity, t is the time after excitation, A₁ and A₂ are the relative amplitudes, and τ_1 and τ_2 are lifetimes for fast and slow decays. The effective decay times are calculated to be 52.9 and 185.0 µs, respectively.

Furthermore, such highly emissive bulk crystals and powders were examined to exhibit considerable thermal stability. TGA curve suggests that (1-mPQBr)₂MnBr₄ does not lose any mass until 300°C (**Figure 7**), which is comparatively higher than that of other organic-manganese halides reported in literature. This is hypothesized to be due to the large amounts of hydrogen bonding interaction between the organic and inorganic components in the 0D structure benefiting from the extra presence of free bromide ions. It possesses a two-step decomposition including the evaporation of organic parts and MnBr₂, respectively. In DSC scan, a sharp endothermic peak, which occurred at 250°C, corresponds to the melting point of (1-mPQBr)₂MnBr₄.

More importantly, the stability of hybrid metal halides is deemed as an important criterion for evaluation of their



green emission at 520 nm was observed for this novel organicinorganic hybrid material, which should be resulted from the spin-forbidden internal transition (${}^{4}T_{1}(G)$ to ${}^{6}A_{1}$) of tetrahedrally coordinated Mn^{2+} ions. We believe the superior photophysical properties and high stability makes it potential for light-emitting applications.

DATA AVAILABILITY STATEMENT

The datasets presented in this study can be found in online repositories. The name of the repository and accession number can be found below: Cambridge Crystallographic Data Centre (CCDC #1979443).

AUTHOR CONTRIBUTIONS

ZC and XT devised the project and proof outline. XJ synthesized the single crystals and conducted all the characterizations. All authors contributed to manuscript revision, read, and approved the submitted version.

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SUPPLEMENTARY MATERIAL

The Supplementary Material for this article can be found online at: https://www.frontiersin.org/articles/10.3389/fchem. 2020.00352/full#supplementary-material

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intensity (Figure S4).

CONCLUSIONS

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potential for practical applications. Therefore, we evaluated the thermal stability of (1-mPQBr)₂MnBr₄ single crystals by

annealing them at 150°C for 12 h on a hotplate. Notably, negligible change can be observed in the PXRD pattern

(Figure S3). Moreover, after exposure to ambient conditions

for 2 months, it still remain 96.3% of the original PL

In summary, we have synthesized a novel lead-free

organic-manganese halide compound (1-mPQBr)₂MnBr₄

(1-mPQ=1-methylpiperazine, 1- C5H14N2), with 0D structure

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Conflict of Interest: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

The reviewer JD declared a past co-authorship with the authors XJ, ZC, and XT to the handling editor.

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