



Hybrid Co₃O₄@Co₉S₈ Electrocatalysts for Oxygen Evolution Reaction

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It is very essential to design an efficient and low-cost electrocatalyst for oxygen evolution reaction (OER). In this work, we report synthesis of $Co_3O_4@Co_9S_8$ heterostructures (Co-HSs) by employing Co_3O_4 nanowires (Co-NWs) as the precursor on nickel foam though a facile hydrothermal method. The unique structures can combine the advantages of one-dimensional nanowires and two-dimensional nanosheets, simultaneously, which possess plentiful defect atoms and active sites. The as-prepared Co-HSs exhibit excellent electrocatalytic performance for OER, which present a low overpotential of 80 mV at 10 mA cm^{-2} during OER process and small Tafel slope of $107.2 \text{ mV} \text{ dec}^{-1}$.

Keywords: nanosheets, $Co_3O_4@Co_9S_8$, heterostructures, electrocatalysis, OER

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INTRODUCTION

In general, overall water splitting can be divided into two half reactions, namely OER (Zhao et al., 2016; Xuan et al., 2019) and hydrogen evolution reaction (HER) (Deng et al., 2017). However, OER usually presents the disadvantage of high potential and low efficiency. To date, many attempts have been made to improve the electrocatalytic performance (Hu et al., 2018; Liu Y. et al., 2019). Of all electrocatalytic materials, RuO₂ (Abbasi et al., 2016) and IrO₂ (Lee et al., 2012) exhibit the best electrocatalytic activities, but their practical application is restricted to some extent because of their scarcity and high cost. Therefore, it is important to develop rich and low-cost alternatives.

Transition metal oxides have been extensively studied due to their excellent electrocatalytic properties, such as MCo2O4 (Zhao et al., 2019a), NiFeOx (Li et al., 2019), CoMoO4 (Liu H. Q. et al., 2019), NiCo₂O₄ (Zhao et al., 2019b), Co₃O₄ (Tan et al., 2019). Among them, Co_3O_4 is a representative example owning to its earth-abundance, durability and high efficiency (Xu et al., 2016). In previous reports, Wang et al. constructed a large number of defect atoms using a plasma-engraved Co₃O₄ nanosheets method with high specific activity. Compared with a pristine sample, its activity is increased by 10 times. The overpotential of the as-obtained samples is 300 mV at 10 mA cm⁻² (Zhao et al., 2016). Li et al. prepare ultrathin Co₃O₄ nanomeshes with many defect atoms and large surface area by an etching-free one-step method. The prepared sample exhibited the overpotential of 307 mV at $10 \text{ mA} \text{ cm}^{-2}$ (Li et al., 2018a). Ma and coworkers fabricate Co₃O₄ nanocrystals grown on nitrogen-doped grapheme oxide with the overpotential of 380 mV at 10 mA cm^{-2} (Han et al., 2017). However, the catalytic performance of Co₃O₄ electrode materials is not satisfactory due to its poor conductivity. Moreover, the OER process is a four-step electron transfer one, which leads to kinetic sluggishness during the reaction process compared to HER with a two-step process. Therefore, many efforts have been made to improve its electrocatalytic efficiency, such as etching (Liu et al., 2017; Lu et al., 2017), doping (Chai et al., 2017; Wang et al., 2018), and facet control (Yang et al., 2017; Li et al., 2018b). Among them, etching is an effective method to increase oxygen defects, which

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make the as-obtained catalysts thin or porous. Therefore, heterostructures are considered to be an effective method because they combine the advantages of different materials.

In this work, we report ultrathin $Co_3O_4@Co_9S_8$ heterostructures (Co-HSs) as electrocatalysts by a simple etching Co_3O_4 nanowires (Co-NWs) approach, revealing a low overpotential of 80 mV at 10 mA cm⁻², low Tafel slope of 107.2 mV dec⁻¹. The prepared samples show excellent cycle activity and still keep 100% current retention after 13 h. The unique structure can provide many defect atoms and active sites. It speeds up electron transfer and then enhances electrocatalytic performance for OER.

EXPERIMENTAL

Material Preparation

All the chemicals were of analytical grade and used as purchased. Prior to typical synthesis, a piece of nickel foam $(4 \times 4 \text{ cm}^2)$ was immersed in 0.5 M hydrochloric acid for 0.5 h, followed by ultrasonic rinsing repeatedly with distilled water and alcohol to remove grease, nickel oxide, and other impurities from surface. Next, the nickel foam was dried in an oven at 60°C for 10 h. In a typical procedure, 1.443 g Co(NO₃)₂, 1.800 g urea and 0.900 g NH₄F were dissolved in 60 ml deionized water, and then stirred for 30 min to obtain a pink clear solution. The solution and the pre-treated Ni foam were transferred to a 100 ml autoclave heated to 120° C for 6 h. After naturally cooling to room temperature, the as-obtained samples were washed repeatedly with deionized water and alcohol, respectively. Subsequently, the sample was dried at 60° C for 10 h in oven. Finally, the as-synthesized samples were annealed at 350° C for 2 h.

The Co-HSs were synthesized by vulcanizing the prepared Co-NWs. 0.40 g Na₂S was dissolved into 40 ml of deionized water and stirred until a clear solution appeared. Next, the abovementioned solution was put into a 100 ml autoclave. Then the autoclave was maintained at 120° C for 6 h. After cooling to room temperature, the as-obtained products were washed several times, and dried at 60° C overnight.

Structure Characterization

The morphology and structure of the as-prepared products were studied by scanning electron microscopy (SEM, Gemini 300-71-31), X-ray diffraction analyzer (XRD, 7000, Shimadzu) by Cu K α radiation ($\lambda = 1.5406$ Å) at 40 kV and transmission electron microscopy (TEM, FEI Tecnai F20, 200kV). X-ray photoelectron spectroscopy (XPS) was used to investigate elemental composition of the sample surface by ESCALAB250 with an Al K α sources.

Electrochemical Characterization

The electrochemical properties of the as-synthesized products were conducted by using an electrochemical workstation (Shanghai Chenhua CHI660e) in a three-electrode system. The as-prepared sample is used as the work electrode, Pt plate as the auxiliary electrode and the saturated calomel electrode (SCE) as the reference electrode. All potentials were converted to reversible hydrogen electrode (RHE) potentials according to the Nernst equation: $E_{RHE} = E_{SCE} + 0.198 + 0.059 \times pH$. E_{SCE} is measured potential. The overpotential (η) was calculated with the equation: $\eta = E_{RHE} - 1.23$.

RESULTS AND DISCUSSION

XRD patterns were used to analyze the phase composition and crystal structure of the Co-NWs and Co-HSs samples. **Figure 1A** shows the characteristic diffraction peaks of the samples. The peak positions located at 44.6°, 52.0°, 76.6° are the peaks of nickel foam (JCPDS NO. 42-0712). The other peaks at 31.3°, 36.9°, 55.7°, 59.4°, 65.2° are assigned to (220), (311), (422), (511), and (440) crystal planes of spinel Co₃O₄ samples (JCPDS NO. 42-1467). No other diffraction peaks were detected, suggesting their high purity. **Figure 1B** is XRD pattern of the sample after vulcanizing. In addition to the diffraction peaks from Co₃O₄ sample, the peaks at 29.7°, 31.1°, 39.4°, 40.6°, 61.7°, 65.1°, and 76.4° correspond to (311), (222), (331), (420), (622), (444), and (800) crystal planes of Co₃S₈ (JCPDS No. 02-1459). It reveals that Co₃O₄ samples were vulcanized partially.

To further confirm the composition of the sample, XPS measurements are carried out. Figure 1C shows five elements (Ni, Co, O, C, and S) in the full spectrum. The binding energies at 887.2, 807.2, 540.2, and 172.1 eV correspond to Ni 2p, Co 2p, O 1s, and S 2p, respectively. Figure 1D shows Co 2p XPS spectra with two sharp $2p_{1/2}$ and $2p_{3/2}$ doublet peaks of Co^{2+} and Co^{3+} with separation of 16.4 eV and a pair of shake-up satellites. The fitting peaks at 797.2 and 781.7 eV correspond to Co^{2+} , whereas the fitting peaks at 795.7 and 779.4 eV are identified as Co³⁺. The O 1s XPS spectra (Figure 1E) are identified by four peaks at 529.5, 531.2, 532.5, 533.5 eV, which correspond to the binding energy between oxygen atoms and basal plane, oxygen atoms binding to defect atoms, oxygen atoms in hydroxyl groups, and adsorbed water molecules, respectively. The O signal between oxygen atoms and defect atoms indicates the asprepared products possess many oxygen defect atoms (Zhuang et al., 2017). In addition, S 2p fitting spectra at 169.4 and 167.6 eV can be ascribed to S $2p_{1/2}$ and S $2p_{3/2}$, respectively, as shown in Figure 1F. The results are consistent with XRD results.

The morphologies of the as-obtained products were observed by SEM first. **Figure 2a** shows that the many nanowires are uniformly covered on the surface of nickel foam. Further observation finds that the average diameter of each nanowire is tens of nanometers (**Figure 2b**). The shapes of the hybrid products present sheet-like structures after vulcanizing, as shown in **Figures 2c,d**. The spatial structure can combine the advantages of one dimensional nanowires and two-dimensional nanosheets with large specific surface area and active sites. The low TEM image in **Figure 2e** further proves the sheet-like structure of the products. HRTEM image shows the lattice spacings between adjacent lattice fringes are 0.243 and 0.191 nm, respectively, which correspond to (311) and (511) planes for Co₃O₄ and Co₉S₈ material (**Figure 2f**).

Cyclic voltammetry (CV) curve is an important indicator to investigate the electrochemical performance of the product in a traditional three-electrode system. From **Figures 3A,B**, it





FIGURE 2 | Morphology characterization, (a,b) SEM image of Co-NWs, (c,d) Co-HSs, and (e,f) TEM images of the Co-HSs.



can be found that the shape of CV curves does not change with the increasing of scan rate. And the charging current of Co-HSs is larger than that of Co-NWs at the same scan rate, indicating that the prepared samples possess larger areas than Co-NWs. Then, the electrochemical double-layer capacitances $(C_{\rm dl})$ of the electrodes are estimated by using the following equation: $i = vC_{dl}$ (*i* is current density, v refers to scan rate). Through the linear relationship between *i* and v, the C_{dl} of Co-HSs and Co-NWs are 0.2198 and 0.1280 F cm⁻², respectively (Figure 3C). The electrochemically active surface area (ECSA) plays the crucial role in evaluating the ability of water oxidation of electrocatalyst. The as-prepared composite shows a high ECSA and many active sites, suggesting that the Co-HSs presents better electrocatalytic performance than that of Co-NWs. EIS is studied in a frequency range from 0.01 to 100,000 Hz with an amplitude of 5 mV (Figure 3D). Charge transfer resistance (R_{ct}) can be obtained by the diameter of the semicircle at the high frequency region in Nyquist plots. The smaller the diameter, the faster the charge transfers. It can be found that the resistance of Co-HSs is much smaller than that of Co-NWs.

Then, the electrocatalytic activities of the samples are investigated further by linear sweep voltammetry (LSV) at 2 mV s⁻¹ for OER. Co-HSs show lower OER onset potential (1.28 V) than Co-NWs (1.39 V) (**Figure 4A**). After a run time of 13 h, Co-HSs still show a low potential of 1.32 V, indicating that the Co-NWs show excellent cycle stability. The overpotential is very important to evaluate oxygen evolution ability. Co-HSs require merely 80 mV at 10 mA cm⁻² to trigger OER, which is lower than Co-NWs. It shows that the synthesized samples are more

prone to water splitting than Co-NWs at the same current density. At 100 mA cm⁻², Co-HSs exhibit the overpotential of 350 and 380 mV after cycling, respectively. However, the Co-NWs require 420 mV to achieve 100 mA cm⁻² (Figure 4B). It is the case that the interfaces of hybrid structures cause the redistribution of charge, which could cause the change of the Gibbs' free energy (Wang et al., 2017). The electronegativity of sulfur is stronger than that of oxygen, the charge can transfer from Co₃O₄ nanowires to Co₉S₈ nanosheets until the ΔG_H tends thermo-neutral state (Wang et al., 2017). In alkaline environment, a four-step reaction for OER can be expressed by the following mechanisms:

$$OH^- +^* \to OH^* + e^- \tag{1}$$

$$OH^* + OH^- \rightarrow O^* + H_2O + e^-$$
(2)

$$O^* + OH^- \to OOH^* + e^- \tag{3}$$

$$OOH^* + OH^- \rightarrow O_2 + H_2O + e^-$$
(4)

Where * refers active site. The performance of the samples is mainly affected by the second process (14). The charge transfer reduces Fermi energy, which reduces the filling of antibonding states and increases O* absorption. The overpotential in this work is lower than those of many other Co-based electrocatalysts (**Table 1**; Mendoza-Garcia et al., 2016; Yuan et al., 2016; Zhao et al., 2016; Chai et al., 2017; Han et al., 2017; Tahir et al., 2017; Yang et al., 2017; Li et al., 2018a,b; Liu et al., 2018; Wang et al., 2018).

Tafel slope is used to reflect the catalytic mechanism of OER, which bridges the relationship between overpotential and current



TABLE 1 Comparison of the as-prepared Co-HSs electrochemical performance with literature reported the Co-based electrocatalysts.

Catalysts	Electrolyte	η (10 mA cm ⁻²)	References
Co-HSs	1 M KOH	80 mV	This work
Co ₃ O ₄ @MoS ₂	1 M KOH	90 mV	Liu et al., 2018
Co ₃ O ₄ Nanomeshes	1 M KOH	307 mV	Li et al., 2018a
Co hydroxide nanosheets	1 M KOH	326 mV	Tahir et al., 2017
Co ₃ O ₄ -NP/N-rGO	1 M KOH	380 mV	Li et al., 2018b
Co ₃ O ₄ @N-rGO Nanosheets	1 M KOH	490 mV	Han et al., 2017
P-Co ₃ O ₄ /NF nanowires	1 M KOH	260 mV	Wang et al., 2018
Co-B nanosheet	0.1 M KOH	520 mV	Yang et al., 2017
Co ₃ O ₄ nanosheets	0.1 M KOH	300 mV	Zhao et al., 2016
Co-Fe phosphide	0.1 M KOH	370 mV	Mendoza-Garcia et al., 2016
CoO nanosheet/graphene	0.1 M KOH	330 mV	Yuan et al., 2016
P, N Co-doped graphene	0.1 M KOH	320 mV	Chai et al., 2017

density. Low value indicates excellent kinetic characteristics. In **Figure 4C**, Co-HSs show lower Tafel slope (107.2 mV dec⁻¹) than Co-NWs (116.2 mV dec⁻¹), revealing their fast kinetics transfer process. The slope of the as-synthesized samples still increases even after 13 h, suggesting their excellent cycling performance.

Finally, the long term durability of Co-HSs is measured through chronoamperometry potential with a current density of 52 mA cm^{-2} (Figure 4D). As time increases, current intensity gradually increases. It reaches a maximum at 12 h, and then begins to decrease. After 13 h, current density for OER is 63 mA cm⁻², indicating the increasing of 21%. It may be attributed to

the reason that gradual increasing of current intensity before 12 h induces slow wetting of the electrode material during the cycling process, which results in a gradual increasing of the reaction interface and the active site. Current intensity gradually decreases after 12 h due to the mechanical loss and activate sites during gas evolution process. Co-HSs show excellent cycle durability. It could be explained as follows. Firstly, active materials directly grown on the substrate can ensure good electron contact and mechanical adhesion with the conducting substrate. Secondly, the synergistic effect between Co_3O_4 nanaowires and Co_9S_8 nanosheets is beneficial to speed up the dynamics process.

Thirdly, the as-prepared Co-HSs show abundant active sites, which can provide adequate chemical reaction spaces occurring between the active materials and electrolyte. Therefore, the asprepared Co-HSs provide excellent durability and high oxygen evolution performance.

CONCLUSION

In summary, Co-HSs have been synthesized by employing Co-NWs as precursor though a facile hydrothermal method. The as-obtained samples present many active sites and low charge transfer resistance. As a electrocatalyst for OER, Co-HSs show high electrocatalytic activity compared to Co-NWs. Meanwhile, the obtained Co-HSs exhibit low overpotential and outstanding cyclic performance, revealing that the Co-HSs might be a promising candidate in water splitting.

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DATA AVAILABILITY STATEMENT

The datasets generated for this study are available on request to the corresponding author.

AUTHOR CONTRIBUTIONS

Y-LT makes substantial contributions to conception and design. LX and M-ZD make contributions to acquisition of data. XW makes contributions to analysis and interpretation of data.

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Conflict of Interest: The authors declare that the research was conducted in the absence of any commercial or financial relationships that could be construed as a potential conflict of interest.

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